

A consistent phase-field-regularised partition of unity method for fracture analysis

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Abstract

Recent advancements in phase-field models have significantly reshaped the landscape of fracture mechanics, which was dominated by the partition of unity method in the early 21st century. In this study, we aim to leverage the advantages of the two approaches by adopting a novel phase-field-regularised partition of unity method to improve computational efficiency, robustness and physical consistency. Specifically, we establish a connection between early phase-field models and the partition of unity method for cohesive fracture. To this end, we replace the standard discontinuous Heaviside enrichment in the partition of unity method with a regularised and continuous Heaviside function, leveraging the phase-field approximation of the Dirac- δ function. The proposed formulation effectively resolves ill-conditioning issues in the traditional partition of unity method while retaining the key advantages of discrete fracture representations, offering a distinct contrast to traditional phase-field approaches for smeared crack models. These advantages include eliminating the need for extremely fine meshes and providing an unambiguous and physically consistent representation of the displacement jump across a crack. Furthermore, by integrating Non-Uniform Rational B-Splines (NURBS) for spatial discretisation, the approach enhances solution accuracy compared to standard finite element formulations. Compatibility enforcement is also modified to accommodate the crack diffused by the phase-field approximation. Through numerical examples, including stationary and propagating cracks, mesh refinement studies, and sensitivity analyses of the

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phase-field length scale, we establish an optimal prescription for the internal length scale based solely on the element size. The examples compare the results obtained via the presented formulations with exact solutions and other numerical techniques, demonstrating the accuracy, conditioning stability, and computational efficiency of the methodology. The proposed methodology thus presents a robust alternative to conventional fracture models, combining key advantages offered by discrete and smeared approaches.

Keywords: Phase-field model, extended finite element analysis, isogeometric analysis, partition of unity method

1. Introduction

Fracture mechanics has always been an important topic of study in engineering, and has attracted various numerical approaches to simulate the fracture process effectively. At the early stages, the theorem of minimum energy was developed for a predefined crack in an elastic solid [1] inspired by the observation of unbounded (singular) stresses in the vicinity of a sharp notch [2], rendering the stresses useless for determination of crack propagation. Griffith's theory of brittle fracture was expanded to ductile materials [3], and later generalised for elastic-plastic solids [4]. A common feature of all the previously mentioned contributions is the relatively small fracture process zone compared to the size of the structure, which limits the focus to Linear Elastic Fracture Mechanics (LEFM). Additionally, they fail to model crack nucleation in an undamaged area away from the discontinuity and to predict the direction of the crack growth except through *a posteriori* studies [5], which required remeshing at the crack front after propagation. New techniques were developed to address these issues in numerical fracture modelling, following two main trends: smeared approaches, which account for deterioration of the stiffness of the material in the micro-structure; and discrete models, which explicitly represent fractures as distinct geometrical entities, generating a discontinuous displacement field.

The smeared approaches involve embedding strain localisation within an element [6, 7], for instance, through the presence of high deformation gradients due to fracture. The displacement discontinuity is represented through an additional, incompatible, Dirac- δ strain term [8], where the so-called embedded crack is diffused within the element. Different variants of the embedded discontinuity approach have been put forth [9], and comparative studies

have been conducted [10]. One of the main issues which adhere to the embedded approach is the ill-posedness in the presence of strain softening, which necessitates using extra information, such as incorporating higher-order derivatives of displacement in the kinematics [11, 12] or adding non-local terms [13], which are typically referred to as regularisation models. However, the proposed remedies often do not yield a symmetric stiffness matrix, and retrieving the crack opening displacement is non-trivial [14, 15].

In the context of discrete fracture models, a displacement discontinuity can be incorporated into continuum mechanics through a rigorous definition of the fracture by extra degrees of freedom (DOFs). This can be obtained either through interface elements located in between standard continuum elements [16, 17], or via Partition of Unity Method (PUM) [18], which allows for local enrichment of nodal points of the cracked element with extra DOFs, known as extended finite element analysis [19]. In this manuscript, we distinguish eXtended Finite Element Method (XFEM), where Lagrange polynomials are utilised as shape functions, from the broader term extended finite element analysis, referring to a larger family of enriched approaches including eXtended IsoGeometric Analysis (XIGA) [20]. The enriched approach decouples the crack path and the mesh layout, allowing both inter- and intra-element fractures which enable arbitrary crack propagation. This flexibility is absent in interface elements as the crack extension path should be known *a priori*. Moreover, since interface elements are inserted in advance, a dummy stiffness must be adopted to avoid premature crack growth, which can lead to spurious stress oscillations if the dummy stiffness is too high [17]. The cohesive zone model, an extension of LEFM in the presence of non-negligible fracture process zone [21, 22], is widely used in discrete models, including interface elements [23] and XFEM [24].

More recently, efforts have been primarily devoted to phase-field models to simulate the fracture process. These models trace back to the minimisation of the total energy based on Griffith’s theory of brittle fracture [25] and its regularised form [26]. Afterwards, a majority of the phase-field literature has been devoted to brittle fracture [27, 28, 29, 30, 31, 32, 33] and, by way of analogy to gradient-enhanced damage mechanics [34, 35], expanded to ductile fracture [36, 37, 38]. Regarding the cohesive-zone phase-field models, two alternative approaches can be identified: (i) the traction-separation relationship is applied to a crack diffused by the phase-field approximation of the Dirac- δ , without any explicit degradation function used for the deterioration of the bulk material [32, 39, 40, 41, 42]; (ii) a tailored degradation function,

characterised and tuned through a uniform tension test, is applied to the bulk material to replicate the mechanical response of a certain traction-separation relationship [43, 44], analogous to gradient damage models [45]. In this work, we aim at enhancing the former through Partition of Unity Method (PUM), while involving a degradation function will be explored in a future work. It will be shown in the remainder that the former is reminiscent of regularised discrete fracture models, for instance the regularised XFEM [46].

Extended finite element analysis requires crack tracking strategies, such as level-set [47, 48] and fast marching [49, 50] methods, to define the tangential and normal vectors on the crack profile needed for the Heaviside function. However, crack branching or coalescence pose significant challenges, such as non-unique normal vectors at crack intersections, which becomes even more difficult when accounting for three-dimensional problems [51]. In contrast, phase-field models eliminate the need for crack tracking through adding a separate differential equation for the crack profile, allowing for simulating crack propagation, coalescence and branching, particularly in three-dimensional cases. However, this comes at the expense of utilising a very fine mesh [40]. To leverage the advantages of XFEM and phase-field models, several hybrid approaches have been developed. For instance, in the context of LEFM, a multi-resolution global-local enrichment is utilised for extended/generalised finite element method. The phase-field model is only used for crack advancement at the fine scale in the vicinity of the crack tip, while XFEM handles the discontinuity elsewhere [52]. Another approach involves using XFEM for the large scale displacement field, while a local phase-field solution determines the crack propagation at the crack tip(s) for LEFM [53]. Taking advantage of the known exponential form of a typical phase-field solution, exponential shape functions were proposed to reduce the number of elements perpendicular to the crack path, while a fine mesh is still needed in the parallel direction [54]. The ansatz presented in [54] for phase-field was recently transformed to resolve compatibility issues and combined with XFEM for brittle fracture [55]. Additionally, Thick Level-Set (TLS) method, a level-set-based damage model [56], was successfully compared with phase-field models for brittle and quasi-brittle fracture [57].

A well-known issue in extended finite element analysis is the ill-conditioning of the stiffness matrix in certain scenarios, which can lead to the loss of accuracy or, in case of linear dependency, results in a singular stiffness matrix [58]. This typically occurs when the crack (i) aligns very close to the element edge or (ii) divides the element into highly disproportionate sections,

103 making it impossible to place sufficient integration points on one side of the
 104 crack to properly account for the Heaviside enrichment [58, 59]. The mesh-
 105 independent nature of extended finite element analysis generally makes it
 106 difficult to avoid these scenarios, particularly in arbitrary propagating crack
 107 problem. Several approaches have been proposed to mitigate the conditioning
 108 issue, such as perturbation of the stiffness matrix [60, 61] or preconditioning
 109 based on domain decomposition [62]. Retrieving the convergence rate of fi-
 110 nite element analysis is the objective of a few other contributions [58, 63, 64].
 111 All of these attempts, however, fail to directly resolve the robustness issue as-
 112 sociated with the Heaviside enrichment in the ill-conditioning scenarios [65].
 113 Instead, *a posteriori* re-arrangement of the mesh layout is the typical indi-
 114 rect solution to both scenarios. Node snapping is an example, where nodes
 115 are moved away from the crack path to resolve the conditioning issue.[59]
 116 However, this approach nullifies the computational advantages given by de-
 117 coupling the crack path from the mesh layout in XFEM. Another remedy
 118 involves deviating the crack path [66, 67], even though extra care must be
 119 taken to limit the deviation to avoid other conditioning issues [59]. An alter-
 120 native to avoid the linear dependency is orthonormalisation of shape func-
 121 tions, as applied in XFEM [68] and in XIGA [69], however this approach does
 122 not solve ill conditioning issues associated with cracks that split an element
 123 disproportionately.

124 On the other hand, IsoGeometric Analysis (IGA) tends to outperform
 125 the standard Finite Element Analysis (FEA) in accurate representation of
 126 the geometry [70], versatility in providing arbitrary degree of (higher-order)
 127 continuity at element boundaries [71], and a per-degree-of-freedom superi-
 128 ority in accuracy and robustness [72, 73]. Non-Uniform Rational B-splines
 129 (NURBS) are adopted in the present work. Regarding fracture, IGA was first
 130 cast in an extended finite element analysis format, coined XIGA, for LEFM
 131 [20], and later expanded to cohesive fracture [74]. IGA has been incorporated
 132 in cohesive interface elements approach for fracture simulation as well [75].
 133 Recently, blending XIGA with a meshfree method has successfully mitigated
 134 the conditioning issue of the stiffness matrix associated with the Heaviside
 135 function [76], as no fixed notion of spatial discretisation exist. Instead, the
 136 overlapping support clouds of points provide sufficient space for the Heaviside
 137 function integration in extreme scenarios of disproportionate fracture.

138 Herein, we aim at regularising XIGA using phase-field approximation of
 139 the Dirac- δ , enabled through the use of continuous exponential decay function
 140 to regularise the Heaviside and the signed-distance level set functions by

141 taking into account the relationship between these quantities. The concept
 142 of regularised Heaviside function has already been implemented for XFEM
 143 [46, 77, 78], but these formulations fail to enforce compatibility constraints
 144 and provide suitable guidelines for defining an appropriate regularization
 145 length scale, all of which are further complicated in the context of IGA. For
 146 instance, the inter-element sharing of control points complicates enrichment,
 147 rendering the use of the standard enrichment prescription for the regularised
 148 formulation questionable. In addition, it is imperative to investigate whether
 149 the regularised formulation mitigates the conditioning issue of the stiffness
 150 matrix associated with the Heaviside enrichment.

151 This work addresses the above-mentioned challenges by developing a for-
 152 mulation that regularises XIGA using a phase-field approximation of the
 153 Dirac- δ , enforces compatibility for diffused crack descriptions, accommodates
 154 nonlinear traction-separation relationships, and defines a spatially varying
 155 length scale based on the characteristic element size. The article is struc-
 156 tured as follows. Section 2 presents the ingredients at the bulk and the dis-
 157 continuity for a phase-field regularised PUM for fracture analysis, followed
 158 by the variational forms of the displacement and the phase-field in Section 3.
 159 Next, the discretisation of the weak forms are discussed in Section 4, along
 160 with a succinct introduction of NURBS formulated on Bézier-extraction to
 161 comply with finite element data-structure [79]. Section 5 discusses implemen-
 162 tation aspects, including compatibility enforcement of the regularised formu-
 163 lation, the enrichment scheme and the integration strategy utilised. Finally,
 164 some case studies, comprised of mesh refinement and singularity analysis for
 165 traction-free cracks and cohesive fractures, are examined to demonstrate the
 166 efficacy of the proposed formulation. Stationary and progressive fracturing
 167 are explored in Section 6, while rate-independent materials are assumed and
 168 isotropic elasticity is utilised for the bulk and the regularised layer.

169 **2. Phase-field regularised partition of unity method**

170 The partition of Unity Method (PUM) has been proven effective in ap-
 171 proximating non-uniform fields with local high gradients. This is, for in-
 172 stance, the case when dealing with fractures, as a discontinuity Γ_d is shown
 173 within a continuous body Ω in Figure 1. PUM [60] approximates the ad-
 174 ditional field by means of an enrichment function γ and the set of shape
 175 functions φ ,

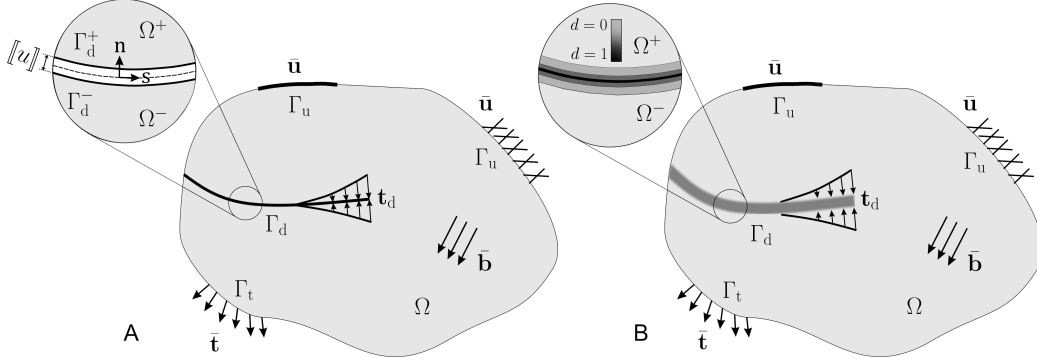


Figure 1: Boundary value problem Ω with the discontinuity Γ_d and cohesive tractions $\mathbf{t}_d$: A) a sharp crack B) a diffused description of the discontinuity.

$$a_{\text{PUM}}^h = \sum_i \varphi_i(x) \left(\sum_j a_{ij} \gamma_j(x) \right) \quad \text{where} \quad \sum_i \varphi_i(\mathbf{x}) = 1 \quad \forall \mathbf{x} \in \Omega \quad (1)$$

176 where $\bullet^h$ denote the approximated $\bullet$ field. Enriching the Finite Element
 177 Analysis (FEA) with PUM leads to Generalised Finite Element Method
 178 (GFEM) [60, 80] in terms of standard and (extrinsically[81]) enriched discrete
 179 nodal values, $\hat{\bullet}$ and $\tilde{\bullet}$, respectively

$$a^h = \sum_i \hat{\varphi}_i(x) \hat{a}_i + \sum_i \tilde{\varphi}_i(x) \left(\sum_j \tilde{a}_{ij} \mathcal{H}_j(x) \right). \quad (2)$$

180 Using the Heaviside sign function, GFEM is capable of representing a frac-
 181 tured body similar to Figure 1 through

$$\mathbf{u}^h = \hat{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \tilde{\mathbf{u}}^h \quad \text{with} \quad \mathcal{H}_{\Gamma_d}(x_n) = \{1 \mid \forall x_n > 0; -1 \mid \forall x_n < 0; 0 \mid x_n = 0\}, \quad (3)$$

182 where $\mathbf{u}$, $\hat{\mathbf{u}}$ and $\tilde{\mathbf{u}}$ indicate the total, continuous and discontinuous displace-
 183 ment fields, respectively. Dealing with cracks and voids, GFEM is referred
 184 to as extended finite element analysis, and has been widely used for different
 185 applications using Lagrange polynomials [81, 82, 83], or by means of spline
 186 technologies for isogeometric analysis [73, 76, 84].

187 In the absence of the acceleration, the quasi-static equilibrium equations
 188 read

$$\begin{cases} -\nabla \cdot \boldsymbol{\sigma} = \bar{\mathbf{b}} & \mathbf{x} \in \Omega \\ \mathbf{u}^h = \bar{\mathbf{u}} & \mathbf{x} \in \Gamma_u \\ \boldsymbol{\sigma} \cdot \mathbf{n}_t = \bar{\mathbf{t}} & \mathbf{x} \in \Gamma_t \\ \boldsymbol{\sigma} \cdot \mathbf{n}_d = \mathbf{t}_d & \mathbf{x} \in \Gamma_d \end{cases} \quad (4)$$

where $\boldsymbol{\sigma}$ is the Cauchy stress tensor, $\mathbf{n}_d$ and $\mathbf{n}_t$ are the vectors normal to the fracture surface Γ_d and the external traction surface Γ_t , respectively. The prescribed values for displacements, tractions and body forces are referred to as $\bar{\mathbf{u}}$, $\bar{\mathbf{t}}$ and $\bar{\mathbf{b}}$, respectively. We apply the weighted residual method to the strong form in Equation (4) and utilise the divergence theorem to define a general weak form of the potential energy as

$$\Psi_{\text{pot}} = \overbrace{\int_{\Omega} \psi(\boldsymbol{\varepsilon}(\mathbf{u}^h)) \, d\Omega}^{\mathcal{E}_{\text{int}}} - \overbrace{\left(\int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathbf{u}^h \, d\Gamma + \int_{\Omega} \bar{\mathbf{b}} \cdot \mathbf{u}^h \, d\Omega \right)}^{\mathcal{P}_{\text{ext}}} \quad (5)$$

where $\mathcal{E}_{\text{int}}$ and $\mathcal{P}_{\text{ext}}$ indicate internal and external energy functions. The strain energy function ψ reads,

$$\psi = \frac{1}{2} \boldsymbol{\sigma} : \boldsymbol{\varepsilon}. \quad (6)$$

Unlike explicit description of fracture by Griffith's potential energy function [39, 25, 26], which allocates a distinct energy term to the fracture, we aim to construct the fracture energy term from the displacement discontinuity, which is characterised by the Heaviside function $\mathcal{H}_{\Gamma_d} = \mathbf{n}_d \cdot \nabla \mathcal{D}_{\Gamma_d}$. Here, $\mathcal{D}_{\Gamma_d}$ denotes the sign distance function, as illustrated in Figure 2.

Adopting the small displacement assumption, the infinitesimal strain field becomes

$$\boldsymbol{\varepsilon} = \nabla^s \mathbf{u}^h = \underbrace{\nabla^s \hat{\mathbf{u}}^h}_{\text{continuous}} + \underbrace{\mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h + 2\delta_{\Gamma_d} (\tilde{\mathbf{u}}^h \otimes^s \mathbf{n}_d)}_{\text{discontinuous}} \quad (7)$$

204

where $\nabla^s \square = (\nabla \square + \nabla \square^T)/2$, $\otimes^s$ indicates the symmetric tensor product, and δ_{Γ_d} is the Dirac-delta at Γ_d , *i.e.*, $\delta_{\Gamma_d} = \delta_{\text{Dirac}}(\mathbf{x} - \mathbf{x}_c)$ with $\mathbf{x}_c = \text{argmin}(\|\mathbf{x}^* - \mathbf{x}\|) \, \forall \mathbf{x}^* \in \Gamma_d$ [39], which follows the identity $\mathbf{n}_d \cdot \nabla \mathcal{H}_{\Gamma_d} = 2\delta_{\Gamma_d}$.

Equation 7 can be expressed for strain components at the bulk and the discontinuity, *i.e.*, $\boldsymbol{\varepsilon} = \boldsymbol{\varepsilon}^b + \boldsymbol{\varepsilon}^{\Gamma_d}$

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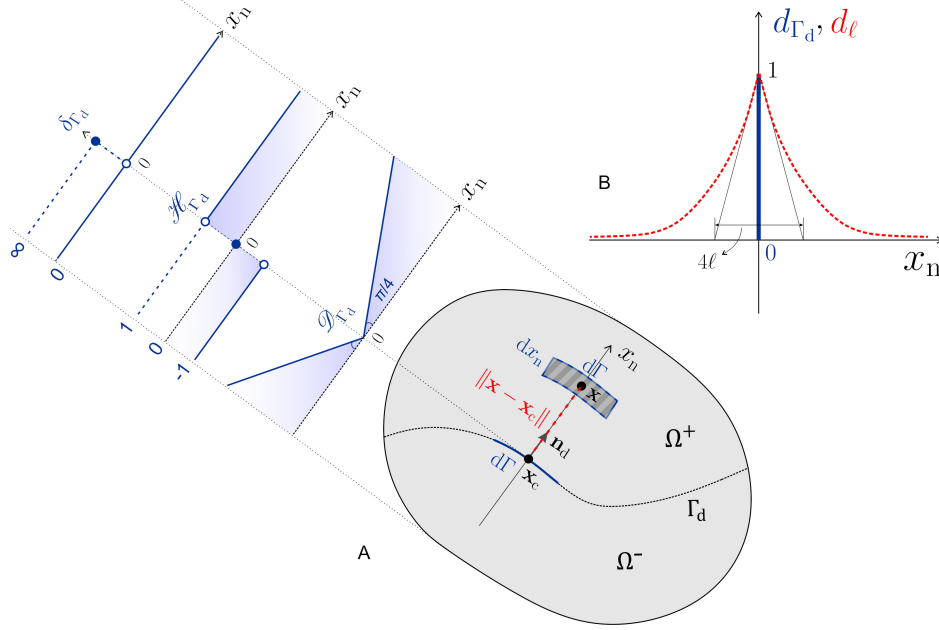


Figure 2: Consistent definition of the discontinuity: A) signed distance function $\mathcal{D}_{\Gamma_d}$, Heaviside sign function $\mathcal{H}_{\Gamma_d}$ and Dirac-delta function δ_{Γ_d} , B) Sharp and diffused definition of the phase-field variable, d_{Γ_d} and d_ℓ respectively.

$$\boldsymbol{\varepsilon}^b = \nabla^s \tilde{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h \quad \text{and} \quad \boldsymbol{\varepsilon}^{\Gamma_d} = 2\delta_{\Gamma_d} (\tilde{\mathbf{u}}^h \otimes^s \mathbf{n}_d). \quad (8)$$

Inserting Equation (8) into (5) and utilising the identity $(\square \otimes^s \bullet) : \bigcirc = \bigcirc \cdot \bullet \cdot \square$ leads to the internal potential energy of the form

$$\mathcal{E}_{\text{int}} = \int_{\Omega} \psi^b(\boldsymbol{\varepsilon}^b(\mathbf{u}^h)) \, d\Omega + \int_{\Omega} \delta_{\Gamma_d} \mathbf{t}_d \cdot \tilde{\mathbf{u}}^h \, d\Omega \quad (9)$$

where $\mathbf{t}_d = \boldsymbol{\sigma} \cdot \mathbf{n}_d$ indicates the traction vector at the discontinuity and

$$\psi^b = \frac{1}{2} \boldsymbol{\sigma} : \boldsymbol{\varepsilon}^b \quad \text{with} \quad \boldsymbol{\sigma} = \mathbb{C} \boldsymbol{\varepsilon}^b \quad (10)$$

following the Hooke's law for an isotropic linear elastic material where $\mathbb{C}$ indicates the stress-strain relationship.

