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Significance of particle size and charge capacity in TiO₂ nanoparticle-lipid interactions

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Abstract

Hypothesis

The activity of submicron sized titanium oxide (TiO₂) particles towards biomembrane models is coupled to their charge carrying capacity and their primary particle size.

Experiments

Electrochemical methods using a phospholipid layer on a mercury (Hg) supported membrane model have been used to determine the phospholipid monolayer activity of TiO₂ as an indicator of biomembrane activity. The particles were characterised for size by dynamic light scattering (DLS) and scanning electron microscopy (SEM) and charge by acid-base titration.

Findings

TiO₂ nanoparticles aggregate in 0.1 mol dm⁻³ solutions of KCl. The charge capacity of TiO₂ nanoparticles depends on their primary particle size and is unaffected by aggregation. TiO₂ particles of < 20 nm primary particle size interact significantly with phospholipid layers. Aggregation of these particles initially has a small effect on this interaction but long term aggregation influences the interaction whereby the aggregates penetrate the lipid layer rather than adsorbing on the surface. Fulvic acid does not inhibit the <20 nm particle/phospholipid interaction. P25 TiO₂ particles of larger particle size interact less strongly with phospholipid layers and the interaction is alleviated following particle aggregation. The semiconductor properties of TiO₂ are evident in voltammograms showing electron transfer to TiO₂ adsorbed on uncoated Hg.

Keywords: Titanium dioxide nanoparticles; Phospholipid monolayers; Rapid cyclic voltammetry; Particle size; Particle aggregation, Semiconductor properties.

Introduction

1 Titanium dioxide (TiO₂) nanoparticles are one of the most widely used nanomaterials in current
2 application and have extensive use as sunscreens and paints. Accordingly they have a
3 considerable release into environmental systems. Recent estimations have predicted surface
4 water concentrations of discharged TiO₂ to be 21 ng dm⁻³ [1] although these values have yet to
5 be confirmed experimentally. TiO₂ is refractory with a very low water solubility as well as showing
6 a strong tendency to aggregate [2-6]. In spite of this, its biological activity remains uncertain with
7 conflicting reports as to its hazard to environmental and human health [7-10]. A specific feature of
8 TiO₂ is its semiconductor properties which means that the material can absorb UV which enables
9 its photoactivity [11] and photocatalytic activity [12] and may enhance its incipient toxicity [13].
10 Another property of TiO₂ is its existence in three mineral phases anatase, rutile and brookite
11 respectively [11]. The metastable anatase and brookite phases convert irreversibly to the
12 equilibrium rutile phase upon heating above temperatures in the range 600°-800 °C [14]. Rutile
13 and anatase are the more common forms of TiO₂ with rutile being the most common [11]. In all
14 mineral phases, Ti is octahedrally co-ordinated to oxygen atoms [11].

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17 The aggregation of TiO₂ dispersions in aqueous conditions is mainly due to their lyophobicity
18 promoted by their mineralisation and surface features as well as their overall dimensions [15].
19 Previous studies in this laboratory have looked at SiO₂ [16,17], ZnO [18] and CdTe [19]
20 nanoparticles correlating their chemical and physical characteristics with their activity towards
21 biological membranes (biomembranes) and/or biomembrane-like layers. Biomembrane activity is
22 defined as the tendency of nanoparticles to structurally modify and /or permeate in, biomembranes
23 and/or biomembrane-like layers. Although SiO₂, ZnO and CdTe nanoparticles differ considerably
24 from each other in their structural properties and functionality, it is shown that for all classes their
25 biomembrane activity is dependent on their particle size. This study continues the theme of
26 relating the physical and chemical properties of nanoparticles to both their electrochemical activity
27 and their activity towards phospholipid model membranes working with representative research
28 grades of TiO₂. The aim of the work is to develop a general model describing the relation between
29 inorganic oxide particle characteristics and their biomembrane activity.
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Experimental

Materials

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36 All nanoparticle dispersion concentrations are presented as weight per volume. In the following:
37 TiO₂ nanoparticles employed in this study are described together with their particle sizes in
38 parentheses as quoted by supplier. P25 TiO₂ nanoparticles which are 73 - 85 % anatase, 14 - 17
39 % rutile and 0 - 18 % amorphous TiO₂ (20-30 nm) [20] was sourced from Degussa. Rutile (30 nm),
40 and a 4:1 mix of anatase and rutile (20 nm) powders respectively and rutile and anatase
41 nanoparticle (5-15 nm) dispersions in water (15% w/v, 99.9% purity) were obtained from US
42 Research Nanomaterials, Inc. The anatase and rutile dispersions are referred to as "stock
43 dispersions" in the remainder of the paper. 1,2-dioleoyl-sn-glycero-3-phosphocholine (DOPC) was
44 purchased from Avanti Polar Lipids and had a purity of >99%. Analytical grade NaOH, KNO₃ and
45 70% HNO₃ were purchased from Sigma-Aldrich. All glassware was rinsed before use with MilliQ
46 18.2MΩ water (Millipore, U.K). Piranha solution used to treat the Pt electrodes consists of H₂SO₄
47 (Fisher Scientific) and H₂O₂ (Sigma-Aldrich) in a 3:1 ratio respectively.
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Dynamic light scattering (DLS) and scanning electron microscopy (SEM)

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52 The stability of TiO₂ nanoparticles in various media was measured using DLS on a nanoZS
53 zetasizer (Malvern Instruments Ltd.). The media tested consisted of MilliQ water and 0.1 mol dm⁻³
54 KCl. Samples for DLS assay were prepared by adding 10 μL of 10% w/v TiO₂ dispersions into a
55 cuvette containing 1 cm³ of the desired media to give a testing concentration of 0.1% TiO₂. All
56 dispersions were sonicated for 1 minute prior to DLS analysis. In the SEM measurements, a 0.1
57 cm³ sample was deposited on to a SEM stub and dried within a stream of nitrogen gas (Air
58 Products). The nanoparticle samples were sputter coated with a 3 nm thick Pt/Pd layer prior to
59 SEM imaging. SEM imaging was carried out at 3 kV on a LEO 1530 Gemini FEG-SEM (Carl Zeiss
60 ZMT) with an Oxford Instruments AZtecEnergy EDX system attached. Anatase and rutile "stock
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dispersions" were also tested but were not mixed with any media as they were already fully dispersed.

