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**Article:**

Aende, A, Gardy, J [orcid.org/0000-0003-1806-4056](https://orcid.org/0000-0003-1806-4056), Aslam, Z et al. (5 more authors) (2022) A novel highly osmotic K/Fe<sub>3</sub>O<sub>4</sub>/CNF magnetic draw solution for salty water desalination. *Desalination*, 538. 115903. ISSN 0011-9164

<https://doi.org/10.1016/j.desal.2022.115903>

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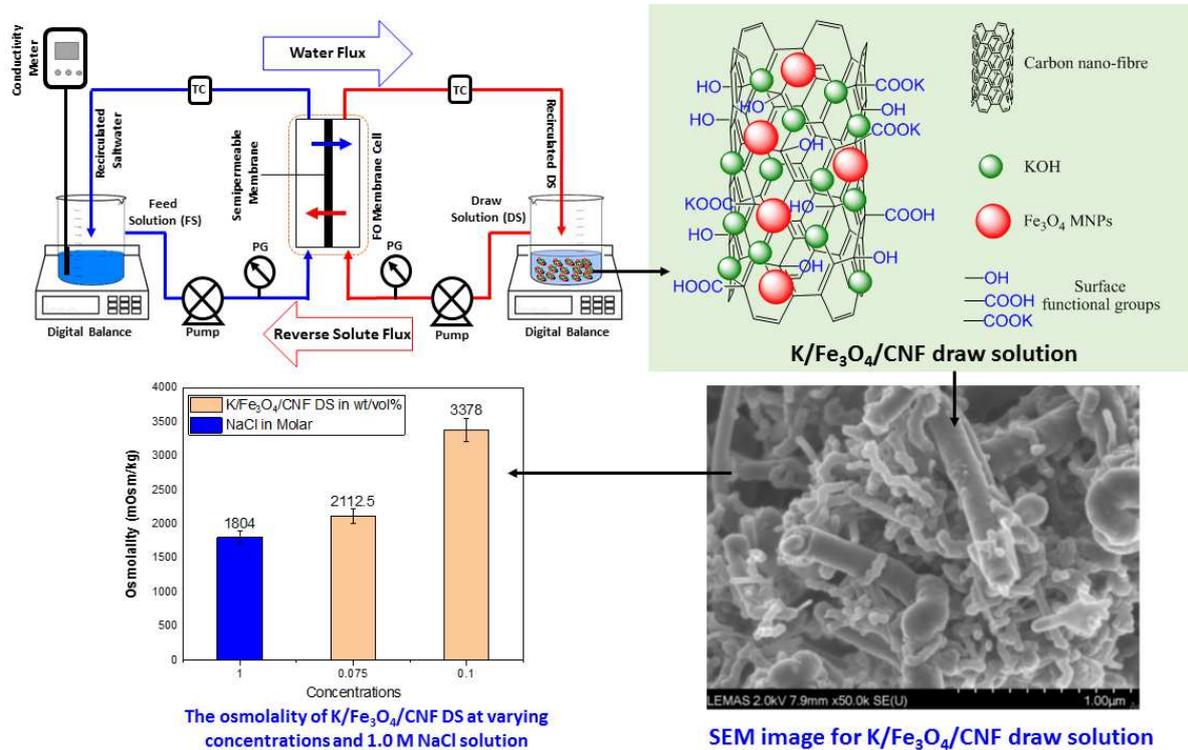
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1 **Graphical abstract:**

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6 **Highlights:**

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- MNP DS appraisal against high salinity feed streams is more rational.
- The DS preparation protocol enhanced MNP hydrophilicity without polymers.
- K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS produced high osmotic pressure (86.1 bar) and water flux (24 LMH).
- K-species considerably improved the DS hydrophilicity.
- DS material feasibly recoverable via magnetic field and solar-thermal separability.

14 **A novel highly osmotic K/Fe<sub>3</sub>O<sub>4</sub>/CNF magnetic draw solution for salty water**  
15 **desalination**

16  
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27 **Abstract:**

28 Forward osmosis (FO) is increasingly being studied as an alternative desalination technique to  
29 the other conventional desalination technologies, owing primarily to its low energy potential.  
30 However, the co-opted energy limitations in the draw solution (DS) regeneration stage in FO  
31 desalination processes and the lack of effective DS have hampered FO's implementation for  
32 potable water application on an industrial scale. In this work, we explored the Donnan principle  
33 to engineer a DS material having the duality of magnetic and solar-thermal separability  
34 functionalities from a sustainability viewpoint whilst exploiting a careful selection of material  
35 properties. A novel potassium functionalised iron oxide doped carbon nanofibres  
36 (K/Fe<sub>3</sub>O<sub>4</sub>/CNF) magnetic nanoparticles (MNPs) was successfully synthesised for FO  
37 desalination applications, utilising an eco-friendly strategy that improves hydrophilicity of DS  
38 without polymers. The novel DS obtained a significant osmotic pressure (86.1 bar), while its  
39 FO performance showed a small reverse salt flux (RSF) and specific reverse salt flux (SRSF)  
40 values of 0.56 gMH and 0.02 g/L, respectively. These values compare to at least less than 10%  
41 of most RSF and SRSF values reported in the literature. The facile DS synthesis strategy  
42 adopted herein will potentially open a new route to preparing other DS nanomaterials with  
43 unique multi functionalities and enhanced hydrophilicity devoid of polymers. Whilst magnetic  
44 DS re-concentration may be achievable, additional research is required to appraise this

45 methodology's overall energy implications and economic advantages over existing DS  
46 recovery methods.

47

48 **Keywords:** Water desalination; Forward osmosis; Magnetic nanoparticles; Draw solutions;  
49 Osmotic pressure; Carbon nanofibres.

50

## 51 **1. Introduction**

52 In response to addressing the global challenge of freshwater insecurity, desalination of  
53 saltwater as a means of increasing global freshwater supplies via technology has been advanced  
54 amongst other solutions due to the preponderance of saltwater on the earth's surface [1].  
55 Notable technologies, including multi-stage filtration (MSF), multi-effects distillation (MED),  
56 vapour compression (VP), reverse osmosis (RO), electrodialysis, etc., have been exploited [2,  
57 3]. However, it is mostly held that desalination as a practice is energy-intensive [4]. Forward  
58 osmosis (FO), a natural process, has gained tremendous research interest in recent years as an  
59 alternative desalination technique [2]. This is because the FO process is often considered as  
60 having a low fouling propensity relative to pressurised membrane processes like reverse  
61 osmosis (RO), good salt rejection ability, and, more importantly, its supposedly low energy  
62 potentialities [2, 5]. The FO process is propelled by the concentration gradient (osmotic  
63 pressure difference) across a semipermeable membrane between two solutions. As an emerging  
64 desalination technique, FO has demonstrated some tried-and-true abilities to desalinate  
65 saltwater [5] and is deployable in treating highly vitiated source waters [6, 7]. However, FO's  
66 supposedly low energy potentials as a desalination technique are seriously challenged during  
67 DS recovery and regeneration. The energy involved at this stage often exceeds the minimum  
68 thermodynamic energy of separation for seawater ( $1.56 \text{ kWh/m}^3$ ) at 50% recovery [8]. This is  
69 because more than one process or a secondary technique (s) is required at the DS regeneration  
70 stage to obtain pure water successfully. At this point, any attendant energy deficiencies of these

71 secondary techniques get arrogated to the overall FO process. Over the years, different  
72 categories of DS materials, including organic salts, inorganic salts, volatile compounds,  
73 stimuli-responsive materials, polyelectrolytes, and carbon-based materials, have been  
74 exploited across various research studies for potential application in FO desalination systems  
75 [9]. However, most of these materials, such as organic and inorganic salts, volatile compounds,  
76 and polyelectrolytes, suffer from severe reverse salt flux, occasioning concentration  
77 polarisation (CP) effects. [2] Consequently, the osmotic pressure difference across the  
78 membrane and, by extension, water flux is negatively influenced. The potential cost implication  
79 of replenishing the lost DS materials is considered expensive [2, 9]. Furthermore, the  
80 requirement of membrane-based downstream processes such as RO, membrane distillation  
81 (MD), nanofiltration (NF), and ultrafiltration (UF) to achieve DS re-concentration adversely  
82 impacts the FO system's energy efficiency, which is a downside [2]. On the other hand, low  
83 water recovery rates associated with stimuli-responsive DS materials render them unsuitable  
84 for industrial applications. Therefore, recent research efforts to engineer new and improved DS  
85 materials, particularly those that mitigate the energy limitation imposed during the DS recovery  
86 stage in FO desalination systems, led to considering the application of MNPs as DS materials  
87 [9]. Magnetic separation involving a magnetic field is gaining considerable research attention  
88 as a potentially viable DS regeneration mechanism that could address the energy deficiencies  
89 in FO desalination imposed by the secondary recovery processes [9]. This is primarily due to  
90 the potentially low energy and convenience of applying this methodology, particularly the  
91 ability of MNPs to be separated by low-cost magnets [10, 11]. Nanoparticles (NPs) of iron  
92 oxide preponderate studies for applications as MNPs DS in FO desalination research studies  
93 [9]. Principal amongst the increased focus on iron oxides MNPs as DS is their chemical  
94 stability, biological compatibility, insignificant toxicity, and the fabrication of  $\text{Fe}_3\text{O}_4$  and  $\text{Fe}_2\text{O}_3$   
95 into NPs facilely [12]. Several studies have fabricated a host of surface functionalised MNPs