The key difference between the discrete and regularised approaches emerges in Equation (9), depending on how the last term is interpreted,

$$\overbrace{\int_{\Gamma} \phi(\mathbf{x}) \, d\Gamma}^{\text{discrete}} \longleftarrow \int_{\Omega} \delta_{\Gamma_d}(\mathbf{x}) \phi(\mathbf{x}) \, d\Omega \longrightarrow \overbrace{\int_{\Omega} \delta_{\ell} \phi(\mathbf{x}) \, d\Omega}^{\text{smeared}} \quad (11)$$

217 where $\phi(\mathbf{x})$ is an arbitrary continuous function, and δ_{ℓ} represents the diffused
 218 estimate of the Dirac- δ function with the internal length scale parameter
 219 $\ell \in \mathbb{R}^+$. Fracture energy, the energy dissipated upon creation of a unit
 220 fracture surface, reads

$$\mathcal{G}_f = \mathbf{t}_d \cdot \tilde{\mathbf{u}}^h = \frac{1}{2} \mathbf{t}_d \cdot \llbracket \mathbf{u} \rrbracket \quad (12)$$

221 where

$$\llbracket \mathbf{u} \rrbracket = (\mathbf{u}^h)^+ - (\mathbf{u}^h)^- = (\mathcal{H}_{\Gamma_d}^+ - \mathcal{H}_{\Gamma_d}^-) \tilde{\mathbf{u}}^h = 2\tilde{\mathbf{u}}^h \quad (13)$$

222

223 denotes the displacement jump, derived by taking advantage of the explicit
 224 definition of the discontinuity in the discrete approach through the Heaviside
 225 sign function, see Equation (3).

226 Utilising these findings, Equation (5) becomes

$$\Psi_{\text{pot}} = \int_{\Omega} \psi^b(\boldsymbol{\epsilon}^b(\mathbf{u}^h)) \, d\Omega + \int_{\Omega} \delta_{\Gamma_d} \mathcal{G}_f \, d\Omega - \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathbf{u}^h \, d\Gamma - \int_{\Omega} \bar{\mathbf{b}} \cdot \mathbf{u}^h \, d\Omega \quad (14)$$

227 which is identical to the Griffith potential energy used in other cohesive
 228 phase-field models [25, 32, 39, 41, 43, 52].

229 2.1. Dirac-delta approximation

230 Now, we provide a smeared approximate of δ_{Γ_d} by means of an exponential
 231 decay [29, 30, 39]

$$\delta_{\Gamma_d}(x_n) \approx \delta_{\ell}(x_n) = \frac{1}{2\ell} \exp\left(-\frac{|x_n|}{\ell}\right), \quad (15)$$

232 where $x_n = (\mathbf{x} - \mathbf{x}_c) \cdot \mathbf{n}_d(\mathbf{x}_c)$, and $\mathbf{n}_d(\mathbf{x}_c)$ is the unit vector normal to the
 233 interface at $\mathbf{x}_c$, which is in-line with the signed distance function $\mathcal{D}_{\Gamma_d} =$
 234 $|(\mathbf{x} - \mathbf{x}_c) \cdot \mathbf{n}_d| = \|\mathbf{x} - \mathbf{x}_c\| \text{sign}((\mathbf{x} - \mathbf{x}_c) \cdot \mathbf{n}_d)$, see Figure 2, with the Euclidean
 235 norm shown by $\|\square\|$.

236 In order to avoid the ambiguities associated with generalisation of Equa-
 237 tion (15) to multi dimensions [35, 39], the phase-field variable, $d \in [0, 1]$,
 238 was proposed which describes the state of the material: $d = 0$ for intact and
 239 $d = 1$ for fully broken materials. The phase-field variable $d = \exp\left(-\frac{|x_n|}{2\ell}\right)$ is
 240 the solution of the Euler-Lagrange equation subjected to the constraints

$$\begin{cases} d - 4\ell^2 \nabla^2 d = 0 & \mathbf{x} \in \Omega \setminus \Gamma_d \\ d = 1 & \mathbf{x} \in \Gamma_d \\ \nabla d \cdot \mathbf{n}_{\partial\Omega} = 0 & \mathbf{x} \in \partial\Omega = \{\Gamma_t, \Gamma_u\} \end{cases}. \quad (16)$$

241 that is associated with

$$d^h = \arg \left\{ \inf_{d^h \in \mathcal{S}_d} \int_{\Omega} \delta_{\ell}(d^h) d\Omega \right\}, \quad (17)$$

242 where d^h denotes the approximated phase-field variable, the phase-field space
 243 $\mathcal{S}_d = \{d^h | d^h(\mathbf{x}) = 1 \ \forall \mathbf{x} \in \Gamma_d\}$ and

$$\delta_{\ell} = \frac{1}{4\ell} \left(d^h \cdot d^h + 4\ell^2 \frac{\partial d^h}{\partial x_n} \cdot \frac{\partial d^h}{\partial x_n} \right) = \frac{1}{4\ell} (d^h \cdot d^h + 4\ell^2 \nabla d^h \cdot \nabla d^h) \quad (18)$$

244 with $\partial d^h / \partial x_n = \mathbf{n}_d \cdot \nabla d^h$ and $\mathbf{n}_d^T \cdot \mathbf{n}_d = 1$.

245 In a similar manner we can derive the Heaviside function approximation,
 246 $\mathcal{H}_{\ell}$. Recalling the identity $\mathbf{n}_d \cdot \nabla \mathcal{H}_{\Gamma_d} = 2\delta_{\Gamma_d}$ and Equation (15) we can write

$$\begin{aligned} \mathcal{H}_{\ell}(x_n) &= 2 \int_0^{x_n} \frac{1}{2\ell} \exp\left(-\frac{|\varsigma|}{\ell}\right) d\varsigma = \text{sign}(x_n) \left(1 - \exp\left(-\frac{|x_n|}{\ell}\right)\right) \\ &= \mathcal{H}_{\Gamma_d}(x_n) (1 - 2\ell\delta_{\ell}) \end{aligned} \quad (19)$$

247 which is identical to the findings by Benvenuti *et al.* [46]. Finally, given
 248 $\mathbf{n}_d \cdot \nabla \mathcal{D}_{\Gamma_d} = \mathcal{H}_{\Gamma_d}$,

$$\begin{aligned} \mathcal{D}_{\ell}(x_n) &= \int_0^{x_n} \text{sign}(\varsigma) \left(1 - \exp\left(-\frac{|\varsigma|}{\ell}\right)\right) d\varsigma \\ &= |x_n| - \ell \left(1 - \exp\left(-\frac{|x_n|}{\ell}\right)\right) = \mathcal{D}_{\Gamma_d}(x_n) - \ell(1 - 2\ell\delta_{\ell}) \end{aligned} \quad (20)$$

249 The effect of the length-scale parameter ℓ on the approximated Dirac-delta,
 250 Heaviside and signed-distance functions has been presented in Figure 3.
 251 As ℓ decreases the solution converges to analytical discrete functions, *i.e.*,
 252 $\lim_{\ell \rightarrow 0} \mathcal{H}_{\ell} = \mathcal{H}_{\Gamma_d}$ and $\lim_{\ell \rightarrow 0} \mathcal{D}_{\ell} = \mathcal{D}_{\Gamma_d}$.

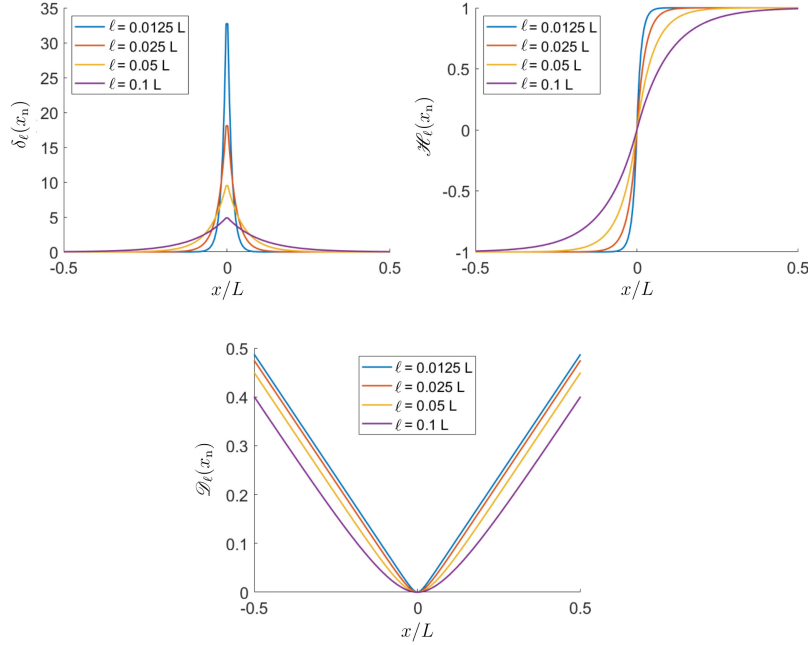


Figure 3: Effect of the phase-field internal length-scale parameter ℓ for the diffused values of Dirac-delta (δ_ℓ), the Heaviside ($\mathcal{H}_\ell$) and the signed-distance ($\mathcal{D}_\ell$) functions. The crack is located at $x = 0$ of a one-dimensional problem with length L

2.2. Constitutive relation at the discontinuity

The potential energy in Equation (14) is written for a non-negligible fracture process zone, which lodges the nonlinear deformations and their gradients. We adopt the cohesive-zone model in the form of tractions at the discontinuity, *i.e.*, $\mathbf{t}_d(\mathbf{x}_c)$ at Γ_d , by means of a the dissipation energy defined

$$\mathcal{G}_f = \mathcal{G}_f(\llbracket \mathbf{u} \rrbracket(\mathbf{x}_c), \boldsymbol{\kappa}), \quad \forall \mathbf{x}_c \in \Gamma_d \quad (21)$$

in terms of the displacement jump, defined in Equation (13), and the history parameter $\boldsymbol{\kappa}$, which stores the maximum experienced displacement jump. In the literature, many options have been proposed for different mechanical behaviours [85], a few of which are explored in our examples. Hereafter, we refer to $\llbracket \mathbf{u} \rrbracket(\mathbf{x}_c)$ simply by $\llbracket \mathbf{u} \rrbracket$, as the notion of the displacement jump is only valid for Γ_d and, therefore, meaningless elsewhere.

Recalling Equation (12) the tractions at Γ_d yield

$$\mathbf{t}_d(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa}) = \frac{\partial \mathcal{G}_f(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa})}{\partial \llbracket \mathbf{u} \rrbracket} \quad (22)$$

265

266 in the global coordinate system while the local values of the displacement
267 jump and the traction read

$$\mathbf{t}_d(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa}) = \mathbf{R}^T \cdot \mathbf{t}_d^{\text{loc}}(\llbracket \mathbf{u}^{\text{loc}} \rrbracket, \boldsymbol{\kappa}^{\text{loc}}), \quad \llbracket \mathbf{u}^{\text{loc}} \rrbracket = \{\llbracket u_s^{\text{loc}} \rrbracket, \llbracket u_n^{\text{loc}} \rrbracket\} = \mathbf{R} \cdot \llbracket \mathbf{u} \rrbracket \quad (23)$$

268 where $\square_s$ and $\square_n$ are the values of $\square$ in the directions tangent and normal
269 to the crack path, respectively. $\mathbf{R}$ denotes the global-local rotation matrix.
270 The Kuhn-Tucker condition is adopted to enforce the irreversibility of the
271 displacement jump,

$$\mathbf{f}(\llbracket \mathbf{u}^{\text{loc}} \rrbracket, \boldsymbol{\kappa}^{\text{loc}}) = \llbracket \mathbf{u}^{\text{loc}} \rrbracket - \boldsymbol{\kappa}^{\text{loc}} \leq 0 \quad \dot{\boldsymbol{\kappa}}^{\text{loc}} \geq 0 \quad \dot{\boldsymbol{\kappa}}^{\text{loc}} \cdot \mathbf{f} = 0 \quad (24)$$

272 where the function $\mathbf{f}$ determines the loading/unloading state based on the
273 maximum local opening experienced until the current time, *i.e.*, $\boldsymbol{\kappa}^{\text{loc}} =$
274 $\max_{\tau \in (-\infty, t]} \llbracket \mathbf{u}^{\text{loc}}(\tau) \rrbracket$.

275 3. Variational formulation

276 In line with the cohesive phase-field formulation [39, 42] which adopts the
277 elastic strain, we exploit the second principle of thermodynamics to formulate
278 the diffused version ($\Gamma_d \leftarrow \ell$) of the internal potential energy in Equation
279 (9),

$$0 \leq \dot{\mathfrak{D}} = \boldsymbol{\sigma} : \dot{\boldsymbol{\varepsilon}} - \dot{\psi}^e = \boldsymbol{\sigma} : (\dot{\boldsymbol{\varepsilon}}^e + \dot{\boldsymbol{\varepsilon}}^d) - \frac{\partial \psi^e}{\partial \boldsymbol{\varepsilon}^e} : \dot{\boldsymbol{\varepsilon}}^e = \boldsymbol{\sigma} : \dot{\boldsymbol{\varepsilon}}^d \quad (25)$$

280 with the total strain $\boldsymbol{\varepsilon}$ decomposed into elastic and diffused parts, *i.e.* $\boldsymbol{\varepsilon} =$
281 $\boldsymbol{\varepsilon}^e + \boldsymbol{\varepsilon}^d$. On the other hand, the explicit derivation of the energy dissipation
282 reads

$$\begin{aligned} \dot{\mathfrak{D}} &= \frac{d\dot{\boldsymbol{\varepsilon}}^d}{dt} = \frac{d}{dt} (\mathcal{H}_\ell \nabla^s \tilde{\mathbf{u}}^h + \delta_\ell \mathbf{t}_d \cdot \tilde{\mathbf{u}}^h) \\ &= \mathcal{H}_\ell \nabla^s \dot{\tilde{\mathbf{u}}}^h + \nabla^s \tilde{\mathbf{u}}^h \frac{\partial \mathcal{H}_\ell}{\partial d} \dot{d} + 2\delta_\ell \mathbf{t}_d \cdot \dot{\tilde{\mathbf{u}}}^h + \mathcal{G}_f \frac{\partial \delta_\ell}{\partial d} \dot{d} \end{aligned} \quad (26)$$

where the first and the third terms represent the energy dissipated by further opening of the existing crack, *i.e.* $\dot{\tilde{\mathbf{u}}}^h$. The second and the fourth terms indicate the energy dissipated through the extension of the cohesive zone by an increment d . In the cohesive-zone modelling, the newly progressed cohesive zone due to $\dot{d}$ is initially closed, *i.e.* $[[\mathbf{u}]] = \tilde{\mathbf{u}}^h = \mathbf{0}$, and consequently $\mathcal{G}_f = 0$ as well as $\nabla^s \tilde{\mathbf{u}}^h = 0$, which remove the second and the fourth terms. The elastic strain tensor yields

$$\boldsymbol{\varepsilon}^e = \boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^d = \boldsymbol{\varepsilon} - \mathcal{H}_\ell \nabla^s \tilde{\mathbf{u}}^h - 2\delta_\ell (\tilde{\mathbf{u}}^h \otimes^s \mathbf{n}_d) = \nabla^s \hat{\mathbf{u}}^h. \quad (27)$$

Finally, the total potential strain energy function becomes

$$\Psi_{\text{pot}} = \int_{\Omega} \psi^e(\boldsymbol{\varepsilon}^e(\hat{\mathbf{u}}^h)) \, d\Omega + \int_{\Omega} \psi^d(\boldsymbol{\varepsilon}^d(\tilde{\mathbf{u}}^h, \boldsymbol{\kappa})) \, d\Omega - \mathcal{P}_{\text{ext}}. \quad (28)$$

Plugging $\mathbf{t}_d = \boldsymbol{\sigma} \cdot \mathbf{n}_d$ and Equation (27) into (28)

$$\begin{aligned} \Psi_{\text{pot}} = \int_{\Omega} \left(\frac{1}{2} \boldsymbol{\sigma} : (\nabla^s \hat{\mathbf{u}}^h + \mathcal{H}_\ell \nabla^s \tilde{\mathbf{u}}^h) + \delta_\ell \mathbf{t}_d([[\mathbf{u}]], \boldsymbol{\kappa}) \cdot \tilde{\mathbf{u}}^h \right) d\Omega \\ - \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathbf{u}^h \, d\Gamma - \int_{\Omega} \bar{\mathbf{b}} \cdot \mathbf{u}^h \, d\Omega. \end{aligned} \quad (29)$$

Remark 1: Potential strain energy of a discrete Heaviside function

Had we not utilised the regularised description of the Heaviside function and instead only focused on δ_ℓ , the energy dissipation in Equation (26) would become

$$\dot{\mathfrak{D}} = \frac{d\boldsymbol{\varepsilon}^d}{dt} = \frac{d}{dt} (\delta_\ell \mathbf{t}_d \cdot \tilde{\mathbf{u}}^h) = 2\delta_\ell \mathbf{t}_d \cdot \dot{\tilde{\mathbf{u}}}^h + \mathcal{G} \frac{\partial \delta_\ell}{\partial d} \dot{d}. \quad (30)$$

After applying $[[\mathbf{u}]] = \tilde{\mathbf{u}}^h = \mathbf{0}$ for the newly progressed cohesive zone due to $\dot{d}$, the elastic strain becomes

$$\boldsymbol{\varepsilon}^e = \boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^d = \boldsymbol{\varepsilon} - 2\delta_\ell (\tilde{\mathbf{u}}^h \otimes^s \mathbf{n}_d) = \nabla^s \hat{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h = \boldsymbol{\varepsilon}^b. \quad (31)$$

Finally, according to Equation (28), the total potential strain energy function becomes

$$\begin{aligned} \Psi_{\text{pot}} = \int_{\Omega} \left(\frac{1}{2} \boldsymbol{\sigma} : (\nabla^s \hat{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h) + \delta_\ell \mathbf{t}_d([[\mathbf{u}]], \boldsymbol{\kappa}) \cdot \tilde{\mathbf{u}}^h \right) d\Omega \\ - \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathbf{u}^h \, d\Gamma - \int_{\Omega} \bar{\mathbf{b}} \cdot \mathbf{u}^h \, d\Omega. \end{aligned} \quad (32)$$

300 which is identical to Equation (29) except for $\mathcal{H}_\ell \leftarrow \mathcal{H}_{\Gamma_d}$, and therefore
 301 the same formulation can be used for discrete Heaviside formulation $\mathcal{H}_{\Gamma_d}$.
 302 However, the condition number issue adhered to the discontinuous Heaviside
 303 enrichment will certainly remain when the crack divides an element dispro-
 304 portionately. Therefore, remedies identical to those of XFEM, referred in
 305 Introduction (Section 1), must be adopted to circumvent the issue.