Charge-pH measurements

10% dispersions of each TiO₂ sample were prepared by the addition of 20 g of nanoparticles to 200 cm³ of MilliQ water (Millipore, U.K) in plastic beakers. The dispersion was then brought to pH 2 by the dropwise addition of 75% HNO₃ and sonicated in a Branson 2100 sonicator for 2 hours. Samples were stored in plastic bottles at 25°C for a maximum of 1 week before being purified by dialysis. Purification of nanoparticle dispersions is necessary since any impurities and salts that may be present as a result of the manufacturing process must be removed in order to determine surface charge accurately. Dialysis was carried out in MilliQ water using dialysis tubing (Visking 1350/2) that had been heated to 100°C in MilliQ water for two minutes and allowed to cool. All Visking tubes were rinsed successively with MilliQ water to remove any major impurities. TiO₂ dispersions were ultrasonicated for one hour prior to dialysis to ensure maximum disaggregation. Dialysis was carried out against MilliQ water for three days. Following dialysis the dispersions were re-adjusted to pH 2 by addition of HNO₃. They were then ultrasonicated for one hour to ensure stability. All samples were used within 1 month of purification. 25 cm³ of 10% w/v nanoparticle dispersion was diluted with 25 cm³ of 0.2, 0.02 and 0 mol dm⁻³ KNO₃ to give 5% TiO₂ in 0.1, 0.01 and 0 mol dm⁻³ KNO₃ respectively in a 100 cm³ beaker. The TiO₂ dispersions were titrated with 0.1 mol dm⁻³ NaOH to pH 9-10 under conditions of constant stirring. The solution pH was measured using a Corning pH Meter 240. In the analysis, the H⁺ ions titrated as estimated from the pH value were subtracted from the moles of NaOH added to give the titrated H⁺ ions adsorbed on the TiO₂. The error in this estimation due to the depression of the H⁺ activity by the ionic strength is within the error of the pH measurement [21]. An additional complication is the fact that the H⁺ activity can be increased in the presence of colloidal dispersions [22] contributing an error if the pH of the dispersion is compared to that of a pure solution. An advantage therefore of estimating the H⁺ titrated from the pH value of the TiO₂ dispersion is that it represents an internal control. Relative charge per gram TiO₂ was plotted against the solution pH. The absolute charge values were obtained by subtracting the charge value found at the solution pH equivalent to the position of zero charge (PZC) of TiO₂, from the relative charge value. The PZC of TiO₂ is the point where zero charge resides on the particle surface. The pH value of the TiO₂ dispersion equivalent to the PZC of TiO₂ was determined by measuring the pH of a concentrated (~15%) pure dispersion [23]. To do this a TiO₂ dispersion acidified to pH 2 was ultrasonicated and dialysed as above. It was then brought to pH 7 by addition of 0.1 mol dm⁻³ NaOH. Following this, it was dialysed until the dispersion pH reached a constant value. For all dispersions, this was recorded as pH 6.3 +/- 0.15. This value was generally in agreement with the literature value for the pH value in equilibrium with the PZC of TiO₂ [23] and was used throughout to obtain absolute charge values for the TiO₂ samples.

Electrochemical measurements

The electrochemical testing system used consisted of a peristaltic pump (Cole-Parmer Instrument Co.), a 10 dm³ electrolyte reservoir, a 25 cm³ sample cell and a 0.75 cm³ flow cell constructed of Plexiglas [16,18]. Two universal valve switch modules (Anachem Ltd.) were used to control the supply of either electrolyte or test sample to the pump and all mechanical components were connected with rubber tubing. The contents of both the electrolyte reservoir and sample cell were constantly stirred by magnetic stirrers and aerated with argon (Air Products) prior to each test. The working electrode consisted of a microfabricated one mm Pt electrode base (Tyndall National Institute, Ireland) on a silicon wafer with an electrodeposited Hg surface as Hg film electrode (MFE) [24]. The auxiliary electrode was a rectangular Pt film fabricated on the wafer. The Pt electrodes were first cleaned with piranha solution and rinsed with MilliQ water. They were then dried with a stream of N₂ and the wafer was placed in the flow cell. 50 mmol dm⁻³ Hg(NO₃)₂ solution was passed through the flow cell at a rate of 5 cm³ min⁻¹. Reduction of Hg on to the Pt electrode occurred at -0.4V vs Ag/AgCl. Subsequent to electrodeposition, the electrode was washed with a stream of MilliQ water. A Ag/AgCl 3 mol dm⁻³ KCl REF 201 electrode (VWR International Ltd.) was placed in the flow cell as reference and all potentials in this paper are quoted against this. All electrodes were connected to a PGSTAT 30 Autolab potentiostat (Ecochemie, Utrecht, The Netherlands). Rapid cyclic voltammetry (RCV) was carried out in 0.1

1 mol dm⁻³ KCl + 0.01 mol dm⁻³ phosphate (PBS) electrolyte at pH 7.4 with a total ionic strength of
2 0.125 mol dm⁻³. Additional experiments in 0.1 mol dm⁻³ KCl showed that the phosphate had no
3 effect on the TiO₂ interaction with the DOPC layers. 300 μL of DOPC dispersion of concentration
4 0.254 μm cm⁻³ was injected into the flow cell at a rate of 1 cm³ min⁻¹ applying RCV from -0.4V to -
5 3V at 80 Vs⁻¹. Following the appearance of the DOPC voltammetric peaks, RCV was restarted at a
6 voltage excursion of -0.4 to -1.625V to test the integrity of the DOPC monolayer. Subsequently the
7 potential was switched off and electrolyte was replaced with a 0.1% dispersion of TiO₂ in MilliQ
8 water in the flow cell. The electrode was exposed for 30 seconds to the TiO₂ dispersion. In one
9 experiment the electrode was exposed to 0.1% TiO₂ dispersion in soil water containing 5 mg dm⁻³
10 humic acid. The TiO₂ dispersion was replaced with electrolyte, the electrode was tested by RCV
11 from -0.4V to -1.125V and in some cases to -1.8 V at 40Vs⁻¹ and scans were taken at 1 second
12 intervals for 30 seconds. RCV was then terminated and the system washed through with
13 electrolyte at a flow rate of 10 cm³ min⁻¹. RCV was restarted from -0.4V to -3V at 80Vs⁻¹ to clean
14 the Hg electrode, at which point all voltammetric peaks disappeared indicating monolayer removal
15 and only the uncoated Hg and electrolyte remained. Following this, the DOPC layer was re-
16 deposited as described above. This monitoring/cleaning/redeposition process was repeated as
17 described above for as long as it was necessary to take measurements. Measurements on an
18 uncoated Hg electrode were carried out in the same way with the same TiO₂ dispersions in MilliQ
19 water. In this case the Hg was cleaned in between measurements using the same procedure as
20 described above for removing the DOPC layer.