96 with polymers as draw agents for applications in FO systems [11, 13-20]. However, despite  
97 some appreciation in the study of MNPs as DS, particularly in desalination research, a key  
98 challenge remains the somewhat inability of most of these materials to develop considerable  
99 osmotic pressure enough to facilitate water transfer across a semipermeable membrane. For  
100 example, Ling et al. [21] studied a FO system combining a low-intensity magnetic field to  
101 facilitate magnetic separation of DS and mitigate agglomeration, The poly(N-isopropyl  
102 acrylamide) and tri-ethylene glycol functionalised MNPs (PNIPAM/TRI) as the representative  
103 DS in this study generated a very low water flux (< 2LMH) after five cycles which is a  
104 downside. Moreover, the DS fabrication methodology is seemingly complicated, which can  
105 constitute cost increases [9]. Additionally, according to research, magnetic fields are not always  
106 able to entirely capture MNPs, particularly those of decreasing particle sizes [22]. As a result,  
107 ultrafiltration is considered a suitable post-treatment to concentrate MNPs draw solutes because  
108 its pore size may be small enough to capture MNPs without the attendant effects presumably  
109 of using NF and RO, the commonly utilised techniques to intercept salts and organic chemicals  
110 [23]. However, based on research in the literature, it is established that regardless of any DS  
111 type and any pressure-driven recovery technique employed, the energy consumption of any  
112 such hybrid FO system is effectively indistinguishable from any other [24]. In addition, another  
113 investigation revealed that MNPs of smaller sizes than a UF membrane pores uncaptured by a  
114 magnetic field reconcentration could permeate through the membrane [25]. This can pose a  
115 hindrance to industrial application due to potential health implications, thus necessitating  
116 further research to optimise this deficiency. Concentrating the uncaptured MNPs via solar-  
117 thermal evaporation will offer sustainability. Furthermore, whilst polymer compounds confer  
118 the needed hydrophilicity in MNPs, some polymers can be hazardous in water, even in trace  
119 amounts [9]. Consequently, the complicity of separating these polymers from water often  
120 using processes like RO undermines the FO's main beneficence. Therefore, from an

121 environmental sustainability point of view it will be well-thought to explore sustainable means  
122 of inducing hydrophilicity in MNPs [26]. Many investigations using MNPs as DS in FO  
123 desalination use DI water as the FS [11, 13-20] There is insufficient research on MNPs DS  
124 against saline FS since few studies use FS in saltwater environments [15, 26-28], thus, creating  
125 a gap in understanding how these materials operate in veritable saltwater settings. Testing these  
126 materials' osmotic action in saltwater conditions would be more well-advised.  
127 Therefore, this work aims to feasibly explore, from a sustainability viewpoint, approaches that  
128 can facilitate the enhanced application of MNPs as DS in effectuating the FO process. To meet  
129 our objective, we attempt to exploit the tenets of the Donnan mechanism to synthesise  
130 engineered DS materials with dual functionality of magnetic and solar-thermal separability/re-  
131 concentration to facilitate improved membrane water flux. Specifically, the work will: (i)  
132 exploit a blend of unique material properties for draw solutes fabrication to foster high osmotic  
133 pressure, improved water flux and facilitate DS re-concentration via a magnetic field and solar  
134 irradiation (ii) conduct the material characterisation and osmotic performance evaluation of the  
135 as-fabricated draw solutes relative to modelled saltwater solution as feed. The performance  
136 would be examined relative to parameters including osmotic pressure, water flux, reverse salt  
137 flux, and specific reverse salt flux. Therefore, this work presents a novel K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs  
138 as DS for application in FO desalination operations. The novel DS not only exhibits high  
139 osmolality and, by extension, high osmotic pressure that can be tested against feed streams  
140 with high salinity but is also feasibly recoverable via a magnetic field and solar energy.

141

## 142 **2. Materials and methods**

### 143 **2.1. Design strategy and material consideration**

144 The strategy is to exploit a unique blend of material properties to effectuate the Donnan effect  
145 in the FO process in anticipation that when in aqueous media, the DS material will generate

146 significant charges to repel the NaCl ions' activity around the membrane's active layer.  
147 Consequently, this will allow water as the primary eluent in the FS to permeate with ease,  
148 increasing the membrane water flux. We premised the material choice on the following  
149 consideration:

- 150 1. Carbon in CNF, because of its outstanding solar-thermal properties coupled with its  
151 unique physicochemical properties, allows it to be multi-functionalized by covalent or  
152 non-covalent modifications or a combination to achieve multiple vital functionalities.
- 153 2. The strong hydrophilic disposition of KOH, the excellent water solubility offered by  
154 the high hydration energy of the K<sup>+</sup> ions, and the ability of K species to reversibly  
155 intercalate
- 156 3. Along with its strong magnetism compared to other MNPs, iron oxide MNPs offer a  
157 high reactive surface that readily fosters modification with biocompatible coatings. In  
158 addition, it can be controlled and easily separated by a magnetic field for further and  
159 subsequent applicable reuse.

160 Other advantages, including low cytotoxicity, natural abundance, and low cost, contribute to  
161 our choice of these materials.

162

### 163 **2.1. Synthesis of iron oxide (Fe<sub>3</sub>O<sub>4</sub>) nanoparticles**

164 Fe<sub>3</sub>O<sub>4</sub> NPs were synthesised via the co-precipitation method [29] as can be seen in the reaction  
165 below:



167 Briefly, an ultrasonic probe was used to separately disperse 0.2 mol of FeCl<sub>2</sub>·4H<sub>2</sub>O (≥99.99%,  
168 Sigma-Aldrich) and 0.68 mol of FeCl<sub>3</sub>·6H<sub>2</sub>O (≥98%, Sigma-Aldrich) in a 25 mL of a 1:1 vol%  
169 ethanol (Sigma-Aldrich) to deionised water solution. The obtained clear resultant solutions  
170 were then transferred into a 250 mL round-bottomed flask and subsequently subjected to

171 heating at 80 °C under the flow of N<sub>2</sub> atmosphere and continuous stirring (250 rpm) for 6 h.  
172 Whilst the reaction mixture was under heating, 10 mL of ammonia solution (28-30 vol%,  
173 NH<sub>4</sub>OH, Sigma Aldrich) was simultaneously added dropwise to maintain the pH of the  
174 solutions mixture at 12. The formed iron oxide NPs were then aged for 24 h at room  
175 temperature before being isolated utilising an external magnetic field. The as-obtained NPs  
176 (dark reddish particles) were then washed several times with 1:1 vol% aqueous ethanol solution  
177 until no traces of chloride ions were detected using AgNO<sub>3</sub> solution as an indicator. The  
178 prepared magnetic NPs were then oven-dried overnight at 120 °C and calcined at 550 °C for 3  
179 h to obtain Fe<sub>3</sub>O<sub>4</sub> MNPs.

180

## 181 **2.2. Synthesis of K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs**

182 An 80 mL aqueous solution of a molar ratio of 1:3 of nitric acid (HNO<sub>3</sub>) to sulfuric acid  
183 (H<sub>2</sub>SO<sub>4</sub>) was carefully transferred into a round bottom flask. 5 g of carbon nano-fibres (CNF)  
184 powder [graphitised (iron-free) compressed of conical platelets, Aldrich Chemistry, USA] was  
185 then slowly added into the acid mixture and then refluxed at 70 °C for 6 h under continuous  
186 stirring at 250 rpm. The solution mixture was then filtered and repeatedly washed with  
187 deionised (DI) water and dried overnight at 80 °C using a vacuum oven. The as-treated CNF  
188 powder was then dispersed in a 50 mL aqueous solution of 0.05 M sodium dodecyl (SDS) using  
189 a probe sonicator for 30 min. The dispersed Fe<sub>3</sub>O<sub>4</sub> MNPs and as-treated CNF were then  
190 transferred into a round-bottomed flask and put under reflux at 80 °C for 4 h whilst stirring  
191 continuously at 250 rpm. Following this, a 250 mL aqueous solution of 0.5 M KOH (Sigma-  
192 Aldrich) was then slowly added dropwise into the above mixture whilst stirring continuously  
193 under reflux for another 4 h. The resultant mixture is then oven-dried at 60 °C overnight and  
194 washed several times using deionised water to obtain the final product labelled as  
195 K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs. Consequently, 1.0 wt/vol% stock solution was then prepared from the

196 prepared above DS by dispersing 1.0 g of K/Fe<sub>3</sub>O<sub>4</sub>/CNF into 100 mL of DI water, after which  
197 7.5 mL (0.075 wt/vol%) and 10 mL (0.1 wt/vol%) respectively from the stock solution were  
198 subsequently measured and set aside for FO performance evaluations.

199

### 200 **3. Characterisation techniques**

201 Analysis of the structural and crystal phase of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs was performed using  
202 the Bruker D8 X-ray diffraction (XRD) with Cu K $\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ). The instrument  
203 was operated and maintained at a voltage of 40 kV and an applied current of 40 mA,  
204 respectively. The sample was scanned at a  $2\theta$  angle range from 10 to 70° with step size 0.0495°  
205 at 1 sec per step.

