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- Title

**Synthesis, structure and dielectric properties of a new family of phases,  $ABC_3O_{11}$ : A = La, Pr, Nd, Sm, Gd; B = Zr, Hf; C = Ta, Nb**

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Abstract:

Eight new phases with the general formula of  $ABC_3O_{11}$  with different rare-earth, (Zr, Hf), (Nb, Ta) combinations, have been prepared by solid-state reactions at a temperature range of 1200° – 1500° C. The new phases: LaHfTa, LaHfNb, LaZrNb, PrHfTa, NdHfTa, NdHfNb, SmHfTa and GdHfTa are characterised by X-ray and neutron diffraction data at room temperature and variable frequency impedance measurements. They are isostructural with  $LaZrTa_3O_{11}$  which consist of alternating single layers of  $UO_7$  pentagonal bipyramids and octahedra as shown by Rietveld refinement of X-ray and neutron powder diffraction data. Lattice parameters decrease with decreasing size of rare earth elements substitution at A-site and of all, Gd is the smallest rare earth that formed  $LaZrTa_3O_{11}$  analogues. Detailed attempts of attained and unattainable  $LaZrTa_3O_{11}$  analogues with different temperatures are included in this paper. All phases are highly insulating with temperature-independent bulk permittivities in the range 17 to 50; LaHfNb demonstrates the highest permittivity. Arrhenius plot shows that the activation energies are in the range 0.8 to 1.94 eV.

Keywords:  $U_3O_7$  pentagonal bipyramids; Rare-earth; Low permittivity; Rietveld refinement

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## Introduction

The crystal structure of  $\text{LaZrTa}_3\text{O}_{11}$  was first reported [1] in 1991 and was indexed on an orthorhombic unit cell,  $a = 10.890(3) \text{ \AA}$ ,  $b = 12.450(3) \text{ \AA}$  and  $c = 6.282(2) \text{ \AA}$ . However, its true symmetry was subsequently shown to be hexagonal with a geometrically-related unit cell and space group  $\text{P6}_322$  ( $N^\circ 182$ ) with  $a = 6.2824(2) \text{ \AA}$ ,  $c = 12.4469 \text{ \AA}$  and  $Z = 2$  [2].

$\text{LaZrTa}_3\text{O}_{11}$  is isostructural with  $\text{CaTa}_4\text{O}_{11}$  [3] which belongs to a series of compounds [4] with general formula  $\text{Me}_x(\text{Nb,Ta})_{3n+1}\text{O}_{8n+3}$ , e.g.  $n = 1$ :  $\text{CaTa}_4\text{O}_{11}$ ,  $\text{Na}_2\text{Nb}_4\text{O}_{11}$ ,  $\text{Na}_2\text{Ta}_4\text{O}_{11}$ ,  $\text{K}_2\text{Ta}_4\text{O}_{11}$ ,  $\text{Ag}_2\text{Nb}_4\text{O}_{11}$ ,  $\text{Ag}_2\text{Ta}_4\text{O}_{11}$  and “ $\text{Cu}_2\text{Ta}_4\text{O}_{11}$ ”;  $n = 2$ :  $\text{RETa}_7\text{O}_{19}$  where  $\text{RE} = \text{La} - \text{Eu}$ ,  $\text{Y}$  and  $\text{Bi}$ ; intergrowth of 1 and 2:  $\text{Cu}_5\text{Ta}_{11}\text{O}_{30}$ . These phases are structurally-related to  $\alpha\text{-U}_3\text{O}_8$  whose structure exhibits layers of  $\text{UO}_7$  pentagonal bipyramids sharing edges in the equatorial plane and apical oxygens [5]. These layers are repeated identically in the perpendicular direction. However, the  $\text{LaZrTa}_3\text{O}_{11}$  structure has an alternating single layer of  $(\text{Ta, Zr})\text{O}_7$  pentagonal bipyramids with layers containing  $(\text{Ta, Zr})\text{O}_6$  octahedra and  $\text{LaO}_8$  polyhedron; consecutive  $(\text{Ta, Zr})\text{O}_7$  layers are rotated by  $180^\circ$  relative to each other. La occupies an octahedral  $2c$  site and a 3:1 disordered distribution of Ta and Zr occupy  $6g$  pentagonal bipyramidal and  $2d$  octahedral sites. Figure 1 illustrates the layers of pentagonal bipyramids alternating with octahedra.  $\text{LaZrTa}_3\text{O}_{11}$  is highly insulating with electrical conductivity as low as  $3 \times 10^{-7} \text{ ohm}^{-1} \text{ cm}^{-1}$  at  $800 \text{ }^\circ\text{C}$ [1].

Given the ferroelectric and antiferroelectric nature of some structurally-related materials [6–9], namely  $\text{Ag}_2\text{Nb}_4\text{O}_{11}$ ,  $\text{Na}_2\text{Nb}_4\text{O}_{11}$  and its solid solution  $(\text{Na}_{1-x}\text{Ag}_x)\text{Nb}_4\text{O}_{11}$ , it has been of interest to seek other members of this structural family and to evaluate their electrical properties. In this work, the synthesis of phases isostructural with  $\text{LaZrTa}_3\text{O}_{11}$  with the general formula of  $\text{ABC}_3\text{O}_{11}$ ;  $\text{A} = \text{La}^{3+}$ ,  $\text{Pr}^{3+}$ ,  $\text{Nd}^{3+}$ ,  $\text{Sm}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{B} = \text{Zr}^{4+}$ ,  $\text{Hf}^{4+}$  and  $\text{C} = \text{Ta}^{5+}$ ,  $\text{Nb}^{5+}$  is reported. Structure determination by Rietveld refinement of several of the new phases were determined using X-ray and neutron powder diffraction data. A preliminary measurement of their electrical properties was made using impedance spectroscopy.

## Experimental

Starting materials were Ta<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>, ZrO<sub>2</sub>, HfO<sub>2</sub>, La<sub>2</sub>O<sub>3</sub>, Pr<sub>6</sub>O<sub>11</sub>, Nd<sub>2</sub>O<sub>3</sub> (99.9% pure, Stanford Materials) and Sm<sub>2</sub>O<sub>3</sub>, Gd<sub>2</sub>O<sub>3</sub> (99.9% pure, Sigma Aldrich). They were dried at 800° – 1000° C and stored under vacuum. Mixtures totalling 3 – 5 g were weighed, mixed with acetone using an agate mortar and pestle, dried, pressed into pellets and fired in Pt crucibles in the air in electric muffle furnaces. The mixtures were heated at different times and temperatures in the range 1000° – 1500° C for up to 3 days; intermittently. Pellets were removed from the furnace, crushed into a fine powder, repelleted and returned to the furnace. Reaction conditions were determined by trial and error, taking account of the refractory nature of the oxide reagents which meant that long period at high temperatures were required for the reaction to occur. Not any of the compositions, Table 1 are single phase products. The residual amounts of secondary phases remained, even after repeated firing. However, a relatively small amount of impurity is negligible for some compositions since they barely affect the results.

Phase analysis by X-ray powder diffraction (XRD) was carried out using a STOE Stadi P diffractometer, CuK $\alpha_1$  radiation ( $\lambda = 1.54056 \text{ \AA}$ ). NIST 640d Si was used as an external standard for line position calibration. Data were processed and compared against reference data in the ICDD's PDF-2 database using the WinX<sup>POW</sup> software package. Room temperature time-of-flight (ToF) neutron powder diffraction (ND) data were collected on the General Materials diffractometer, (GEM) at the ISIS facility, Rutherford Appleton Laboratory, RAL, Oxford for LaHfNb<sub>3</sub>O<sub>11</sub> powder in a vanadium can. Data were collected in the backscattering detector bank (50.07° - 74.71°), bank 4 with resolution  $\Delta Q/Q \sim 0.79 \times 10^{-2}$ . Rietveld refinement was performed using the EXPGUI interface for GSAS and both XRD and ND data.

For electrical measurements, cylindrical sintered pellets were coated with Pt paste electrodes on opposite faces and hardened by heating in the air at 800° C for 2 h. The samples, with electrodes attached, were placed in a ceramic compression jig inside a furnace controlled and measured to  $\pm 1^\circ \text{C}$ . Impedance spectroscopy (IS) measurements were performed using Hewlett Packard 4192A and Solartron 1260/1286 impedance analysers at frequencies 10 Hz - 1 MHz and 1 kHz - 1 MHz,

respectively. Impedance data were recorded between room temperature and 800° C in the air with 100 mV applied ac voltage. Data were corrected for overall pellet geometry and for the blank cell capacitance (jig correction) and analysed using the software program ZView.

## Results and Discussion

### Reaction Conditions and Products

Results of the attempted synthesis of new phases with different rare-earth – Zr, Hf – Nb, Ta combinations are summarised in Table 1. Eight new phases, in addition to  $\text{LaZrTa}_3\text{O}_{11}$ , were prepared and for each, the synthesis conditions are detailed. Attempted synthesis of a further 12 new phases was unsuccessful; details of the reaction conditions and products, in which there was no evidence of formation of a phase with the  $\text{LaZrTa}_3\text{O}_{11}$  structure, are given in Table 1(b).

The new phases were readily recognised by the similarity of their XRD patterns to that of  $\text{LaZrTa}_3\text{O}_{11}$ . Powder XRD patterns of these new phases are shown in Figures 2 and 3 for the Ta-based and Nb-based compositions, respectively. Indexed data for one analogue,  $\text{LaHfTa}_3\text{O}_{11}$ , are given in Table 2; data for all the analogue have been submitted for inclusion in the ICDD powder diffraction file. Unit cell data are summarised in Table 3.