21 Results and discussion

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24 Figure 1 summarises the DLS results for all the samples. It is noted that the particle sizes as
25 determined by DLS bear little relation to those quoted by the supplier. SEM results for the
26 particles displayed in Figure 2 also show little agreement as seen by eye with the supplier's values
27 and show varying degrees of aggregation in particular the rutile (Figure 2(b)) and anatase/rutile
28 (Figure 2(c)) powders. The DLS results indicate that in the presence of 0.1 mol dm⁻³ KCl, TiO₂
29 particles aggregate and the extent of aggregation is variable. Interestingly the anatase "stock
30 dispersion" showed the smallest primary particle size at 16 nm confirmed by SEM in Figure 2(e)
31 which suggests a relatively monodispersed sample. This sample was the most stable aggregating
32 to 67 nm after 30 minutes incubation in 0.1 mol dm⁻³ KCl. There is some correlation between the
33 samples in Figures 1 and 2 in particular the primary peak in DLS and the SEM images but the
34 drying effects in SEM samples will have an effect. The secondary peak in DLS is not observed at
35 this SEM magnification. The charge-pH plots are displayed in Figure 3. Their derivation from the
36 titration is summarised in the inset to Figure 3(a) which displays plots of moles NaOH added
37 versus moles H⁺ titrated calculated from *in-situ* pH measurements. It is evident that at low pH
38 values the plots overlay each other since the TiO₂ is saturated with protons and only the solution
39 H⁺ is titrated whereas at higher pH values the H⁺ dissociated from the particle surface is titrated.
40 The overlay of the results at the low dispersion pH value confirms the validity of the experimental
41 analysis taken. The main plot in Figure 3(a) indicates that the P25 dispersion has a charge
42 capacity of more than twice as much as the rutile "stock dispersion" and the rutile and
43 anatase/rutile powder dispersions. This charge-capacity relates to its particle size as shown by
44 DLS which is smaller than that of the other particle samples except for the anatase "stock
45 dispersion". Figure 3(b) shows that the anatase "stock dispersion" holds the highest charge on its
46 surface relating to its smallest particle size as indicated by DLS. Interestingly for all particle
47 dispersions, the charge carried increases with increasing ionic strength (Figure 3(c)) as shown
48 previously due to increased counter-ion association [23].

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53 Figure 4 shows the effect of the particle dispersions on the voltammograms of uncoated Hg. In this
54 case an increase in current is seen in relation to an increase in negative potential commencing at
55 ~ -0.8 V. This increase in current is significant when the uncoated electrode is exposed to P25
56 (Figure 4(a)) and is highest when exposed to the anatase "stock dispersion" (Figure 4(e)). The
57 ranking of current increase is related to the particles' charge carrying capacity. The semiconductor
58 effect is similar to that observed when uncoated Hg electrodes were exposed to ZnO particle
59 dispersions [18]. The increase in current can therefore be related to the semiconductor
60 properties of TiO₂ and the injection of electrons under applied potential into the conduction band of
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1 TiO₂. This effect has been observed previously on nanocrystalline TiO₂ electrodes in aqueous
2 electrolyte [25]. The TiO₂ is adsorbed on the electrode since following incubation with the TiO₂
3 dispersion in MilliQ water, the electrode was monitored by RCV with pure electrolyte in the flow
4 cell. Figure 5 displays the voltammograms of the DOPC coated electrode exposed to the TiO₂
5 dispersions. As noted previously [16], adsorption of particles on the DOPC surface is seen as a
6 depression of the capacitance current peaks on the voltammogram. The P25 dispersion interacts
7 to a small extent with the lipid layer (Figure 5(a)) correlating with its increased charge capacity
8 while three of the other TiO₂ dispersions show insignificant interaction with the DOPC layer
9 (Figures 5(b), 5(c) and 5(d)). The anatase "stock dispersion" interacts strongly with the DOPC layer
10 as evidenced by the almost complete suppression of the capacitance current peaks after 30 s
11 exposure to a 0.1% TiO₂ dispersion in MilliQ water (Figure 5(e)). The P25 interaction was probed
12 more extensively by recording voltammograms with a greater voltage excursion to -1.8 V. This is
13 displayed in Figure 6(a) where the interaction of the P25 dispersion in MilliQ water with the DOPC
14 is clearer, exemplified by the increase in current at potentials more negative than those
15 characterising the DOPC reorientations. This is caused by electron transfer to TiO₂ brought close
16 to the electrode surface following the DOPC structural transitions. Interestingly following incubation
17 of the dispersion in 0.1 mol dm⁻³ KCl for one minute, the interaction is greatly decreased (Figure
18 6(b)) and is similar to that of the remaining TiO₂ dispersions in MilliQ water as exemplified by the
19 effect of the rutile powder dispersion on DOPC layers (Figure 6(c)). This indicates that P25
20 nanoparticle aggregation in 0.1 mol dm⁻³ KCl decreases its activity towards the DOPC layer.

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22 Figure 7 shows the effect of the incubation of the anatase "stock dispersion" for 30 minutes in 0.1
23 mol dm⁻³ KCl where there is some particle aggregation to 67 nm (Figure 1(e)). A strong
24 interaction of the dispersion with the DOPC layer remains as shown in Figure 7(a). Interestingly
25 after 8 hours incubation with 0.1 mol dm⁻³ KCl, the interaction of an aggregated anatase "stock
26 suspension" with the DOPC layer is altered whereby the depression of the capacitance peak
27 current is lessened but an increase in the baseline capacitance current with increasing negative
28 potential is observed (Figure 7(b)). This baseline increase can be better interpreted following sight
29 of the same in the RCVs in Figure 4 which arises from electron transfer to electrode-adsorbed
30 TiO₂ particles. Accordingly the baseline increase in Figure 7(b) corresponds to electron transfer to
31 TiO₂ aggregates which have penetrated through the lipid layer to the Hg surface. The aggregates
32 have lower affinity for the monolayer surface due to their increased particle size which allows their
33 penetration into the layer to take place. Of interest also is that TiO₂ particle dispersions in fulvic
34 acid media display strong interaction with the DOPC layer (Figure 7(c)) which is in contrast to ZnO
35 particle dispersions where the fulvic acid alleviates interactions between the particles and the lipid.
36 Stabilisation of the small TiO₂ particle size by fulvic acid has been reported [26] and this would
37 enhance interaction with DOPC as observed by the effect of dispersant on ZnO interaction [18].
38 On the other hand, it is considered that the solubility of ZnO to Zn²⁺ ion will facilitate complexation
39 of Zn(II) fulvate complex [27] on the surface of ZnO which may inhibit the interaction of ZnO with
40 the lipid. Figure 7(d) confirms that the fulvic acid itself does not interact with the DOPC layer [18].