206 The elemental composition, surface morphologies and location of elements for the  
207 K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs sample were examined using a cold field-emission scanning electron  
208 microscopy (Hitachi SU8230 CFE-SEM). Also utilised was a transmission electron  
209 microscopy (FEI Titan3 Themis 300) coupled with the Super-X EDX system with windowless  
210 4-detector design and the Gatan OneView 16 Megapixel CMOS digital camera for collecting  
211 bright-field TEM images. A droplet of dispersed K/Fe<sub>3</sub>O<sub>4</sub>/CNF in acetone was placed over a  
212 carbon-coated copper grid for examinations using a micropipette.

213 A Thermo Scientific™ Nicolet™ iS10 Fourier transform infrared (FTIR) spectrometer using  
214 the attenuated total reflection (ATR) method was employed at room temperature to examine  
215 the functional groups present in the synthesised K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs, in the range of 525 –  
216 4000 cm<sup>-1</sup> at a resolution of 4 cm<sup>-1</sup>. The total output was in the percentage of transmittance.

217 A thermo-gravimetric analysis (TGA) instrument (Mettler Toledo, TGA/DSC-3) was used to  
218 investigate the thermal stability of the prepared K/Fe<sub>3</sub>O<sub>4</sub>/CNF sample. The TGA analysis was  
219 performed under a constant flow rate (50 mL/min) of nitrogen purge gas and temperature range  
220 of 30 to 900 °C and at a 10 °C/min heating rate.

221 The zeta potential of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs was investigated using the Zetasizer (Malvern  
222 Instruments, UK) to ascertain their charge stability in colloidal solutions. A 1 mL syringe was  
223 used to fill an aqueous K/Fe<sub>3</sub>O<sub>4</sub>/CNF sample into a folded capillary cuvette (Malvern DTS  
224 1070), and the cuvette was then appropriately inserted into the instrument. Consequently, the  
225 device is set to run after all the appropriate measuring conditions have been imputed, whilst  
226 the measurements are collected in the software window.

227 The 3250 model osmometer (Advanced Instruments, USA), which measures samples  
228 osmolality based on the freezing point depression principle, was deployed to measure the  
229 osmolality and, by extension, the osmotic pressure of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF.

230 The magnetic characteristics of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF sample powder were investigated using a  
231 superconducting quantum interference device (SQUID) vibrating sample magnetometer  
232 (VSM). The magnetisation (M) of the powder was measured at room temperature (300 K)  
233 between -1.5 and 1.5 T.

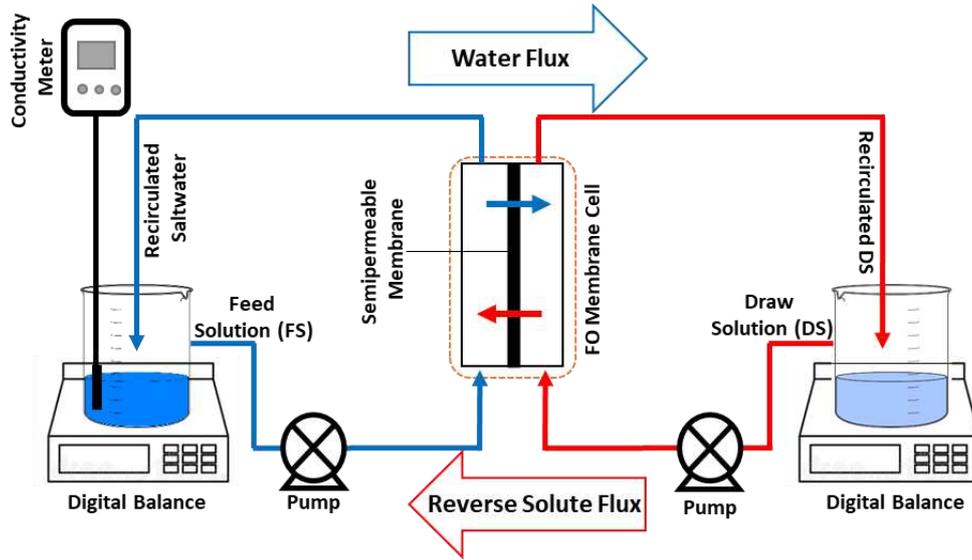
234

#### 235 **2.4. Forward osmosis setup**

236 A lab-scale cross-flow membrane cell filtration unit (CFO42D) purchased from Sterlitech  
237 Corporation Inc, USA, having an active membrane area of 42 cm<sup>2</sup> was assembled and used in  
238 performing the FO experiment. The CF042D cross-flow cell unit comprises tightening screws,  
239 top and bottom plates, cell top and cell bottom, sintered metal support, membrane filter, and  
240 O-rings. Whilst assembling the CF042D cell unit, it was ensured that the O-rings were wet with  
241 water before being firmly fitted flat into the groove on the cell bottom. Ensuring that the O-  
242 rings lie flat into the grooves helps avoid damage to the O-rings and leakages that may occur  
243 during system operation if the O-rings are not correctly installed.

244 The FO performance of the novel K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS was conducted in the FO mode with the  
245 dense active layer (smooth side) of the commercial FTSH2O<sup>TM</sup> flat sheet CTA membrane

246 always placed facing the FS side and the support layer (rough side) always facing the DS side.  
 247 The FS and DS flow volumes were moderated utilising two Masterflex Console Gear Pumps  
 248 (Cole Palmer, USA) at a steady flow rate of 1280 mL/min. The overall schematic of the setup  
 249 is shown in Fig. 1.



250

251 **Fig. 1.** The overall schematic of the FO setup.

252 The water fluxes across the membrane were evaluated by measuring the weight change  
 253 occurring on the feed tank containing a given volume of modelled saltwater (1.0 M NaCl  
 254 solution) every 5 min using an electronic balance (A&D EX300i, USA) over a 60 min period  
 255 per run. Readings of the first five minutes were not considered as the system was allowed to  
 256 equilibrate the flow fluctuations occasioned at the start of the experimental runs. The  
 257 membrane water flux is given by:

$$258 \quad J_w = \frac{\Delta m}{\rho_f A \Delta t} = \frac{\Delta V}{A \Delta t} \quad \text{Eqn. (1)}$$

259 Where  $J_w$  is the permeate flux in L/m<sup>2</sup>/h (LMH),  $\Delta m$  is the change in mass of the FS over time,  
 260  $\rho_f$  is the density of water,  $\Delta t$  is the change in time (h),  $\Delta V$  is the volumetric change (L) of the  
 261 FS, and  $A$  is the active membrane area (m<sup>2</sup>) of the CTA membrane. Over the course of the FO  
 262 runs, the FS's electrical conductivity (EC) measurements were obtained using a calibrated  
 263 benchtop conductivity meter (VWR symphony™) every 5 min interval. The measured

264 conductivities (mS/cm) were then converted to concentrations [30] to evaluate reverse salt flux  
265 (RSF). The RSF was evaluated from the equation:

$$266 \quad J_s = \frac{(C_t V_t - C_o V_o)}{A \Delta t} \quad \text{Eqn. (2)}$$

267 Where  $J_s$  is the reverse salt flux ( $\text{g/m}^2/\text{h}$ ), also expressed as  $\text{gMH}$ ,  $C_o$  and  $C_t$  are FS  
268 concentrations at the start of the experiment and  $\Delta t$  respectively,  $V_o$  and  $V_t$  are the FS volumes  
269 (L) at the beginning and  $\Delta t$ , respectively [31].

270

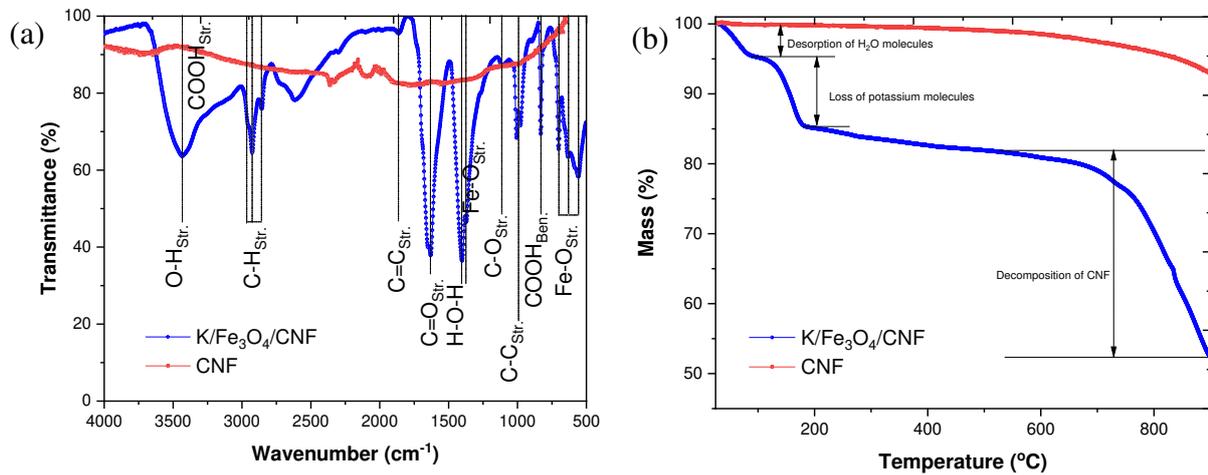
### 271 **3. Results and discussions**

#### 272 **3.1. Characterisation of K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs**

273 The particle size of the CNF and the fabricated K/Fe<sub>3</sub>O<sub>4</sub>/CNF was analysed utilising dynamic  
274 light scattering (DLS). The average size of the CNF was 5.0  $\mu\text{m}$ . After the doping reaction of  
275 the CNF with potassium functionalised iron (III) oxide, the average particle diameter of the  
276 K/Fe<sub>3</sub>O<sub>4</sub>/CNF material was 4.5  $\mu\text{m}$ , demonstrating the bonding of the K species and Fe<sub>3</sub>O<sub>4</sub> on  
277 the CNF.

