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# Impurity band assisted carrier relaxation in Cr doped topological insulator $\text{Bi}_2\text{Se}_3$

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## Abstract

Topological insulators (TIs) with unique band structures have wide application prospects in the fields of ultra-fast optical and spintronic devices. The dynamics of hot carriers plays a key role in these TI based devices. In this work, using time and angle resolved photoemission spectroscopy (TR-ARPES) technique, the relaxation process of the hot carriers in Cr doped  $\text{Bi}_2\text{Se}_3$  has been systematically studied, where the ferromagnetic TI is one of the key building blocks for the next generation spintronics. It is found that the electronic temperature ( $T_e$ ) and chemical potential ( $\mu$ ) decrease faster with the increase of the Cr doping concentration. Similarly, the lifetime ( $\tau$ ) of the excited electrons also decreases with more Cr doped into TIs. The results suggest a new mechanism of the impurity bands assisted carrier relaxation, where the impurity bands within the bulk band gap introduced by Cr doping provide significant recombination channels for the excited electrons. This work directly illustrates the dynamic process of the carriers in Cr doped  $\text{Bi}_2\text{Se}_3$ , which is expected to promote the applications of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  in ultrafast optical and spintronic devices.

**Keywords:** TR-ARPES, Cr doped  $\text{Bi}_2\text{Se}_3$ , hot carriers, ultrafast optical and spintronic devices.

TIs are a class of quantum materials featuring with a bandgap in its bulk and unique Dirac-like metallic states on the surface, which have abroad application prospects in the fields of ultrafast optical and spintronic devices<sup>1-3</sup>. The performance of these devices depends on the dynamics of the hot carriers in TIs, which can be observed directly by TR-ARPES<sup>4-6</sup>. So far, a number of works have been carried on illustrating the recovery process of the excited electrons in TIs due to various mechanisms: such as the phonons assisted,<sup>7,8</sup> the surface states assisted,<sup>9</sup> and the electron-electron scattering assisted.<sup>10,11</sup>

Recently, numerous work have reported that the doping with transition metals can introduce novel physical phenomena in TIs, such as quantum anomalous Hall effect (QAHE)<sup>12</sup>, the giant magneto-optical Kerr effect<sup>13</sup>, and chiral Majorana fermions<sup>14</sup>. In addition, the doping with transition metals changes the band structure of TIs by weakening the surface state and increasing the bulk bandgap. This suggests that it may also affect the carrier relaxation. However, this has not been systematically studied experimentally, though it is crucial to the application of TIs in the field of ultrafast optical and spintronic devices.

In this work, a series of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  ( $x = 0 \sim 0.074$ ) samples were grown by molecular beam epitaxy (MBE). Then, the dynamic processes of the excited electrons in  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  are directly studied by the TR-ARPES. It is found that as the Cr doping concentration increases, the relaxation rate of  $T_e$  and  $\mu$  increase, and  $\tau$  of the excited electrons decreases. This has been attributed to the mechanism of the impurity bands

assisted carrier relaxation. The impurity bands within the bulk band gap introduced by Cr provide more recombination channels for the excited electrons. This provides direct insight into the relaxation mechanisms of the Cr doped TIs, and therefore has profound implications for future works exploring the potential of these materials for ultrafast optical and spintronic devices.

## Results

The  $\text{Bi}_2\text{Se}_3$  and  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  ultrathin films are grown on fluorophlogopite  $[\text{KMg}_3(\text{AlSi}_3\text{O}_{10})\text{F}_3]$  substrates by MBE. Compared with conventional epitaxy, van der Waals epitaxy (VDW) growth significantly relaxes the stringent conditions of lattice match between the substrate and the epitaxial layer<sup>15-17</sup>. And, atomically flat fluorophlogopite with pseudo-hexagonal layered structure is suitable for VDW epitaxy growth of layered material due to no dangling bonds associated with the surface<sup>18,19</sup>. The Cr dopants are evaporated simultaneously with the Bi and Se atoms and deposited on fluorophlogopite at 200°C. The growth was carried out under the Se-rich environment with a nominal Se to Bi ratio of 20:1, which is beneficial to reduce the Se vacancy defects<sup>20</sup>. The Cr doping concentration ( $x=0.013\sim 0.074$ ) is determined by the Cr deposition flux and the X-ray photoelectron spectroscopy (XPS). Samples were transferred into ARPES chamber through a high vacuum tube ( $<3.0\times 10^{-9}$  mbar) to ensure the clean surfaces. The band structure of the  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  films measured by static ARPES (He-I, 21.2eV) at room temperature is shown in Figure 1. It is found that the BVB, surface state and BCB can be clearly distinguished in undoped  $\text{Bi}_2\text{Se}_3$ .