41 Conclusions

42 *A summary of the findings.*

43 TiO₂ nanoparticles show a strong tendency to aggregate in media of ionic strength ~ 0.1 mol dm⁻³
44 KCl. Their charge holding capacity depends on their primary particle size and is unaffected by their
45 aggregation. TiO₂ particles below primary particle size 20 nm interact strongly with DOPC layers.
46 Aggregation of these particles has a small effect on this interaction but long term aggregation
47 influences the interaction whereby the aggregates penetrate the lipid layer rather than adsorbing
48 on the surface. The significant interaction of the P25 dispersion with the DOPC layer as compared
49 to the remaining dispersions is related to its charge holding capacity which is higher and its
50 primary particle size which is smaller than the other dispersions excepting the anatase "stock
51 dispersion". The semiconductor properties of TiO₂ are shown in the voltammograms of uncoated
52 Hg and can be used as a probe to indicate penetration of the DOPC layers by the TiO₂ in particular
53 at higher applied potentials than those characterising the DOPC reorientations. The
54 semiconductor properties are also related to the particles' charge carrying capacity and inversely
55 related to their particle size.

56 *A synopsis of your new concepts and innovations.*

1 A proven concept in this study is the coupling of TiO₂ particle biomembrane activity, charge
2 carrying capacity and primary size. An additional significant concept is that aggregates of the
3 smallest size TiO₂ particles can retain their phospholipid layer activity and their charge carrying
4 capacity since they are necessarily loosely bound at least initially.

5 *A brief restatement of your hypotheses.*

6 The biomembrane activity of TiO₂ particles is coupled with their charge carrying capacity and
7 primary particle size.

8 *A comparison with findings by other workers [give references].*

9 The aggregation of TiO₂ nanoparticles follows that of ZnO nanoparticle aggregation [18]. The
10 biomembrane activity of TiO₂ nanoparticles with primary particle size below 20 nm can be
11 compared to that of SiO₂, ZnO and CdSe and CdTe nanoparticles [16-19]. A significant difference
12 between TiO₂/DOPC and ZnO/DOPC interaction is that solution fulvic acid and phosphate does
13 not inhibit the <20 nm particle interaction which is probably associated with the insolubility of TiO₂
14 compared with the ability of ZnO to release Zn²⁺ ion to aqueous solution [18]. The semiconductor
15 properties of TiO₂ observed in their voltammetry is similar to that of ZnO nanoparticles and has
16 been observed by other workers using similar electrochemical methods [25].

17 *Your vision for future work.*

18 Future work in this area should compare the effects of TiO₂ nanoparticles on the electrochemical
19 membrane model with other membrane models in particular vesicles to exactly elucidate the
20 interaction mechanism using direct imaging methods such as confocal microscopy.

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23 NMP-229244. Many thanks to project student Harris Jammaluddin for carrying out the initial TiO₂
24 titrations. Characterised soil water was provided by Dr Claus Svendsen from the NERC Centre for
25 Ecology and Hydrology, Wallingford, UK.

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Figure legends

Figure 1

DLS number particle size distribution of 0.1% dispersions of: (a) P25, (b) rutile powder, (c) mixed anatase/rutile (4:1) powder, (d) rutile "stock dispersion" and, (e) anatase "stock dispersion" in, MilliQ water (black line), 0.125 mol dm⁻³ PBS (red line) and (e) in 0.125 mol dm⁻³ PBS, incubation time: 1 (red dash line), 15 (purple line) and 30 min (blue line).

Figure 2

SEM images of dried 0.1% dispersions of (a) P25, (b) rutile powder, (c) mixed anatase/rutile (4:1) powder, (d) rutile "stock dispersion" and, (e) anatase "stock dispersion".

Figure 3

Charge per mass (coulomb g⁻¹) versus solution pH of 5% (a) P25 (solid triangle), rutile powder (solid circle), mixed anatase/rutile (4:1) powder (open square) and rutile "stock dispersion" (cross), in 0.1 mol dm⁻³ KCl, (b) anatase (open triangle) and rutile (cross) "stock dispersions" and, (c) P25 in 0.1 (solid triangle), 0.01 (open circle) and 0 (diagonal cross) mol dm⁻³ KCl. Inset to (a) moles of added NaOH (solid triangle) and moles titrated solution H⁺ (open circle) calculated from *in-situ* pH, versus pH of 5% rutile "stock dispersion".

Figure 4

RCV (scan rate 40 Vs⁻¹) of uncoated MFE in 0.125 mol dm⁻³ PBS at pH 7.4 before (black line) and after (red line) exposure to 0.1% (a) P25, (b) rutile powder, (c) mixed anatase/rutile (4:1) powder, (d) rutile "stock dispersion" and, (e) anatase "stock dispersion" in MilliQ water.

Figure 5

RCV (scan rate 40 Vs⁻¹) of DOPC coated MFE in 0.125 mol dm⁻³ PBS at pH 7.4 before (black line) and after (red line) exposure to 0.1% (a) P25, (b) rutile powder, (c) mixed anatase/rutile (4:1) powder, (d) rutile "stock dispersion" and, (e) anatase "stock dispersion" in MilliQ water.

Figure 6

RCV (scan rate 40 Vs⁻¹) of DOPC coated MFE in 0.125 mol dm⁻³ PBS at pH 7.4 before (black line) and after (red line) exposure to (a) 0.1% P25 in MilliQ water, (b) 0.1% P25 incubated for one minute in 0.125 mol dm⁻³ PBS at pH 7.4 (c) rutile powder in MilliQ water.