278 The obtained FTIR spectra for CNF and K/Fe<sub>3</sub>O<sub>4</sub>/CNF samples is shown in Fig. 2 (a). The  
279 broad peak at  $\sim 3434 \text{ cm}^{-1}$  corresponds to the O-H stretching vibrations of the -C-OH and -  
280 COOH functional groups [32, 33]. The peaks at  $\sim 2925$  and  $\sim 2855 \text{ cm}^{-1}$  correspond to the  
281 asymmetric and symmetric stretching in saturated aliphatic groups, respectively [32]. The  
282 broad peak at  $\sim 2615 \text{ cm}^{-1}$  could be due to the O-C-O group, whilst a weak peak that followed  
283 at  $\sim 1950 \text{ cm}^{-1}$  is related to C=C stretching vibrations [34]. A sharp peak at  $\sim 1629 \text{ cm}^{-1}$   
284 represents the presence of free H-O-H molecules absorbed on the sample's surface [31, 34].  
285 Another sharp peak that follows at  $\sim 1402 \text{ cm}^{-1}$  is related to O-H bending in carboxylic acid and  
286 alcohol [32, 34]. A weak characteristic peak at  $\sim 1106 \text{ cm}^{-1}$  could be attributed to a C-O  
287 stretching vibration in aliphatic ether and secondary alcohols [32]. Two other bands at  $\sim 1005$   
288 and  $\sim 979 \text{ cm}^{-1}$  are assigned to C-C vibrations in saturated aliphatic [32]. Subsequently, the

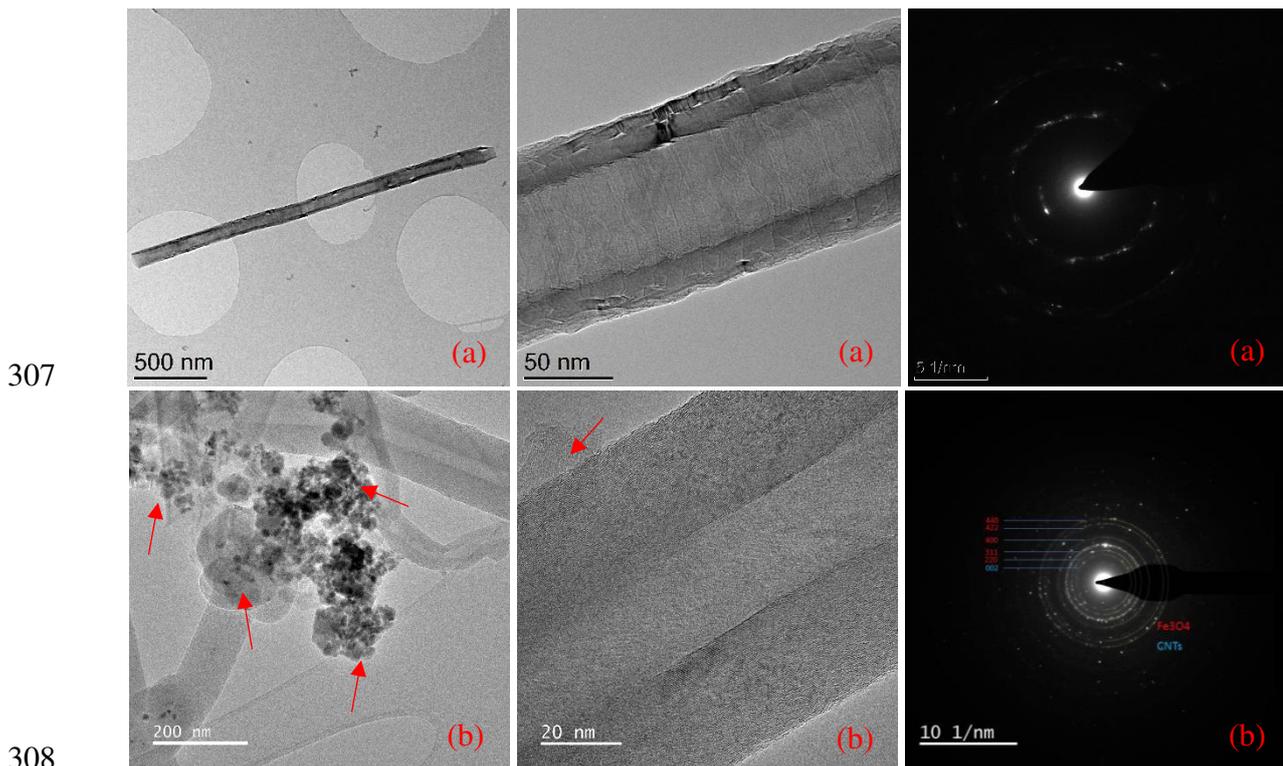
289 peak at  $\sim 830\text{ cm}^{-1}$  is attributed to contributions from aromatic C-H out-of-plane bending [32],  
 290 whilst the absorption band at  $\sim 701$ ,  $\sim 632$ , and  $\sim 558\text{ cm}^{-1}$  respectively could be associated with  
 291 the Fe-O stretching [34]. In the case of CNF, no clear absorbed peaks were detected as the  
 292 materials acted as a very effective black body absorber [35].



293  
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 296

**Fig. 2.** (a) FTIR spectra and (b) TGA profiles for CNF and K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs samples.

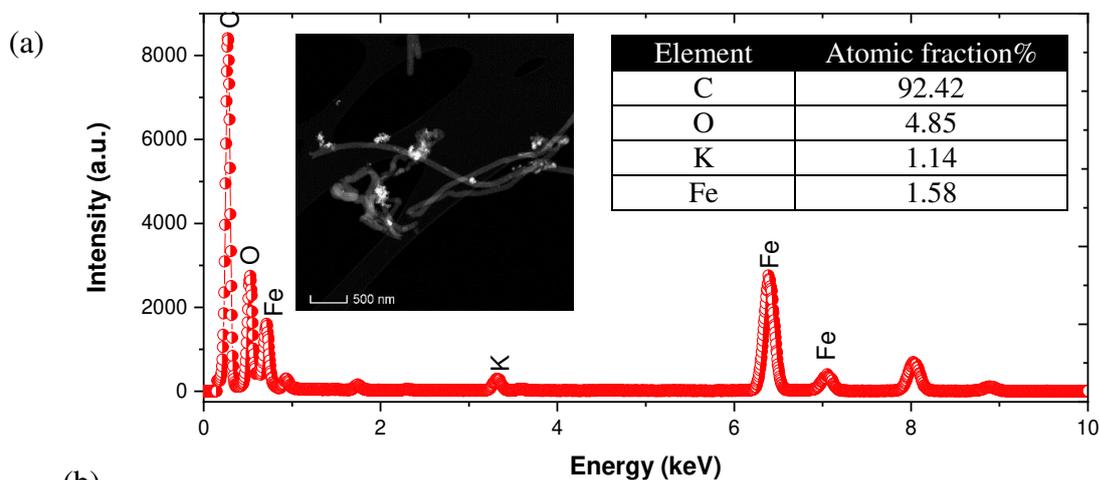
297 The thermal stability of CNF and K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs samples is shown in Fig. 2 (b). There  
 298 was only one stage of mass loss observed in the CNF sample, which is related to the slight  
 299 decomposition of CNF at a very high temperature (started  $\sim 600\text{ }^{\circ}\text{C}$ ). However, three significant  
 300 stages of mass loss are observed in the case of K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs. In the first stage, an initial  
 301 mass loss ( $\sim 4.8\%$ ) is noted to have occurred around  $50\text{ }^{\circ}\text{C}$ , which could be attributed to the  
 302 loss of free water attached to the sample's surface. The second mass loss is observed in the  
 303 temperature range of  $90$  to  $170\text{ }^{\circ}\text{C}$ , attributed to potassium loss. After that, no significant mass  
 304 loss was noted as the sample remained relatively stable over the temperature of  $170$  to  $540\text{ }^{\circ}\text{C}$ .  
 305 The final mass loss stage at higher temperatures spanning from  $540$  to  $900\text{ }^{\circ}\text{C}$  owing to the  
 306 decomposition of CNF.