However, the surface state decreases and the bulk band gap increases with the increase of Cr doping concentration. When the Cr doping concentration is increased to  $x=0.074$ , the surface states cannot be observed at all in the band structure. This is consistent with the previous results<sup>21,22</sup>.

TR-ARPES measurements were performed using a pump-probe scheme. The laser pulse with 800 nm ( $\sim 1.55$  eV) wavelength and 1 kHz repetition is split into two beams. One beam is used as the pump light to stimulate the sample, and the other is used for high harmonic generation (HHG)<sup>23,24</sup>, and the 13th harmonic extreme ultraviolet (EUV) at  $\sim 20.15$  eV has been chosen as the probe light<sup>25</sup>. Detailed description of the HHG system will be found in our previous work<sup>26</sup>. The overall energy resolution is 190 meV and time resolution is 60 fs. Pump light is *s* polarized, which was maintained at  $0.21$  mJ/cm<sup>2</sup> during the measurement. Probe pulse is *p* polarized, and the flux of probe pulse is chosen to minimize the space charge effect. All the experiments were carried out at room temperature.

Figure 2(a)-(c) demonstrates the various differential band structures after pumping of Cr ( $x=0.013$ ) doped Bi<sub>2</sub>Se<sub>3</sub> obtained by EUV by subtracting the un-excited band structure at  $-0.2$  ps. The red and blue represents the increased or decreased electrons at different *E-K* positions. It's found that the intensity above the Fermi level is strongest at  $0.7$  ps because of the thermalization of the electrons after inelastic scattering process<sup>27</sup>. Then, it starts to reduce at  $1.5$  ps due to the recombination of excited electrons and holes assisted by phonons. The band structure recovers to un-excited state at  $8$  ps.

Figure 2e-2g and Figure 2i-2k are the differential band structures of the higher doped samples with  $x=0.056$  and  $0.074$ , respectively. The similar dynamic process of excited electrons could be seen in these three samples. Namely, after the pump light of  $0.7$  ps, electrons are excited above the Fermi level, then quickly fall back around the Fermi level, and finally return to its original state<sup>8</sup>.

The relaxation dynamics of the hot electrons around the Fermi level after pumping can be quantified by the Fermi-Dirac (FD) distribution convoluted with a Gauss distribution, which considers the energy resolution of the TR-ARPES system:

$$I(E, t) = A(t) \int_{-\infty}^{\infty} [F(\varepsilon, T_e(t), \mu(t))D(\varepsilon)]G(E - \varepsilon, \sigma) d\varepsilon \quad (1)$$

Here,  $I$  represent the energy dispersion curve (EDC) as a function of time delay,  $F$  is the FD distribution,  $T_e$  and  $\mu$  denote the electron temperature and chemical potential of the electrons.  $G$  is the Gauss distribution to take the overall energy resolution of the system (mainly depending on EUV) into account.  $A$  and  $D$  are the scale factor<sup>28</sup> and the product of photoemission matrix element and the density of state, respectively<sup>29</sup>. Due to the complexity to determine the  $D$ <sup>30</sup>, in the fitting process, a narrow range around the  $\Gamma$  point ( $\pm 0.01 \text{ \AA}^{-1}$  around the center) has been used to extract the leading edge of the EDCs (as shown in Figure 2d, 2h and 2l). With this way,  $D$  can be treated as a constant. Although this simple treatment neglects the fine structure in  $D$ , it is helpful to compare the dynamic process of different samples quantitatively.