The lattice parameters decreased on the substitution of La by smaller rare earth elements. Gd was the smallest rare earth that formed  $\text{LaZrTa}_3\text{O}_{11}$  analogue. With three cationic components in each phase: rare earth, Zr/Hf and Nb/Ta, there is a complex interplay between ion size, possible formation of  $\text{LaZrTa}_3\text{O}_{11}$  analogue and its thermal stability. It was found that a standard set of synthesis conditions could not be used for all the new phases. In general, higher reaction temperatures ( $\sim 1500^\circ\text{C}$ ) were required to obtain Ta-based phases compared to the reaction temperatures needed for the Nb analogues ( $1200^\circ - 1300^\circ\text{C}$ ), consistent with the lower reactivity of  $\text{Ta}_2\text{O}_5$  compared to  $\text{Nb}_2\text{O}_5$ . However, fewer Nb-based family members were prepared successfully, and those that did form had less thermal stability than the Ta analogues and decomposed at above  $\sim 1400^\circ\text{C}$ .

For Ta-based phases, LaHfTa<sub>3</sub>O<sub>11</sub> was the easiest to prepare and obtain as a phase with LaZrTa<sub>3</sub>O<sub>11</sub> structure, whereas the smaller rare earths require increasingly longer reaction times and lower temperatures. SmHfTa<sub>3</sub>O<sub>11</sub> and GdHfTa<sub>3</sub>O<sub>11</sub> started to form at 1200° C, as shown by the traces of LaZrTa<sub>3</sub>O<sub>11</sub> patterns in the XRD data. The impurity phases remained similar until 1400° C, although peak intensities decreased with time, Sm, Gd target phases were still not phase-pure and decomposed after increasing the temperature to 1500° C.

In the niobate systems, only three compositions produced LaZrTa<sub>3</sub>O<sub>11</sub> analogues: LaHfNb<sub>3</sub>O<sub>11</sub>, LaZrNb<sub>3</sub>O<sub>11</sub> and NdHfNb<sub>3</sub>O<sub>11</sub>, which are all formed at temperatures in the range 1150° – 1300° C. These niobates have lower thermal stability than their corresponding tantalates. LaHfNb<sub>3</sub>O<sub>11</sub> has an upper limit of stability at 1300° C before melting at 1350° C, NdHfNb<sub>3</sub>O<sub>11</sub> appears to have an upper limit of stability at ~ 1200° C and LaZrNb<sub>3</sub>O<sub>11</sub> appears to be on the edge of stability at 1150° C. The presence of LaZrTa<sub>3</sub>O<sub>11</sub> phase was detected after ~ 36 h but decomposed to LaNb<sub>3</sub>O<sub>9</sub> after another 12 h of reaction time.

### Structure Refinements

Rietveld refinement using XRD data was carried out for three compositions: LaHfTa<sub>3</sub>O<sub>11</sub>, PrHfTa<sub>3</sub>O<sub>11</sub> and NdHfTa<sub>3</sub>O<sub>11</sub>. For the fourth composition, LaHfNb<sub>3</sub>O<sub>11</sub>, combined refinement using XRD and ND data was conducted. For each refinement, the crystal structure of LaZrTa<sub>3</sub>O<sub>11</sub> was used as the starting model with La on octahedral 2c sites, Ta and Hf disordered in a fixed 1:3 ratio on octahedral 6g and 2d sites and oxygen on 4f, 6g and 12i sites. Total occupancies for all sites were fixed at unity; background and peak profile parameters were refined first using the Chebyshev function with ten terms for the background. This was followed by the single positional variable for the 6g site and simultaneous refinement of the isotropic thermal parameters,  $U_{iso}$ , of the cation sites until convergence, with negligible shifts in atomic variables.

The statistical measures and visual fits shown in Figures 4 – 7 were relatively good. For LaHfTa<sub>3</sub>O<sub>11</sub>,  $\chi^2 = 4.704$ , R factors < 7% and thermal, or  $U_{iso}$ , parameters were all refined to acceptable, positive

values. For PrHfTa<sub>3</sub>O<sub>11</sub> and NdHfTa<sub>3</sub>O<sub>11</sub>, good fits were also obtained with acceptable statistical measures. U<sub>iso</sub> values were positive for all atoms, indicating that the statistical occupancies of Ta, Hf over both sites is reasonable; with the  $\chi^2$  value for PrHfTa<sub>3</sub>O<sub>11</sub> at 7.475 compared to 5.996 for NdHfTa<sub>3</sub>O<sub>11</sub>. The final atomic coordinates and bond distances are presented in Tables 4 and 5. The unreacted HfO<sub>2</sub> peak in LaHfTa<sub>3</sub>O<sub>11</sub> and PrHfTa<sub>3</sub>O<sub>11</sub> was so small that it could barely affect the refinements with the weight fraction 0.165% and 0.047%, respectively. As for NdHfTa<sub>3</sub>O<sub>11</sub>, impurities of NdTa<sub>3</sub>O<sub>9</sub> and un-reacted HfO<sub>2</sub> are taken into account with weight fraction 1.236% and 0.774%, respectively; with the lattice parameters obtained for NdTa<sub>3</sub>O<sub>9</sub>: a = 3.9127(9) Å, c = 7.865(3) Å.

For LaHfNb<sub>3</sub>O<sub>11</sub>, the ToF ND data were collected at room temperature. The background, diffractometer constant DIFA, and peak profiles were refined for the ToF histogram with fixed values of lattice parameters and unit cell volume, which were then refined together with atomic positions and U<sub>iso</sub>, with initial values of 0.005 Å<sup>2</sup>. The whole process was repeated until convergence. Combined structure refinement gave a good visual fit and statistical measures (Figure 7, Table 4), although  $\chi^2$  was inconsiderably high, 5.878.

These new phases are isostructural with LaZrTa<sub>3</sub>O<sub>11</sub> which contains two layers of pentagonal bipyramids related by the 6<sub>3</sub> symmetry axis sharing edges. Both octahedral layers, i.e. (Ta/Nb, Hf/Zr)(1)O<sub>7</sub> and (Ta/Nb, Hf/Zr)(2)O<sub>6</sub> contain a disordered 3:1 ratio of Ta/Nb and Hf/Zr, where (Ta/Nb, Hf/Zr)(1)O<sub>7</sub> at z = 0 and ½ corner-share four of their oxygens, O(1) and O(2) with each other and have two apical oxygens, O(3) along the z-axis. Meanwhile, (Ta/Nb, Hf/Zr)-O<sub>6</sub> and REO<sub>6</sub> octahedra that are located at z = ¼ and ¾ are formed by the apical oxygens; three in the layer below and above (Ta/Nb, Hf/Zr)(1)-O<sub>7</sub> octahedra.

### Electrical Properties

Impedance measurements for several of the new phases over the range from room temperature to 800° C showed that are all excellent insulators. The permittivities of ABC<sub>3</sub>O<sub>11</sub> are in the range  $17 \leq \epsilon' \leq 50$  and are essentially frequency – independent over the range 10<sup>4</sup> to 10<sup>7</sup> Hz, Figure 8. The highest



permittivity is shown by LaHfNb<sub>3</sub>O<sub>11</sub>, ~ 50, whilst the lowest is SmHfTa<sub>3</sub>O<sub>11</sub>, ~ 17. The capacitance data of the tantalates from room temperature to 400 °C were independent of frequency and temperature. The conductivity of all the samples was too low to measure at room temperature with the available instrumentation, indicative of highly insulating materials with  $R \gg 10^9 \Omega\text{cm}$ . At higher temperatures, the conductivity data were obtained from IS measurements and are shown as Arrhenius plots in Figure 9. The highest conductivity was found for LaHfNb<sub>3</sub>O<sub>11</sub>, ~  $10^{-5} \Omega^{-1}\text{cm}^{-1}$  at 1000 K. The activation energy,  $E_a$  for conduction obtained from the gradient of the Arrhenius plots was in the range 0.8 and 1.94 eV.

Figure 10 shows the bulk permittivity,  $\epsilon_\infty$ , of the tantalates phases at room temperature plotted against ionic radii of rare earth elements. The bulk permittivity decreased with decreasing ionic radius; replacing Ta<sup>5+</sup> by Nb<sup>5+</sup> in one case increased the permittivity significantly.

## Conclusions

A new family of phases based on LaZrTa<sub>3</sub>O<sub>11</sub> with the general formula of ABC<sub>3</sub>O<sub>11</sub>: A = La, Pr, Nd, Sm; B = Zr, Hf and C = Ta, Nb, but not all combinations, has been prepared by solid state reaction. The successful elements combinations are LaHfTa, LaHfNb, LaZrNb, PrHfTa, NdHfTa, NdHfNb, NdZrNb, SmHfTa and GdHfTa. However, none were completely in phase pure as ATa<sub>3</sub>O<sub>9</sub> and un-reacted raw materials were often present. Similar to LaZrTa<sub>3</sub>O<sub>11</sub>, all new phases were indexed on the hexagonal space group P6<sub>3</sub>22; unit cell parameters and cell volume decreased with decreasing size of A.

The temperatures necessary for the sample synthesis depend on the size of A and C. Most AHfTa combinations were prepared at 1500° C except for SmHfTa and GdHfTa which were prepared at 1400° C and decomposed at 1500° C. AHfNb<sub>3</sub>O<sub>11</sub> compounds were prepared at 1150° – 1300° C, which was significantly lower than required to synthesise the tantalates. With smaller rare earth ions, it was generally found to be increasingly difficult to eliminate secondary phases. Gd appears to be the smallest rare earth that allows formation of LaZrTa<sub>3</sub>O<sub>11</sub> analogue phases.