309 **Fig. 3.** TEM micrograph images and selected area electron diffraction (SAED) of the (a) CNF  
 310 and (b) K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs samples  
 311

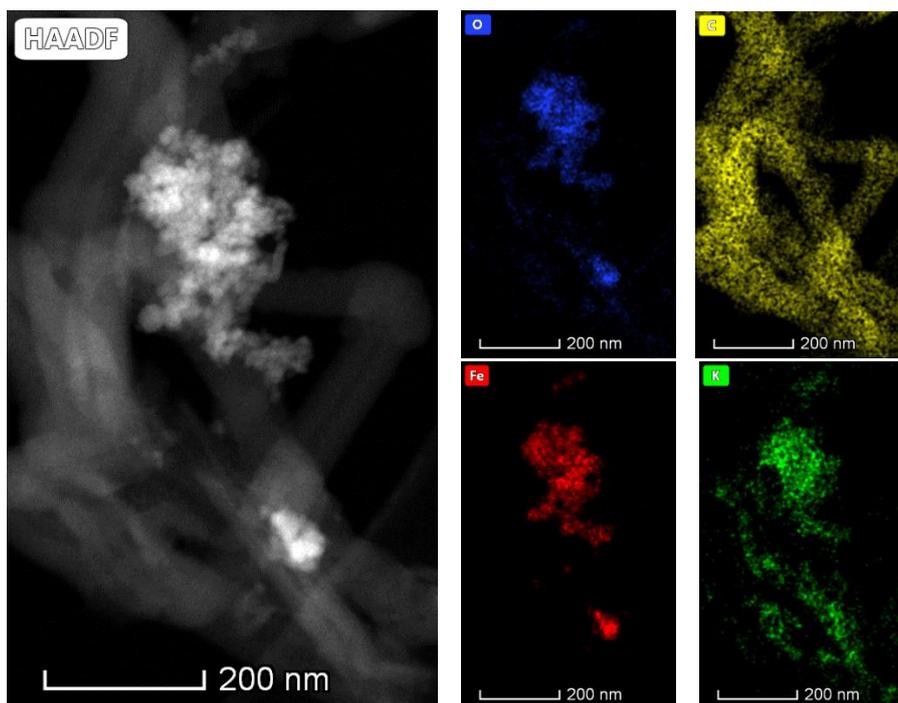
312 Fig. 3 represents the bright-field TEM images and a selected area electron diffraction (SAED)  
 313 of the CNF and K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs samples at different magnifications. The SAED pattern  
 314 of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF (Fig. 3 b) shows some discrete rings corresponding to the CNF crystalline  
 315 plane (002) and Fe<sub>3</sub>O<sub>4</sub> crystalline planes, (220), (311), (400), (422), and (440), which identifies  
 316 the CNF and Fe<sub>3</sub>O<sub>4</sub> NPs in the synthesised material [36]. The TEM images for K/Fe<sub>3</sub>O<sub>4</sub>/CNF  
 317 MNPs (Fig. 3b) show some cylindrical fibres with inwardly curved centres between some  
 318 rough-like exposed edged planes. The fibres are those of the CNF, and the edge planes present  
 319 sites for reactions that enable the chemical modification of the CNF surface to form composites  
 320 It can also be seen from Fig. 3 (b) the formation of some rough-spherical shapes and dark spots  
 321 bound on the CNF. These qualify the metal particles from the Fe<sub>3</sub>O<sub>4</sub> owing to the highly  
 322 ordered nature of the NPS, which gives diffraction contrasts, whilst the K-species appears  
 323 matched up on the surfaces of the Fe<sub>3</sub>O<sub>4</sub> and CNF in the sample. This confirms the presence of

324 K and Fe<sub>3</sub>O<sub>4</sub> in the synthesised magnetic NPs. The presence of K-species in the sample  
 325 enhances the hydrophilicity of the novel DS material, which could foster a rise in osmotic  
 326 pressure [31]. In the case of CNF (Fig. 3a), the surfaces of fibres are relatively smooth and long  
 327 filled cylindrical chains which can be directly related to the nature of CNF synthesis.



328  
329

(b)



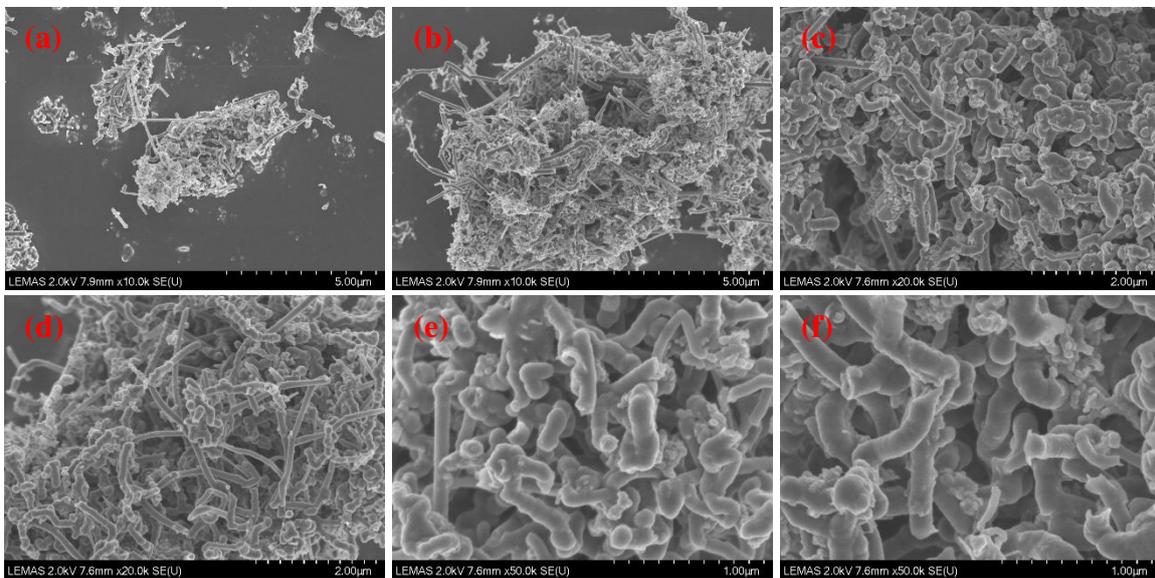
330

331 **Fig. 4.** (a) TEM-EDS spectrum showing the elemental composition and (b) EDS mapping of  
 332 K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs sample

333

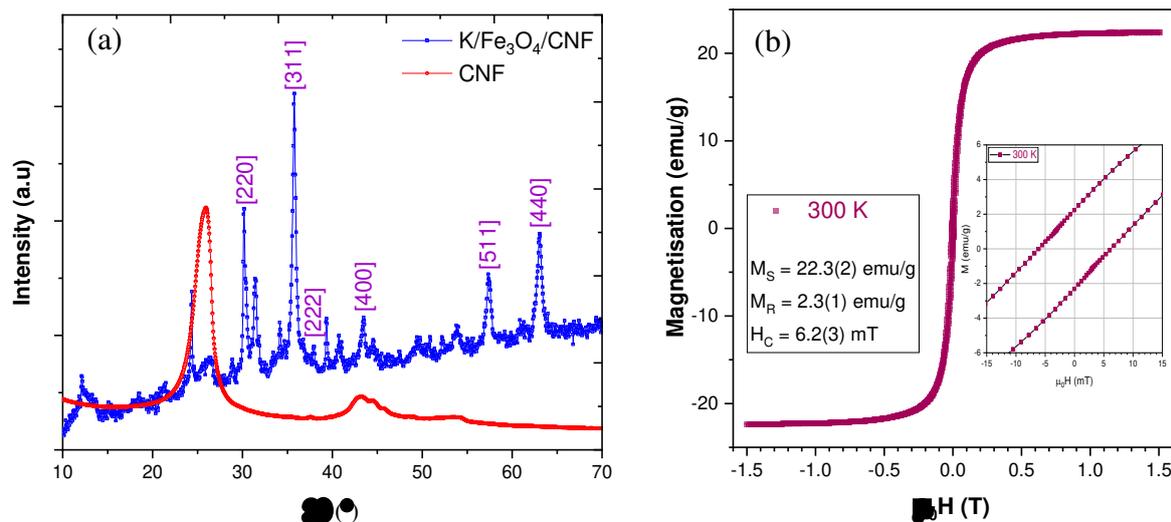
334 The elemental analysis of K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNP using EDS revealed that the synthesised sample  
 335 consisted of carbon, oxygen, iron, potassium elements, as shown in Fig. 4(a). The elemental

336 mapping, shown in Fig. 4b), depicts clustered nanoparticles with the K-species appearing  
337 evenly distributed in the Fe NPs whilst on the other hand appearing to have overlapped with  
338 the CNF, which potentially would have modified its surface. This further confirms the  
339 successful modification of the CNF with Fe<sub>3</sub>O<sub>4</sub> NPs and potassium hydroxide. The morphology  
340 of K/Fe<sub>3</sub>O<sub>4</sub>/CNF was further investigated using SEM analysis. Fig. 5 depicts the SEM  
341 micrographs of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF nanocomposites at different magnifications, showing some  
342 spherical-like morphologies characteristics of Fe<sub>3</sub>O<sub>4</sub> closely bounded together with some  
343 cylindrical morphologies representative of CNF.