Figure 2(d) demonstrates EDCs of  $x=0.013$  at the different delay time. It is found that the EDC at  $0.7$  ps (black line) shifts to the higher energy compared to  $-0.2$  ps (red

line), which means the  $T_e$  and  $\mu$  of the former are higher. This is due to the thermalization of electrons in CB after absorbing the energy from pumps photons. The EDC at 1.5ps shows intermediate features compared with above two due to the energy exchange between excited electrons and phonons help the relaxation of electrons. At 8 ps after the arrival of pump photons, the excited electrons almost recombine with the holes and the sample recovers to the state before excited. EDCs at the same delay time of  $x=0.056$  (Figure 2h) and  $0.074$  (Figure 2l) showed a similar recovery process.

Moreover,  $T_e$  and  $\mu$  as a function of delay time with different  $x$  were extracted using Eq. (1) to fit the EDCs, as shown in Figure 3a and 3b. The  $T_e$  of  $x=0$  reveal a quick relaxation from the peak value to around 500~590 K, and maintain around that temperature within the detection window (Figure 3a). The reason is that the weak coupling of phonons and surface electrons. It should be noted that the  $\Delta\mu$  versus delay time does not satisfy exponential decay function at low Cr doping concentration, suggesting that the activity of phonons at the surface of pure  $\text{Bi}_2\text{Se}_3$  is weak<sup>31</sup>. Interestingly, with the increase of  $x$  ( $x=0.013\sim 0.074$ ),  $T_e$  and  $\Delta\mu$  recover to the state before pump pulse more quickly, which also demonstrate the ability of Cr doping to assist the relaxation of excited electrons.

In order to reveal the  $\tau$  of the excited carriers, a black box was fixed at 0.1 eV above the Fermi level (as shown in Figure 2a) to indicate the integral domain. The integral intensities as a function of pump-probe delay time for various  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  films are shown in Figure 3c. After pumping light, electrons are immediately excited



above the Fermi level, and then quickly fall back around the Fermi level<sup>32</sup>. After that, electrons relax to its original state facilitated by electron-phonon scattering process<sup>8</sup>. Thus, the excited electrons have the same recovery tendency as  $T_e$  and  $\Delta\mu$ . Meanwhile, the electron relaxation rate of undoped  $\text{Bi}_2\text{Se}_3$  is slower than that of Cr doped  $\text{Bi}_2\text{Se}_3$ . Namely, the Cr doping reduces the  $\tau$  of excited electrons. By use of single exponential decay function to fit the curves in Figure 3c, the  $\tau$  of excited electrons can be extracted as a function of  $x$ , as shown in Figure 3d. It is found that  $\tau$  decreases monotonically with the increase of doping concentrations  $x$ .

## Discussion

Two factors have been reported to influence the  $\tau$  of the excited electrons in TIs<sup>4,7</sup>. First, the existing of surface state, which provides additional paths (2 and 3) for the electrons through the coupling of TSS and BCB (Figure 4a)<sup>4</sup>, helps the relaxation of excited electrons. Thus, higher Cr doped samples should have longer  $\tau$ , because of the disappearing of the TSS<sup>33</sup>. Second, doping Cr enlarges the bulk bandgap as shown in Figure 3d, which are extracted by a multi-peaks Gaussian fitting around the  $\Gamma$  point in Figure 1. And this is consistent with our calculated band structure by VASP package (Supporting Information Figure S1). However, the reports demonstrate that the enlarged bulk band gap can only increase  $\tau$  of the excited carriers<sup>22</sup>, which is again controversy to this work.

To explain our results, we propose here a new mechanism of the impurity band assisted carrier relaxation. The band structures of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  have been shown in

Figure S1. It is found that the impurity bands (mainly composed of Cr-*d* states) appear at the bottom of BCB and the top BVB after doping Cr, as shown by the red dashed lines in Figure 4b and Figure S1. Moreover, as more Cr atoms are doped into the lattice, the density of states of impurity bands increases. Due to the existence of impurity band, the excited electrons in the BCB can move to the impurity band through thermal relaxation process. And they then can jump down to the impurity band above the top of the BVB (path 3) or the BVB (path 4), as shown in Figure 4b. Thus, these impurity bands provide additional recombination channels, which assist the relaxation of the excited carriers<sup>34,35</sup>.