306 *Remark 2: Retrieving cohesive phase-field formulation given by Verhoosel*
 307 *and de Borst*

308 Verhoosel and de Borst [39] proposed a cohesive phase-field formulation,
 309 which was later utilised by other contributions for cohesive fracture [32,
 310 40, 41, 42]. Here we prove that the formulation is a special form of our
 311 phase-field-regularised PUM for fracture analysis. We start by rewriting the
 312 infinitesimal strain field in Equation (7), while the last term is defined on the
 313 displacement jump in Equation (13),

$$\boldsymbol{\varepsilon} = \nabla^s \mathbf{u}^h = \nabla^s \hat{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h + \delta_{\Gamma_d} (\llbracket \mathbf{u} \rrbracket \otimes^s \mathbf{n}_{\Gamma_d})$$

314

315 with the elastic strain $\boldsymbol{\varepsilon}^e = \nabla^s \hat{\mathbf{u}}^h + \mathcal{H}_{\Gamma_d} \nabla^s \tilde{\mathbf{u}}^h$. Adopting the diffused Dirac- δ
 316 given by analytical phase-field solution the elastic strain can be rewritten in
 317 terms of the total displacement $\mathbf{u}$ as

$$\boldsymbol{\varepsilon}^e = \nabla^s \mathbf{u}^h - \delta_\ell (v^h \otimes^s \mathbf{n}_d),$$

318

319 where $\llbracket \mathbf{u} \rrbracket$ is substituted with v , a one-dimensional auxiliary jump field which
 320 is not associated with the displacement field $\mathbf{u}^h$. Using the newly derived
 321 strain field, the total potential energy identical to Verhoosel and de Borst
 322 [39] yields,

$$\begin{aligned} \Psi_{\text{pot}} = \int_{\Omega} \left(\frac{1}{2} \boldsymbol{\sigma} : (\nabla^s \mathbf{u}^h - \delta_\ell (v^h \otimes^s \mathbf{n}_d)) + \delta_\ell \mathbf{t}_d (v^h, \boldsymbol{\kappa}) \cdot v^h + \frac{1}{2} \alpha \left| \frac{\partial v^h}{\partial x_n} \right|^2 \right) d\Omega \\ - \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathbf{u}^h d\Gamma - \int_{\Omega} \bar{\mathbf{b}} \cdot \mathbf{u}^h d\Omega. \end{aligned}$$

323 where the last term in the internal potential energy function, *i.e.*, $\frac{1}{2} \alpha \left| \frac{\partial v^h}{\partial x_n} \right|^2$
 324 has been weakly imposed by a penalty factor α to enforce a constant dis-

325 placement jump in the direction normal to the crack profile [32, 39, 41, 42]

$$\frac{\partial v^h}{\partial x_n} = 0,$$

326 with $[[\mathbf{u}]](x_c) \approx v^h(x_c)$.

327 In other words, the formulation by Verhoosel and de Borst is retrieved
 328 by removing the definition of the total displacement field, $\mathbf{u}^h = \hat{\mathbf{u}}^h + \mathcal{K}_{\Gamma_d} \tilde{\mathbf{u}}^h$,
 329 from the energy function and utilising additional constraints to compensate
 330 for the lack of inherent jump in phase-field models, in contrast to the PUM.
 331 Inspired by the formulation proposed by Verhoosel and de Borst, the auxil-
 332 iary jump field approximation has been upgraded using a first-order Taylor
 333 series expansion on the total displacement field $\mathbf{u}$ [40, 86], addressing the
 334 absence of any explicit relationship between the displacement and the aux-
 335 iliary jump field. In our formulation, however, the explicit definition of the
 336 displacement jump in Equation (13) obviates the need for the auxiliary jump
 337 field v , thereby eliminating the additional constraint imposed by the penalty
 338 method and reducing the likelihood of perturbations in equilibrium. More-
 339 over, the resulting jump complies with the PUM, while expansion series are
 340 merely approximation tools.

341 *Remark 3: Inclusion of damage-dependent degradation functions*

342 In future developments, the presented formulation can be extended to accom-
 343 modate a variety of damage-dependent degradation functions. This can help
 344 achieve a more straightforward definition of the cohesive phase-field than
 345 that given in available degradation-dependent contributions. Moreover, it
 346 can help establish a direct relationship between the phase-field variable and
 347 the level set function, which enables tracking of the crack by the phase-field
 348 and accounting for crack propagation and branching. The latter is a major
 349 shortcoming in discrete approaches which can be circumvented by adopting
 350 phase-field, while using PUM on the other hand preserves the functionality
 351 of the proposed approach with coarse meshes.

352 3.1. Displacement weak forms

353 We obtain the admissible displacement field by

$$\mathbf{u}^h = \arg \left\{ \inf_{\mathbf{u}^h \in \mathcal{S}_u} \Psi_{\text{pot}} \right\}, \quad (33)$$

354 where $\mathcal{S}_u = \{\mathbf{u}^h | \mathbf{u}^h(\mathbf{x}) = \bar{\mathbf{u}} \quad \forall \mathbf{x} \in \Gamma_u, \mathbf{u} \in \mathbb{H}^1(\Omega)\}$, and $\mathbb{H}^1$ denotes the first
 355 Hilbert space. Taking the variation of the newly derived potential energy
 356 functions with respect to the displacement:

$$\begin{aligned} \partial \Psi_{\text{pot}, u} = 0 \implies & \int_{\Omega} (\boldsymbol{\sigma} : (\nabla^s \delta \hat{\mathbf{u}}^h + \mathcal{H}_\ell \nabla^s \delta \tilde{\mathbf{u}}^h) + 2\delta_\ell \mathbf{t}_d(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa}) \cdot \delta \tilde{\mathbf{u}}^h) \, d\Omega = \\ & \int_{\Gamma_t} \bar{\mathbf{t}} \cdot (\delta \hat{\mathbf{u}}^h + \mathcal{H}_\ell \delta \tilde{\mathbf{u}}^h) \, d\Gamma + \int_{\Omega} \bar{\mathbf{b}} \cdot (\delta \hat{\mathbf{u}}^h + \mathcal{H}_\ell \delta \tilde{\mathbf{u}}^h) \, d\Omega. \end{aligned} \quad (34)$$

357 Equation (34) can be further decomposed into continuous and discontinuous
 358 parts

$$\int_{\Omega} \boldsymbol{\sigma} : \nabla^s \delta \hat{\mathbf{u}}^h \, d\Omega = \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \delta \hat{\mathbf{u}}^h \, d\Gamma + \int_{\Omega} \bar{\mathbf{b}} \cdot \delta \hat{\mathbf{u}}^h \, d\Omega \quad (35a)$$

$$\begin{aligned} & \int_{\Omega} (\boldsymbol{\sigma} : \mathcal{H}_\ell \nabla^s \delta \tilde{\mathbf{u}}^h + 2\delta_\ell \mathbf{t}_d(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa}) \cdot \delta \tilde{\mathbf{u}}^h) \, d\Omega = \\ & \int_{\Gamma_t} \bar{\mathbf{t}} \cdot \mathcal{H}_\ell \delta \tilde{\mathbf{u}}^h \, d\Gamma + \int_{\Omega} \bar{\mathbf{b}} \cdot \mathcal{H}_\ell \delta \tilde{\mathbf{u}}^h \, d\Omega \end{aligned} \quad (35b)$$

360 for the continuous and discontinuous equilibrium equations of the mechanical
 361 problem. Integrating by parts and adopting the Dirac-delta for the test
 362 functions $\delta \hat{\mathbf{u}}$ and $\delta \tilde{\mathbf{u}}$ similar to the collocation method [87], the updated
 363 strong forms read

$$\begin{cases} \nabla \cdot \boldsymbol{\sigma}(\mathbf{x}) = \bar{\mathbf{b}} & \mathbf{x} \in \Omega \\ \boldsymbol{\sigma}(\mathbf{x}) \cdot \mathbf{n}_t = \bar{\mathbf{t}} & \mathbf{x} \in \Gamma_t \end{cases} \quad (36a)$$

$$2\delta_\ell (\mathbf{t}_d(\llbracket \mathbf{u} \rrbracket, \boldsymbol{\kappa}) - \boldsymbol{\sigma} \cdot \mathbf{n}_d) = \mathcal{H}_\ell \bar{\mathbf{b}} \quad \mathbf{x} \in \Omega \quad (36b)$$

365 It is noteworthy that the diffused (continuous) definition of the discontinuity,
 366 $\mathcal{H}_\ell$, allows for the use of the Dirac- δ as the test function, which is identical to
 367 the collocated phase-field [88]. This is due to the requirement of the sifting
 368 property of the Dirac- δ at x^* , *i.e.*, $\int_{\Omega} \phi(x) \delta_{\text{Dirac}}(x - x^*) \, d\Omega = \phi(x^*)$, which
 369 is violated if (discontinuous) discrete Heaviside function ($\mathcal{H}_{\Gamma_d}$) is used. For
 370 further discussion and alternative solutions, the interested reader is referred
 371 to Fathi *et al.* [87].

372 *3.2. Phase-field weak form*

373 Taking the variation of the newly derived potential energy function in
374 Equation (29) with respect to the phase-field variable d yields

$$\begin{aligned} \partial \Psi_{\text{pot},d} = 0 &\implies \int_{\Omega} (\boldsymbol{\sigma} : \partial \mathcal{H}_{\ell,d} \nabla^s \tilde{\mathbf{u}}^h + 2 \partial \delta_{\ell,d} \mathbf{t}_d ([\mathbf{u}], \boldsymbol{\kappa}) \cdot \tilde{\mathbf{u}}^h) \, d\Omega \\ &= \int_{\Gamma_t} \partial \mathcal{H}_{\ell,d} (\bar{\mathbf{t}} \cdot \tilde{\mathbf{u}}^h) \, d\Gamma + \int_{\Omega} \partial \mathcal{H}_{\ell,d} (\bar{\mathbf{b}} \cdot \tilde{\mathbf{u}}^h) \, d\Omega \end{aligned} \quad (37)$$

375 with

$$\partial \delta_{\ell,d} = \frac{\delta d^h \cdot d^h}{2\ell} + 2\ell \nabla \delta d^h \cdot \nabla d^h, \quad (38a)$$

376

$$\partial \mathcal{H}_{\ell,d} = -2\ell \mathcal{H}_{\Gamma_d} \partial \delta_{\ell,d} = -\mathcal{H}_{\Gamma_d} (\delta d^h \cdot d^h + 4\ell^2 \nabla \delta d^h \cdot \nabla d^h). \quad (38b)$$

377

378 By applying the divergence theorem on the external traction ($\bar{\mathbf{t}} = \boldsymbol{\sigma} \cdot \mathbf{n}_d$)
379 term the final form, after some rewriting, becomes [39]

$$\begin{aligned} \partial \Psi_{\text{pot},d} = 0 &\implies \\ \int_{\Omega} \partial \delta_{\ell,d} \underbrace{(2\ell \mathcal{H}_{\Gamma_d} (\boldsymbol{\sigma} : -\nabla^s \tilde{\mathbf{u}}^h + \nabla \cdot (\boldsymbol{\sigma} \cdot \tilde{\mathbf{u}}^h) + \bar{\mathbf{b}} \cdot \tilde{\mathbf{u}}^h) + 2 \mathbf{t}_d ([\mathbf{u}]_{\ell}, \boldsymbol{\kappa}) \cdot \tilde{\mathbf{u}}^h)}_{\text{constant} \neq 0} \, d\Omega &= 0 \\ &\implies \int_{\Omega} \partial \delta_{\ell,d} \, d\Omega = 0 \end{aligned} \quad (39)$$

380 In this paper, the Dirichlet constraints of the phase-field for pre-existing
381 cracks, *i.e.* $d^h|_{\Gamma_d} = 1$ in Equation (16), are imposed weakly [29, 31, 39].
382 This obviates manipulating the mesh in order to accommodate the Dirichlet
383 constraints [32]. Applying these conditions in the weak form leads to

$$\int_{\Omega} (\delta d^h d^h + 4\ell^2 \nabla \delta d^h \cdot \nabla d^h) \, d\Omega + 2\ell c_0 \int_{\Omega} \delta d^h (d^h - 1) \delta_0(x_n) \, d\Omega = 0 \quad (40)$$

384 where c_0 is the non-negative coefficient weighing the weak imposition of the
385 Dirichlet constraint term [39]. Rewriting Equation (40) leads to

$$\int_{\Omega} ((1 + \mathcal{F}) \delta d^h \cdot d^h + 4\ell^2 \nabla \delta d^h \cdot \nabla d^h) \, d\Omega = \int_{\Omega} \mathcal{F} \delta d^h \, d\Omega \quad (41)$$

386 with $\mathcal{F}$, the driving force [39],

$$\mathcal{F} = 2\ell c_0 \delta_0(x_n), \quad \delta_0(x_n) = \begin{cases} \frac{2}{h} \left(1 - \frac{2|x_n|}{h}\right) & -\frac{h}{2} \leq x_n \leq \frac{h}{2} \\ 0 & \text{otherwise} \end{cases}, \quad (42)$$

387 where h indicates the mesh size.

388 4. Discretisation

389 Herein, we focus on finite element implementation by discretising the
390 weak forms, in which the domain Ω is subdivided into non-overlapping smaller
391 sections, also known as elements

$$\Omega = \bigcup_e^{n_{\text{elm}}} \Omega^e. \quad (43)$$

392 While Lagrange polynomials are typically used as the set of basis functions
393 for the customary finite element analysis [89, 90], we adopt spline technology
394 used in isogeometric analysis (IGA) [70], particularly Non-uniform rational
395 B-splines (NURBS).

396 4.1. Bézier-extraction-based NURBS

397 We construct a NURBS surface on a univariate B-spline basis function,
398 *i.e.* $\mathbf{N}^{\text{B-spline}}$,

$$\mathbf{S}(\boldsymbol{\xi}) = \sum_{k=1}^{n_{\text{IGA}}} \mathbf{N}_k(\boldsymbol{\xi}) \mathbf{P}_k, \quad N_{k,p}(\boldsymbol{\xi}) = \frac{w_k N_{k,p}^{\text{B-spline}}}{\mathcal{W}(\boldsymbol{\xi})}. \quad (44)$$

399 that is defined by the Cox-de Boor formula[89] with $\mathcal{W}(\boldsymbol{\xi}) = \sum_{k=1}^n N_k^{\text{B-spline}}(\boldsymbol{\xi}) w_k$
400 which utilise the weights $\mathbf{w}$, and p denotes the order of the underlying knot
401 vector. n_{IGA} is the number of control points $\mathbf{P}$.

402 Use of Bézier extraction provides a direct transformation from physical
403 space to the parametric domain identical to the finite element data-structure.
404 Therefore, we use the Bézier extraction operator $\mathbf{C}$

$$(\mathbf{N}^{\text{B-spline}})^e = \mathbf{C}^e \mathbf{B}. \quad (45)$$

405 The superscript "e" denotes the element index, which is different from "e"
 406 used for the elastic term, and $\mathbf{B}$ indicates a univariate Bernstein polynomial
 407 within the input domain $[-1 \ 1]$,

$$\mathcal{B}_{k,p}(\xi) = \frac{1}{2}(1 - \xi)\mathcal{B}_{k,p-1}(\xi) + \frac{1}{2}(1 + \xi)\mathcal{B}_{k-1,p-1}(\xi) \quad (46a)$$

$$\mathcal{B}_{1,0}(\xi) \equiv 1 \quad (46b)$$

$$\mathcal{B}_{k,p}(\xi) \equiv 0 \quad \text{if } k < 1 \text{ or } k > p + 1. \quad (46c)$$

410 Generalisation to multi dimensions is possible via tensor product.

411 4.2. Phase-field discretisation

412 The phase-field weak form in Equation (41) is now discretised using finite
 413 elements

$$d^h = \mathbf{N}_d \hat{\mathbf{d}}, \quad (47)$$

414 where $\hat{\mathbf{d}}$ denotes the set of phase-field DOFs and $\mathbf{N}_d$ is the NURBS set of
 415 basis functions utilised for phase-field. Accordingly, the phase-field weak
 416 form defined in Equation (41) leads to

$$\overbrace{\int_{\Omega} [(1 + \mathcal{F})\mathbf{N}_d^T \mathbf{N}_d + 4\ell^2 \mathbf{B}_d^T \mathbf{B}_d] \, d\Omega}^{\mathbf{f}_d^{\text{int}}} \hat{\mathbf{d}} = \overbrace{\int_{\Omega} \mathcal{F} \mathbf{N}_d^T \, d\Omega}^{\mathbf{f}_d^{\text{ext}}} \quad (48)$$

417 where $\mathbf{B}$ contains the derivatives of the shape function, and $\mathcal{F}$ is given in
 418 Equation (42). The tangent term corresponding to this equation yields

$$\mathbf{K}_{\hat{d}\hat{d}} = \frac{\partial \mathbf{f}_d^{\text{int}}}{\partial \hat{\mathbf{d}}} = \int_{\Omega} ((1 + \mathcal{F})\mathbf{N}_d^T \mathbf{N}_d + 4\ell^2 \mathbf{B}_d^T \mathbf{B}_d) \, d\Omega. \quad (49)$$

419 The regularised Dirac- δ in Equation (18) is also discretised as

$$\delta_{\ell} = \hat{\mathbf{d}}^T \left(\frac{1}{4\ell} \mathbf{N}_d^T \mathbf{N}_d + \ell \mathbf{B}_d^T \mathbf{B}_d \right) \hat{\mathbf{d}}. \quad (50)$$

420 A convergence study on Equation (48) for the problem of dimensions
 421 $L \times W = 2 \times 1$ —uniformly discretised into elements of size h —is presented in
 422 Figure 4, where the Γ -error is defined

$$\Gamma\text{-error} = \int_{\Omega} \delta_{\ell} d\Omega - \Gamma_d = W \left(\int_{-L/2}^{L/2} \delta_{\ell} dx_n - 1 \right) \quad (51)$$

423 with $\Gamma_d = W$ for this example. Moreover, L^2 norm and H^1 semi-norm errors
 424 are used in this article as well,

$$L^2\text{-error} = \left(\int_{\Omega} (\bullet^h - \bullet_{\text{exct}})^2 d\Omega \right)^{\frac{1}{2}} \quad H^1\text{-error} = \left(\int_{\Omega} \left(\frac{\partial \bullet^h}{\partial \mathbf{x}} - \frac{\partial \bullet_{\text{exct}}}{\partial \mathbf{x}} \right)^2 d\Omega \right)^{\frac{1}{2}} \quad (52)$$

425 The results for the linear and quadratic NURBS discretisations are sepa-
 426 rated, in which L^2 and Γ errors are explored for different scenarios. L^2 norm
 427 error compares the estimates with exact phase-field values $d_{\ell} = \exp\left(-\frac{|x_n|}{2\ell}\right)$,
 428 and Γ -error assesses the Dirac- δ identity which is directly used in the reg-
 429 ularisation, *i.e.*, $\delta_{\ell} = \frac{1}{2\ell} \exp\left(-\frac{|x_n|}{\ell}\right)$. First, we explore the errors against
 430 h -refinement for different length-scales at the fixed $c_0 = 9$. While the L^2 -
 431 errors in Figures 4A and G recommend larger length-scales, insensitivity
 432 of the Γ -errors in D and J are evident (except for the anomaly observed
 433 for $\ell = h/20$ in J). The results corresponding to different c_0 s, at the fixed
 434 $\ell = 2h/15$ reveal different prescriptions for coarse and fine meshes. Never-
 435 theless, $\forall c_0 \in \{8, 9, 10\}$ an approximately stable behaviour is observed for all
 436 discretisations, which is supported by both L^2 and Γ errors presented in Fig-
 437 ures 4B, E, H and K. To consider c_0 and ℓ simultaneously, three-dimensional
 438 surfaces are presented for a fixed discretisation (25 elements) in Figure 4.
 439 Blue and orange surfaces indicate linear and quadratic NURBS, respectively.
 440 L^2 - and Γ -error surfaces are sliced at particular length-scales in the subplots
 441 C, F, I and L for better clarity. It is worth mentioning that, although the
 442 above-mentioned optimum values for c_0 proved to be effective in this work,
 443 c_0 might generally be expected to be problem-dependent.