344  
345 **Fig. 5.** SEM images for K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs sample at different magnifications.  
346

347  
348 Fig. 6(a) shows the XRD profile of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs sample. The XRD graph revealed  
349 reminiscent characteristics of the cubic spinel structure of Fe<sub>3</sub>O<sub>4</sub> (JCPDS-ICDD: 04-008-  
350 8146), which often emerges in XRD patterns of Fe<sub>3</sub>O<sub>4</sub> materials. The set of values of 2θ at 30°,  
351 35°, 37°, 43°, 57°, and 62.8° are attributed to reflections at (220), (311), (222), (400), (511) and  
352 (440) planes which are common with Fe<sub>3</sub>O<sub>4</sub> [37]. The other peaks could not be found in the  
353 International Centre for Diffraction Data (ICDD) database. This could result from the treatment  
354 of the CNF with strong acids and modification with KOH solution.



355

356 **Fig. 6.** (a) XRD patterns for CNF and K/Fe<sub>3</sub>O<sub>4</sub>/CNF samples and (b) magnetisation hysteresis

357 loop of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF sample at room temperature.

358

359 Fig. 6(b) shows the behaviour of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF sample under the influence of a magnetic

360 field at room temperature. Here, the applied field has been corrected against a palladium

361 reference sample to mitigate the effect of trapped flux in the superconducting coil. The soft

362 ferromagnetic behaviour required to separate the K/Fe<sub>3</sub>O<sub>4</sub>/CNF from their media effectively is

363 presented. This is evidenced by the low values of magnetic remanence ( $M_R = 2.3(1)$  emu/g)

364 and coercive field ( $H_C = 6.2(1)$  mT), which has also previously been shown to be sufficient

365 for limiting inter-particle magnetic interaction [38]. Meanwhile, the saturation magnetisation

366 of  $M_S = 22.3(2)$  emu/g is comparable to other Fe<sub>3</sub>O<sub>4</sub> based NPs, which undergo magnetic

367 separation in solution by an applied external field [39].

368

### 369 3.2 Osmotic pressure analysis

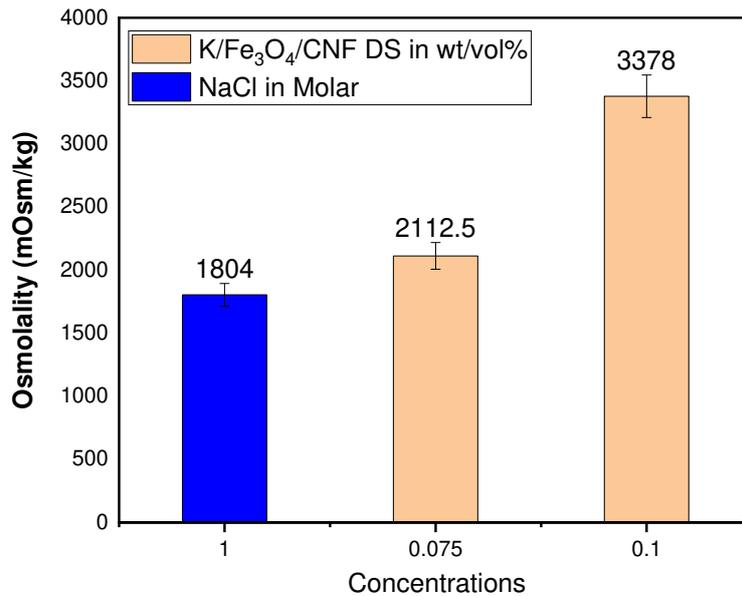
370 DS osmotic pressure is the most essential property other than membrane properties that

371 capacitate FO processes. It is the most critical condition that a functional material must fulfil

372 relative to the feed stream for such material to function effectively as a draw agent in FO

373 processes [2]. Fig. 7 represents the osmolality of varying concentrations of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF

374 MNPs and that of 1.0 M NaCl solution, respectively. It can be seen that the osmolality of the  
375 samples is dependent on their various concentrations. Two different osmolality measurements  
376 were read for each sample concentration, and the average was computed as the final osmolality  
377 reading. The osmolality of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs samples concentrations of 0.075 and 0.1  
378 wt/vol% were 2112.5 and 3378 mOsmol/kg, which relates to 53.8 and 86.1 bar, respectively.



379

380 **Fig. 7.** The osmolality of K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS at varying concentrations and 1.0 M NaCl  
381 solution.

382

383 Relatedly, the osmolality of 1.0 M NaCl solution was measured to be 1804 mOsmol/kg,  
384 corresponding to 45.96 bar. It is pertinent to note that the DS concentrations of 0.075 and 0.1  
385 wt/vol% generated considerable osmotic pressures, respectively. This is in great contrast to  
386 other DS functional materials explored in other FO studies. For instance, a study by Dey et al.  
387 [40] could only register an osmotic pressure of ~11.37 atm (i.e. ~11.5 bar) with a DS  
388 concentration of 0.078 wt%. Different studies [15, 17-19, 26, 28, 41, 42] could only achieve  
389 some reasonable osmotic pressures at much higher DS concentrations between 0.35 - 74 wt%  
390 relative to the DS concentrations applied in this work. This indicates that the K/Fe<sub>3</sub>O<sub>4</sub>/CNF  
391 MNPs could generate even greater osmotic pressure and water flux with measured

392 concentrations higher than those utilised in this work. It further lends credence that the DS  
393 functional material in this study is potentially deployable in FO saltwater desalination on the  
394 strength of its osmotic pressures.

395

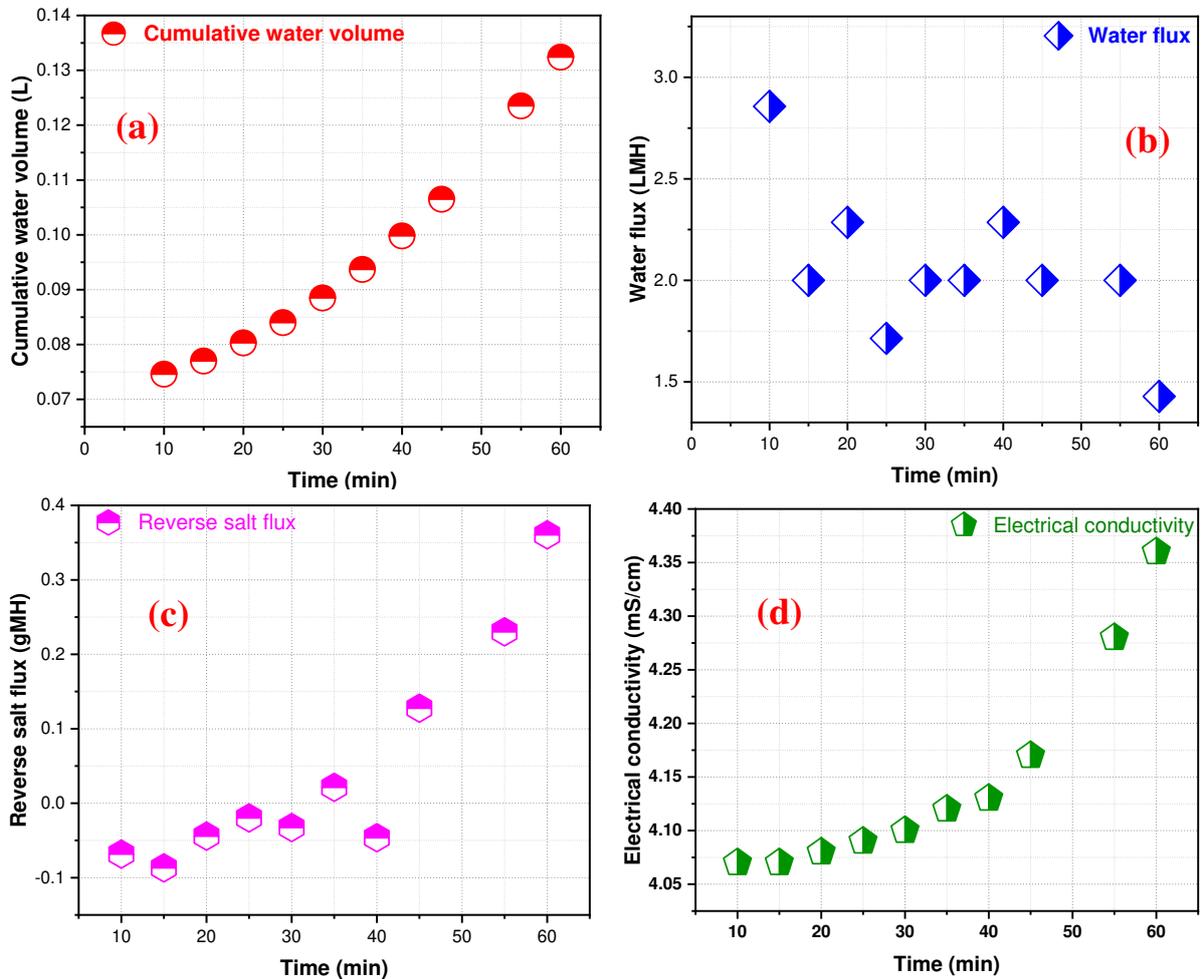
### 396 **3.3 FO evaluation**

397 The FO properties of K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS were examined for different parameters, including  
398 osmotic pressure, water flux over time, average water flux, reverse salt flux, and specific  
399 reverse salt flux (SRSF) with the view to understanding its performance. Fig. 8(a) represents  
400 the cumulative permeate volume change over time. Incremental permeate volume change rose  
401 continuously, reflecting the volumetric weight loss in the feed tank containing the 1.0 M NaCl  
402 solution over time. Fig. 8(b, c and d) illustrates the water flux, reverse salt flux, and the change  
403 in electrical conductivity (feed tank) over time during the FO experiment with the  
404 K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS. The water flux herein shown tend to be minimally impacted by the RSF,  
405 which is not unexpected considering obtained RSF value. The RSF exhibited some negative  
406 values at the initial stages, connoting bidirectional solutes diffusion. Negative ion flux  
407 highlights diffusion from the FS into the DS [43]. At high water flux, FS solutes move by  
408 advection to the membrane, at which point some may permeate through the membrane whilst  
409 the majority get rejected.