According to shock-read-Hall recombination theory, considering n-type conduction and low optical injection situation in this work, the recombination rate ( $R_I$ ) of excess carriers passing through path 2, path 3 and path 4 can be written as:  $R_I = (C_2N_I + C_3N_I + C_4N_I)\delta_p$ , quantitatively.  $C_2$ ,  $C_3$  and  $C_4$  are constants representing the hole capture cross-section of impurity bands.  $\delta_p$  is the number of excess holes, which can be treated as a constant due to the invariable pump power density.  $N_I$  is the number of the trap center, and it can be estimated by the integral of the density of states of the impurity band. Meanwhile, the recombination rate of the bulk (path 1) ( $R_B$ ) can be expressed as:  $R_B = C_1N_B\delta_p$ .  $C_1$  is a constant representing the hole capture cross-section of CB.  $N_B$  is the density of states at the bottom of the conduction band.

Here,  $\tau$  of the excited carriers is proportional to  $1/R$ , so it can be written as:

$$\tau = \frac{1}{R_I + R_B} = \frac{1}{(C_2 + C_3 + C_4)N_I + C_1N_B} \quad (2)$$

Because  $C_1$ ,  $C_2$ ,  $C_4$  and  $N_B$  are constants,  $\tau$  can be simplified as:  $\tau = \frac{1}{C_2'N_I + C_1'}$ .

As  $C_2'$  and  $C_1'$  are constants, the  $\tau$  vs. impurity band density can be fitted by equation

2 as shown in Figure 4c. Here, a background constant  $\tau_0$  has been added as:  $\tau =$

$\tau_0 + \frac{1}{C_2'N_I + C_1'}$ . The fitting results agree well within the experimental data, suggesting

that  $\tau$  of the hot carriers shortens with the increase of the density of states of impurity

band. Thus, the impurity bands induced by the Cr doping provide additional

recombination paths, overcome the previous two factors, and effectively lower  $\tau$  of the

hot electrons.

## Conclusion

In summary, we have systematically studied the dynamic process of photoexcited

carriers in high quality  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  with different Cr doping concentrations by TR-

ARPES. It is found that as the Cr doping concentration increases, the  $T_e$  and  $\mu$  decrease

faster. Secondly,  $\tau$  of the excited electrons decreases with the increase of Cr doping

concentration. This has been attributed to the new relaxation mechanism that the

impurity bands induced by doping provide additional recombination paths, and

effectively lower  $\tau$  of the hot electrons. This work has found that the doping can tune

the carrier relaxation in TIs, which may promote their applications in ultrafast optical

and spintronic devices.

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**Captions:**

Figure 1. (Color online) The band structures of a series of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  with  $x$  from 0 to 0.074. The bulk bandgap increases with the increase of Cr doping concentration, and the surface states gradually disappear.

Figure 2. (Color online) (a)-(c) Differential band structures of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  ( $x=0.013$ ) around  $\Gamma$  point at  $t= 0.7$  ps, 1.5 ps and 8 ps, respectively, which have subtracted the band structure of  $t= -0.2$  ps. (e)-(g) and (i)-(k) are the differential band structures of  $(\text{Bi}_{1-x}\text{Cr}_x)_2\text{Se}_3$  with  $x= 0.056$  and 0.074, respectively. (d), (h) and (l) are the EDCs integrated between  $\pm 0.01 \text{ \AA}^{-1}$  around  $\Gamma$  point at different delay time with  $x= 0.013$ , 0.056 and 0.074, respectively. Using a convoluted Gaussian and Fermi-Dirac distribution function,  $T_e$  and  $\Delta\mu$  can be extracted.

Figure 3. (Color online) (a) The  $T_e$  and (b) the  $\Delta\mu$  as a function of the delay time with different  $x$ . And, the  $T_e$  and  $\Delta\mu$  recover more quickly as the increase of Cr doping concentration. (c) Normalized integral intensities within the black boxes in Figure 2a, as a function of the delay time for different samples. (d) The lifetime ( $\tau$ ) of the excited electrons and bulk bandgap as the function of Cr doped concentration ( $x$ ).