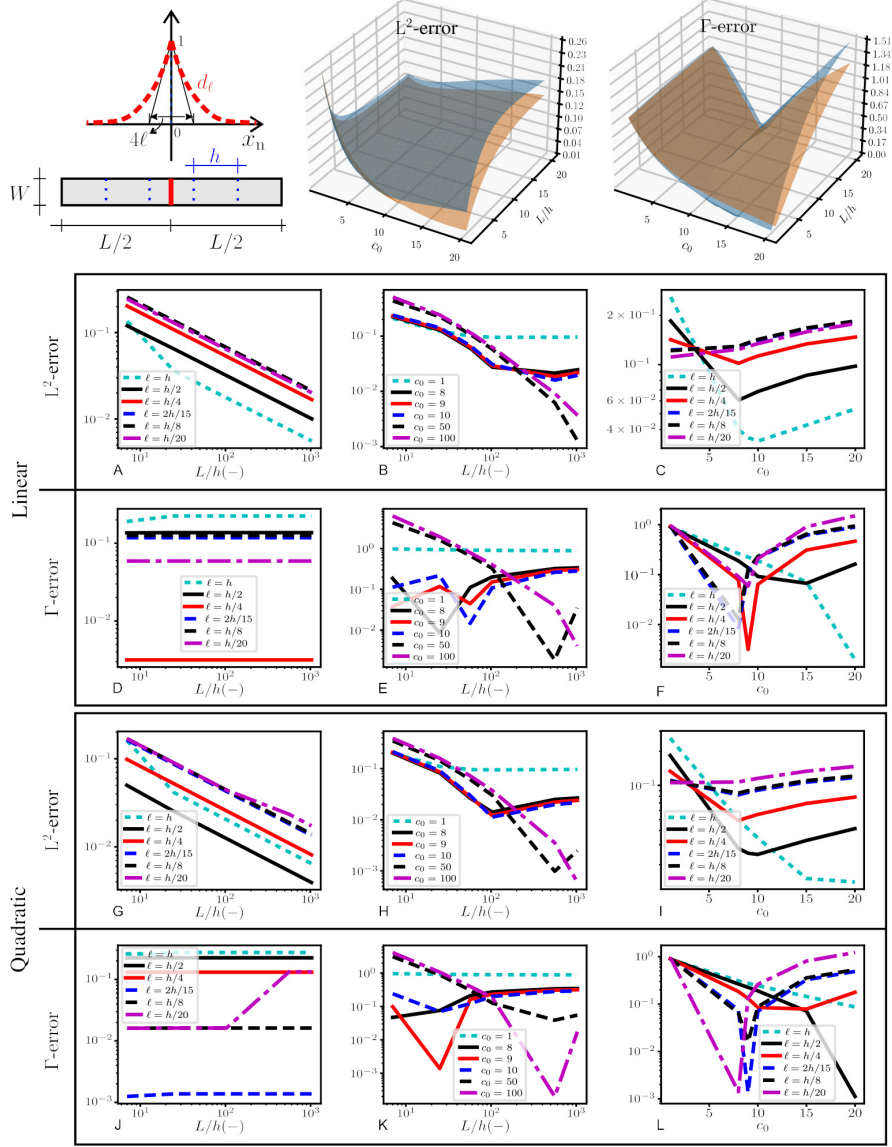


Figure 4: Sensitivity analysis of the phase-field solution with respect to the length-scale parameter ℓ and the constant c_0 . L^2 -norm error with respect to the exact phase-field solution and Γ -error are reported for different ℓ s in A, D, G and J, adopting $c_0 = 9$. B, E, H and K explore different c_0 s in the form of L^2 and Γ errors when the constant length-scale $\ell = 2h/15$ is adopted for 25 elements. To explore the effect of c_0 and ℓ simultaneously, L^2 and Γ errors are shown by three-dimensional surfaces on the top, where blue and orange surfaces indicate linear and quadratic NURBS shape functions, respectively. The surfaces are further sliced for various ℓ s in C, F, I and L.

444 *4.3. Displacement discretisation*

445 Next, we discretise the displacement field $\mathbf{u}^h$ for a regularised extended
 446 finite element analysis using a Bubnov-Galerkin approach as

$$\mathbf{u}^h = \hat{\mathbf{u}}^h + \mathcal{H}_\ell \tilde{\mathbf{u}}^h = \mathbf{N}_{\hat{u}} \hat{\mathbf{u}} + \mathcal{H}_\ell \mathbf{N}_{\tilde{u}} \tilde{\mathbf{u}}. \quad (53)$$

447 Discretisation of the governing weak form in Equations (35a) and (35b)
 448 yields

$$\overbrace{\int_{\Omega} \mathbf{B}_{\hat{u}}^T \boldsymbol{\sigma} \, d\Omega}^{\mathbf{f}_{\hat{u}}^{\text{int}}} = \overbrace{\int_{\Gamma_t} \mathbf{N}_{\hat{u}}^T \bar{\mathbf{t}} \, d\Gamma + \int_{\Omega} \mathbf{N}_{\tilde{u}}^T \bar{\mathbf{b}} \, d\Omega}^{\mathbf{f}_{\tilde{u}}^{\text{ext}}} \quad (54a)$$

$$\overbrace{\int_{\Omega} \left((\mathbf{B}_{\tilde{u}}^{\text{enr}})^T \boldsymbol{\sigma} + 2\delta_\ell \mathbf{N}_{\tilde{u}}^T \mathbf{R}^T \mathbf{t}_d^{\text{loc}} \right) d\Omega}^{\mathbf{f}_{\tilde{u}}^{\text{int}}} = \overbrace{\int_{\Gamma_t} (\mathbf{N}_{\tilde{u}}^{\text{enr}})^T \bar{\mathbf{t}} \, d\Gamma + \int_{\Omega} (\mathbf{N}_{\tilde{u}}^{\text{enr}})^T \bar{\mathbf{b}} \, d\Omega}^{\mathbf{f}_{\tilde{u}}^{\text{ext}}} \quad (54b)$$

449 with $\square^{\text{enr}} = \mathcal{H}_\ell \square$. The equations are then linearised in a Newton-Raphson
 450 iterative scheme with the tangent terms

$$\mathbf{K} = \begin{bmatrix} \mathbf{K}_{\hat{u}\hat{u}}^\Omega & \mathbf{K}_{\hat{u}\tilde{u}}^\Omega \\ \mathbf{K}_{\tilde{u}\hat{u}}^\Omega & \mathbf{K}_{\tilde{u}\tilde{u}}^\Omega \end{bmatrix} \quad (55)$$

451 where

$$\mathbf{K}_{\hat{u}\hat{u}}^\Omega = \frac{\partial \mathbf{f}_{\hat{u}}^{\text{int}}}{\partial \hat{\mathbf{u}}} = \int_{\Omega} \mathbf{B}_{\hat{u}}^T \mathbb{C} \mathbf{B}_{\hat{u}} \, d\Omega \quad (56a)$$

$$\mathbf{K}_{\hat{u}\tilde{u}}^\Omega = (\mathbf{K}_{\tilde{u}\hat{u}}^\Omega)^T = \frac{\partial \mathbf{f}_{\tilde{u}}^{\text{int}}}{\partial \tilde{\mathbf{u}}} = \left(\frac{\partial \mathbf{f}_{\tilde{u}}^{\text{int}}}{\partial \hat{\mathbf{u}}} \right)^T = \int_{\Omega} \mathbf{B}_{\tilde{u}}^T \mathbb{C} \mathbf{B}_{\tilde{u}}^{\text{enr}} \, d\Omega \quad (56b)$$

$$\mathbf{K}_{\tilde{u}\tilde{u}}^\Omega = \frac{\partial \mathbf{f}_{\tilde{u}}^{\text{int}}}{\partial \tilde{\mathbf{u}}} = \int_{\Omega} \left((\mathbf{B}_{\tilde{u}}^{\text{enr}})^T \mathbb{C} \mathbf{B}_{\tilde{u}}^{\text{enr}} + 4\delta_\ell \mathbf{N}_{\tilde{u}}^T \mathbf{R}^T \frac{\partial \mathbf{t}_d^{\text{loc}}}{\partial \llbracket \mathbf{u} \rrbracket} \mathbf{R} \mathbf{N}_{\tilde{u}} \right) d\Omega. \quad (56c)$$

454 Noteworthy is that the displacement jump in Equation (13) is directly used
 455 in deriving the stiffness term presented in Equation (56c).

456 5. Implementation aspects

457 Herein, we mention the main implementation requirements for a PUM
 458 regularised by phase-field for fracture analysis. The staggered solution scheme,
 459 solving phase-field first to determine the state of the crack and then assess-
 460 ing the balance of momentum for the determined crack state, is the well-
 461 established choice in phase-field models [26, 30, 39], which is also adopted
 462 here for the numerical solution of phase-field.

463 5.1. Compatibility enforcement

464 Compatibility enforcement is a necessary component in extended finite
 465 element analysis, which has been studied for different aspects of isogeometric
 466 analysis in finite element [73, 74, 84] and meshfree approaches [76]. Two
 467 numerical techniques have usually been adopted in the extended framework,
 468 the shifting and the blending. The shifting localises the enriched field into a
 469 narrow band in the directions perpendicular to the crack profile, mitigating
 470 the error arisen from the additional enhanced term at nodal/control points.
 471 Rewriting the shifted version of the displacement field in Equation (3) reads

$$u^{\text{h,SH}}(\mathbf{x}) = \sum_{i \in \mathcal{I}} N_{\hat{u}_i}(\mathbf{x}) \hat{u}_i + \sum_{i \in \mathcal{I}^{\text{enr}}} \underbrace{(\mathcal{H}_{\Gamma_d}(x_n) - \mathcal{H}_{\Gamma_d}(x_{n_i}))}_{\mathcal{H}_{\Gamma_d}^{\text{SH}}(x_n)} N_{\tilde{u}_i}(\mathbf{x}) \tilde{u}_i. \quad (57)$$

472 where $\mathcal{I}$ denotes the set of all control points in the domain and $\mathcal{I}^{\text{enr}}$ indicates
 473 the enriched subset of control points, *i.e.*, $\mathcal{I}^{\text{enr}} \subset \mathcal{I}$. Figure 5 illustrates
 474 $\mathcal{H}_{\Gamma_d}^{\text{SH}}(x_n)$. It is noted that a $\mathcal{C}^0$ -continuity at element boundaries, accompa-
 475 nied with the shifting technique, guarantees the enriched field confined by
 476 the cracked elements boundaries only. This can be either achieved by the use
 477 of Lagrange polynomials within a customary finite element framework or by
 478 knot insertion in isogeometric analysis.[71]

479 The blending technique, on the other hand, removes the effect of disconti-
 480 nuity at crack front arisen from the inter-element share of the control points.
 481 This is imposed by employing a Heaviside step function, $\mathcal{H}_{\Gamma_d}^{\text{BL}} = 0$ in front of
 482 the crack tip and $\mathcal{H}_{\Gamma_d}^{\text{BL}} = 1$ otherwise, in XIGA [74]. Applying the blending
 483 technique to the shifted displacement field yields

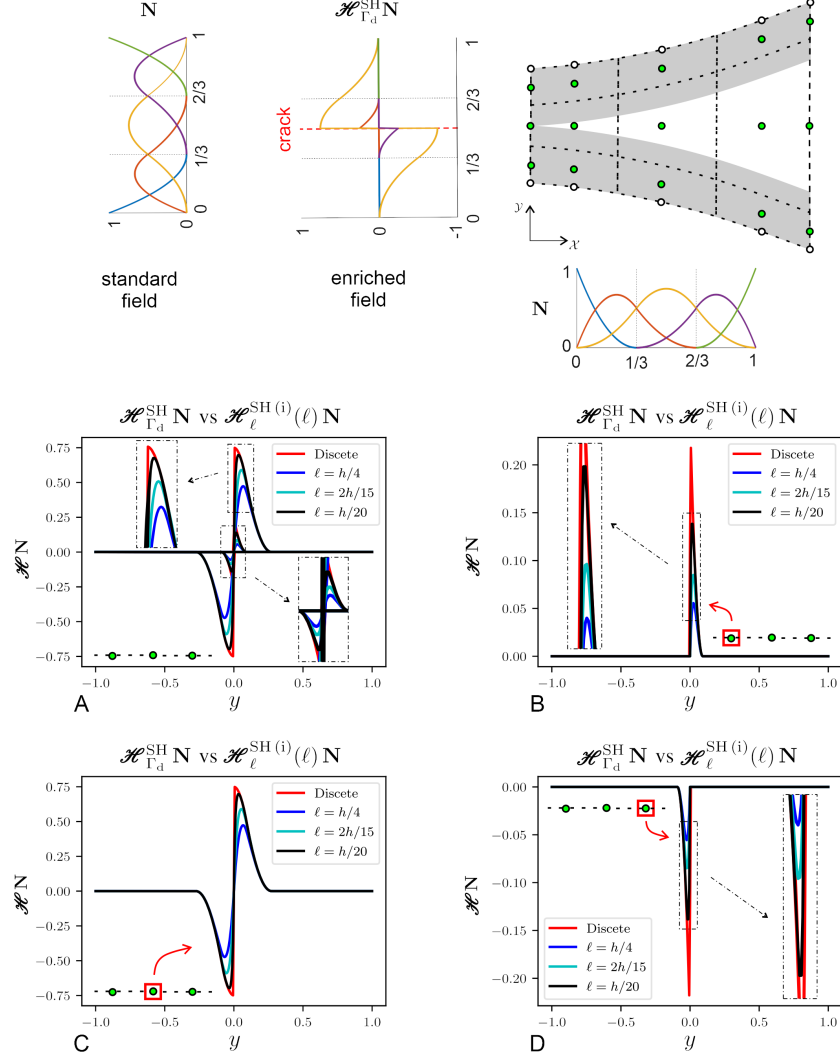


Figure 5: Shifting technique comparison between the discrete and the regularised Heaviside functions, $\mathcal{H}_{\Gamma_d}^{SH}$ and $\mathcal{H}_{\ell}^{SH(i)}$. Shifting technique is shown for $\mathcal{H}_{\Gamma_d}^{SH}$ in a Mode-I fracture of quadratic NURBS, following the XIGA's enrichment recipe (the green circles denote the enriched control points). Univariate standard and enriched fields are illustrated for the discontinuity in the $\mathcal{Y}$ -direction, while the medium is continuous along the $\mathcal{X}$ -direction. The results of the shifting techniques for an eleven-element discretisation are compared for different length-scales in A. For further clarification, the set of shifted basis functions associated with the enriched control points are represented separately in B, C and D.

$$u^{\text{h,SH}}(\mathbf{x}) = \sum_{i \in \mathcal{I}} N_{\hat{u}_i}(\mathbf{x}) \hat{u}_i + \sum_{i \in \mathcal{I}^{\text{enr}}} \mathcal{H}_{\Gamma_d}^{\text{BL}}(x_s) \underbrace{(\mathcal{H}_{\Gamma_d}(x_n) - \mathcal{H}_{\Gamma_d}(x_{n_i}))}_{\mathcal{H}_{\Gamma_d}^{\text{SH}}(x_n)} N_{\tilde{u}_i}(\mathbf{x}) \tilde{u}_i. \quad (58)$$

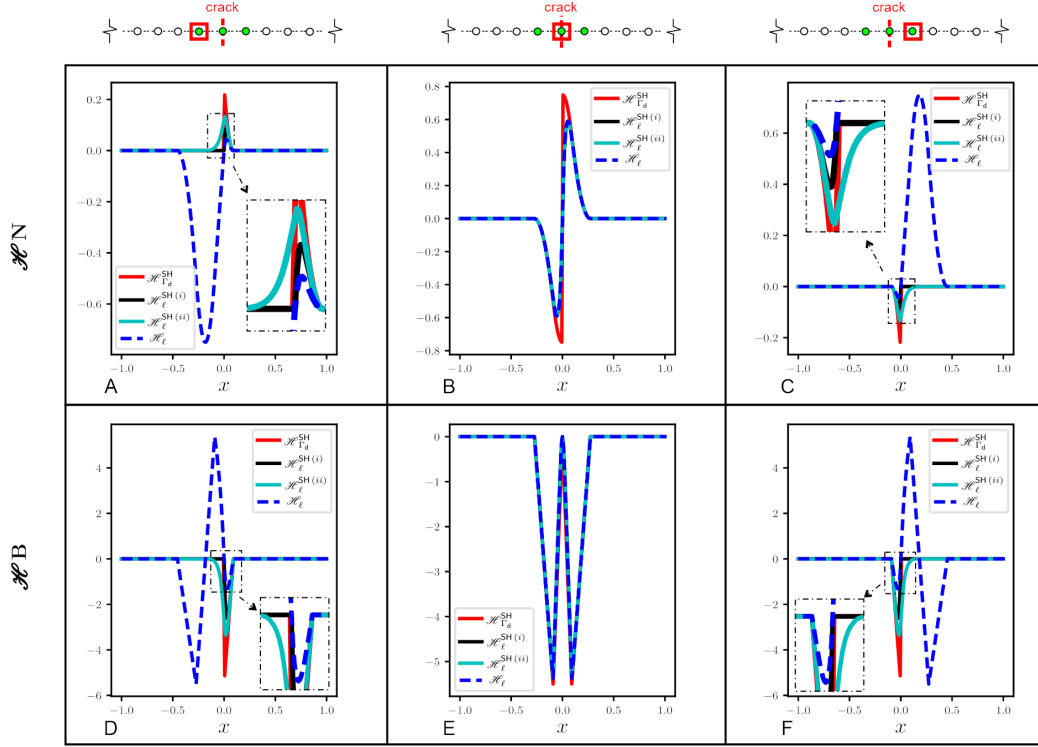


Figure 6: Comparison among the diffused Heaviside function and its shifted candidates. Enriched set of shape functions (A, B and C) and their gradients (D, E and F) are presented for the enriched control points, indicated by green circles, individually.

484 5.1.1. Regularised compatibility enforcement technique

485 Now we derive compatibility enforcement on the regularised Heaviside
 486 function in Equation (19).

$$u_{\ell}^{\text{h}}(\mathbf{x}) = \sum_{i \in \mathcal{I}} N_{\hat{u}_i}(\mathbf{x}) \hat{u}_i + \sum_{i \in \mathcal{I}^{\text{enr}}} \underbrace{\mathcal{H}_{\Gamma_d}(x_n) (1 - 2\ell\delta_{\ell})}_{\mathcal{H}_{\ell}(x_n)} N_{\tilde{u}_i} \tilde{u}_i \quad (59)$$

487 We employ the shifting technique for the diffused displacement field, lead-
 488 ing to two candidates for the shifted, regularised, Heaviside functions:

$$u_\ell^{\text{h,SH(i)}}(\mathbf{x}) = \sum_{i \in \mathcal{I}} N_{\hat{u}_i}(\mathbf{x}) \hat{u}_i + \sum_{i \in \mathcal{I}^{\text{enr}}} \underbrace{(\mathcal{H}_{\Gamma_d}(x_n) - \mathcal{H}_{\Gamma_d}(x_{n_j})) (1 - 2\ell\delta_\ell)}_{\mathcal{H}_\ell^{\text{SH(i)}}(x_n)} N_{\tilde{u}_i} \tilde{u}_i \quad (60)$$

489 and

$$u_\ell^{\text{h,SH(ii)}}(\mathbf{x}) = \sum_{i \in \mathcal{I}} N_{\hat{u}_i}(\mathbf{x}) \hat{u}_i + \sum_{i \in \mathcal{I}^{\text{enr}}} \underbrace{(\mathcal{H}_{\Gamma_d}(x_n) (1 - 2\ell\delta_\ell) - \mathcal{H}_{\Gamma_d}(x_{n_i}) (1 - 2\ell\delta_{\ell_i}))}_{\mathcal{H}_\ell^{\text{SH(ii)}}(x_n)} N_{\tilde{u}_i} \tilde{u}_i. \quad (61)$$

490 To confirm $\lim_{\ell \rightarrow 0} \mathcal{H}_\ell^{\text{SH}} = \mathcal{H}_{\Gamma_d}^{\text{SH}}$, a comparison has been made between
 491 $\mathcal{H}_{\Gamma_d}^{\text{SH}}$ and $\mathcal{H}_\ell^{\text{SH(i)}}$ in Figure 5. Next, another comparison is made between the
 492 Heaviside functions candidates, *i.e.*, $\{\mathcal{H}_{\Gamma_d}^{\text{SH}}, \mathcal{H}_\ell, \mathcal{H}_\ell^{\text{SH(i)}}, \mathcal{H}_\ell^{\text{SH(ii)}}\}$ in Figure
 493 6. Shifting candidates are further applied to the equations in Section 4.3
 494 by simply replacing the Heaviside function with the shifted candidates, *i.e.*,
 495 $\mathcal{H}_\ell \leftarrow \mathcal{H}_\ell^{\text{SH(i)}} \vee \mathcal{H}_\ell^{\text{SH(ii)}}$.

496 As previously discussed, the inter-element share of the control points un-
 497 desirably extends the discontinuous field to the crack front. This has been
 498 prevented through the blending technique used for sharp cracks, *e.g.*, $\mathcal{H}_{\Gamma_d}^{\text{BL}}$ in
 499 Equation (58), to retrieve the intact behaviour of crack front. For the regu-
 500 larised formulation, however, the contribution of the discontinuous (enriched)
 501 field at the crack front, see the purple shade in Figure 7, is disregarded, ren-
 502 dering use of additional Heaviside step function redundant. Consequently,
 503 the notion of displacement jump becomes meaningless at the crack front (see
 504 the purple shade in Figure 7), leading to cohesive traction elimination.

505 5.2. Integration and crack extension direction

506 Greville abscissae are naturally used in IGA integration of intact bod-
 507 ies. Following the comprehensive study on extended isogeometric collocation
 508 method, however, they are proven incompatible with PUM for fracture anal-
 509 ysis, as enough integration points are needed at cracked sections for the
 510 Heaviside function evaluation [87]. Hence, Gauss quadrature is used as the
 511 standard integration scheme in this article. It is worth noting that neither
 512 quadrature sub-cell strategy, such as tessellation typically used in XFEM, nor
 513 equivalent polynomial approximants [46] are utilised in this study. Instead,

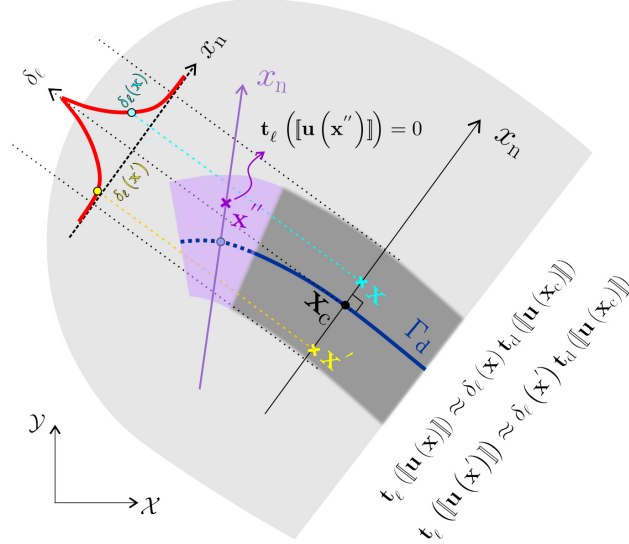


Figure 7: Regularised evaluation of cohesive tractions. Points are first projected onto the crack path (the solid blue line) where the displacement jump $[[\mathbf{u}]]$ is explicitly defined via PUM. Accordingly, points $\mathbf{x}$ and $\mathbf{x}'$ share the same central point on the crack path $\mathbf{x}_c$, therefore sharing the displacement jump and cohesive traction which is finally diffused with their respective δ_ℓ values. Considering a fictitious crack extension profile distinguished by the dotted blue line, cohesive tractions at $\mathbf{x}''$ are disregarded, as points in front of the crack tip, indicated by the purple shade, are intact.