Figure 7

RCV (scan rate 40 Vs⁻¹) of DOPC coated MFE in 0.1 mol dm⁻³ PBS before (black line) and after (red line) exposure to 0.1% anatase "stock dispersion", incubated for (a) 30 minutes and (b) 480 minutes, in 0.1 mol dm⁻³ KCl and (c) in 10% soil water containing 5 mg dm⁻³ fulvic acid, and DOPC coated MFE exposure to (d) 10% soil water containing 5 mg dm⁻³ fulvic acid.

Figure 1

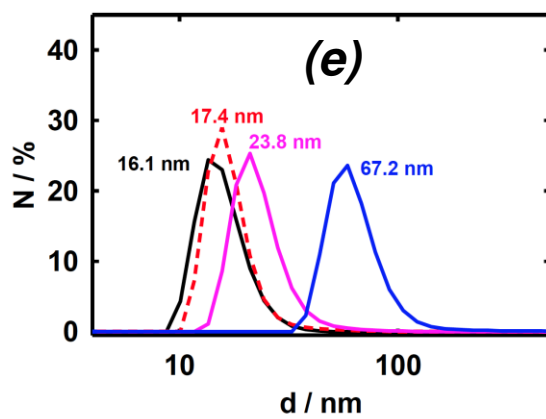
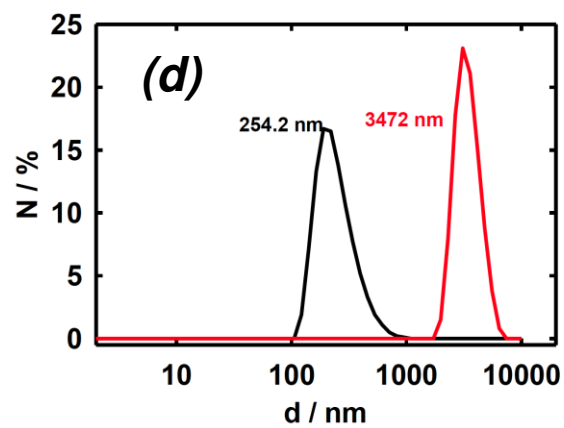
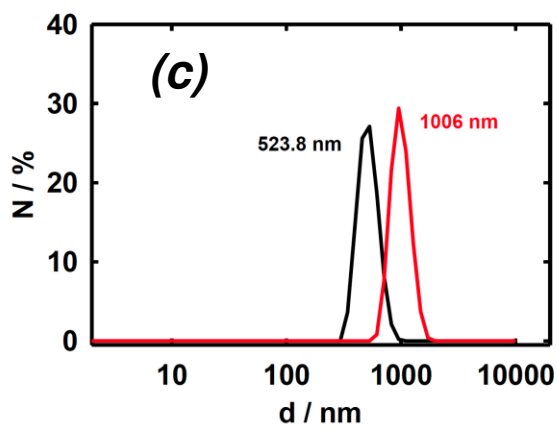
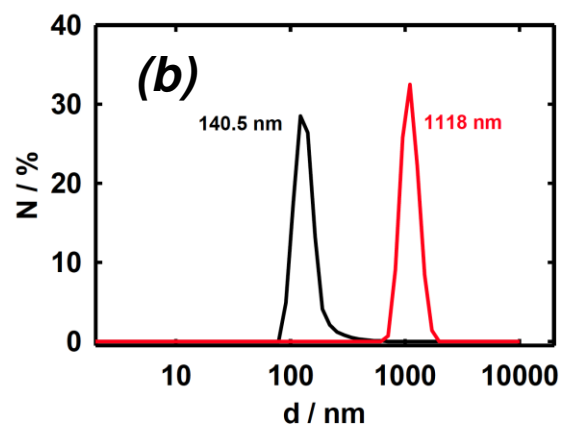
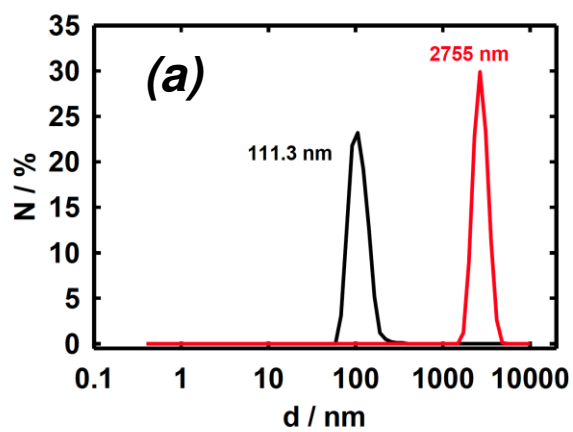
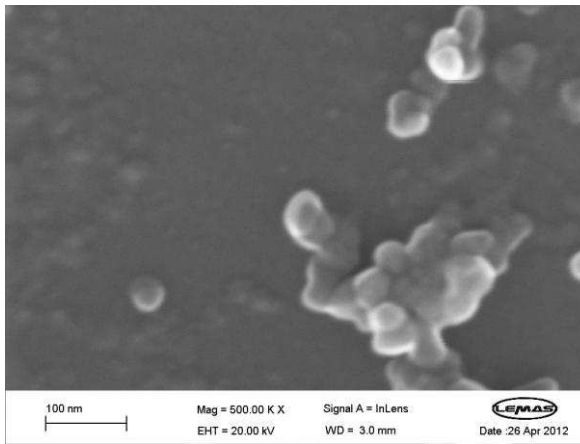
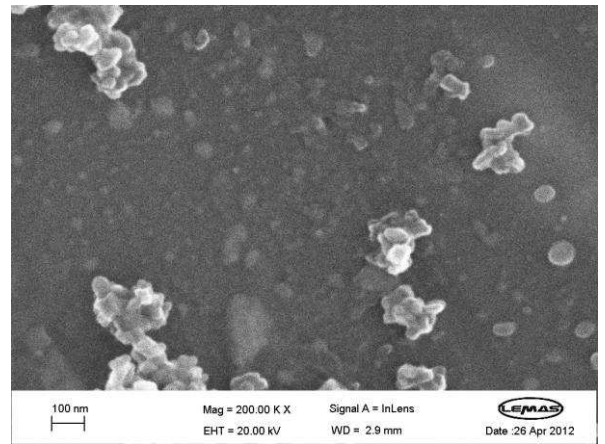