Figure 4. (Color online) Mechanism diagram of the non-equilibrium carrier recombination process of the (a) undoped and (b) Cr doped  $\text{Bi}_2\text{Se}_3$ . The results show that the impurity band can provide additional carrier recombination channels. Green lines represent the surface states. (c) The Carrier's lifetime ( $\tau$ ) as a function of the density of states of impurity band. The results show that the  $\tau$  of hot carrier reduces with the increase of the density of states of impurity band.

Figure 1

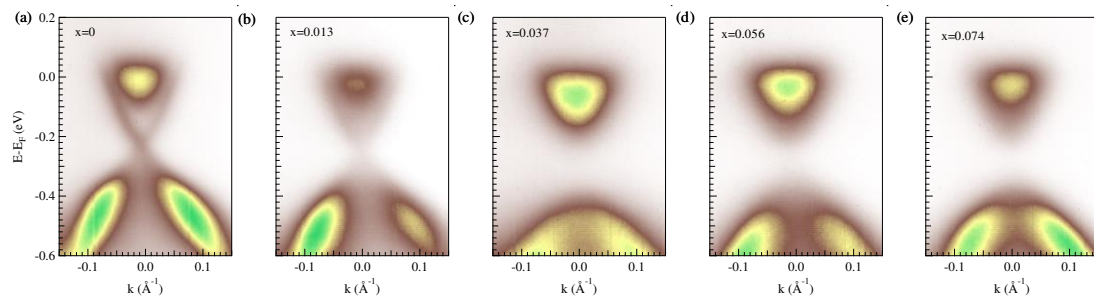


Figure 2

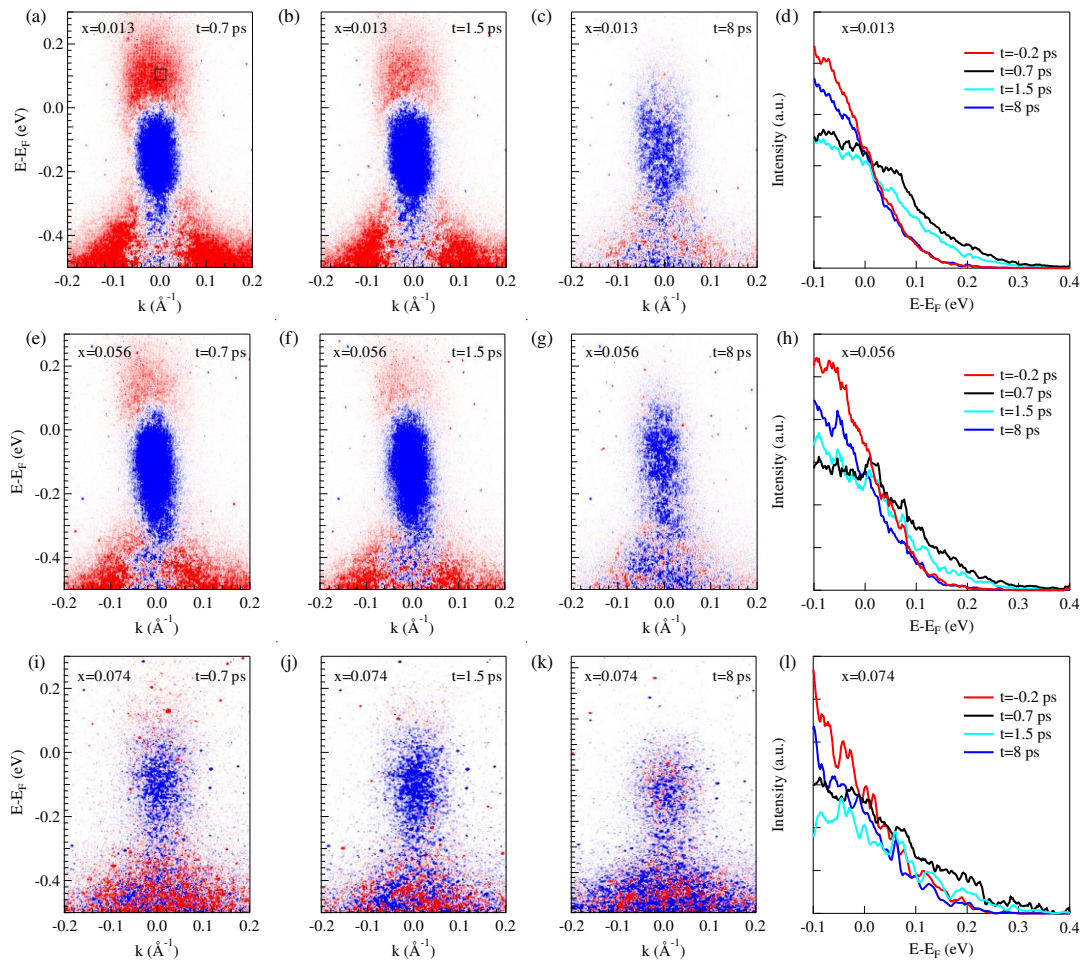


Figure 3

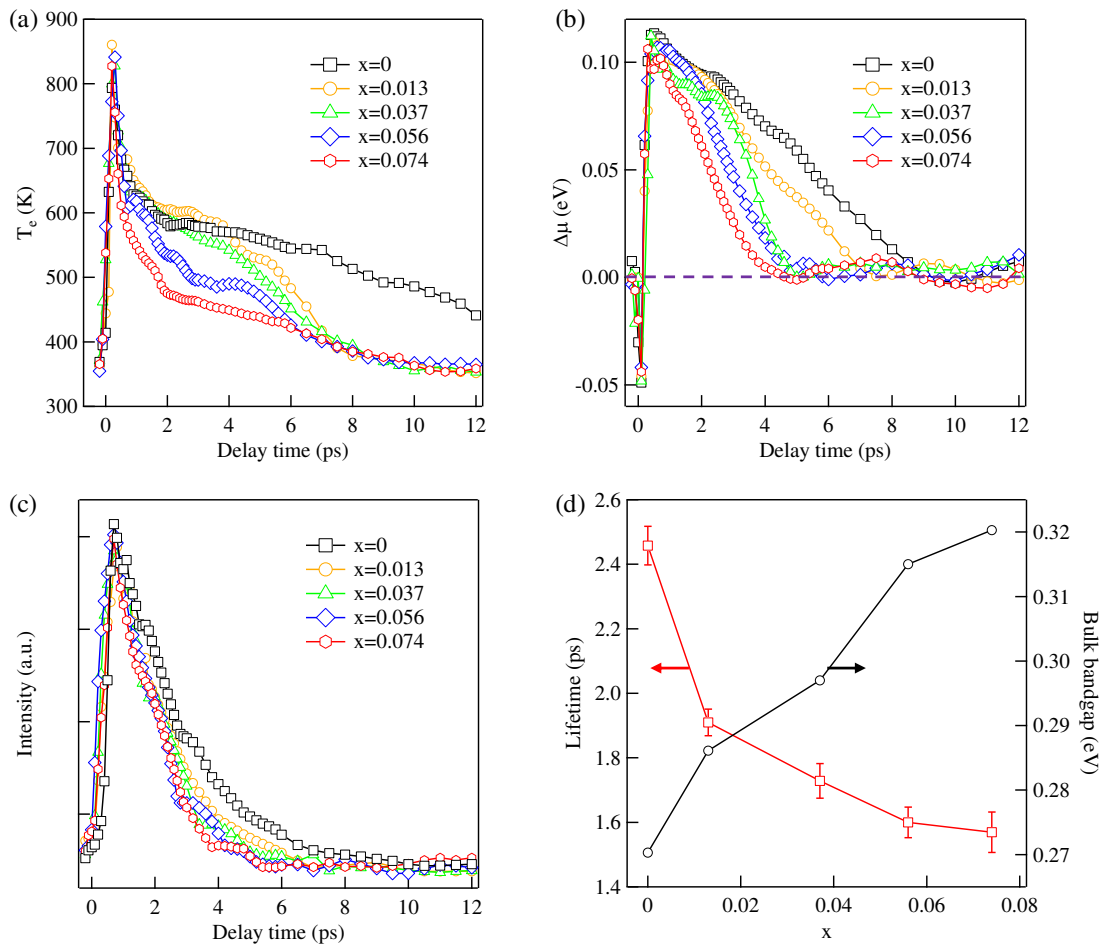


Figure 4