The crystal structures have been confirmed by Rietveld refinement using both XRD and NPD data in space group  $P6_322$ . All four structures that were refined contain a random distribution of (Ta, Nb) and Hf in the ratio 3:1 over the two sites of edge-sharing octahedra.

All samples were insulating at room temperature and higher temperatures were required to obtain conductivity data. The activation energies are in the range 0.8 – 1.74 eV, consistent with the insulating nature of the materials. The capacitances of all the samples, in the form of sintered pellets, are frequency – independent at low temperatures, and gradually increased as the temperature increased. Of all the samples,  $\text{LaHfNb}_3\text{O}_{11}$  exhibits the highest permittivity  $\sim 50$ . There was no evidence of any ferroic transitions, but it would be interesting to measure the low temperature electrical properties of  $\text{LaHfNb}_3\text{O}_{11}$ .

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X-ray powder diffraction data for LaHfTa<sub>3</sub>O<sub>11</sub> (LHT)

<i>2θ(obs)</i>	<b>h k l</b>	<i>2θ(calc)</i>	<b>Obs - calc</b>	<b>Int.</b>	<b>d (obs) Å</b>	<b>d (calc) Å</b>
14.240	0 0 2	14.229	0.0108	31.0	6.2149	6.2196
16.280	1 0 0	16.287	-0.0066	49.0	5.4402	5.4380
17.797	1 0 1	17.787	0.0106	6.6	4.9798	4.9827
21.709	1 0 2	21.691	0.0187	19.8	4.0904	4.0939
27.050	1 0 3	27.020	0.0296	4.0	3.2937	3.2973
28.728	0 0 4	28.683	0.0454	57.0	3.1050	3.1098
29.328	1 1 1	29.315	0.0131	89.6	3.0428	3.0442
31.910	1 1 2	31.904	0.0054	100.0	2.8023	2.8028
33.170	1 0 4	33.159	0.0119	12.2	2.6986	2.6996
35.868	1 1 3	35.847	0.0217	31.8	2.5016	2.5030
39.613	2 0 3	39.605	0.0076	2.1	2.2733	2.2737
40.819	1 1 4	40.808	0.0103	2.9	2.2089	2.2094
43.638	0 0 6	43.622	0.0157	4.8	2.0725	2.0732
44.029	2 1 0	44.021	0.0085	6.2	2.0550	2.0554
44.647	2 1 1	44.649	-0.0018	8.4	2.0280	2.0279
46.549	1 1 5	46.538	0.0113	27.5	1.9494	1.9499
46.879	1 0 6	46.861	0.0185	3.1	1.9365	1.9372
49.459	2 1 3	49.453	0.0055	4.2	1.8413	1.8415
50.299	3 0 0	50.295	0.0034	32.8	1.8126	1.8127
52.549	3 0 2	52.544	0.0051	4.9	1.7401	1.7403
52.880	1 1 6	52.878	0.0015	31.0	1.7300	1.7301
53.389	2 1 4	53.389	-0.0000	6.3	1.7147	1.7147
55.728	2 0 6	55.710	0.0181	6.5	1.6481	1.6436
58.174	2 1 5	58.173	0.0016	4.0	1.5845	1.5846
58.938	3 0 4	58.928	0.0104	35.6	1.5658	1.5651
59.299	2 2 1	59.285	0.0140	11.6	1.5571	1.5575
59.749	1 1 7	59.747	0.0016	13.4	1.5455	1.5455
60.799	2 2 2	60.806	-0.0076	14.1	1.5223	1.5221
61.400	3 1 0	61.425	-0.0245	5.1	1.5038	1.5032
61.964	3 1 1	61.924	0.0399	4.1	1.4954	1.4973
63.289	2 2 3	63.294	-0.0046	7.9	1.4632	1.4631
63.709	2 1 6	63.705	0.0043	3.1	1.4595	1.4596
65.869	3 1 3	65.840	0.0294	2.1	1.4168	1.4174
68.749	3 0 6	68.732	0.0174	2.7	1.3643	1.3646
69.168	3 1 4	69.170	-0.0013	5.0	1.3571	1.3571
69.928	2 1 7	69.922	0.0060	2.8	1.3442	1.3443
70.932	4 0 2	70.898	0.0332	6.0	1.3276	1.3281
73.344	3 1 5	73.347	-0.0023	2.2	1.2898	1.2897
75.020	1 1 9	75.025	-0.0054	4.9	1.2651	1.2650
75.979	2 2 6	75.975	0.0039	8.8	1.2515	1.2515
76.369	4 0 4	76.396	-0.0269	2.6	1.2450	1.2457
76.790	2 1 8	76.806	-0.0153	2.8	1.2403	1.2400
78.050	3 2 2	78.062	-0.0120	4.4	1.2234	1.2232
78.889	1 0 10	78.876	0.0132	1.8	1.2124	1.2126

Crystal symmetry: hexagonal

Refined cell:  $a = 6.28319(6)$  Å,  $\alpha = 90.0^\circ$ , ( $a = b$ ) $c = 12.4358(2)$  Å,  $\gamma = 120.0^\circ$ , $V = 425.173(8)$  Å<sup>3</sup>



X-ray powder diffraction data for NdHfTa<sub>3</sub>O<sub>11</sub> (NHT)

<i>2θ(obs)</i>	<i>h k l</i>	<i>2θ(calc)</i>	Obs - calc	Int.	<i>d (obs) Å</i>	<i>d (calc) Å</i>
14.285	0 0 2	14.279	0.0063	29.2	6.1952	6.1979
16.344	1 0 0	16.346	-0.0019	48.1	5.4191	5.4185
17.852	1 0 1	17.851	0.0012	6.9	4.9645	4.9649
21.768	1 0 2	21.769	-0.0009	20.2	4.0795	4.0793
27.109	1 0 3	27.118	-0.0085	4.5	3.2866	3.2856
28.790	0 0 4	28.785	0.0047	53.7	3.0985	3.0990
29.418	1 1 1	29.423	-0.0049	84.9	3.0337	3.0332
32.019	1 1 2	32.022	-0.0032	100.0	2.7930	2.7828
33.281	1 0 4	33.279	0.0018	13.7	2.6899	2.6901
35.980	1 1 3	35.979	0.0005	31.6	2.4941	2.4941
39.770	2 0 3	39.753	0.0171	3.2	2.2647	2.2656
40.963	1 1 4	40.960	0.0033	3.9	2.2014	2.2016
43.784	0 0 6	43.783	0.0006	5.8	2.0660	2.0660
44.173	2 1 0	44.188	-0.0144	7.2	2.0486	2.0480
44.359	2 0 4	44.377	-0.0188	4.6	2.0405	2.0397
44.809	2 1 1	44.819	-0.0100	9.8	2.0210	2.0206
46.702	2 1 2	46.673	0.0294	28.5	1.9434	1.9446
47.030	1 0 6	47.035	-0.0058	3.9	1.9306	1.9304
49.638	2 1 3	49.643	-0.0050	4.6	1.8351	1.8350
50.477	3 0 0	50.490	-0.0130	34.1	1.8066	1.8062
52.743	3 0 2	52.748	-0.0045	5.3	1.7342	1.7340
53.079	1 1 6	53.080	-0.0007	33.5	1.7240	1.7240
53.585	2 1 4	53.595	-0.0099	7.4	1.7089	1.7086
55.914	2 0 6	55.924	-0.0104	7.7	1.6431	1.6428
58.395	2 1 5	58.400	-0.0053	4.8	1.5791	1.5789
59.150	3 0 4	59.160	-0.0099	37.6	1.5607	1.5605
59.548	2 2 1	59.520	0.0276	13.5	1.5512	1.5519
59.977	1 1 7	59.980	-0.0027	14.0	1.5411	1.5411
61.038	2 2 2	61.049	-0.0112	16.1	1.5169	1.5166
61.655	3 1 0	61.671	-0.0153	6.2	1.5031	1.5028
62.195	3 1 1	62.172	0.0230	5.4	1.4914	1.4919
63.542	2 2 3	63.548	-0.0055	8.8	1.4630	1.4629
63.954	2 1 6	63.958	-0.0043	4.0	1.4546	1.4545
66.090	3 1 3	66.107	-0.0171	2.8	1.4126	1.4123
69.006	3 0 6	69.011	-0.0057	3.7	1.3599	1.3598
69.434	3 1 4	69.454	-0.0193	6.3	1.3525	1.3522
70.202	2 1 7	70.207	-0.0050	3.9	1.3396	1.3395
71.211	4 0 2	71.193	0.0181	6.6	1.3231	1.3234
73.639	3 1 5	73.653	-0.0134	3.2	1.2853	1.2851
75.337	1 1 9	75.335	0.0015	6.4	1.2605	1.2606
76.287	2 2 6	76.295	-0.0075	10.0	1.2472	1.2471
76.728	4 0 4	76.720	0.0082	3.8	1.2411	1.2412
77.104	2 1 8	77.128	-0.0242	3.5	1.2360	1.2357
78.384	3 2 2	78.397	-0.0136	5.8	1.2190	1.2188
79.203	1 0 10	79.207	-0.0042	2.8	1.2084	1.2084

Crystal symmetry: hexagonal

Refined cell:  $a = 6.2578(1) \text{ \AA}$ ,  $\alpha = 90.0^\circ$ , ( $a = b$ ) $c = 12.3938(3) \text{ \AA}$ ,  $\gamma = 120.0^\circ$ , $V = 420.32(2) \text{ \AA}^3$