514 a 9×9 standard Gauss quadrature for fixed grid points are considered here,
 515 see Figure 8, unless stated otherwise. This automatically forms a band of
 516 improved integration around the crack path, as shown in Figure 8, which
 517 becomes narrower with mesh refinement. Next, employing phase-field has
 518 relaxed the line integration typically needed for cohesive-zone discrete frac-
 519 ture models, which subsequently eliminates point projection [74, 91]. As an
 520 alternative, the displacement jump at a given point $\mathbf{x}$ is evaluated at the clos-
 521 est point on the crack profile, *i.e.*, $\mathbf{x}_c = \operatorname{argmin}(\|\mathbf{x}^* - \mathbf{x}\|)$ with $\mathbf{x}^* \in \Gamma_d$, see
 522 Figure 7. This is then weighted by the value of $\delta_\ell(\mathbf{x})$, identical to the constant
 523 displacement jump at the diffused region used in Verhoosel and de Borst [39];
 524 however, without additional constraint enforced on the displacement jump.

525 Crack initiation and propagation are introduced in the same manner as in
 526 extended finite element analysis, through the level set technique and enrich-
 527 ment. In this regard, our proposed approach offers no improvement over the
 528 classical extended finite element analysis. However, it can still be improved
 529 in terms of crack initiation and tracking by incorporating a degradation func-

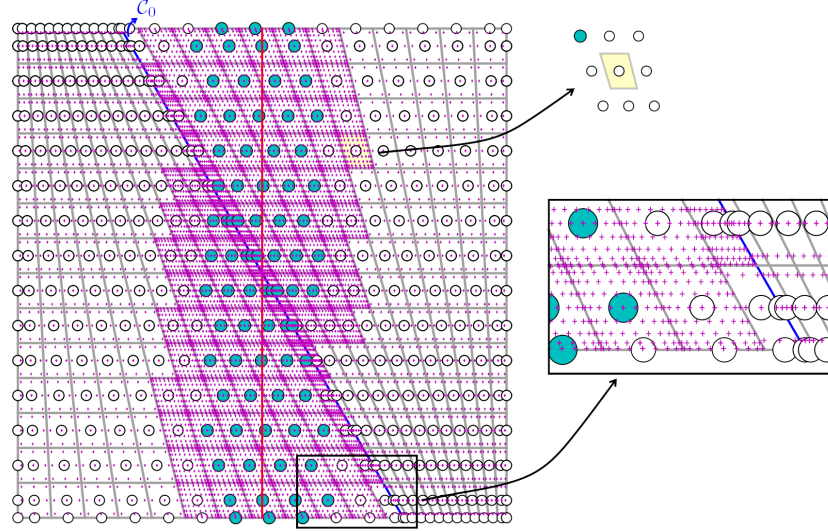


Figure 8: Integration point distribution. For a NURBS of orders p and q in the $\mathcal{X}$ and $\mathcal{Y}$ directions, a normal $(p+1) \times (q+1)$ Gauss quadrature is utilised for non-enriched elements, while a 9×9 is adopted for the elements with enriched control points, see the yellow element with one enriched point. Here, the integration scheme is shown for the quadratic NURBS, *i.e.*, $p = q = 2$. The green circles denote the enriched control points, the red line indicate the crack profile, and a C_0 -continuity is introduced through knot insertion and is illustrated by the blue line.

tion — as discussed in Remark 3 — which would enable automated crack path monitoring via a phase-field-dependent level-set function. This will be the subject of our future work.

In this article, the crack extension direction is determined by the nonlocally-averaged maximum principal stress at the crack tip — an approach commonly used in the extended finite element analysis. The reason behind the non-local averaging [92] is the stress oscillations around the crack tip [24, 74], despite the improvements in stress estimation by spline technologies. Crack nucleation occurs when the majority of the points on the potential crack extension satisfy the evolution criterion on the ultimate tensile strength t_u , *i.e.*, $t_d \geq t_u$. Therefore, no improvement is made here regarding the extension direction and propagation in comparison with the extended finite element analysis. An alternative could be treating the phase-field as damage variable, beyond a mere regularisation term, by considering a degradation function, which will be explored in our future work.

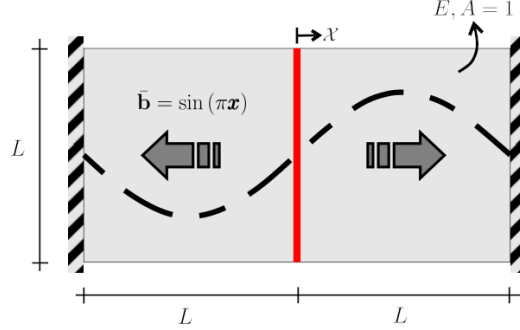


Figure 9: Plate subjected to sinusoidal body forces with a traction-free middle-crack.

545 6. Numerical examples

546 The efficacy of the proposed formulation is examined here at the hand
 547 of multiple numerical examples for stationary and propagating fractures.
 548 Herein, 1 element is assumed in the $\mathcal{Y}$ -direction, unless two values are re-
 549 ported in the format $\bigcirc \times \square$, where $\bigcirc$ and $\square$ indicate the number of elements
 550 used in the $\mathcal{X}$ and $\mathcal{Y}$ directions, respectively.

551 6.1. Traction-free stationary crack

552 A mode-I traction-free fracture is explored in a plate subjected to sinu-
 553 soidal body forces as our first example, see Figure 9. Young's modulus of
 554 $E = 1$ and the cross-section area $A = 1$ are assumed. The exact solution
 555 reads:

$$u^{\text{ex}} = \begin{cases} \frac{1}{\pi^2} \sin(\pi x) - \frac{1+x}{\pi} & \text{if } x < 0 \\ \frac{1}{\pi^2} \sin(\pi x) + \frac{1-x}{\pi} & \text{if } x \geq 0 \end{cases} \quad (62)$$

$$\sigma^{\text{ex}} = \frac{1}{\pi} \cos(\pi x) - \frac{1}{\pi} \quad (63)$$

556

557 The absence of tractions at the crack eliminates $\int_{\Omega} 2\delta_{\ell} \mathbf{N}_{\tilde{u}}^T \mathbf{R}^T \mathbf{t}_d^{\text{loc}} d\Omega$ and
 558 $\int_{\Omega} 4\delta_{\ell} \mathbf{N}_{\tilde{u}}^T \mathbf{R}^T \partial \mathbf{t}_d^{\text{loc}} / \partial [\![\mathbf{u}]\!] \mathbf{R} \mathbf{N}_{\tilde{u}} d\Omega$ in Equations (54)b and (56)c, respectively,
 559 enabling direct investigation on the effect of δ_{ℓ} on the regularised Heaviside
 560 function $\mathcal{H}_{\ell}$, and the associated shifting technique.

561 6.1.1. Enrichment scheme assessment

562 We first explore a potential relationship between the phase-field variable
 563 d and the enrichment of individual control points, with the objective to find
 564 a prescription for the enrichment scheme. Quadratic NURBS is assumed
 565 for the discretisation of 25 uniform elements, and phase-field is evaluated
 566 analytically through direct evaluation of the exponential decay in Equation
 567 (15). Figure 10 shows the displacement (u) and the stress (σ) results, which
 568 are compared with the exact analytical solution for three different enrichment
 569 criteria and length-scales.

570 First we explore the non-shifted displacement field $u_\ell^h(\mathbf{x})$, in Equation
 571 (59), and compare the results of the enrichment criterion $\mathbf{d} \geq 1e - 1$ shown
 572 in Figure 10A. The length-scale $\ell = h/4$ gives the best agreement with re-
 573 spect to the exact solution, even though negligible oscillations are observed
 574 for the stress curve at the peak. In comparison with the enrichment scheme
 575 in XIGA [73, 74], control points corresponding to the cracked element are
 576 enriched, only $\ell = h/4$ matches, while $\ell = 2h/15$ and $\ell = h/20$ are under-
 577 enriched. A similar pattern is also observed for $\mathbf{d} \geq 1e - 4$, where $\ell = 2h/15$
 578 coincides with the XIGA enrichment. It is interesting to see the effect of
 579 over-enrichment in $\ell = h/4$, where displacement agreement is satisfactory,
 580 but oscillations result in stress estimates. $\ell = h/20$ shows under-enrichment,
 581 leading to an incorrect displacement jump at the crack. For $\mathbf{d} \geq 1e - 7$,
 582 XIGA's enrichment with $\ell = h/20$ returns satisfactory results for both dis-
 583 placement and stress comparisons. For other length-scales, however, over-
 584 enrichment exhibit oscillations which are more pronounced for $\ell = h/4$ and
 585 less noticeable for $\ell = 2h/15$ (with only one additional column of control
 586 points enriched at each side of the crack).

587 Instability is an immediate consequence of over-enrichment for $u_\ell^{h,SH(i)}(\mathbf{x})$
 588 in Figure 10B, as the stiffness matrix becomes singular. This is reminis-
 589 cent of the ill-conditioning issue of the discrete fracture analysis with over-
 590 enrichment. While the combination of under-enrichment and XIGA's recipe
 591 renders stable solutions, the errors are significant. Thus, no further investi-
 592 gation is carried out on $u_\ell^{h,SH(i)}(\mathbf{x})$. On the other hand, $u_\ell^{h,SH(ii)}(\mathbf{x})$ renders
 593 stable results for all enrichment scenarios. Again, the best agreement is ob-
 594 served for XIGA's prescription, while, in general, the results are very similar
 595 to those of $u_\ell^h(\mathbf{x})$, though with a completely different set of shape functions
 596 utilised, see Figure 6.

597 Overall, the XIGA enrichment recipe appears to be the most suitable op-

tion, irrespective of the phase-field value. According to XIGA, control points corresponding to the cracked element should be enriched. Hereafter, we only investigate the enrichment scheme proposed by XIGA. Another observation is that, by adopting the XIGA's enrichment scheme, stress oscillations slightly appear for $\ell = h/4$, and disappear with smaller length-scales adopted. It can be deduced that, maintaining the XIGA's prescription, the length-scale parameter ℓ should be smaller than $h/4$ in a uniform mesh of length h . The physical interpretation of this choice is that the regularised phase-field variable d_ℓ majorly occupies the cracked element and vanished elsewhere, see Figure 2B. This is in line with the philosophy of extended finite element analysis, which localises the (discontinuous) enriched field within the cracked element.

6.1.2. $\mathbb{p}$ -refinement: continuity-order assessment

One of the main features of IGA is the higher inter-element continuity provided by NURBS basis functions, which increases with $\mathbb{p}$ -refinement by elevating the continuity-degree of NURBS. A uniform discretisation of 25 elements is adopted following the enrichment scheme prescribed by XIGA. Analytical δ_ℓ defined in Equation (15) is utilised here. The results are presented for displacement and stress estimates, which are compared with the analytical solution, in Figure 11. Two scenarios are explored for displacement field candidates: the non-shifted displacement field $u_\ell^h(\mathbf{x})$ and the shifted $u_\ell^{h, \text{SH}(\text{ii})}(\mathbf{x})$ format.

Similar to the results in Section 6.1.1, no difference between the results of the non-shifted formula $u_\ell^h(\mathbf{x})$ and the shifted version $u_\ell^{h, \text{SH}(\text{ii})}(\mathbf{x})$ is observed. Stress oscillations observed for $\ell = h/4$, which are more pronounced for quartic and less for quadratic and cubic NURBS basis functions, while satisfactory results are obtained for other length-scales. This is reminiscent of the conclusion made in the previous section about confining d_ℓ by choosing $\ell < h/4$. Moreover, the good agreement observed in Figure 10 for $\mathbb{p}$ -refinement is another validation of the hypothesis about the best enrichment scheme. Therefore, the term "enrichment" refers to the XIGA's recipe in the remainder.

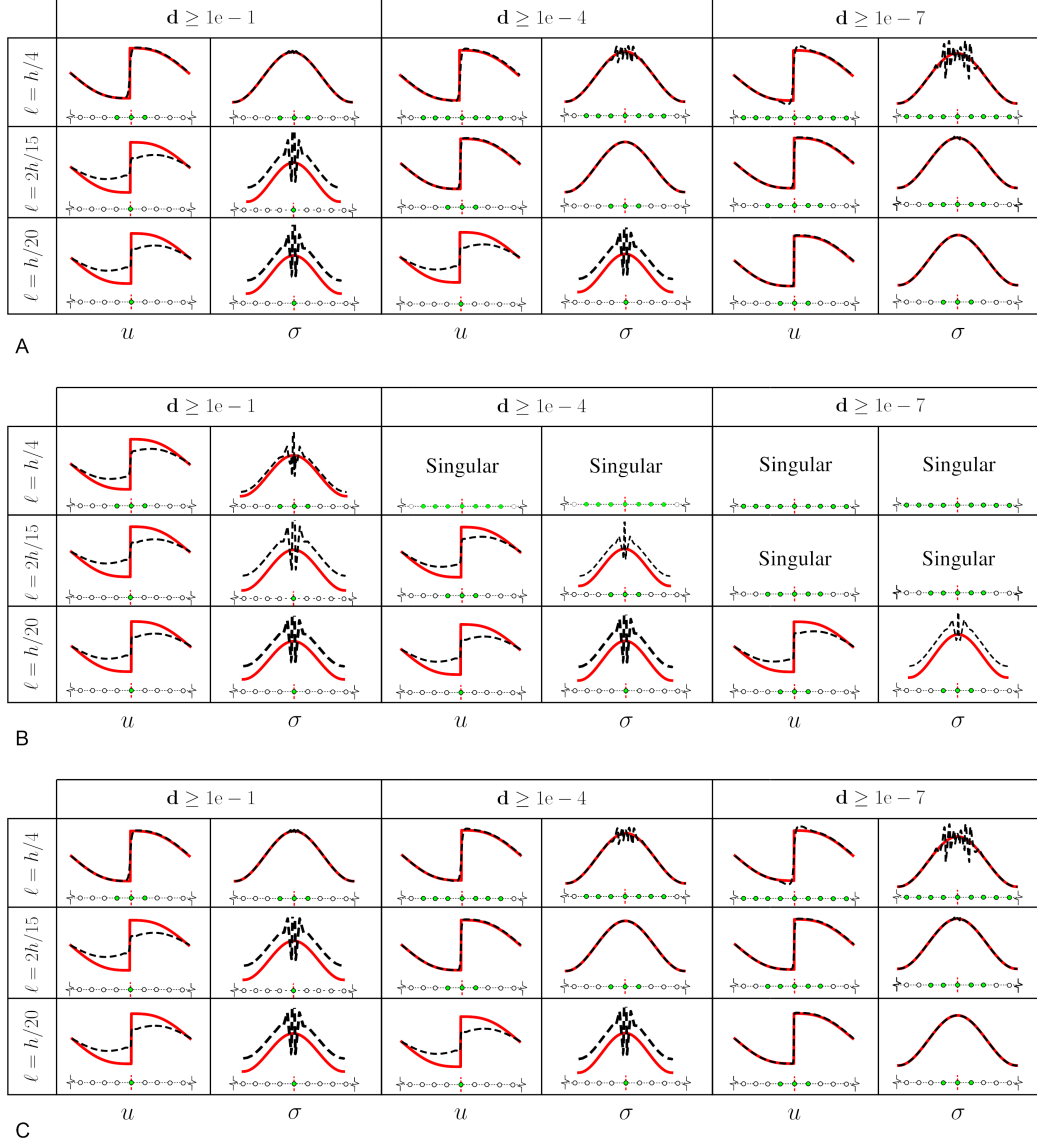


Figure 10: Establishing a relationship between the enriched field and the phase-field variable d for (A) $u_\ell^h(\mathbf{x})$, (B) $u_\ell^{h,SH(i)}(\mathbf{x})$ and (C) $u_\ell^{h,SH(ii)}(\mathbf{x})$. u and σ denote the displacement and the stress results of our proposed approach, indicated by the dashed black line, compared with the analytical solution, given by the solid red line. Enriched control points corresponding to the enrichment criterion defined on the phase-field variable d are illustrated by green circles. Note that the crack is located at the middle of the plate.

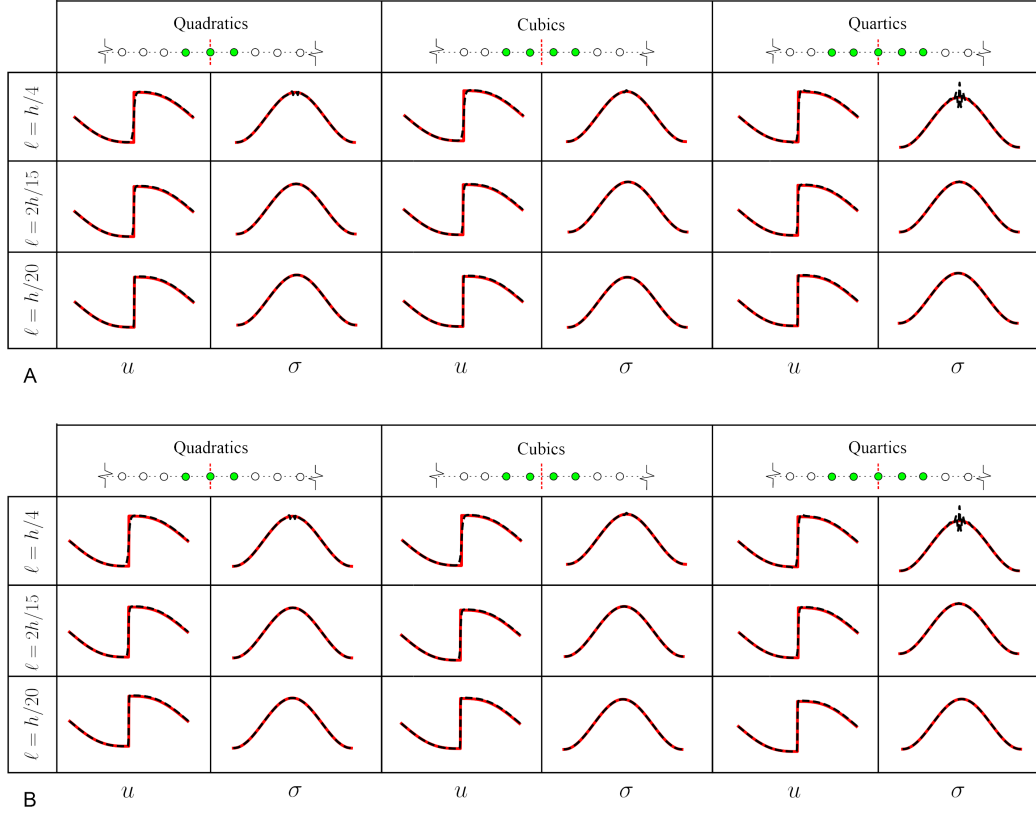


Figure 11: p-refinement comparison for (A) $u_\ell^h(\mathbf{x})$ and (B) $u_\ell^{h,SH(ii)}(\mathbf{x})$. Green circles indicate the enriched control points associated with each NURBS. The results are compared with the analytical solution denoted by the red solid curves, comprised of displacement u and stress σ estimates.

6.1.3. Analytical vs numerical phase-field

So far the analytical δ_ℓ in Equation 15 has been utilised in the previous examples. Now, we explore the numerical evaluation of the phase-field formula given in Sections 3.2 and 4.2. A 25-element discretisation with the length scale $\ell = 2h/15$ and the penalty factor $c_0 = 9$ are adopted based on the investigations carried out in Section 4.2. The displacement and stress results are shown in Figure 12 for various combinations of NURBS utilised for mechanical and phase-field problems. A chamfered response is observed for the displacements at a short distance at each side of the crack (see the zoom sections of u in 12A). No substantial difference between analytical and numerical phase-field solutions is observed for the displacements in Figure 12A, with all

641 solutions showing acceptable accuracy. This is further supported by the L^2
642 norm error on displacements in Figure 12C, where no significant disparity is
643 observed between the analytical and numerical results. On the other hand,
644 the stresses exhibit different responses for the $\mathbb{p}$ -refinement within the me-
645 chanical problem, as shown in Figure 12A (see σ columns of Lin (u), Quad (u)
646 and Cub (u)). Unsurprisingly, regardless of phase-field, oscillatory stresses
647 result for linear NURBS utilised in the mechanical problem, as a quadratic
648 approximate is needed to ensure continuous gradient of displacement.