410 The result revealed a relatively stable water flux over the time of the FO experiment. As shown  
411 in Fig. 9, a cumulative water flux value of 37.8 LMH was attained when DI water was used as  
412 the FS at a DS concentration of 0.1 wt/vol%. On the other hand, 16.43 and 24 LMH water flux  
413 values were attained when 1 M NaCl solution was used as the FS at 0.075 and 0.1 wt/vol% DS  
414 concentrations, respectively. As opposed to the 1.0 M NaCl FS, where the FS might get  
415 concentrated over time as the DS become diluted, the DI water FS produced more water flux  
416 because the DS primarily confers the osmotic pressure gradient for the process without an

417 opposite or equal counteraction. Therefore, evaluating the osmotic activity of MNPs DS  
 418 materials relative to saltwater conditions is more rational. So doing would further help to  
 419 advance the potentialities of these MNPs DS materials in sure-enough conditions contrary to  
 420 evaluating them versus DI water as the feed.

421



422

423

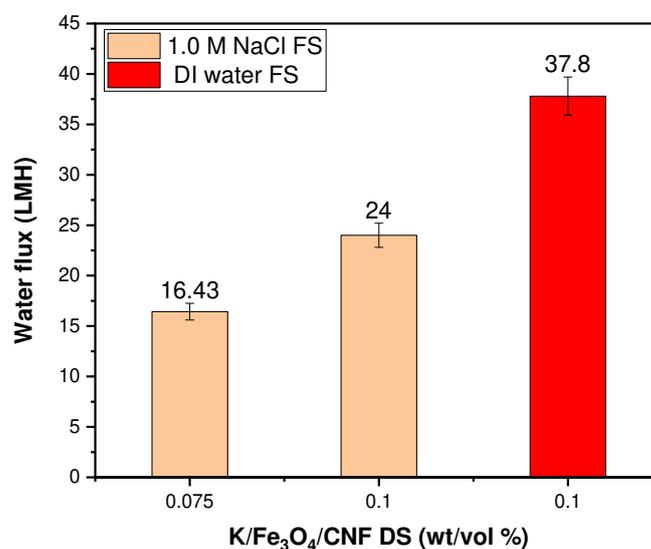
424 **Fig. 8.** a) Cumulative permeate volume over time; (b) water flux; (c) reverse salt flux (d); and  
 425 electrical conductivity of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS as a function of time.

426

427 The reason is that the ionic strength of feed streams, amongst other factors, can encumber water  
 428 flux, especially in osmotic filtration processes where the membrane acts as the primary site of  
 429 activities. Both results indicate a decrease in the water flux which can be attributed to different  
 430 factors including a decrease in the osmotic driving force due to dilution of the DS [11, 27].  
 431 Other factors can be due to the inhibition of the pores of the membrane support layer because

432 of the large particles sizes of the DS material [27, 44]. This inference agrees with the trend of  
433 the water flux data in both cases. Particle size increases are known to adversely impact the  
434 osmotic driving force, which affects the overall FO system negatively [40]. At increased DS  
435 concentration and, by extension, the net osmotic pressure, the water flux increases. Such  
436 increases induce increased dilutive internal concentration polarization (ICP) inside the  
437 membrane support layer, reducing the overall driving force and water flux gains. Controlling  
438 the particle size of DS materials is, therefore, exigent. On the other hand, the hydrodynamic  
439 effects within the FO cell may have caused some NPs to abrade off the DS material whilst  
440 colliding with the membrane surface, depositing within the membrane pores in consequence  
441 [11, 27]. The deposition of the abraded NPs into the membrane pores is known to elicit severe  
442 ICP effects, constituting a limitation on the overall water flux [45]. As with MNPs, because  
443 they are not salts, the high energy potential NPs can migrate easily into the membrane pores,  
444 where they linger without necessarily exiting the pores [11], thereby constituting hindrance to  
445 the water flux. Modifying membranes with smart materials that have a high net surface charge  
446 to repel approaching particles might be plausible in mitigating particles' deposition into pores.  
447 The migrating NPs will bounce away from the membrane if the repulsive force is strong  
448 enough. Consequently, keeping them in a state of dispersion within the DS. In general, MNPs  
449 in colloids tend to aggregate if the interaction is permanent or agglomerate if their commingling  
450 is reversible [46]. These clustering mechanisms are known to have a significant negative  
451 impact on water flux, and may have played a role in this work. Although there are several  
452 methods of addressing issues related to agglomeration, surface modification of MNPs during  
453 material synthesis with suitable and sustainable modifiers like oleic acids should be given good  
454 consideration. It is noteworthy that whilst introducing hydrophilic groups via ligand exchange  
455 reactions helps improve surface hydrophilicity, their subsequent effect on defining size,  
456 modulating size distribution, and particles' effective magnetism requires careful consideration.

457 Because they can change some of the structural and magnetic properties of the modified MNPs,  
458 limiting their efficiency and the system's overall performance.



459  
460 **Fig. 9.** Obtained water flux against varying concentrations of K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS.

461  
462 A glean from Table 1 shows that the K/Fe<sub>3</sub>O<sub>4</sub>/CNF performed remarkably well in water flux  
463 relative to many of the DS, most of which were functionalised with hydrophilic polymer  
464 compounds, and to which DI water was used as the feed in a good number of these studies. At  
465 lower DS concentrations of 0.075 and 0.1 wt/vol%, respectively, the K/Fe<sub>3</sub>O<sub>4</sub>/CNF could  
466 produce significant water flux (16.43 and 24 LMH) relative to what is achievable with a good  
467 number of magnetic DS as reported in the literature at higher concentrations. Whilst  
468 investigating MNPs as DS with DI water as the FS advances their potential application in FO  
469 operations, it is critical to highlight that DI water as the FS does not exhibit osmotic resistance  
470 to the MNPs DS as salt solutions do. Despite being tested against FS with high ionic strength  
471 (1.0 M NaCl solution), the K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS could advance water fluxes higher than those  
472 achieved with existing MNPs as DS in previous studies utilising DI water as feed, which  
473 provide no osmotic resistance to the DS.

474

475 **Table 1.** Comparison of water fluxes of different types of MNPs DS relative to varying DS  
 476 concentrations and source water feed streams.

Draw solutions	Concentration	Osmotic pressure	Initial water flux/time	Average water flux/time	Feed solution	Recovery process	References
PSSS-PNIPAM functionalized MNPs	33 wt%	55.0 (atm)	--	3.7 LMH and 2.7 LMH (1 h)	NaCl solution of 1000 mOsm/kg and 1200 mOsm/kg HTI membrane	Magnetic field & UF	[28]
Gelatin-coated MNPs	27g/L	1.50 (bar)	0.62 LMH (24 h)	--	0.01 M NaCl solution, HTI membrane	Magnetic field	[26]
Pectin coated MNPs	0.5 wt/vol%	--	6.6 LMH (120 min)	--	1.0 g% NaCl solution, Polyamide (PA) membrane	Magnetic field	[27]
Na/PAA MNPs	200 g/l	32.0 (atm)	2.0 LMH	--	Synthetic brackish water, HTI membrane	Electric field and NF	[15]
Poly-sodium-acrylate MNPs (PSA MNPs)	37 wt%	8.88 (atm)	4.3 LMH	--	DI water, CTA flat sheet membrane	Magnetic field and UF	[17]
SHPG-MNPs	400g/L	9.62 (atm)	3.0 LMH	--	DI water, CTA membrane	UF	[47]
Branched poly(deep eutectic solvent)-coated MNPs	3.5 g/l	68.0 (atm)	17.9 LMH (90 min)	--	DI water, CTA flat sheet membrane	Magnetic field	[19]
Poly(sodium acrylate) coated MNPs	0.078 wt%	11.37 (atm)	--	5.6 LMH (4 h)	DI water, Carbon nanotube membrane	Magnetic field	[40]
Citrate coated MNPs	20 mg/l	--	17.3 LMH (35 min)	--	DI water, CTA flat sheet membrane	Magnetic field	[11]