X-ray powder diffraction data for SmHfTa<sub>3</sub>O<sub>11</sub> (SHT)

<i>2θ(obs)</i>	<i>h k l</i>	<i>2θ(calc)</i>	Obs - calc	Int.	<i>d (obs) Å</i>	<i>d (calc) Å</i>
14.305	0 0 2	14.325	-0.0200	28.6	6.1867	6.1782
16.380	1 0 0	16.398	-0.0182	47.9	5.4073	5.4014
17.894	1 0 1	17.908	-0.0142	6.9	4.9531	4.9492
21.841	1 0 2	21.839	0.0021	19.7	4.0660	4.0664
27.188	1 0 3	27.206	-0.0181	6.0	3.2773	3.2752
28.897	0 0 4	28.880	0.0175	52.9	3.0873	3.0891
29.530	1 1 1	29.518	0.0112	82.7	3.0225	3.0237
32.140	1 1 2	32.126	0.0138	100.0	2.7828	2.7839
33.406	1 0 4	33.388	0.0176	14.4	2.6801	2.6815
36.114	1 1 3	36.097	0.0163	32.5	2.4852	2.4862
36.280	2 0 2	36.273	0.0069	15.3	2.4741	2.4746
40.083	1 0 5	40.092	-0.0097	4.1	2.2477	2.2472
41.112	1 1 4	41.096	0.0164	4.7	2.1938	2.1946
43.952	0 0 6	43.930	0.0217	6.5	2.0584	2.0594
44.344	2 1 0	44.335	0.0085	8.1	2.0411	2.0415
44.541	2 0 4	44.526	0.0150	5.4	2.0326	2.0332
44.980	2 1 1	44.969	0.0108	11.1	2.0138	2.0142
46.878	2 1 2	46.829	0.0491	31.1	1.9365	1.9384
47.210	1 0 6	4.195	0.0151	4.7	1.9237	1.9243
49.815	2 1 3	49.811	0.0043	5.9	1.8290	1.8292
50.667	3 0 0	50.661	0.0061	37.7	1.8002	1.8005
52.938	3 0 2	52.928	0.0100	6.4	1.7282	1.7285
53.274	1 1 6	53.262	0.0121	34.2	1.7181	1.7185
53.784	2 1 4	53.779	0.0052	8.9	1.7030	1.7032
56.132	2 0 6	56.118	0.0138	9.1	1.6372	1.6376
58.614	2 1 5	58.604	0.0101	5.9	1.5737	1.5739
59.373	3 0 4	59.366	0.0071	40.0	1.5554	1.5555
59.780	2 2 1	59.728	0.0524	15.7	1.5457	1.5470
60.201	1 1 7	60.191	0.0098	15.7	1.5359	1.5362
61.271	2 2 2	61.263	0.0078	16.7	1.5117	1.5118
61.899	3 1 0	61.887	0.0120	7.6	1.4978	1.4981
62.429	3 1 1	62.391	0.0375	7.0	1.4864	1.4872
63.779	2 2 3	63.773	0.0063	10.5	1.4581	1.4582
64.182	2 1 6	64.186	-0.0041	5.5	1.4499	1.4498
66.336	3 1 3	66.344	-0.0074	4.1	1.4080	1.4078
69.269	3 0 6	69.262	0.0073	4.9	1.3553	1.3555
69.695	3 1 4	69.706	-0.0110	7.5	1.3481	1.3479
70.479	2 1 7	70.464	0.0154	5.3	1.3350	1.3353
71.489	4 0 2	71.453	0.0362	9.2	1.3186	1.3192
73.924	3 1 5	73.925	-0.0012	4.7	1.2811	1.2811
75.648	1 1 9	75.618	0.0297	7.5	1.2561	1.2565
76.595	2 2 6	76.581	0.0136	11.5	1.2429	1.2431
77.018	4 0 4	77.008	0.0096	5.7	1.2372	1.2373
77.427	2 1 8	77.419	0.0073	5.4	1.2316	1.2317
78.709	3 2 2	78.694	0.0150	7.4	1.2148	1.2150

Crystal symmetry: hexagonal

Refined cell:  $a = 6.23885(8) \text{ \AA}$ ,  $\alpha = 90.0^\circ$ , ( $a = b$ ) $c = 12.3569(2) \text{ \AA}$ ,  $\gamma = 120.0^\circ$ , $V = 416.53(1) \text{ \AA}^3$

X-ray powder diffraction data for GdHfTa<sub>3</sub>O<sub>11</sub> (GHT)

<i>2θ(obs)</i>	<i>h k l</i>	<i>2θ(calc)</i>	Obs - calc	Int.	<i>d (obs) Å</i>	<i>d (calc) Å</i>
14.381	0 0 2	14.343	0.0376	27.8	6.1543	6.1703
16.452	1 0 0	16.417	0.0351	48.8	5.3836	5.3950
17.951	1 0 1	17.930	0.0209	7.3	4.9376	4.9433
21.891	1 0 2	21.866	0.0249	19.3	4.0569	4.0615
27.208	1 0 3	27.240	-0.0322	6.0	3.2750	3.2712
28.924	0 0 4	28.917	0.0074	55.3	3.0844	3.0852
29.568	1 1 1	29.554	0.0138	85.4	3.0187	3.0201
32.177	1 1 2	32.165	0.0113	100.0	2.7797	2.7806
33.437	1 0 4	33.431	0.0062	14.3	2.6777	2.6782
34.043	2 0 1	33.992	0.0512	5.1	2.6314	2.6353
36.153	1 1 3	36.143	0.0104	32.4	2.4825	2.4832
40.225	1 0 5	40.145	0.0805	4.8	2.2401	2.2444
41.155	1 1 4	41.149	0.0062	5.2	2.1916	2.1919
43.981	0 0 6	43.989	-0.0083	7.5	2.0571	2.0568
44.400	2 1 0	44.390	0.0102	8.8	2.0387	2.0391
45.031	2 1 1	45.025	0.0063	12.0	2.0116	2.0118
46.924	2 1 2	46.888	0.0363	33.0	1.9347	1.9361
49.873	2 1 3	49.874	-0.0016	6.8	1.8270	1.8270
50.732	3 0 0	50.724	0.0080	37.2	1.7981	1.7983
52.942	3 0 2	52.995	-0.0534	7.4	1.7281	1.7265
53.335	1 1 6	53.334	0.0015	33.8	1.7163	1.7163
53.849	2 1 4	53.849	0.0004	9.0	1.7011	1.7011
56.191	2 0 6	56.194	-0.0033	9.3	1.6357	1.6356
58.675	2 1 5	58.682	-0.0073	7.7	1.5722	1.5720
59.452	3 0 4	59.444	0.0079	39.5	1.5535	1.5537
59.851	2 2 1	59.805	0.0460	17.5	1.5441	1.5452
60.267	1 1 7	60.274	-0.0066	16.0	1.5344	1.5342
61.354	2 2 2	61.343	0.0107	17.4	1.5098	1.5101
61.976	3 1 0	61.968	0.0083	8.0	1.4961	1.4963
62.507	3 1 1	62.473	0.0349	7.8	1.4847	1.4854
63.860	2 2 3	63.858	0.0028	10.9	1.4565	1.4565
64.286	2 1 6	64.274	0.0119	6.4	1.4478	1.4481
66.451	3 1 3	66.432	0.0183	5.1	1.4058	1.4062
69.359	3 0 6	69.358	0.0010	5.8	1.3538	1.3538
69.781	3 1 4	69.801	-0.0193	7.9	1.3466	1.3463
70.570	2 1 7	70.563	0.0067	5.9	1.3335	1.3336
71.584	4 0 2	71.550	0.0343	9.0	1.3171	1.3176
74.034	3 1 5	74.029	0.0048	5.7	1.2795	1.2795
75.717	1 1 9	75.730	-0.0124	8.4	1.2551	1.2550
76.685	2 2 6	76.691	-0.0057	11.5	1.2417	1.2416
77.140	4 0 4	77.116	0.0234	6.2	1.2355	1.2358
78.826	3 2 2	78.804	0.0220	7.7	1.2132	1.2135
79.587	1 0 10	79.631	-0.0444	4.8	1.2036	1.2030

Crystal symmetry: hexagonal

Refined cell:  $a = 6.2293(1) \text{ \AA}$ ,  $\alpha = 90.0^\circ$ , ( $a = b$ ) $c = 12.3421(3) \text{ \AA}$ ,  $\gamma = 120.0^\circ$ , $V = 414.77(2) \text{ \AA}^3$



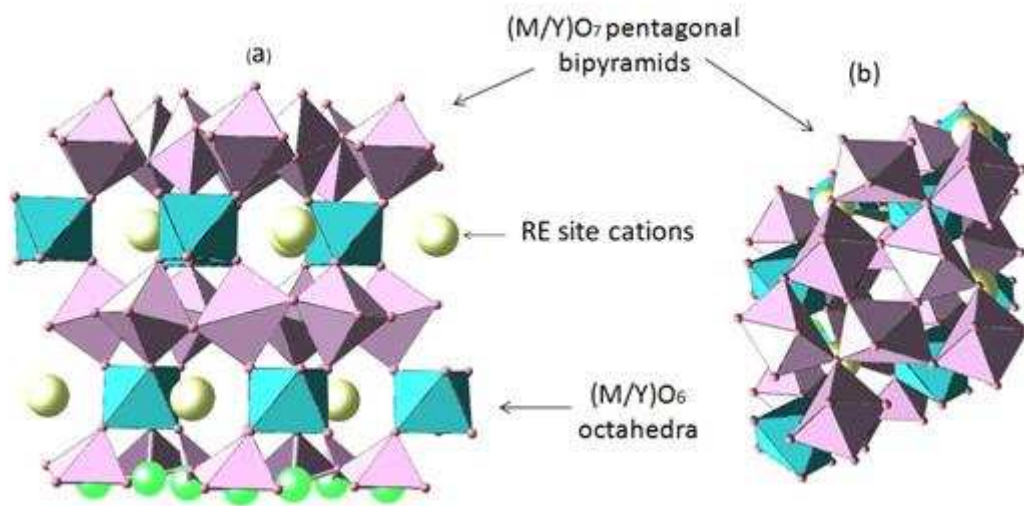


Figure 1: (a) Stacking sequence of single layers of pentagonal bipyramids alternate with single layers of octahedral layers and (b) layer of pentagonal bipyramids at  $z = 0$ .