Figure 2

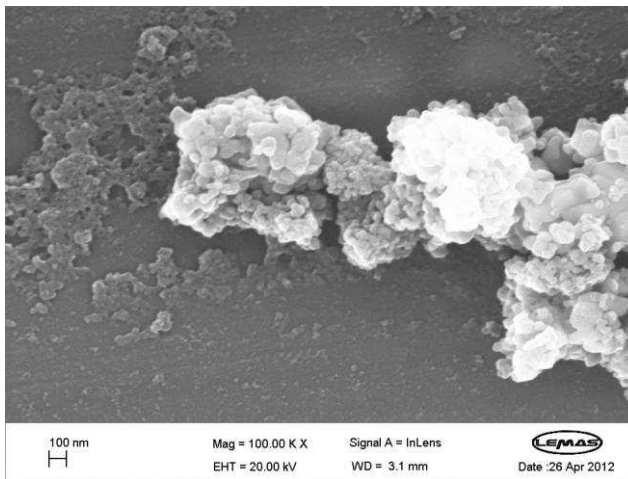
(a)



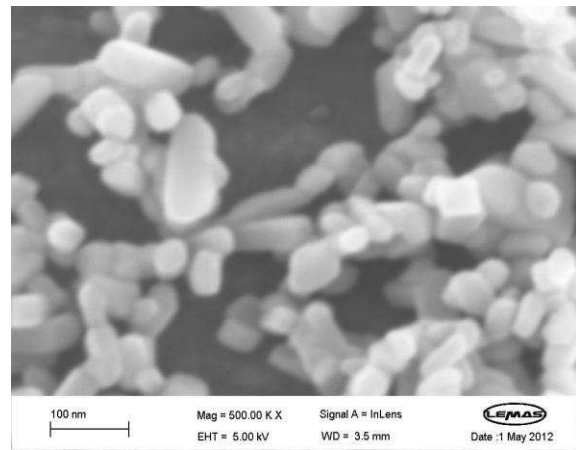
(b)



(c)



(d)



(e)