649 It is best to compare phase-field independently by checking the Γ -error
650 introduced previously, which allows for assessing the quality of δ_ℓ estimates
651 for $\mathbb{p}$ -refinement. Figure 12B compare the approximated regularised Dirac- δ s
652 against the exact values given by the exponential decay in Equation (15). A
653 tortuous solution is observed for the linear order while quadratic and cubic
654 NURBS guarantee a smooth response. The Γ -error in 12B determines the
655 best NURBS order to satisfy the central identity $\int_\Omega \delta_\ell d\Omega = \Gamma_d$, with the
656 orders ranked as quadratic, cubic and linear based on the least error.

657 In a comparison between numerical and analytical solutions of phase-
658 field, those oscillation caused by the former are more pronounced, which
659 are substantially mitigated by $\mathbb{p}$ -refinement, consistently improving as we
660 move away from Lin (u) towards Cub (u), which is also confirmed by the H^1
661 semi-norm error of the mechanical problem given in Figure 12D, where both
662 analytical and numerical solutions improve with $\mathbb{p}$ -refinement. To determine
663 the best combination, H^1 semi-norm errors suggest Cub (u)-Quad (d) by
664 looking at the yellow bars, which is in line the Γ error in Figure 12B for
665 numerical phase-field solutions.

666 6.2. Cohesive fracture: linear traction-separation relationship

667 Now, cohesive tractions are considered for a uniform tension test of a plate
668 cracked in the middle, as depicted in Figure 13A. The effect of the length-scale
669 on analytical estimates of δ_ℓ is presented in Figure 13B, which are compared
670 with the numerical estimates in C. The importance of such example lies in
671 the cohesive terms which rely on the quality of the δ_ℓ estimate, particularly
672 in $\mathbf{f}_u^{\text{int}}$ and $\mathbf{K}_{uu}^{\text{int}}$ of Equations (54)b and (56)c. A linear traction-separation
673 relationship is utilised, *i.e.*,

$$\mathcal{G}_f = \frac{1}{2}k \llbracket u_n \rrbracket^2, \quad t_d = k \llbracket u_n \rrbracket. \quad (64)$$

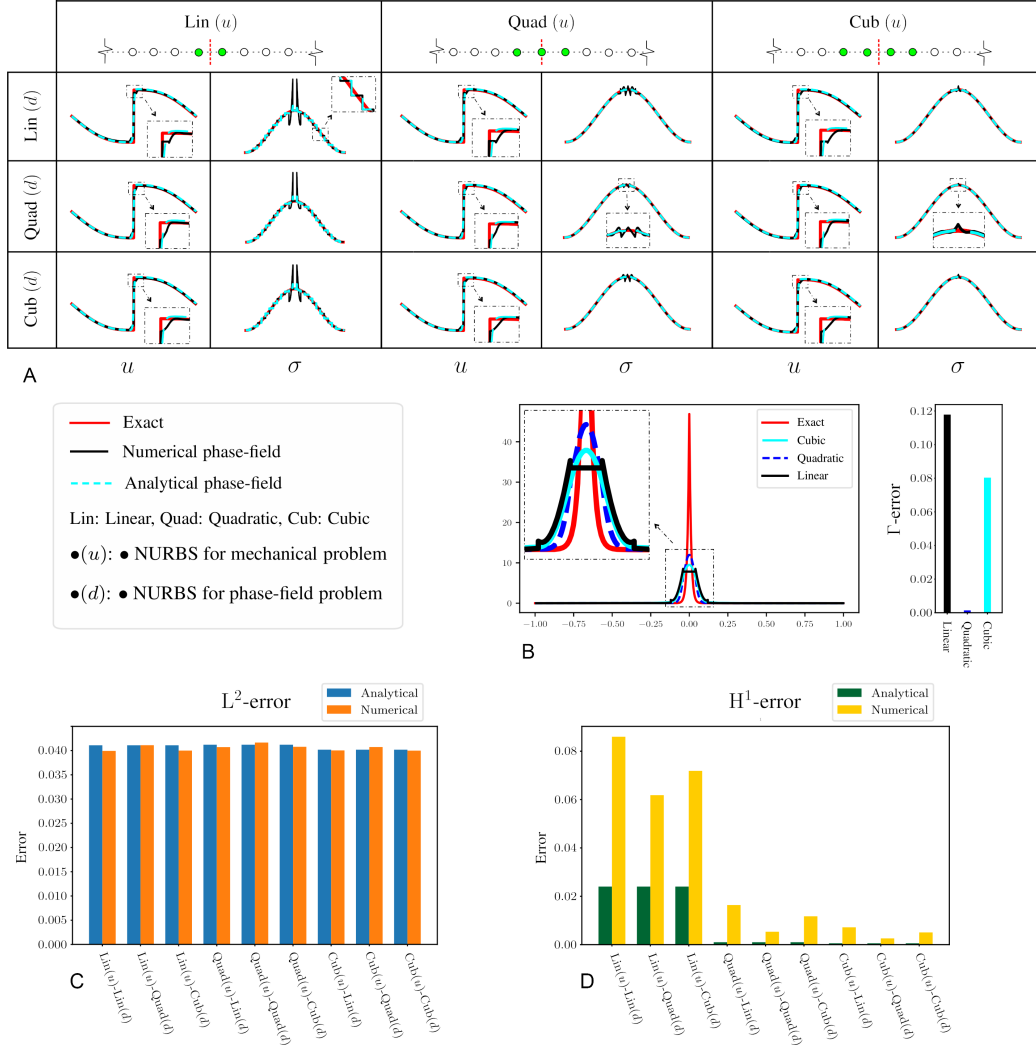


Figure 12: Analytical vs numerical solution for the phase-field problem. Displacement (u) and stress (σ) responses are represented in A for different combinations of NURBS chosen for phase-field and mechanical problems, while B compares numerical δ_ℓ against the analytical (exact) evaluation, assessed by the Γ -error. L^2 norm and H^1 semi-norm errors of the mechanical problem are given in C and D in the form of bar graphs.

674 with a constant tangent $k = 1$. The material parameters read: Young's
675 modulus $E = 1$; Poisson's ratio $\nu = 0$; and the cross-section area $A = 1$. Due
676 to the symmetry with respect to the $\mathcal{X}$ -axis, only 1 element is considered
677 in the $\mathcal{Y}$ -direction, reminiscent of a one-dimensional example. The exact
678 solution is given by

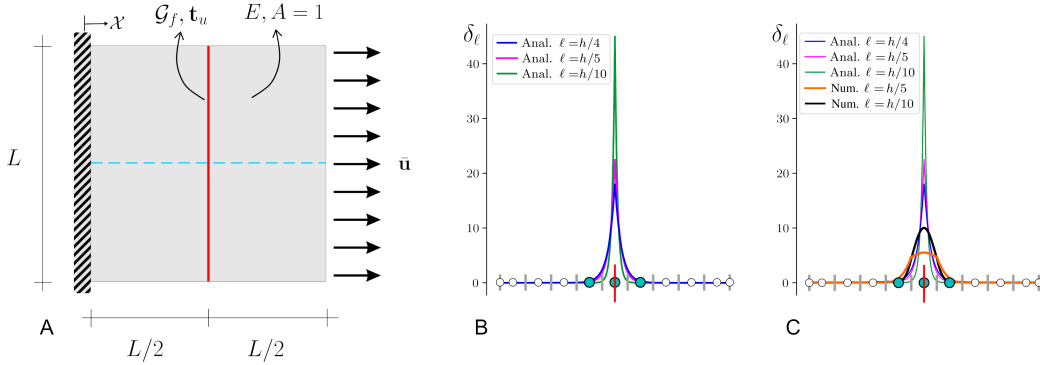


Figure 13: Uniform opening test of (A) a plate cracked at the middle, indicated by the red solid line. Cohesive tractions are characterised by the fracture energy $\mathcal{G}_f$ and tensile strength t_u at the crack location. δ_ℓ values are plotted for a uniform 9-element discretisation at the dashed blue line. Analytical δ_ℓ values are shown in B, and numerical δ_ℓ s are compared to the analytical results in C.

$$u^{\text{ex}} = \begin{cases} \frac{\bar{u}kx}{kL+E} & \text{if } 0 \leq x < L/2 \\ \frac{\bar{u}kx}{kL+E} + \frac{\bar{u}E}{kL+E} & \text{if } L/2 < x \leq L \end{cases} \quad (65)$$

679

$$\sigma^{\text{ex}} = \frac{\bar{u}kE}{kL+E} \quad (66)$$

680 where u^{ex} and σ^{ex} indicate the exact displacement and stress, respectively.

681 Following the results by Section 6.1, two scenarios are considered here:
682 the non-shifted Heaviside function $\mathcal{H}_\ell(x_n)$; the shifted Heaviside function
683 $\mathcal{H}_\ell^{\text{SH(ii)}}(x_n)$. The former is already presented in Section 4.3, whose formula-
684 tions are also valid for the shifted Heaviside by changing $\bullet^{\text{enr}} \leftarrow \bullet^{\text{enr(ii)}}$ (see
685 Appendix Appendix A for the definition of these enriched terms).

686 6.2.1. Length-scale sensitivity analysis

687 First, we conduct a parametric study of the effect of the length-scale
688 on the mechanical response in the presence of cohesive terms by adopting
689 $\ell \in \{h, h/2, h/4, h/5, h/6, 2h/15, h/8, h/10, h/15, h/20\}$. The results are pre-
690 sented for 9 elements in the top two rows of Figure 14. As observed in Figure
691 2B, the tangent of analytical d_ℓ at the peak spans the distance 4ℓ , for which
692 we endeavor to find a physical interpretation with respect to the discrete

fracture model. The results in 14A , B, E and F show the optimum solution when 4ℓ is within the cracked element. Specifically, in a uniform mesh of length h ($h = L/9$ in the $\mathcal{X}$ -direction), choosing $h = 4\ell$ renders the optimum solution for the non-shifted Heaviside formulation, as shown in Figure 14A and B. A slightly larger element length, $h = 5\ell$ in $\mathcal{X}$ -direction, or equivalently a slightly smaller length-scale $\ell = h/5$, exhibits the optimum solution for the shifted Heaviside formulation, $\mathcal{H}_\ell^{\text{SH(ii)}}$, depicted in 14E and F. This is in agreement with the findings in Section 6.1. In the case of numerical phase-field presented in Figure 14C, D, G and H, comprehensively studied in Section 6.1, cubic and quadratic NURBS are adopted for the mechanical and phase-field problems, for which $\ell = h/10$ exhibits the optimum solution for the non-shifted and shifted Heaviside formulations.

The length-scale sensitivity analysis provides a physical interpretation that a majority of d_ℓ should be limited to the cracked element, at least the distance occupied by the gradient of d_ℓ at the peak (4ℓ) in Figure 2B, if not entirely. Therefore, $\ell = h/4$ can be used as a reference value for the phase-field regularised PUM, which can vary according to the quality of δ_ℓ estimates. The disparity between the optimum length-scales found for analytical ($\ell \approx h/4$) and numerical ($\ell \ll h/4$) simulations could be attributed to the blunted peak of the numerical estimates, which leads to a wider tangent (lower gradient values) of d_ℓ at the peak (where crack locates), and consequently a flatter δ_ℓ , see Figure 13C for a comparison between analytical and numerical estimates of δ_ℓ .

6.2.2. h-refinement: discretisation stability

Next, we investigate the effect of the mesh refinement for analytical and numerical phase-field solutions by choosing the discretisation from the list $\{7, 9, 11, 25, 57, 103\}$ in the $\mathcal{X}$ -direction. The results are reported for the optimum length-scales obtained in the previous section, see the bottom two rows in Figure 14. The displacement results, Figures 14I, K, M, O, are difficult to distinguish from the exact solution, while the stress results shown in Figures 14J, L, N and P exhibit clear superiority of the shifted Heaviside formulation, $\mathcal{H}_\ell^{\text{SH(ii)}}$. Unsurprisingly, the analytical phase-field formulation (Figures 14J and N) outperforms the numerical implementation (Figures 14L and P) in terms of accuracy. Regarding the stability, all the cases of discretisations converge to the expected results with the trend of better accuracy yielded for finer meshes, as shown by the arrows in Figures 14I, K, M and O. For the displacements, the finest mesh (103 elements) leads to the best results for

both analytical and numerical phase-field solutions. Regarding the stresses, however, the amplitude of the oscillations increases with mesh refinement, but they are also more concentrated to the crack location, see Figures 14J, L, N and P. Between analytical solutions in Figures 14N and J, the stress oscillations for the shifted Heaviside are negligible in value compared with the non-shifted formulation.

The optimum length-scale results from the analytical and numerical studies are collected in Figure 15. As also supported by L^2 norm and H^1 semi-norm errors in E and F, the shifted Heaviside formulation exhibits better accuracy for both displacements and stresses in A and B, respectively. For the numerical phase-field, however, while L^2 norm errors are very close for quadratic and cubic NURBS, the results by the quadratic discretisation excels from the cubic in terms of the H^1 semi-norm errors in H and the stresses reported in D. Even though lower H^1 semi-norm error is observed for the non-shifted Heaviside function, a more consistent behaviour is observed for the quadratic shifted Heaviside in H. The best results are obtained with the shifted Heaviside function regularised by the analytical phase-field solution, while higher errors and relatively high-amplitude oscillations are observed for non-shifted Heaviside formulation and the stress results of the numerical phase-field solution. Hence, only the shifted Heaviside formulation regularised by analytical phase-field is utilised hereafter.

6.2.3. Sensitivity analysis of singularity

Many approaches have been exploited to mitigate the conditioning issue of the stiffness matrix in extended finite element analysis, as explored in Introduction section. This issue is majorly adhered to two scenarios in which the same reason of inadequate integration of the Heaviside enrichment at one side of the crack can cause the singularity of the stiffness matrix. These scenarios happen when highly disproportionate sections are faced within an element: either the crack path and the element edge align too close to each other; or the crack divides an element too close to a corner. Herein, we explore the hypothesis whether the diffused crack definition can mitigate the ill-conditioning issues. This is motivated by the fact that, unlike discrete fracture models, diffusing the crack leads to a continuous Heaviside function, as explored in Section 2.1. Therefore, within the cracked element, less dependency to proper integration of the Heaviside function is anticipated, since a transition zone for the discontinuity is defined after crack diffusion.

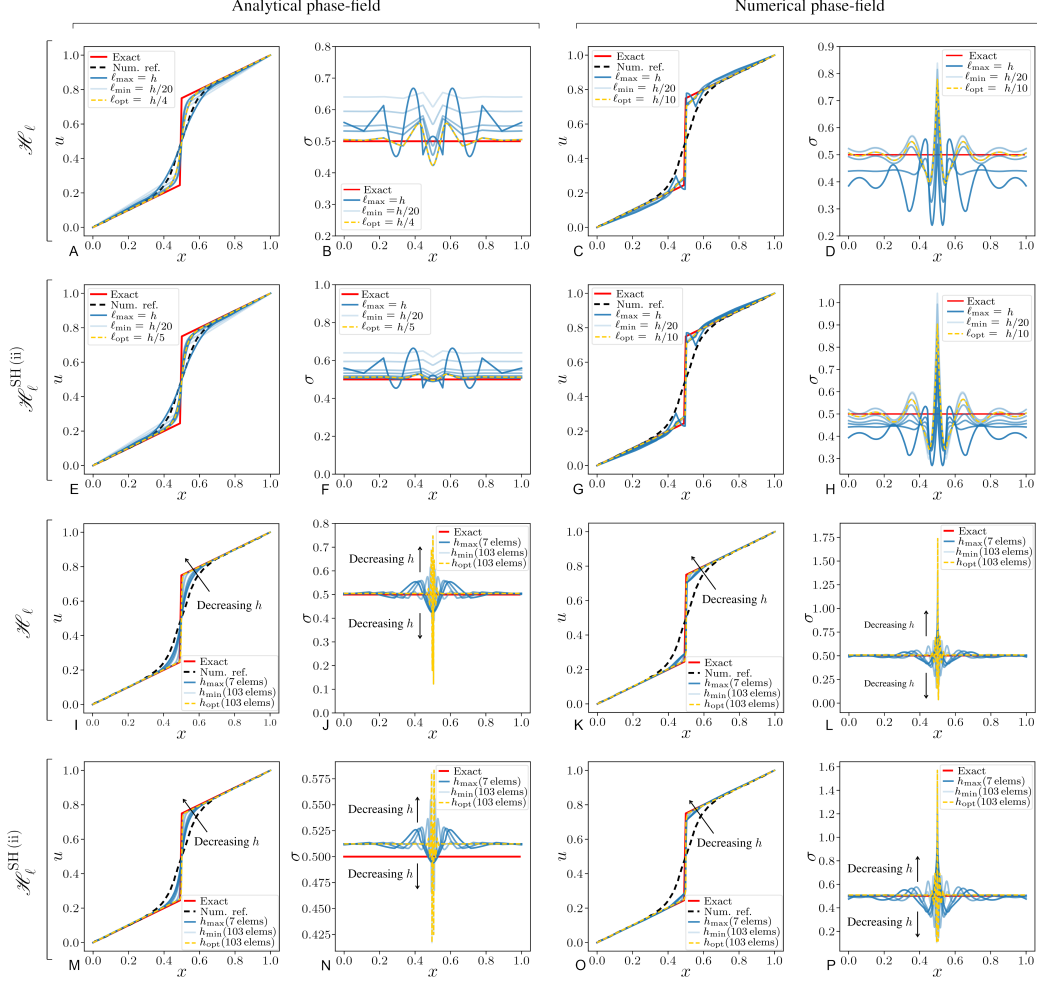


Figure 14: Comparison between the non-shifted ($\mathcal{H}_\ell$) and the shifted ($\mathcal{H}_\ell^{\text{SH(ii)}}$) Heaviside formulation of the displacement (u) and stress (σ) results cast for the constant cohesive traction-separation relationship. The results are compared against the exact solution and the numerical result by Verhoosel and de Borst [39], indicated by the solid red and the dashed black lines, respectively. The analytical solution of the phase-field is assumed for the left two columns, while numerical phase-field approach is utilised for the right two columns. Horizontally, the top two rows represent the effect of the length-scale ℓ for the fixed 9×1 elements, whereas mesh refinement is the focus at the bottom two rows.

766 We have conducted a comprehensive study on the first scenario when
 767 the crack aligns with the edge of element at a very close distance. The
 768 shifted Heaviside formulation and the analytical phase-field description are
 769 adopted for this example with $\ell = \tilde{h}_{\min}/5$, where $\tilde{h}_{\min}$ denotes the min-

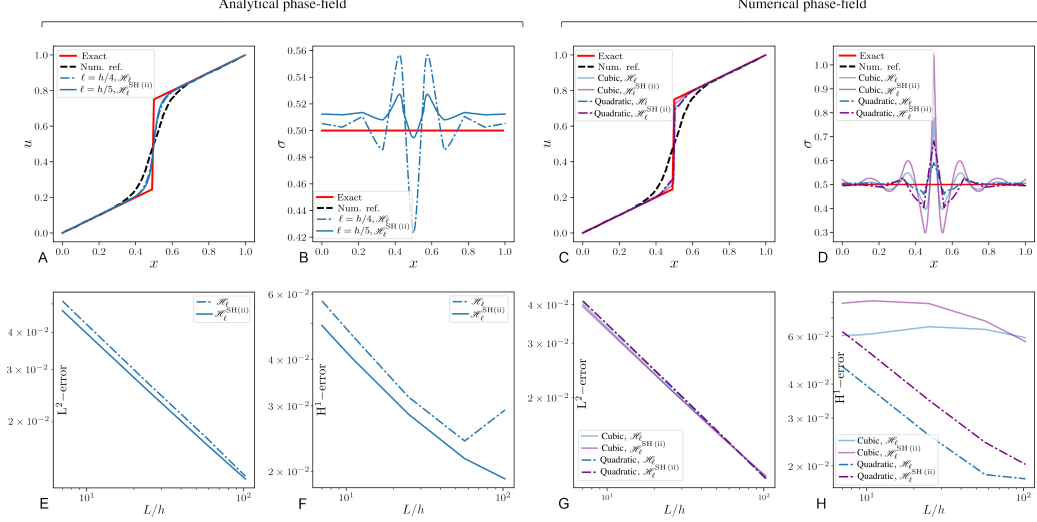


Figure 15: Displacement and stress results of analytical (A and B) and numerical phase-field (C and D) cast for the optimum length-scales specific to the shifted and non-shifted Heaviside functions. 9 elements are considered in the $\mathcal{X}$ -direction. L^2 norm and H^1 semi-norm errors are presented in E and F for the analytical phase-field, while G and H show these error norms associated with cubic and quadratic NURBS used for numerical phase-field. Quadratic displacement field is preserved for all cases.

imum length of the enriched elements. The results are shown in Figure 16. To examine the sensitivity of the distance between the element edge (see the blue edge in A) and the crack path to cause singularity, condition numbers of the stiffness matrix are reported for different distances, *i.e.*, $\mathcal{D}_{\text{singularity}} \in \{0.001, 0.002, 0.005, 0.01, 0.05, 0.07143\}$. The results are shown in B comparing the current study with linear interpolation for XFEM and quadratic NURBS for XIGA. For the normalised distances less than 0.005, both XFEM and XIGA report super high condition numbers, while XFEM exhibits the best performance (the least condition number). The results of the current study yields the most stable condition numbers that is approximately equal to XIGA for $\mathcal{D}_{\text{singularity}}/L = 0.071$ at the beginning.