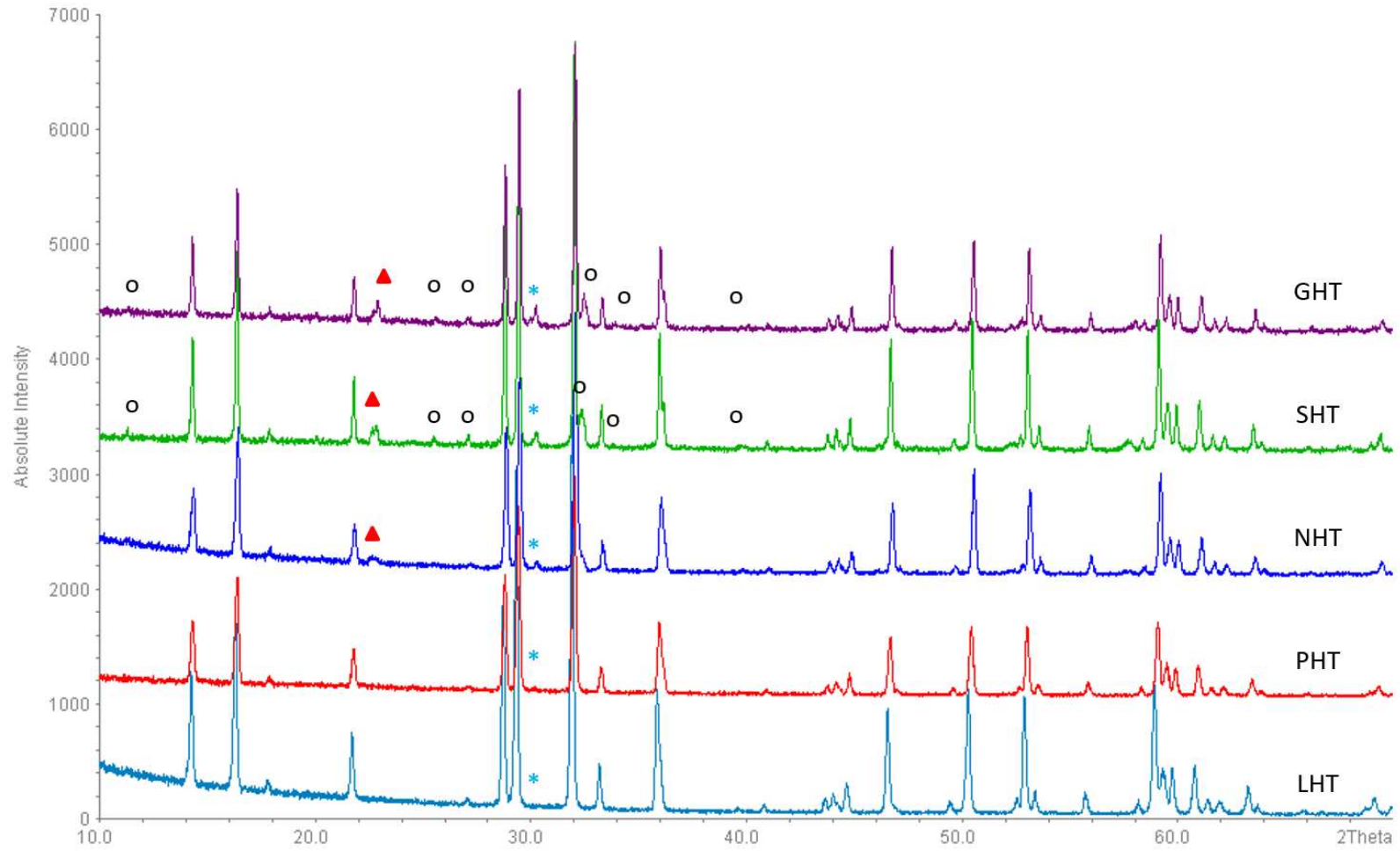


Figure 2: XRD profiles of AHfTa<sub>3</sub>O<sub>11</sub> compositions, (\*) indicates un-reacted HfO<sub>2</sub>, (▲) indicates un-reacted Ta<sub>2</sub>O<sub>5</sub> and (o) indicates ATa<sub>3</sub>O<sub>9</sub>.

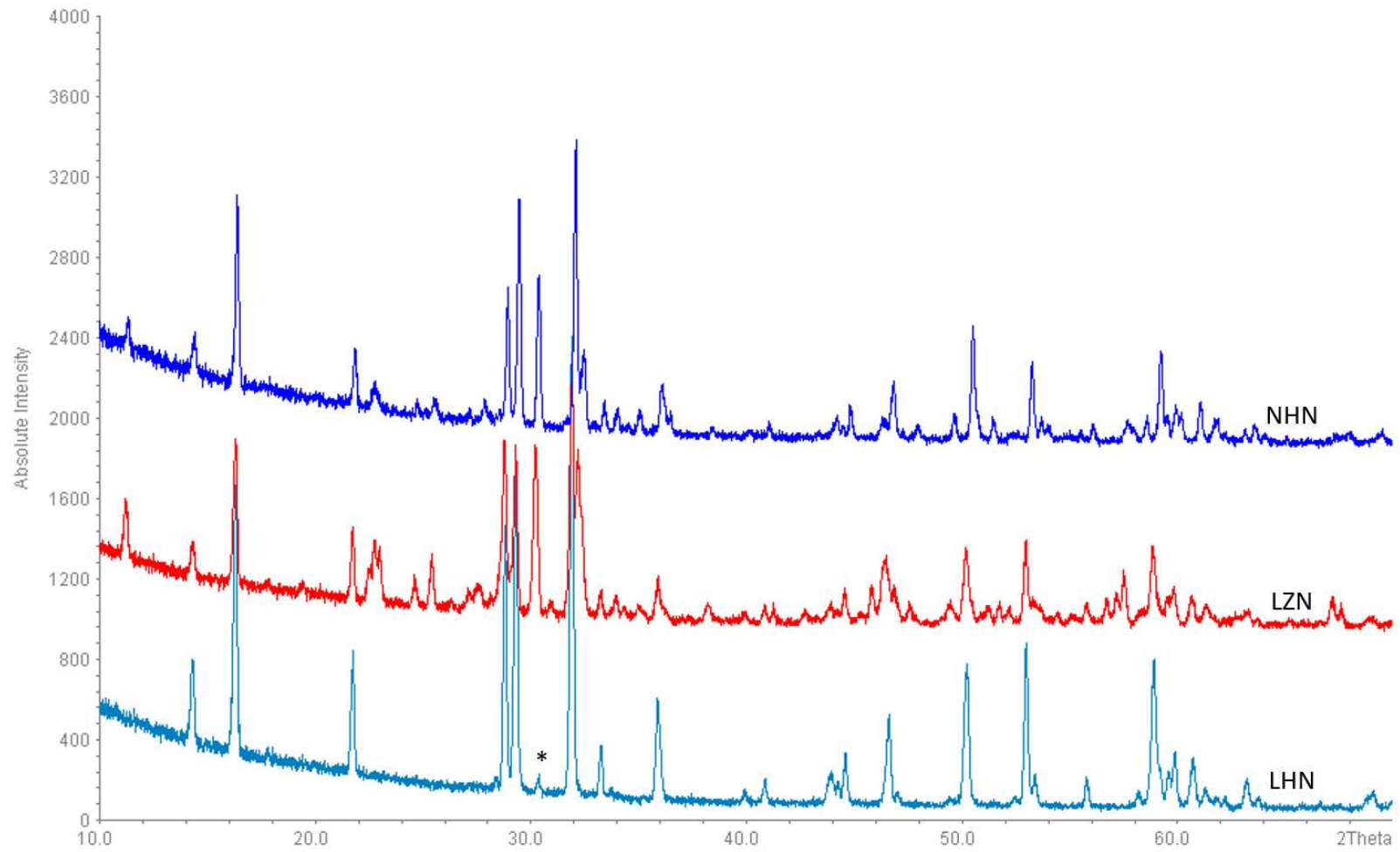


Figure 3: XRD profiles of  $ABNb_3O_{11}$  compositions, LHN contains un-reacted  $HfO_2$  (\*), whereas LZN and NHN are mixed phases.