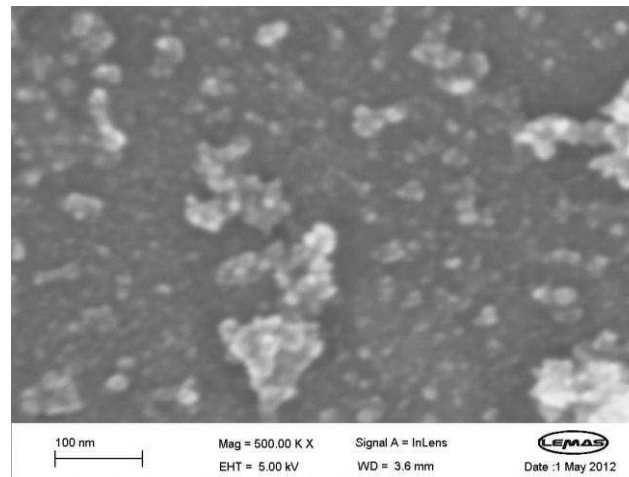


Figure 3

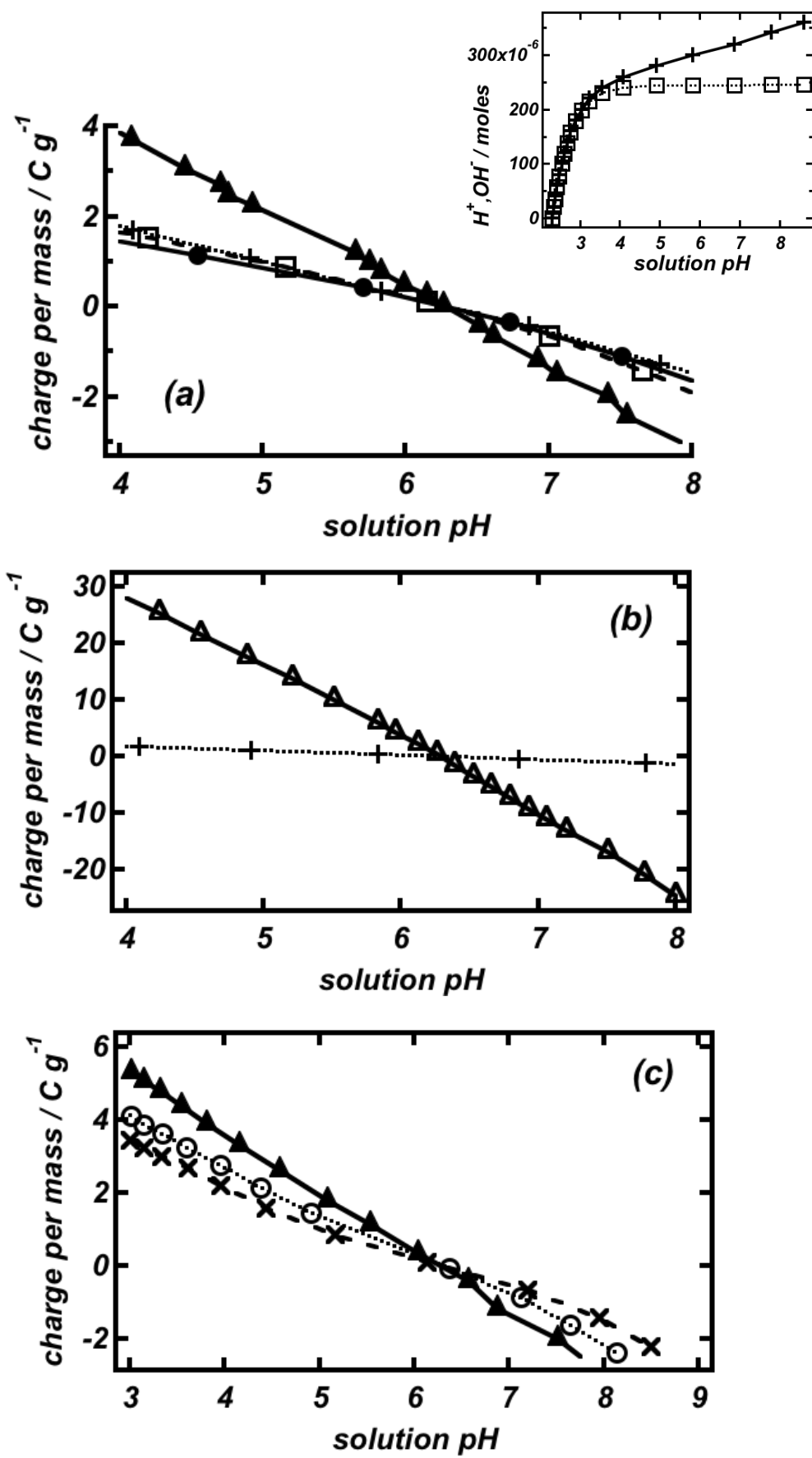


Figure 4

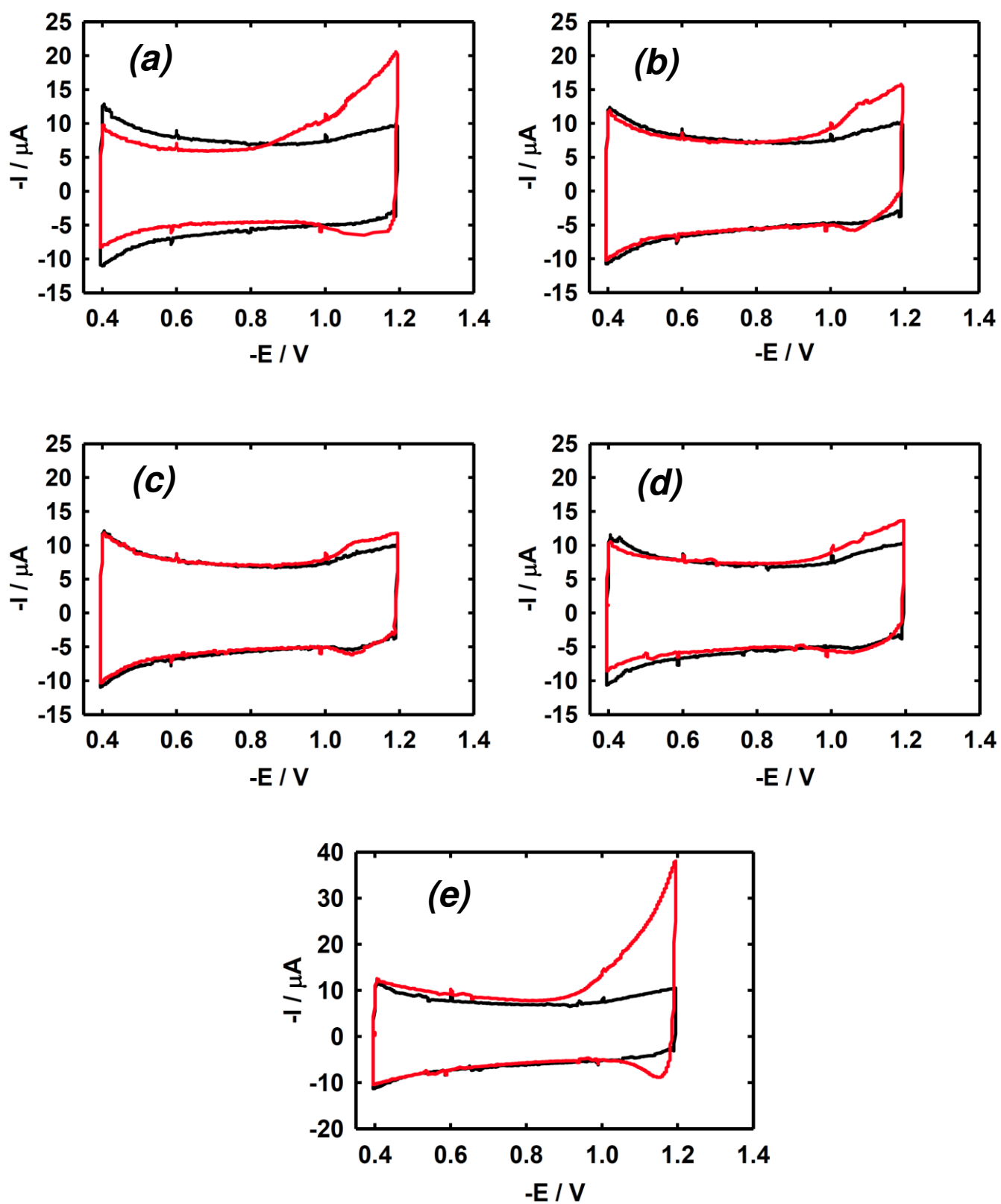


Figure 5