While assessing the condition number of the stiffness matrix is the best indicator of singularity, it is never an assessment of the performance of the mechanical problem. Therefore, we present the displacement and stress results compared with the exact solution and a numerical reference in Figures 16C and D. While displacements return accepted results, the stress plot of the farthest distance are in good agreement with the exact solution. For

787 $\mathcal{D}_{\text{singularity}}/L = 0.001$, however, the stress plot is slightly worse than the case
 788 0.071. Noteworthy is the fact that the quality of the coarse mesh (7 ele-
 789 ments in the $\mathcal{X}$ -direction) also plays a significant role in worse stress results,
 790 see Figure 16B for a comparison between the two meshes. The accuracy is
 791 further assessed by L^2 norm and H^1 semi-norm errors presented in E, where
 792 approximately 4×10^{-2} results for both errors at $\mathcal{D}_{\text{singularity}}/L = 1 \times 10^{-11}$.
 793 From the stability point of view, the condition number remains in the range
 794 $]1.4 \times 10^4 \ 1.7 \times 10^4[$ even for $\mathcal{D}_{\text{singularity}}/L = 1 \times 10^{-16}$.

795 The second scenario of conditioning issue, when the crack divides the
 796 element too close to a corner, happens frequently in extended finite element
 797 analysis since the crack path and the underlying mesh layout are independent,
 798 see the yellow elements in Figure 17B. The results in D and E indicated by
 799 the green dashed line exhibit excellent agreement with the exact solution for
 800 quadratic NURBS, even though the crack exactly passes the corner of the
 801 two yellow elements at the middle of the plate. Noteworthy is the fact that
 802 the absence of Kronecker- δ in $\mathcal{C}^{p-1}$, $p \geq 2$ has no effect on the ill-conditioning
 803 issue, as observed for $\mathcal{C}^1$ in quadratic NURBS.

804 6.2.4. Irregular discretisation

805 One of the main features of the phase-field-regularised PUM developed
 806 here is the capability of handling irregular discretisations even for the coars-
 807 est meshes typically used in discrete fracture models. We have adopted the
 808 shifted regularised Heaviside formulation and the analytical phase-field de-
 809 scription here, while $\ell = \tilde{h}_{\min}/5$ is the length-scale defined on the minimum
 810 element length of the enriched elements. The first case explores moving the
 811 edge of the cracked element onto the crack path, see the mesh in Figure 16B
 812 for $\mathcal{D}_{\text{singularity}}/L = 0.001$. The results are presented by the dotted blue line
 813 in Figures 16C and D, showing excellent agreement for the displacement and
 814 accepted result for the stress against exact results. It is noteworthy that the
 815 results of the stress could be highly improved with the mesh refinement, as
 816 is explored for the next example.

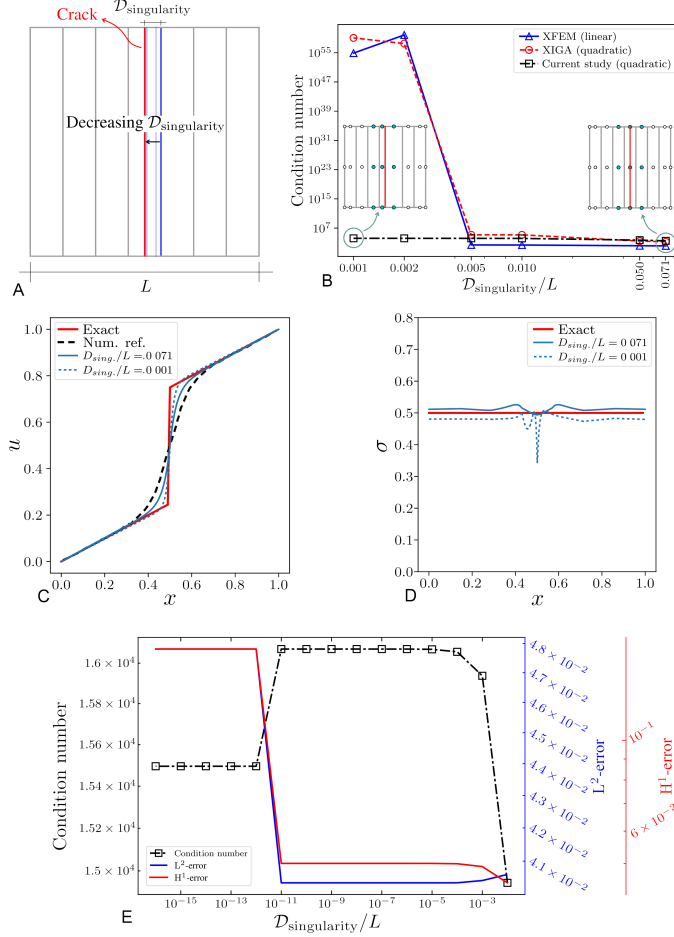


Figure 16: Sensitivity analysis of singularity due to Heaviside enrichment. The blue edge in A gradually approaches the crack and the corresponding condition numbers are plotted for different distances, $\mathcal{D}_{\text{singularity}}$, in B. The results are also compared with XFEM and XIGA. The farthest and closest distances, whose meshes are shown in Figure B, are compared with the exact solution and the numerical reference[39] for the displacement and stress graphs in C and D. L^2 norm and H^1 semi-norm errors are plotted along with their corresponding condition numbers of the current approach in E, in the distance range $[1 \times 10^{-16} \ 1 \times 10^{-2}]$.

817 An inclined mesh is adopted in Figure 17 for the quadratic NURBS (see A
818 and B) to investigate a more complex discretisation example complemented
819 by defining a $\mathcal{C}^0$ -continuous line (see the blue line in A and B) by knot inser-
820 tion [71], while a linear NURBS is also assessed in C. This example provides
821 an additional validation of the enrichment scheme suggested in Section 6.1.1:

control points corresponding to cracked elements are enriched, regardless of the distance to the crack path, which is the only factor in the phase-field variable determination.

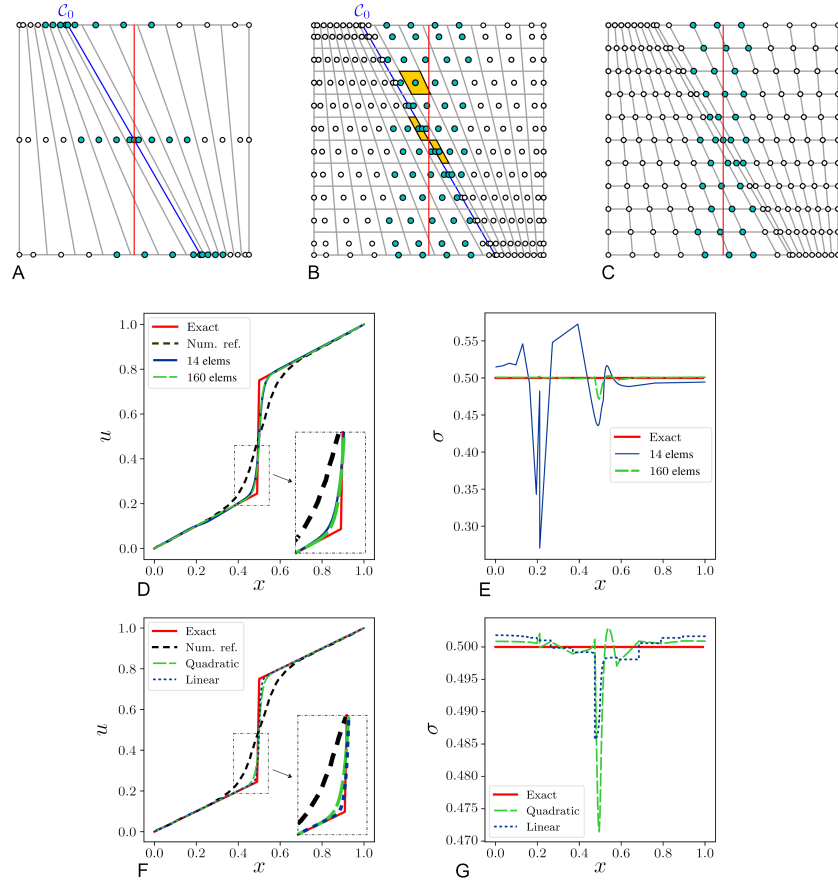


Figure 17: Assessment of phase-field-regularised PUM for irregular meshes shown for (A) 14×1 quadratic, (B) 16×10 quadratic and 16×10 linear NURBS elements. The $\mathcal{C}^0$ -continuous line (see the blue line in A and B) is introduced by means of knot insertion to include all typical features of a general mesh in IGA, while all grids of C are $\mathcal{C}^0$ -continuous due to linear NURBS adopted. The displacement (D) and stress (E) results are compared for mesh refinement of quadratic NURBS shown in A and B. Utilising the same discretisation, 16×10 elements in F (displacement) and G (stress), the effect of NURBS order is assessed for linear (C) and quadratic (B) NURBS. It is noted that the results are plotted for the top edge of the rectangle.

For quadratic NURBS, the results for the displacement in D are almost identical for the meshes proposed in A and B, while there is an obvious

827 difference in the stresses shown in E. As observed in A, use of only 1 element
 828 in the $\mathcal{V}$ -direction causes a non-uniform enrichment spread over the entire
 829 geometry, which is against the localised nature of enrichment. The effect of
 830 mesh refinement in B is twofold: a more uniform mesh is introduced for the
 831 proposed inclined discretisation; a locally distributed enrichment results due
 832 to the smaller support of the refined enriched elements. The stress results
 833 significantly improve for the mesh refinement presented in B. In comparison
 834 with linear NURBS, however, both displacement and stress results improve
 835 when linear NURBS is adopted. Regarding the ill-conditioning of the stiffness
 836 matrix, it is proven that the presence or absence of Kronecker- δ has no effect
 837 on the performance of our proposed formulation, as both $\mathcal{C}^1$ (quadratic) and
 838 $\mathcal{C}^0$ (linear) NURBS have been explored here. Moreover, the satisfactory
 839 results of this example refute any relation between the enrichment scheme
 840 and the phase-field variable d . It is worth noting that a singular matrix
 841 results when the discrete Heaviside formulation, discussed in remark 1 in
 842 Section 3, is utilised.

843 6.2.5. Gauss quadrature sensitivity analysis

844 So far, as shown previously in Figure 8, the number of Gauss points used
 845 for the standard field has been the minimum requirement for full integration
 846 $(p + 1) \times (q + 1)$ using univariate NURBS of orders p and q , while a 9×9
 847 Gauss quadrature has been utilised for the enriched terms to efficiently handle
 848 δ_ℓ , a steep exponential decay function. Therefore, we assess whether the
 849 quality of the enhanced field $\tilde{u}^h$ is majorly dependent on the quantity of
 850 the Gauss points used in the Gauss quadrature. Herein, we examine three
 851 irregular meshes, similar to those utilised in Sections 6.2.3 and 6.2.4, in order
 852 to find the optimum number of Gauss points needed for integration. The
 853 shifted Heaviside formulation given in Equation (61) is only adopted here
 854 with $\ell = \tilde{h}_{\min}/5$. We choose a quadratic bivariate NURBS ($p = q = 2$) for the
 855 mechanical problem and, therefore, a 3×3 Gauss quadrature is adopted for
 856 the standard field. For the phase-field, we exploit the analytical exponential
 857 decay to exclude the possible effect of the errors associated with numerical
 858 phase-field solutions.

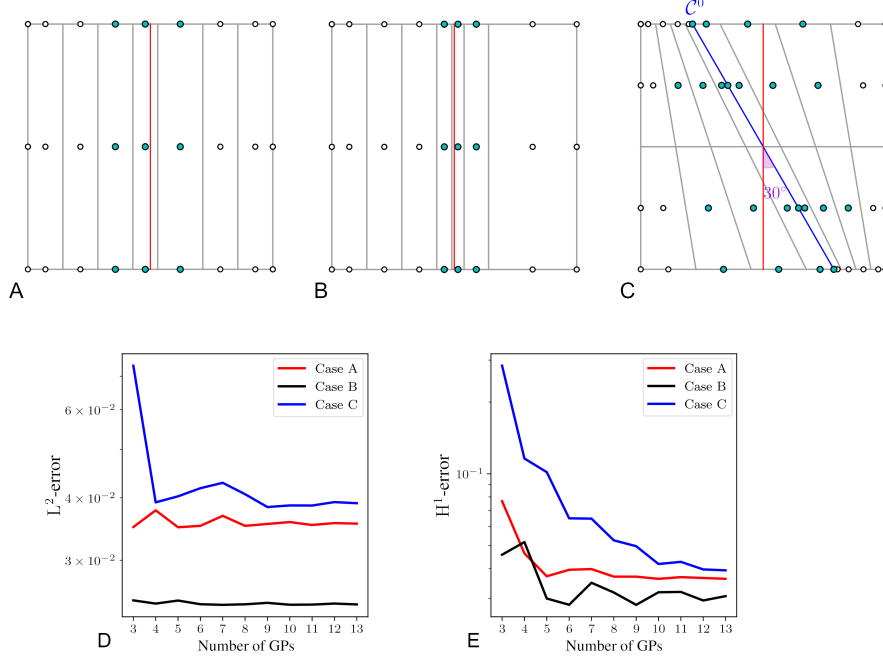


Figure 18: Assessment of the Gauss quadrature for irregular meshes shown for the knot vectors denoted by: $\Xi_x = \{0, 0, 0, 0.1428, 0.2857, 0.4285, 0.53, 0.7142, 0.85714, 1, 1, 1\}$ and $\Xi_y = \{0, 0, 0, 1, 1, 1\}$ in A; $\Xi_x = \{0, 0, 0, 0.1428, 0.2857, 0.4285, 0.49, 0.54, 0.64, 1, 1, 1\}$ and $\Xi_y = \{0, 0, 0, 1, 1, 1\}$ in B; $\Xi_x = \{0, 0, 0, 0.1428, 0.2857, 0.4285, 0.5, 0.5, 0.5714, 0.7142, 0.8571, 1, 1, 1\}$ and $\Xi_y = \{0, 0, 0, 0.5, 1, 1, 1\}$ in C. The L^2 norm and H^1 semi-norm errors are presented in D and E.

859 The results are presented for L^2 norm and H^1 semi-norm errors of the
860 number of Gauss points in Figure 18. The extent of irregularity increases
861 for the discretisations adopted from A to C. The errors in D and E exhibit
862 uniform results for the meshes in A and B. An inclined mesh is examined
863 in C, while a minimum 2 elements is needed in the $\mathcal{Y}$ -direction to prevent
864 the enriched control points from spreading across the entire domain, which
865 can significantly affect the mechanical response, as shown in Section 6.2.4.
866 The results of this case show that a minimum 10 Gauss points are needed
867 to guarantee an almost stable H^1 semi-norm error. The same L^2 norm error
868 exhibits stable errors using a minimum 9 Gauss points used. It is noteworthy
869 that the L^2 norm errors are less restrictive than the H^1 semi-norm errors in
870 terms of the number of Gauss points. In general, it is proved that the number
871 of Gauss points needed are subjective to the quality of discretisation, with

872 local mesh refinement playing the pivotal role (see Figure 18B).

873 6.3. Cohesive fracture: nonlinear traction-separation relationship

874 A nonlinear traction-separation relationship is examined here to better
875 assess the efficacy of the proposed formulation and confirm the conclusions
876 made in previous sections. The geometry and the material properties remain
877 the same as Section 6.2. The energy adopted at the interface reads

$$\mathcal{G}_f = \mathcal{G}_c \left(1 - \left(1 + \frac{\llbracket u_n \rrbracket}{v_n} \right) \exp \left(-\frac{\llbracket u_n \rrbracket}{v_n} \right) \right) \quad (67)$$

878 with $v_n = \mathcal{G}_c / (t_u \exp(1))$. The traction description at the discontinuity yields

$$t_d = \frac{\partial \mathcal{G}_f}{\partial \llbracket u_n \rrbracket} = \mathcal{G}_c \frac{\llbracket u_n \rrbracket}{v_n^2} \exp \left(-\frac{\llbracket u_n \rrbracket}{v_n} \right) \quad (68)$$

879 where the critical fracture energy $\mathcal{G}_c = 1$ and the tensile fracture strength
880 $t_u = 0.75$ are adopted for this example. Dealing with a nonlinear prob-
881 lem, an iterative Newton-Raphson solver is adopted with a non-uniform
882 displacement-controlled increments:

$$\bar{\mathbf{u}}^{\text{new}} = \bar{\mathbf{u}}^{\text{old}} + \Delta \bar{\mathbf{u}}, \quad \begin{cases} \Delta \bar{\mathbf{u}} = 0.066667 & \text{if } \bar{\mathbf{u}}^{\text{old}} \leq 2 \\ \Delta \bar{\mathbf{u}} = 0.15 & \text{if } \bar{\mathbf{u}}^{\text{old}} > 2 \end{cases} \quad (69)$$

883 Based on the previous sections, the optimum length-scale $\ell = \tilde{h}_{\min}/5$ is
884 adopted here for $\mathcal{H}_\ell^{\text{SH(ii)}}$. The analytical phase-field, in the form of the expo-
885 nential decay function, is adopted in this section. The relative energy norm
886 error is calculated by comparing the integral below the force-displacement
887 curve with the energy released by the crack path

$$\text{E-error} = \frac{\left| \int_0^\infty f \, du - \mathcal{G}_c \cdot l_{\text{cr}} \right|}{\mathcal{G}_c \cdot l_{\text{cr}}} \quad (70)$$

888 with l_{cr} denoting the crack length.

889 6.3.1. Irregular discretisation and Gauss points sensitivity analysis

890 Effect of discretisation irregularity is hardly explored for phase-field mod-
891 els in the presence of nonlinear cohesive relationship. Herein, we have adopted
892 two extreme cases of irregular meshes, see Figures 19A and C, for which the
893 mechanical response is presented through a force-displacement curve plotted
894 for various numbers of Gauss points used for the enriched terms. Based on

the results of the linear traction-opening relation, reported in Section 6.2, a minimum number of 10 Gaussian points return satisfactory results. Hence, we have investigated 10, 11 and 12 points for the Gauss quadrature of enriched term, while maintaining the $(p+1) \times (q+1)$ Gauss quadrature for the standard displacement integration.

The force-displacement curves are shown in Figures 19B and D. While both non-uniform (A) and irregular (C) meshes exhibit satisfactory results for all integration points examined, 12 Gauss points successfully capture the behaviour at the peak, particularly for the inclined mesh in D. The errors are reported in Table 1, showing a clear improvement with 12 GPs for the inclined mesh. Regarding the errors of the non-uniform mesh, the poor quality of discretisation, particularly the significant difference in element sizes, cause considerable error after the peak. This is clearly improved by exploiting a uniform discretisation of the same number of elements examined in Figure 20A, which exhibits good agreement with the exact solution in C.

6.3.2. *h-refinement*

To complement the discretisation stability study in Section 6.2, the effect of discretisation is assessed in the presence of nonlinear traction-separation relationship. This example is pivotal in showcasing the role of PUM in compatibility with coarse meshes. Uniform discretisations (7, 11 and 25 elements) are adopted here, see Figure 20. The mechanical responses in the form of the force-displacement curve are shown in C, which are compared with the exact (discrete) solution and the numerical results by Verhoosel and de Borst [39]. The results are presented for 7 and 25 elements using 12 Gauss points showing excellent agreements with the exact solution, which proves the stability of *h*-refinement solution using the proposed length-scale. This is supported by the energy norm errors (Equation (70)) reported in Table 1 as well, which quantifies the errors associated with Figure 20C with values less than 1%. In comparison with other numerical techniques, our proposed formulation is compatible with super coarse discretisations, as the results of the 7-element mesh is closer to the exact results than 800 elements utilised by Verhoosel and de Borst [39]. As discussed in details in Remark 2 (see Section 3), the proposed formulation in Verhoosel and de Borst is a special case of the current formulation in the absence of PUM and the consistent displacement jump definition.

Discretisation	Energy norm error	Gauss quadrature		
		10 GPs	11 GPs	12 GPs
Non-uniform mesh	Relative error (%)	7.7095	8.114	10.482
Inclined mesh	Relative error (%)	4.035	4.090	0.312
Uniform mesh	Relative error (%)	Discretisation		
		7 elems	11 elems	25 elems
		0.826	0.813	0.774

Table 1: Relative energy norm errors for different discretisation cases. Gauss quadrature of the enriched terms is explored by the number of Gauss points (GPs) used for irregular meshes, while mesh refinement is investigated for uniform discretisation.