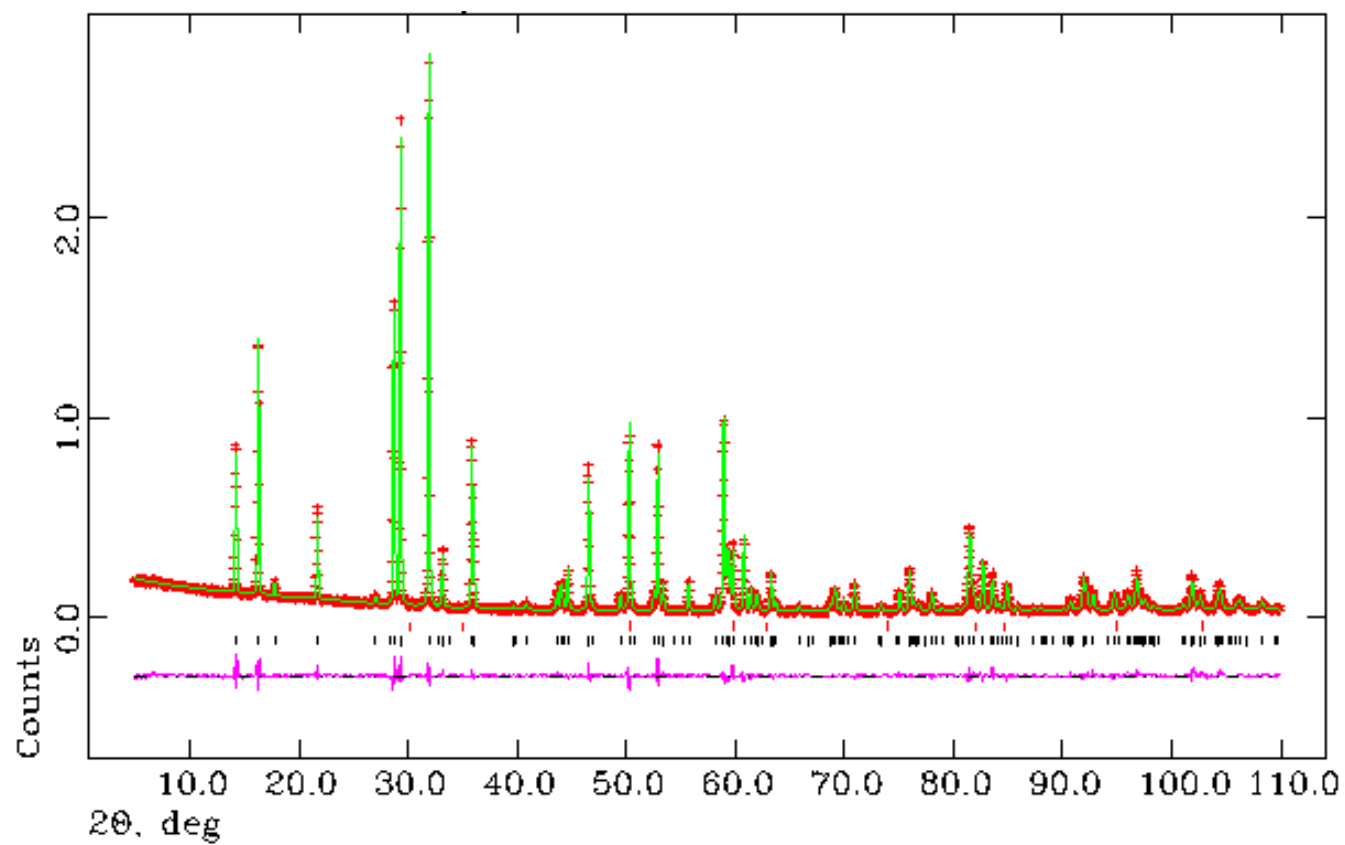


Figure 4: The observed (+), calculated (-) and difference (-) profiles from Rietveld refinement of  $\text{LaHfTa}_3\text{O}_{11}$  with a disordered arrangement of Ta, Hf in the ratio 3:1. Ticks marks represent Bragg peak positions.

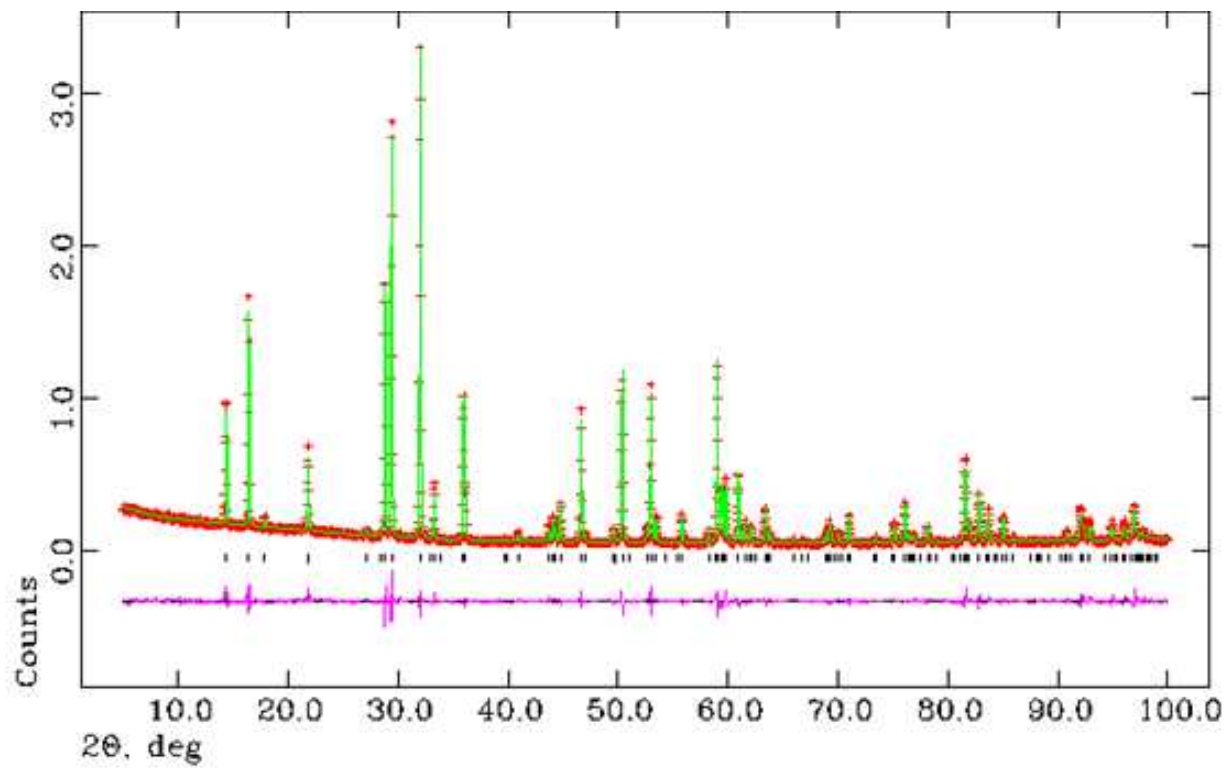


Figure 5: The observed (+), calculated (-) and difference (-) profiles from Rietveld refinement of  $\text{PrHfTa}_3\text{O}_{11}$  with disordered arrangement of Ta, Hf in the ratio 3:1. Ticks marks represent Bragg peak positions.