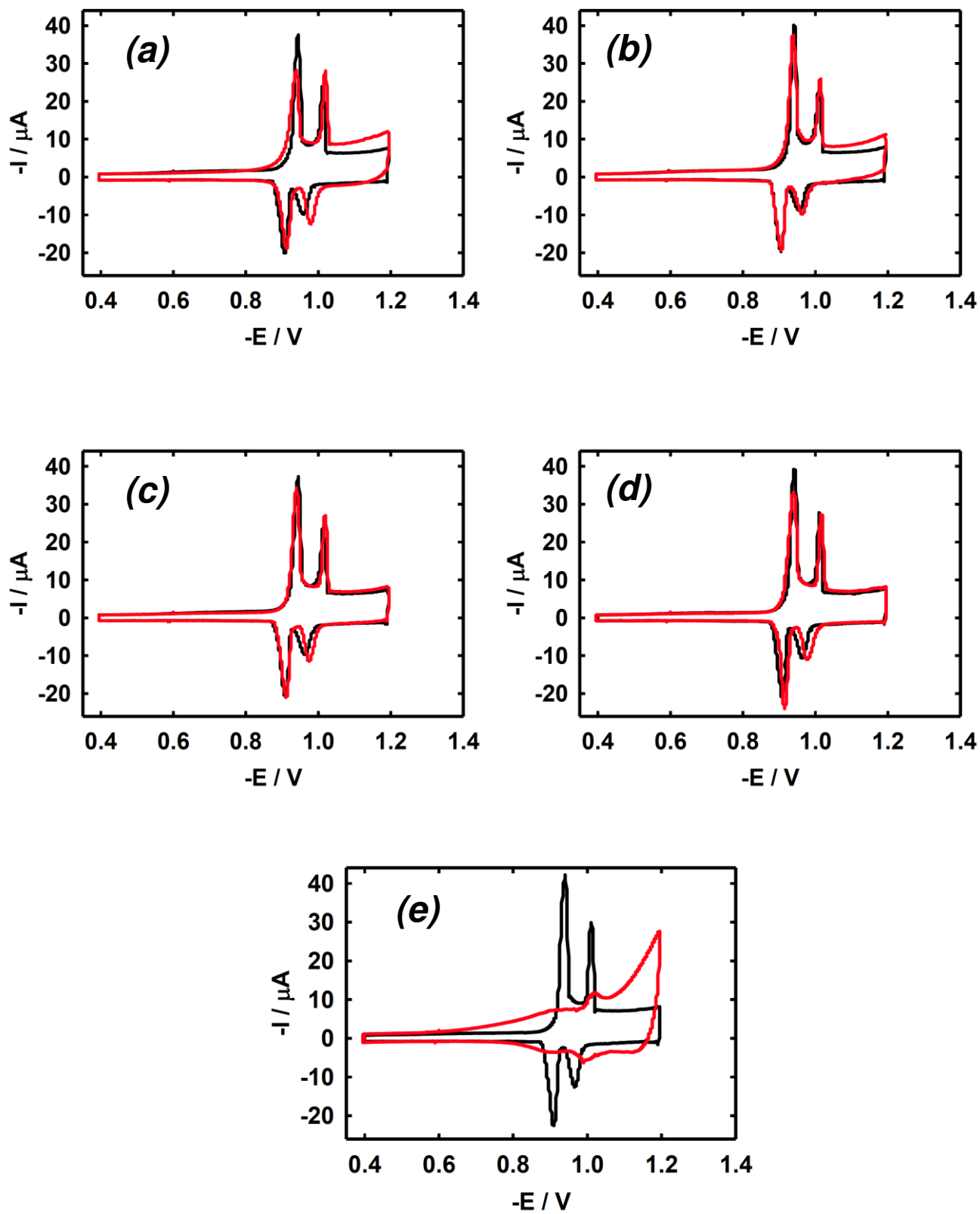


Figure 6

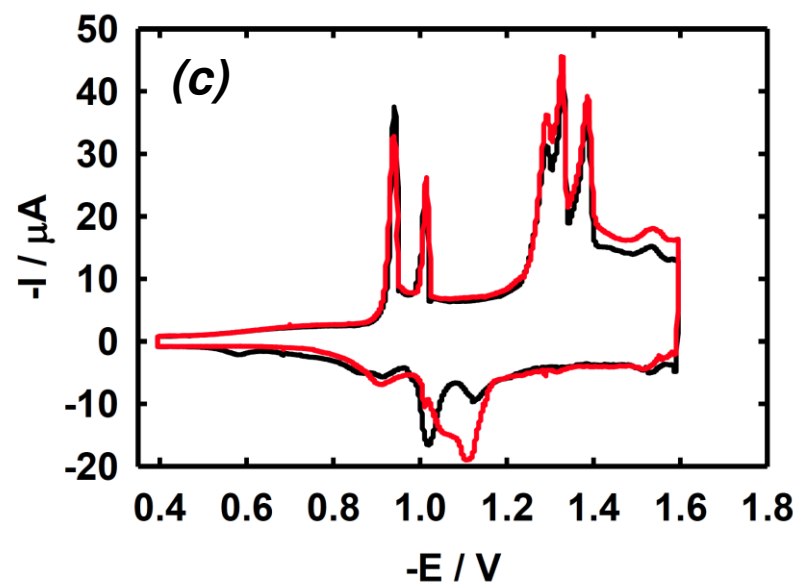
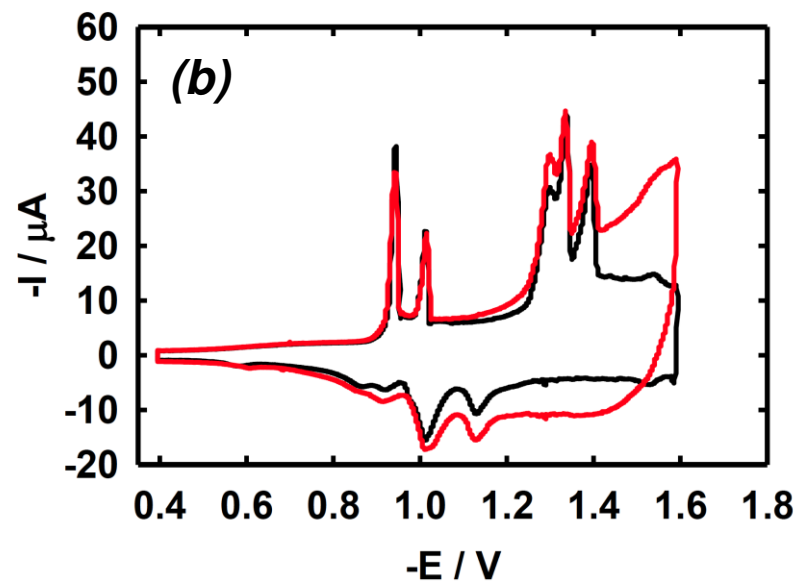
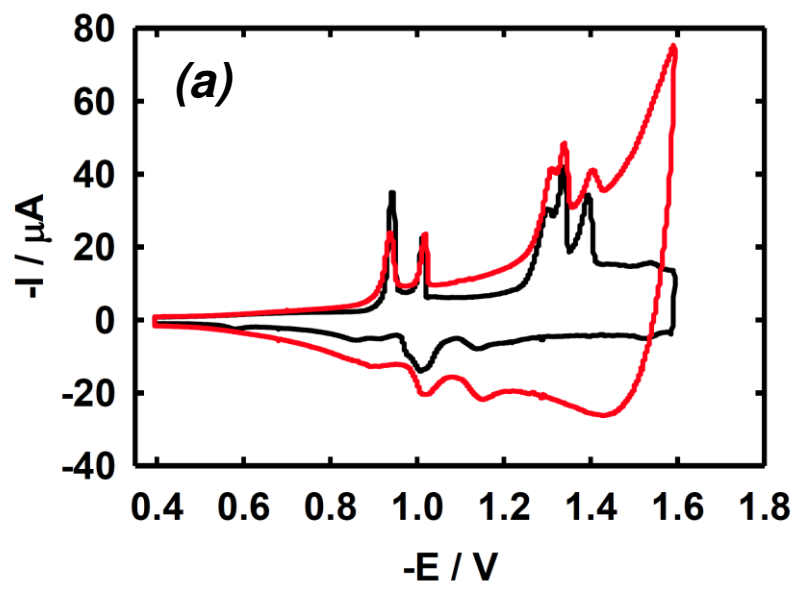


Figure 7