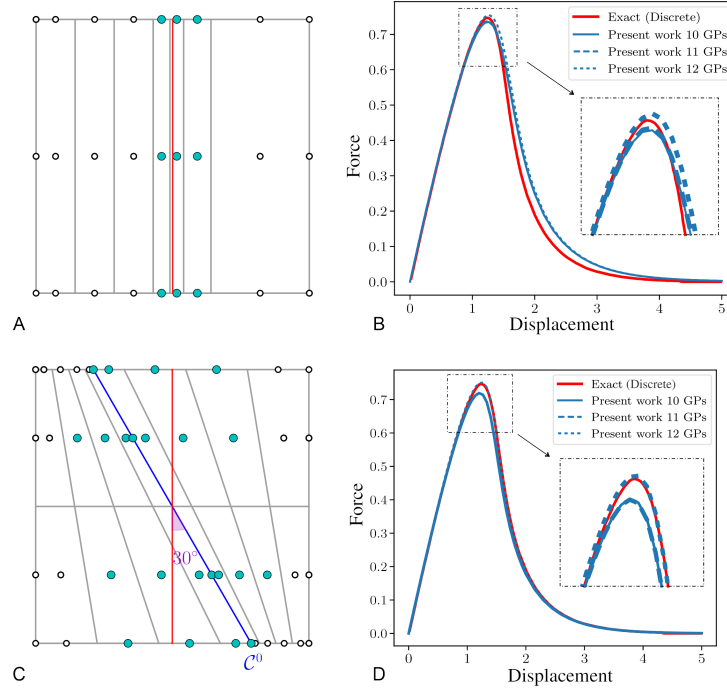


Figure 19: Force-displacement response of irregular discretisation in the presence of cohesive tractions. Two discretisations are investigated: non-uniform mesh $\Xi_x = \{0, 0, 0, 0.14285714, 0.28571429, 0.42857143, 0.49, 0.54, 0.64, 1, 1, 1\}$ and $\Xi_y = \{0, 0, 0, 1, 1, 1\}$ shown in A; inclined mesh $\Xi_x = \{0, 0, 0, 0.14285714, 0.28571429, 0.42857143, 0.5, 0.5, 0.57142857, 0.71428571, 0.85714286, 1, 1, 1\}$ and $\Xi_y = \{0, 0, 0, 0.5, 1, 1, 1\}$ shown in B.

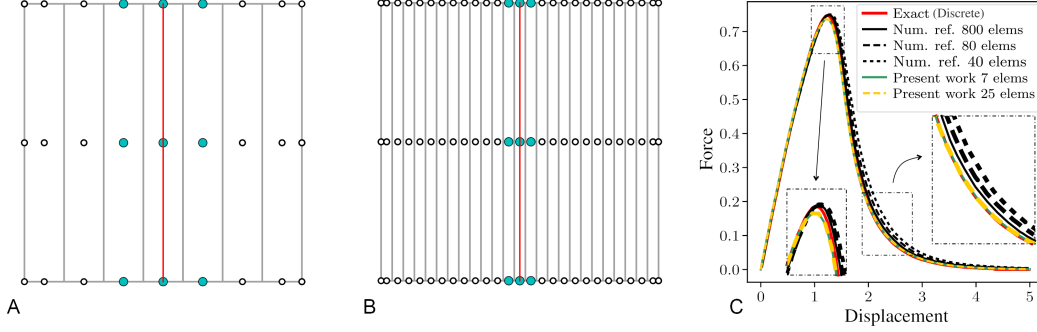


Figure 20: Effect of h -refinement in the presence of non-linear traction-separation relationship. 7 and 25 elements are presented in A and B, while the corresponding force-displacement curves are shown in C, which are compared to those of Verhoosel and de Borst [39] given in black.

6.4. Delamination peel test

Our final example investigates the progressive crack propagation via a peel test, shown in Figure 21, where a traction-free initial slit is considered in a double cantilever beam. The material properties are taken as: the Young's modulus, $E = 100$ MPa; the Poisson's ratio, $\nu = 0.3$; the critical fracture energy, $\mathcal{G}_c = 0.1$ N/mm; and the tensile fracture strength, $t_u = 1$ MPa. The traction-crack opening relationship follows the exponential cohesive-zone model, i.e.,

$$\mathcal{G}_f = -\mathcal{G}_c \exp\left(-\frac{t_u}{\mathcal{G}_c} \llbracket u_n \rrbracket\right), \quad (71)$$

where the cohesive traction relation at the discontinuity yields

$$t_d = \frac{\partial \mathcal{G}_f}{\partial \llbracket u_n \rrbracket} = t_u \exp\left(-\frac{t_u}{\mathcal{G}_c} \llbracket u_n \rrbracket\right). \quad (72)$$

A history parameter κ identical to the one discussed in Section 2.2 replaces the displacement opening $\llbracket u_n \rrbracket$ to ensure irreversibility of fracture opening imposed by the Kuhn-Tucker condition.

As shown in Figure 21A, quadratic NURBS elements are used across different spatial discretisations to support mesh objectivity. The enriched control points are highlighted with green circles. A simple enrichment scheme is adopted for crack propagation: all control points belonging to the cracked elements are enriched regardless of the inter-element share of control points,

947 provided that the enriched terms to be manually removed from the displace-
 948 ment field calculations at all partially-enriched elements in front of the crack
 949 tip. The deformed shape after crack propagation is shown in B and C, illus-
 950 trating displacement and stress fields in the $\mathcal{Y}$ -direction, respectively.

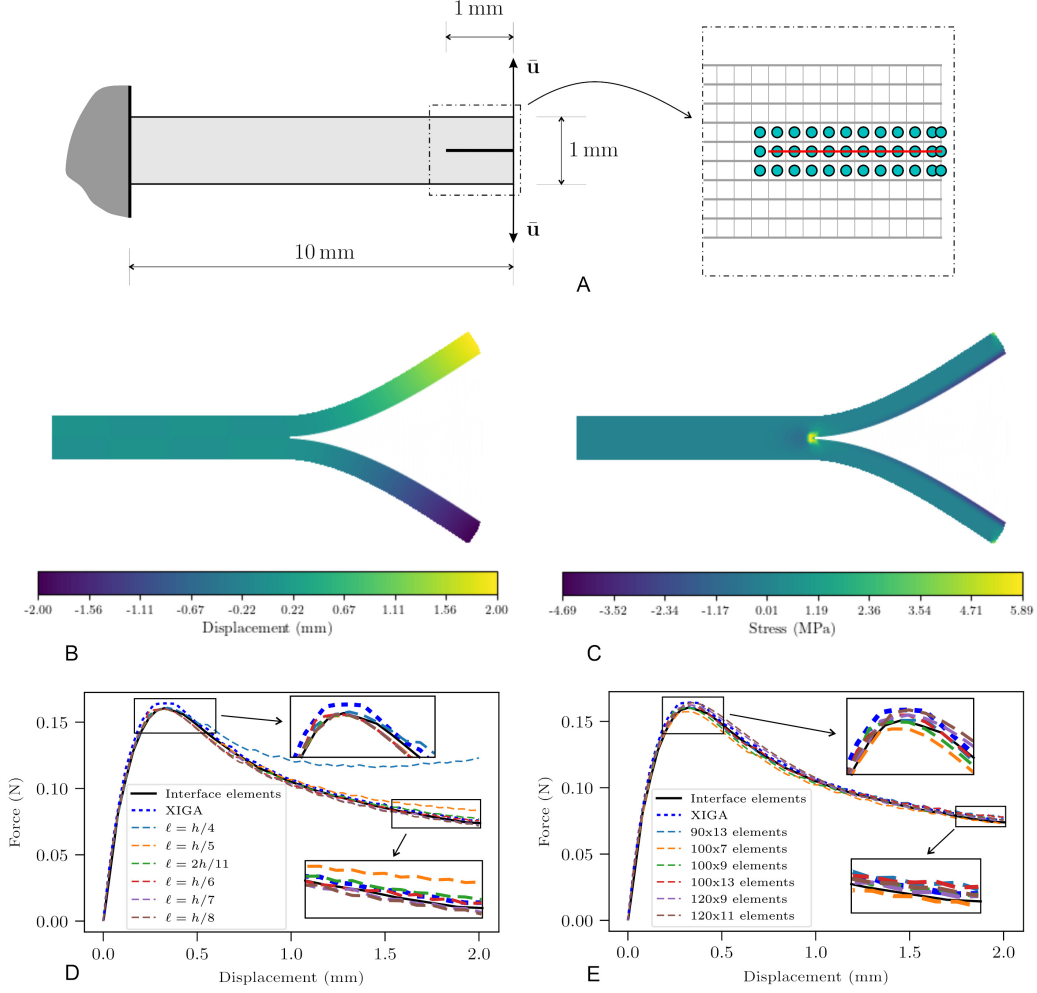


Figure 21: Peel test of (A) a double cantilever beam with an initial slit subjected to prescribed displacement $\bar{u}$. The zoom box illustrates the enrichment scheme utilised (only enriched control points are shown by green circles). The deformed shape is shown in the forms of the (B) displacement and (C) stress fields in the $\mathcal{Y}$ -direction. A parametric study on the internal length scale ℓ is conducted in D, followed by the mesh sensitivity analysis in E. The results are compared with interface elements (90×7) and the XIGA approach (100×9).

951 So far, we have observed that $\ell = h/5$ is a good choice for stationary crack
952 problems explored in the preceding sections. Nevertheless, we have conducted
953 a parametric study for the progressively propagating crack in the peel test, as
954 investigated in Figure 21D for the fixed mesh 100×9 , where the first and the
955 second digits denote the number of elements along the length and the width of
956 the problem, respectively. A converging trend is observed by decreasing the
957 internal length scale ℓ where the factors $\{2/11, 1/6, 1/7, 1/8\}$ render excellent
958 agreement with the results generated by the predefined interface elements
959 and XIGA (see D in Figure 21). This proves that the results are almost
960 insensitive to the length scales below a specific value, a potential alternative
961 to the length scale sensitivity observed in other phase-field contributions.

962 Next, a mesh sensitivity analysis is presented in Figure 21E, while fixing
963 the internal length scale $\ell = h/6$. The results demonstrate good agreement
964 with those of interface elements and XIGA irrespective of the various spatial
965 discretisations examined with minimal effects on the mechanical response.

966 It is worth comparing the computational cost of the present study with
967 that of XIGA [74] and the classical phase field model[39]. The comparison is
968 based on the final configuration of the problem, in which the crack propagates
969 for 34 elements after an initial traction-free slit of 10 elements. Rather than
970 reporting computation time — which is subjective to quality of machine’s
971 computing processor and would requires rerunning all the comparison candi-
972 dates with the same machine — we follow the cost estimation approach used
973 in collocation method [88, 87]. Specifically, we correlate the computational
974 cost with the stiffness matrix assembly effort, measured by the number of
975 integration points.

976 Recalling that the full integration for a bivariate NURBS basis function of
977 order p requires $(p+1) \times (p+1)$, we assume that full integration is employed
978 in [39], even though the exact number of integration points is not reported.
979 Full integration is also used for the standard term in XIGA and our proposed
980 approach. However, the number of points used for the enriched terms differs:
981 XIGA adopts 7×7 integration points, while the current study uses 12×12
982 integration points for the enriched terms. Additionally, XIGA requires a line
983 integration for the enriched crack interface, for which 9 integration points
984 are used.

985 Regarding discretisation, we adopt the minimum number of elements re-
986 ported for this example in [39], namely 50×40 elements, which is comparable
987 to the 100×9 elements used in both XIGA and our proposed approach. XIGA
988 and the current study adopt $\mathcal{C}^1$ -continuous discretisations for the standard

and enriched fields. In contrast, the classical phase field model employed $\mathcal{C}^2$ -continuous elements for the displacement field, and $\mathcal{C}^0$ -continuous elements for both the auxiliary displacement jump and the phase field [39]. It is noted that [39] employed Lagrange polynomials, while the isogeometric version in [41] adopted high-order NURBS with $\mathcal{C}^0$ -continuity at element boundaries, which prevents the typical inter-element share of control points. Moreover, the enriched field $\tilde{u}$ in XIGA and in this study is defined locally, forming a narrow band at the vicinity of the crack path. In contrast, the auxiliary displacement jump used in the classical phase field is defined globally. Consequently, the total number of DOFs in the phase field model reaches 40724, while XIGA and our approach use 2520 DOFs. Further details are provided in Table 2.

Type	Computational cost (DOFs & GPs)				Cost reduction
PF	u	v	d	$\sum_{i=u,v,d} \text{cost}_i$	w.r.t PF (%)
	36542 DOFs	2091 DOFs	2091 DOFs	40724 DOFs	—
	32000 GPs	32000 GPs	8000 GPs	72000 GPs	—
PF-PUM	$\hat{u}$	$\tilde{u}$	d	$\sum_{i=\hat{u},\tilde{u},d} \text{cost}_i$	w.r.t PF (%)
	2244 DOFs	276 DOFs	—	2520 DOFs	93.81
	8100 GPs	32400 GPs	—	40500 GPs	43.75
XIGA	$\hat{u}$	$\tilde{u}$	d	$\sum_{i=\hat{u},\tilde{u},d} \text{cost}_i$	w.r.t PF (%)
	2244 DOFs	276 DOFs	—	2520 DOFs	93.81
	8100 GPs	11331 GPs	—	19431 GPs	73.01

Table 2: Computational cost for stiffness matrix assembly, compared across the phase-field (PF) method [39], the proposed phase-field-regularised partition of unity method (PF-PUM), and extended isogeometric analysis (XIGA) [74]. The comparison is based on the number of integration/Gauss points (GPs) and degrees of freedom (DOFs). In this context, u , v , $\hat{u}$, $\tilde{u}$ and d denote the full displacement, the auxiliary displacement jump, the standard displacement, the enriched displacement and phase fields, respectively.

Table 2 presents the computational cost of each approach, specified in terms of DOFs and number of integration/Gauss points, to provide a fair estimation of the stiffness matrix assembly effort. Notably, the current study and XIGA have the same number of DOFs — representing approximately 94% reduction with respect to that of the phase field model. However, the number of Gauss points reveals a higher computational cost for our proposed approach compared to XIGA. While the phase field model requires 72000 integration points, our approach uses 40500, corresponding to a reduction of

approximately 44%. As expected for a discrete approach, the reduction is even more pronounced for XIGA, around 73%. These results demonstrate that, in addition to the other advantages discussed in this study, our proposed approach is significantly more computationally efficient than the classical phase field model. Particularly, in terms of integration points, the proposed approach corresponds to a computational effort that falls between the classical phase field model and XIGA. Moreover, the computational cost of the present study can potentially approach that of XIGA with improved numerical integration for the local exponential decay function in the enriched displacement field.

7. Concluding remarks

A consistent formulation has been constructed for fracture analysis based on the phase-field-regularised Partition of Unity Method (PUM), in which the Dirac- δ in the form of the exponential decay function is approximated for the crack diffusion. The proposed formulation possesses the advantages of a discrete approach, particularly its efficacy with coarse meshes, which is not achieved with conventional phase-field formulations. Adopting PUM allows for an unambiguous displacement jump definition at the discontinuity based on the existing enriched degrees of freedom, leading to an opening behaviour reflecting the true nature of fracture. This also obviates the need to adopt any auxiliary field to compensate the lack of necessary relation between the jump and displacement fields, as utilised in the customary formulation of the cohesive phase-field models.

Enrichment strategy is an inseparable part of PUM for fracture analysis, and has been explored for our proposed formulation for phase-field regularised PUM. After conducting a comprehensive investigation on the potential relationship between the enrichment scheme and the phase-field variable d , it has been proven that the enrichment identical to that of XIGA renders the most satisfactory results, while under-enrichment leads to a wrong displacement jump at the crack, leading to incorrect displacement and stress evaluations. Over-enrichment, on the other hand, leads to acceptable displacement results. However, oscillatory stress estimates are directly proportional to the extent of over-enrichment. For the progressively fracturing cracks, we have adopted the simplest approach by enriching the control points of the cracked elements entirely, regardless of the inter-element share between the intact and cracked elements. However, the enriched terms must be removed from

the formulations when considering the intact elements in front of the crack tip.

In a comparison between the analytical evaluation of the diffused Dirac δ and the numerical solution of the phase-field problem, quadratic NURBS ($\mathcal{C}^1$ -continuous at element boundaries) renders the minimum error for satisfying the identity $\int_{\Omega} \delta_{\ell} d\Omega = \Gamma_d$. For the coupled phase-field and mechanical problems, however, different combinations have been explored. The best solution pertains to cubic NURBS ($\mathcal{C}^2$ -continuous at element boundaries) for the mechanical field and quadratic for the phase-field problem. Nevertheless, significant stress oscillations are observed for the phase-field numerical approach in general, while the analytical exponential decay function exhibits satisfactory results, provided that enough Gauss points are utilised for the regularised enriched terms. It has been observed that using 12 Gauss points guarantees proper integration of the discontinuous enriched field, while maintaining the full integration for the intact standard terms $((p+1) \times (q+1))$ Gauss points for the NURBS surface of orders $p \otimes q$ suffices.

The optimum length-scale has been assessed against h-refinement in the presence of linear and nonlinear traction-separation relationships at the discontinuity. In the examples tested in this article, the minimum length scale to achieve converging results was found to be in the range $h/6 \leq \ell < h/4$. If length scales smaller than the identified thresholds are adopted, the results do not significantly depend on the length scale. When an adequate length scale is adopted, the results are also shown to be insensitive to mesh refinement.

Next, a comprehensive study has been conducted on the conditioning issue associated with extended finite element analysis. Our formulation has successfully resolved the issue by the continuous description of the Heaviside function generated by the diffused representation of the crack. Irregular discretisation has also been investigated for biased mesh layouts to further prove the efficacy of our formulations. Finally, the approach has been benchmarked against crack propagation, indicating excellent agreement with XIGA and interface elements results.

Regarding arbitrary crack paths, our proposed approach offers no improvement over the classical extended finite element analysis, as it borrows the same crack initiation and propagation strategies used in the extended finite element analysis — through the level-set technique, maximum principal stress, and enrichment. By adopting a degradation function, however, the crack initiation and tracking would benefit from automated crack path

1083 monitoring via a phase-field-dependent level-set function. This enhancement
 1084 will be the focus of our future work.

1085 This work advances the numerical analysis of fracture mechanics, present-
 1086 ing a formulation that delivers robust predictions with relatively low com-
 1087 putational costs. Furthermore, it opens up opportunities for future develop-
 1088 ments and applications. Notably, the current formulation can be extended to
 1089 accommodate conventional damage-dependent degradation functions, lead-
 1090 ing to formulations capable of reproducing a wide range of material behaviors
 1091 and fracture patterns.

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1101 **Appendix A. Enriched shape-function vector and strain-displacement** 1102 **matrix**

1103 For simplicity, we define the following notation for the enriched strain-
 1104 displacement and basis function matrices

$$\mathbf{B}_{u_i}^{\text{enr}} = \mathcal{H}_{\Gamma_d}(x_n)(1 - 2\ell\delta_\ell)\mathbf{B}_{u_i} = \mathcal{H}_{\Gamma_d}(x_n)(1 - 2\ell\delta_\ell) \begin{bmatrix} \frac{\partial N_{u_i}}{\partial X_1} & 0 \\ 0 & \frac{\partial N_{u_i}}{\partial X_2} \\ \frac{\partial N_{u_i}}{\partial X_2} & \frac{\partial N_{u_i}}{\partial X_1} \end{bmatrix} \quad \forall i \in \mathcal{I}^{\text{enr}} \quad (\text{A.1a})$$

$$\mathbf{N}_{u_i}^{\text{enr}} = \mathcal{H}_{\Gamma_d}(x_n)(1 - 2\ell\delta_\ell)\mathbf{N}_{u_i} = \mathcal{H}_{\Gamma_d}(x_n)(1 - 2\ell\delta_\ell) \begin{bmatrix} N_{u_i} & 0 \\ 0 & N_{u_i} \end{bmatrix} \quad \forall i \in \mathcal{I}^{\text{enr}} \quad (\text{A.1b})$$

1107 where x_n denotes the normal signed distance of the point $\mathbf{x}$ from the crack
 1108 path. The formulation for the shifted Heaviside function, *i.e.*, $\mathcal{H}_\ell^{\text{SH(ii)}}$ is
 1109 given next

$$\mathbf{B}_{u_i}^{\text{enr(ii)}} = \left(\mathcal{H}_{\Gamma_d}(x_n) (1 - 2\ell \delta_\ell(x_n)) - \mathcal{H}_{\Gamma_d}(x_n^i) (1 - 2\ell \delta_\ell(x_n^i)) \right) \begin{bmatrix} \frac{\partial N_{u_i}}{\partial X_1} & 0 \\ 0 & \frac{\partial N_{u_i}}{\partial X_2} \\ \frac{\partial N_{u_i}}{\partial X_2} & \frac{\partial N_{u_i}}{\partial X_1} \end{bmatrix} \quad \forall i \in \mathcal{I}^{\text{enr}} \quad (\text{A.2a})$$

1110

$$\mathbf{N}_{u_i}^{\text{enr(ii)}} = \left(\mathcal{H}_{\Gamma_d}(x_n) (1 - 2\ell \delta_\ell(x_n)) - \mathcal{H}_{\Gamma_d}(x_n^i) (1 - 2\ell \delta_\ell(x_n^i)) \right) \begin{bmatrix} N_{u_i} & 0 \\ 0 & N_{u_i} \end{bmatrix} \quad \forall i \in \mathcal{I}^{\text{enr}}. \quad (\text{A.2b})$$

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