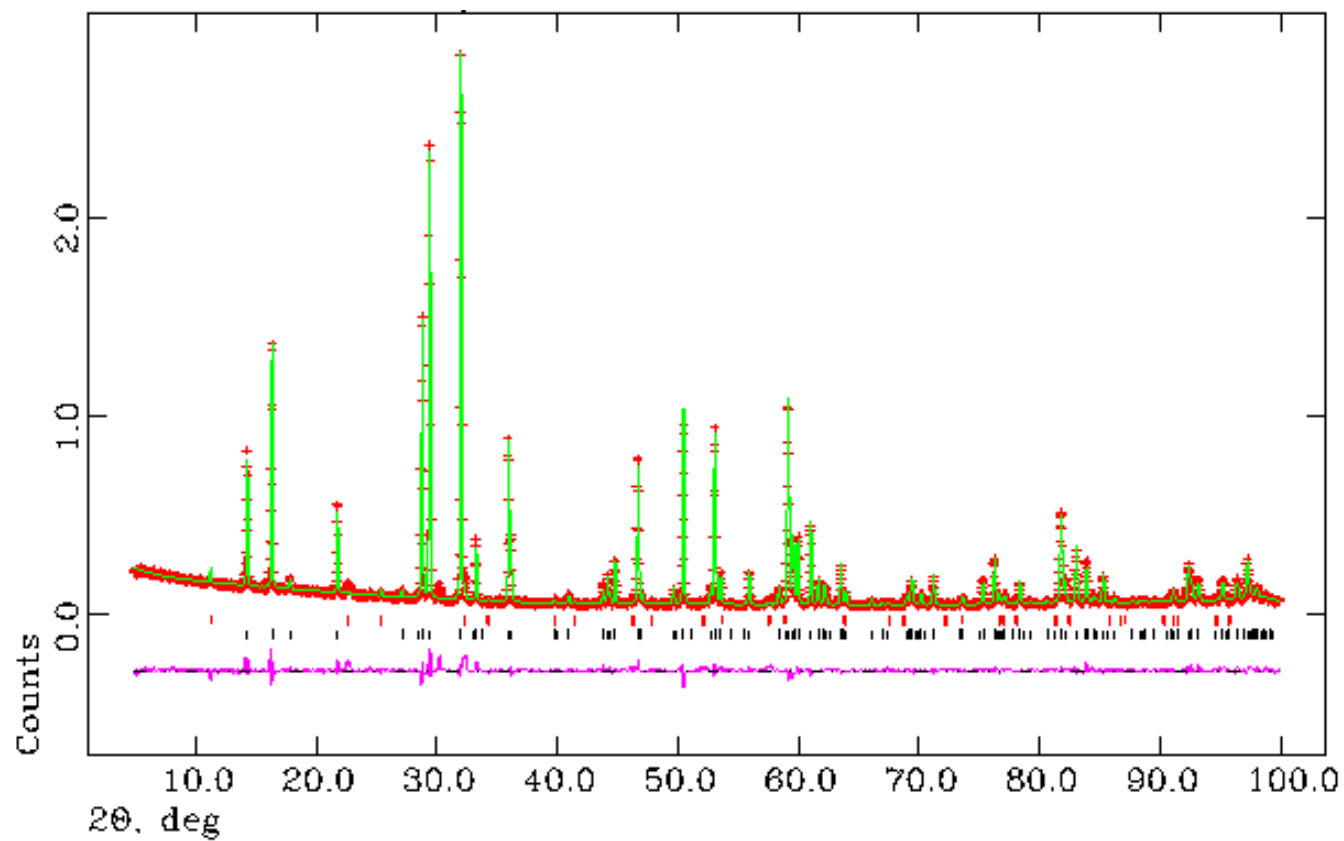


Figure 6: The observed (+), calculated (-) and difference (-) profiles from Rietveld refinement of  $\text{NdHfTa}_3\text{O}_{11}$  with disordered arrangement of Ta, Hf in the ratio 3:1. Ticks marks represent Bragg peak positions.

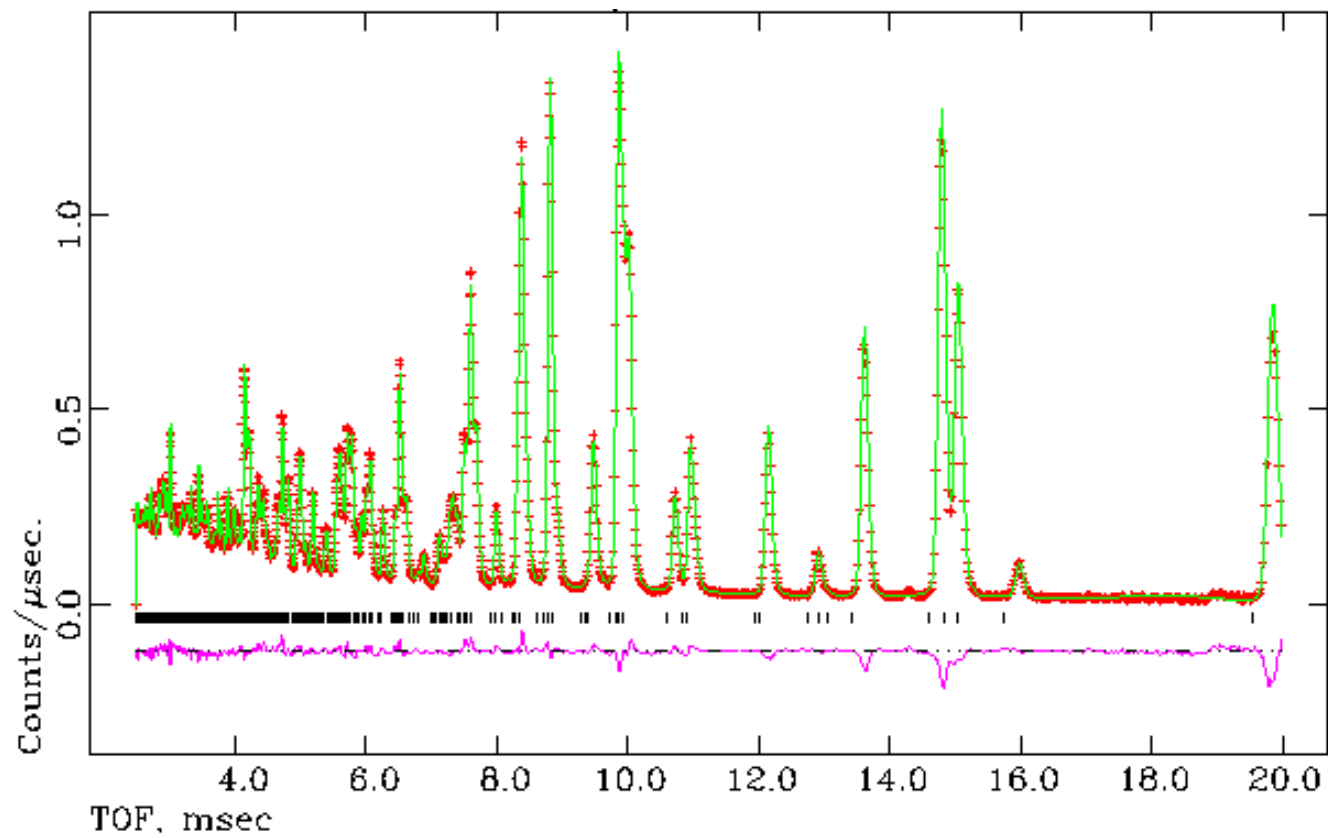


Figure 7: Observed (+), calculated (-) and difference (-) ND profiles at room temperature using constant wavelength ND data of  $\text{LaHfNb}_3\text{O}_{11}$ . Tick marks represent Bragg peak positions.

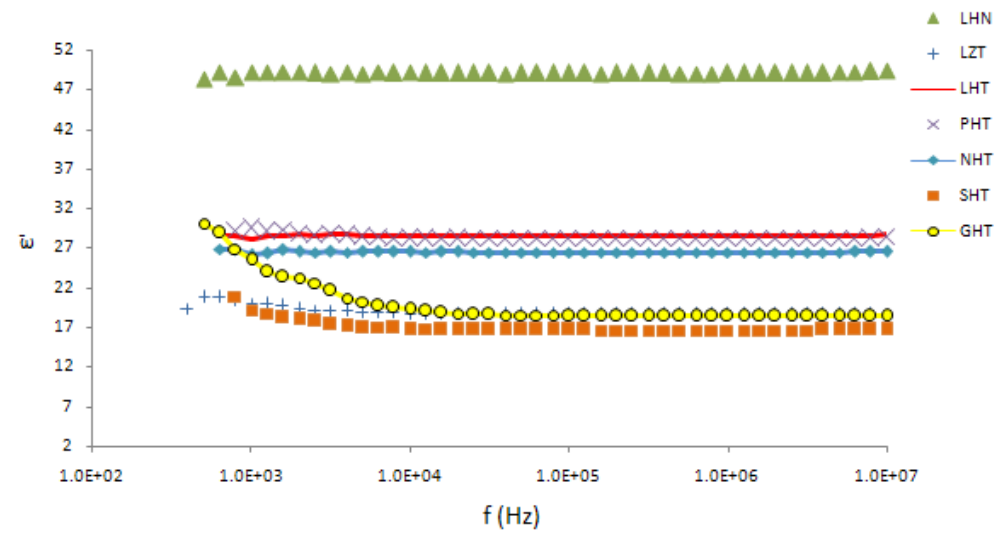


Figure 8: Permittivity,  $\epsilon'$ , at room temperature for  $ABC_3O_{11}$  phases.



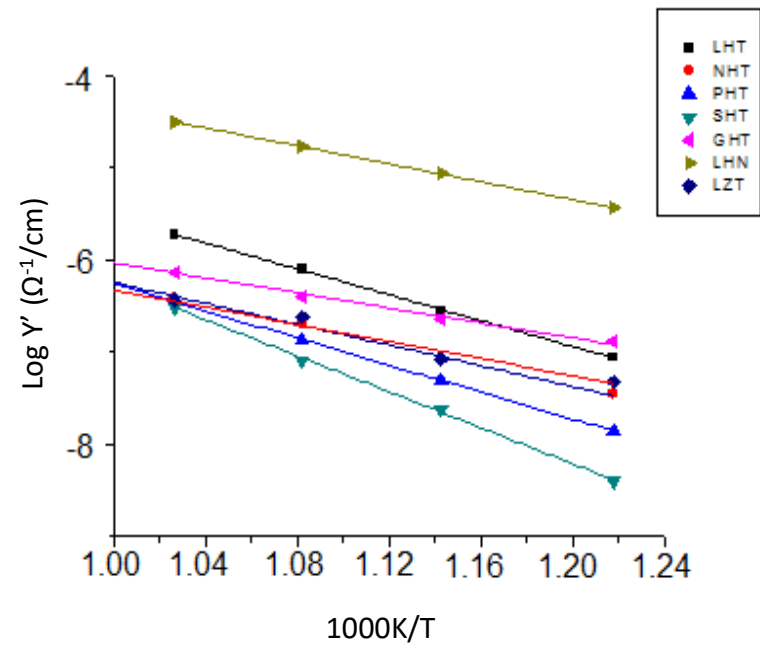


Figure 9: Conductivity data for the new phases.