Sodium alginate sulfate MNPs	60 g/l	117.2 (atm)	--	8.6 LMH (60 min)	DI water, CTA flat sheet membrane	Magnetic field	[18]
K/Fe <sub>3</sub> O <sub>4</sub> /CNF	0.1 wt/vol% and 0.075 wt/vol%	86.1 (bar) 53.8 (bar)	24 LMH and 16.43 LMH (55 min)		1.0 M NaCl solution, CTA flat sheet membrane	Magnetic field	This work

477

478 To further probe the performance of the K/Fe<sub>3</sub>O<sub>4</sub>/CNF MNPs as a functional DS material, the  
479 RSF and SRSF of the DS material are examined. They are essential criteria when considering  
480 the suitability of a functional material as DS in FO processes [48]. The SRSF ( $J_s/J_w$ ) helps with  
481 appraising the degree of draw solutes leakage into the feedstock per unit volume of water  
482 recovered across the membrane into the DS during FO processes [31, 48]. In contexts where  
483 high product quality is considered essential, SRSF value is particularly invaluable in  
484 designating the suitability of a functional DS material being deployed in such situations [42].  
485 Because the 0.1 wt/vol% DS concentration exhibited much higher osmolality and, by  
486 extension, higher water flux, it was employed for elucidating RSF in this report. It can be  
487 observed that the RSF obtained during the experiment with the 0.1 wt/vol% DS concentration  
488 is ~ 0.56 gMH. Furthermore, an inappreciably small SRSF (ratio of  $J_s/J_w$ ) value of 0.02 g/L is  
489 obtained from the maximum DS concentration in the sphere of this study.

490 Instructively, the degree of solute loss within the framework of this study appears to have fared  
491 reasonably well relative to that of varying ionic salts conventionally used as DS in FO  
492 operations, including MgCl, KCl, Ca(NO<sub>3</sub>)<sub>2</sub>, NaCl, NH<sub>4</sub>Cl, CaCl, and NH<sub>4</sub>HCO<sub>3</sub> as presented  
493 in Table 2. The SRSF value obtained in this work with the K/Fe<sub>3</sub>O<sub>4</sub>/CNF is negligible relative  
494 to SRSF values of the salts mentioned above in the range of ~0.57 – 2.48 g/L reported in  
495 previous studies [48]. The smaller SRSF value recorded in this work might not be unconnected  
496 to strong binding activity between the constituent nanoparticles of the functional DS material

497 during synthesis. Such strong binding between ions of the nanocomposite materials would  
 498 likely have increased the diameter of the nanoparticles [28, 49] relative to the membrane pore  
 499 sizes, which would have hindered the reverse diffusion of the draw solutes across the  
 500 membrane into the feed stream. Such an insignificantly small SRSF value obtained in this work  
 501 elicits the novel DS material's potential to be feasibly deployable in long-term operations in  
 502 FO processes [31]. Nonetheless, because ions tend to permeate the membrane bidirectionally  
 503 during FO operations, it will be well-advised to investigate how any diffusion of FS ions into  
 504 the DS can modulate the DS's performance in future works.

505

506 **Table 2.** Reverse salt flux and specific reverse salt flux of commercial DS relative to  
 507 K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS

Draw solution	RSF (J <sub>s</sub> ) g/m <sup>2</sup> h	SRSF (J <sub>s</sub> /J <sub>w</sub> ) g/Lg/L	Reference
MgCl <sub>2</sub> ;	5.6	0.58	[48]
KCl	15.3	1.14	
Ca(NO <sub>3</sub> ) <sub>2</sub>	6.6	0.62	
NaCl	9.1	0.75	
NH <sub>4</sub> Cl	10.2	0.79	
CaCl	9.5	0.82	
NH <sub>4</sub> HCO <sub>3</sub>	18.2	2.48	
K/Fe <sub>3</sub> O <sub>4</sub> /CNF	0.56	0.02	This work

508

509 The DS preparation methodology deployed in this study eliminates the need for MNPs to be  
 510 functionalised with hydrophilic polymers. In contrast to the TEG-K/CNF DS material  
 511 fabricated in an earlier work by our group [31], the in-situ DS fabrication process employed in  
 512 this work is facile, practical, and potentially more effective based on the obtained results. The  
 513 polymer triethylene glycol (TEG) utilised in our previous work was replaced with Fe<sub>3</sub>O<sub>4</sub>. The  
 514 new DS material could yet foster higher osmolality and higher water flux even at lower DS  
 515 concentrations. The constituent materials utilised herein are biocompatible, non-toxic, and  
 516 naturally occurring, which could be easily harnessed. Overall, this offers a potentially

517 sustainable methodology that can further bolster cost reductions applicable to FO desalination  
518 processes. Ordinarily, the colloidal solutions of  $\text{Fe}_3\text{O}_4$  MNPs are understood to exhibit  
519 inappreciable osmotic pressure even at higher concentrations [50], and this is the same case  
520 with CNF. However, it follows that incorporating K-species in the synthesis and some  
521 hydrophilic ligands absorbed into the surfaces of the  $\text{Fe}_3\text{O}_4$  and CNF generated osmotic  
522 pressure build-up within the DS nano-fluid. Consequently, the enhanced osmotic effect  
523 facilitated by the K-species and the hydrophilic groups in the DS ensured a sustained osmotic  
524 diffusion of water molecules across the CTA membrane from the feed stream over time.

525

#### 526 **4. Conclusions**

527 The potential application of K/ $\text{Fe}_3\text{O}_4$ /CNF MNPs as a novel DS in FO desalination was  
528 successfully demonstrated. The DS preparation protocol established in this study is  
529 advantageous as it is eco-friendly, and it eliminates the often recourse to polymer compounds  
530 for the needed hydrophilic functionality in MNPs. Yet, the K/ $\text{Fe}_3\text{O}_4$ /CNF attained enhanced  
531 hydrophilicity. Moreover, the facile DS synthesis strategy will potentially open a new route to  
532 preparing other DS nanomaterials with unique multi functionalities and enhanced  
533 hydrophilicity without recourse to polymers. The novel draw solution generated considerable  
534 osmotic pressure of 86.1 bar, high enough to desalinate highly saline feed streams. It also  
535 occasioned a high cumulative water flux of 24 LMH at a DS concentration of 0.1 wt/vol%  
536 utilising a CTA membrane. Draw solutes eliciting higher water flux at lower DS concentrations  
537 will be preferable, considering that they can contribute to considerable reductions in the overall  
538 capital expenditures in FO operations. Relative to several organic and inorganic salts in  
539 literature commercially used as DS in FO operations, negligibly low reverse salt flux and  
540 specific reverse salt flux values of  $\sim 0.56$  gMH and  $\sim 0.02$  g/L were recorded with the novel  
541 DS. However, to offset the negative impact of bigger particle sizes on water flux, it is necessary

542 to carefully manage the particle size distribution of DS materials during synthesis,  
543 notwithstanding its positive influence on RSF. Furthermore, it will be well-thought for future  
544 research to fully appraise the energy implication and the overall economic merit of magnetic  
545 field DS reconcentration and water recovery relative to other DS recovery methods. This is  
546 considering the often requirement of ultrafiltration with magnetic separation in contexts where  
547 high product water quality is required as the final product. Although the predisposition of  
548 membrane fouling with the K/Fe<sub>3</sub>O<sub>4</sub>/CNF DS is not investigated in this work, it is pertinent to  
549 explore this in future studies to ascertain the extent to which fouling may or otherwise impact  
550 water flux in utilising this novel DS material. In the next phase of this study, our group will  
551 consider the FO system integrating magnetic field and solar-thermal separability to attempt the  
552 feasibility of solute recovery of smaller sized MNPs.

553

#### 554 **Acknowledgements**

555 We acknowledge the technical support provided by Dr Adrian Cunliffe, Ms Karine Alves  
556 Thorne and Dr Ben Douglas for access to the analytical equipment, Mr Stuart Micklethwaite  
557 for performing SEM images, and Mr Mohammed Javed for collecting XRD data. D.H.  
558 acknowledges financial support from the Engineering and Physical Sciences Research Council  
559 (EP/S032797/1). A.A. thanks the Petroleum Technology Development Fund (PTDF) in Nigeria  
560 (PTDF/AAJ/250/17) for funding.

561

#### 562 **CRedit authorship contribution statement:**

563 Aondohemba Aende: Performed the FO experiments, characterisations of DS, formal analysis,  
564 funding, and drafted the first version of the manuscript. Jabbar Gardy: Originated idea,  
565 synthesised the draw solution, methodology, provided support with DS characterisations,  
566 investigation, supervised the work, and writing - review & editing; Zabeada Aslam: Collected

567 data on Titan TEM and provided supports in TEM analysis, investigation, and writing - review  
568 & editing; Matthew Rogers: Collected data on SQUID and provided support with magnetism  
569 analysis, investigation, and writing - review & editing; Mohamed Edokali: Provided supports  
570 with FO experiments and analysis, investigation, writing - review & editing, Oscar Cespedes:  
571 Provided support with magnetism analysis, investigation, and writing - review & editing; David  
572 Harbottle: Supervised the work, funding, investigation, and writing - review & editing; Ali  
573 Hassanpour: Provided supports with FO data analysis, supervised the work, investigation,  
574 project management, and writing - review & editing.

575

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583

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