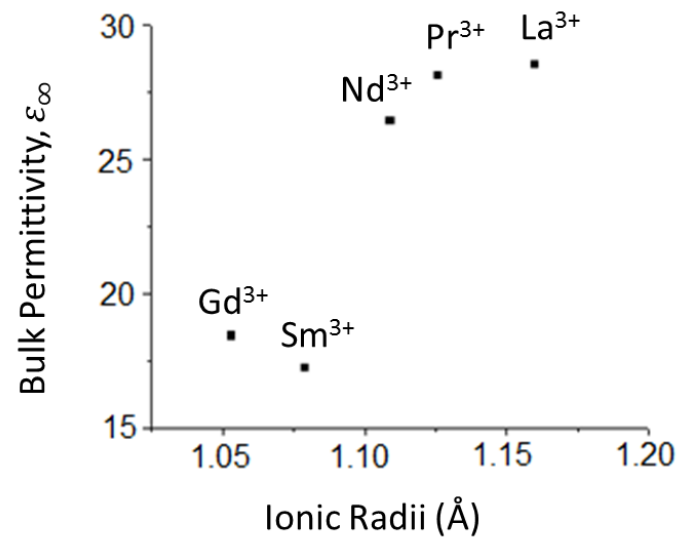


Figure 10: Bulk permittivity of the tantalates phases at room temperature vs. ionic radii of rare earth elements

No.	Compositions	T (°C)	Time (h)	Phases Present
1.	LaZrTa <sub>3</sub> O <sub>11</sub> (LZT)	1500	12	LZT, LaTa <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
		1500	24	LZT, LaTa <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
		1500	36	LZT, LaTa <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
		1500	48	LZT, LaTa <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
		1500	72	LZT, trace of LaTa <sub>3</sub> O <sub>9</sub> and ZrO <sub>2</sub>
2.	LaHfTa <sub>3</sub> O <sub>11</sub> (LHT)	1500	24	LHT, HfO <sub>2</sub>
		1500	48	LHT, trace of HfO <sub>2</sub>
3.	LaHfNb <sub>3</sub> O <sub>11</sub> (LHN)	1150	24	LHN, Nb <sub>2</sub> O <sub>5</sub> , La <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1250	36	LHN, Nb <sub>2</sub> O <sub>5</sub> , La <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1300	96	LHN, trace of HfO <sub>2</sub>
		1350	108	Melt
4.	LaZrNb <sub>3</sub> O <sub>11</sub> (LZN)	1000	12	La <sub>2</sub> Zr <sub>2</sub> O <sub>7</sub> , LZN, Nb <sub>2</sub> O <sub>5</sub> , La <sub>2</sub> O <sub>3</sub> , ZrO <sub>2</sub>
		1150	36	LZN, LaNb <sub>3</sub> O <sub>9</sub> , La <sub>2</sub> O <sub>3</sub> , ZrO <sub>2</sub>
		1150	48	LaNb <sub>3</sub> O <sub>9</sub> , LZN, La <sub>2</sub> O <sub>3</sub> , ZrO <sub>2</sub>
		1200	60	LaNb <sub>3</sub> O <sub>9</sub> , Nb <sub>2</sub> Zr <sub>6</sub> O <sub>17</sub>
5.	PrHfTa <sub>3</sub> O <sub>11</sub> (PHT)	1350	54	PHT, Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	78	PHT, Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	102	PHT, Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	126	PHT, HfO <sub>2</sub>
		1500	150	PHT, trace of HfO <sub>2</sub>
6.	NdHfTa <sub>3</sub> O <sub>11</sub> (NHT)	1000	12	Ta <sub>2</sub> O <sub>5</sub> , NdTa <sub>7</sub> O <sub>19</sub> , HfO <sub>2</sub>
		1500	24	NHT, Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	48	NHT, Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	60	NHT, trace of Ta <sub>2</sub> O <sub>5</sub> and HfO <sub>2</sub>
7.	NdHfNb <sub>3</sub> O <sub>11</sub> (NHN)	1000	24	NdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1100	36	NdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1150	72	NHN, NdNb <sub>3</sub> O <sub>9</sub> , HfO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1200	96	NdNb <sub>3</sub> O <sub>9</sub> , NHN, HfO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1250	108	NdNb <sub>3</sub> O <sub>9</sub> , HfO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
8.	SmHfTa <sub>3</sub> O <sub>11</sub> (SHT)	1200	12	SHT, SmTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
		1300	24	SHT, SmTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
		1350	72	SHT, SmTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1375	84	SHT, SmTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1400	96	SHT, SmTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	108	SmTa <sub>3</sub> O <sub>9</sub> , SHT, HfO <sub>2</sub>
9.	GdHfTa <sub>3</sub> O <sub>11</sub> (GHT)	1200	12	GHT, GdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1300	24	GHT, GdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1350	72	GHT, GdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1375	84	GHT, GdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1400	96	GHT, GdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1500	108	GdTa <sub>3</sub> O <sub>9</sub> , LZT, HfO <sub>2</sub>

(a)

No.	Compositions	T (°C)	Time (h)	Phases Present
1.	DyHfTa <sub>3</sub> O <sub>11</sub>	1000	12	Unknown, DyTaO <sub>4</sub> , Ta <sub>2</sub> O <sub>5</sub> , Dy <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1300	120	Unknown, DyTaO <sub>4</sub> , Ta <sub>2</sub> O <sub>5</sub> , Dy <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
2.	ErHfTa <sub>3</sub> O <sub>11</sub>	1000	12	Ta <sub>2</sub> O <sub>5</sub> , Er <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1150	36	ErTaO <sub>4</sub> , Ta <sub>2</sub> O <sub>5</sub> , Er <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1200	60	ErTaO <sub>4</sub> , Ta <sub>2</sub> O <sub>5</sub> , Er <sub>2</sub> O <sub>3</sub> , HfO <sub>2</sub>
		1250	72	ErTa <sub>7</sub> O <sub>19</sub> , ErTaO <sub>4</sub> , Ta <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
3.	PrHfNb <sub>3</sub> O <sub>11</sub>	1000	15	PrNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Pr <sub>6</sub> O <sub>11</sub>
		1100	40	PrNb <sub>5</sub> O <sub>14</sub> , PrNb <sub>3</sub> O <sub>9</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1150	60	PrNb <sub>5</sub> O <sub>14</sub> , PrNb <sub>3</sub> O <sub>9</sub> , HfO <sub>2</sub>
		1300	72	PrNb <sub>5</sub> O <sub>14</sub> , PrNb <sub>3</sub> O <sub>9</sub> , HfO <sub>2</sub>
4.	PrZrTa <sub>3</sub> O <sub>11</sub>	1350	54	Ta <sub>4</sub> Zr <sub>11</sub> O <sub>32</sub> , ZrO <sub>2</sub> , Ta <sub>2</sub> O <sub>5</sub>
5.	PrZrNb <sub>3</sub> O <sub>11</sub>	1000	15	PrNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Pr <sub>6</sub> O <sub>11</sub>
		1100	41	PrNb <sub>5</sub> O <sub>14</sub> , PrNb <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
		1300	53	PrNb <sub>5</sub> O <sub>14</sub> , PrNb <sub>3</sub> O <sub>9</sub> , ZrO <sub>2</sub>
6.	NdZrTa <sub>3</sub> O <sub>11</sub>	1500	72	NdTa <sub>3</sub> O <sub>9</sub> , Ta <sub>4</sub> Zr <sub>11</sub> O <sub>32</sub> , ZrO <sub>2</sub>
7.	NdZrNb <sub>3</sub> O <sub>11</sub>	1000	24	NdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1050	36	NdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
		1150	48	NdNb <sub>5</sub> O <sub>14</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Nd <sub>2</sub> O <sub>3</sub>
8.	SmZrNb <sub>3</sub> O <sub>11</sub>	1000	24	Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
		1050	36	Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
		1150	48	Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
9.	SmHfNb <sub>3</sub> O <sub>11</sub>	1000	24	Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Sm <sub>2</sub> O <sub>3</sub>
		1100	36	SmNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
		1200	60	SmNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub>
10.	GdHfNb <sub>3</sub> O <sub>11</sub>	1000	24	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1050	36	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1150	48	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , HfO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
11.	GdZrNb <sub>3</sub> O <sub>11</sub>	1000	24	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1050	36	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
		1150	48	GdNbO <sub>4</sub> , Nb <sub>2</sub> O <sub>5</sub> , ZrO <sub>2</sub> , Gd <sub>2</sub> O <sub>3</sub>
12.	YbZrTa <sub>3</sub> O <sub>11</sub>	1500	72	Ta <sub>2</sub> O <sub>5</sub> , Yb <sub>2</sub> O <sub>3</sub> , ZrO <sub>2</sub>

(b)

Table 1: Results of heat treatment on different ABC<sub>3</sub>O<sub>11</sub> combinations; (a) those which yielded an LaZrTa<sub>3</sub>O<sub>11</sub> analogue, (b) those for which an LaZrTa<sub>3</sub>O<sub>11</sub> analogue was not obtained.

Table 2: X-ray powder diffraction data for LaHfTa<sub>3</sub>O<sub>11</sub>

<i>2</i> θ( <i>obs</i> )	<i>h k l</i>	<i>2</i> θ( <i>calc</i> )	<i>Obs - calc</i>	<i>Int.</i>	<i>d (obs) Å</i>	<i>d (calc) Å</i>
14.240	0 0 2	14.229	0.0108	31.0	6.2149	6.2196
16.280	1 0 0	16.287	-0.0066	49.0	5.4402	5.4380
17.797	1 0 1	17.787	0.0106	6.6	4.9798	4.9827
21.709	1 0 2	21.691	0.0187	19.8	4.0904	4.0939
27.050	1 0 3	27.020	0.0296	4.0	3.2937	3.2973
28.728	0 0 4	28.683	0.0454	57.0	3.1050	3.1098
29.328	1 1 1	29.315	0.0131	89.6	3.0428	3.0442
31.910	1 1 2	31.904	0.0054	100.0	2.8023	2.8028
33.170	1 0 4	33.159	0.0119	12.2	2.6986	2.6996
35.868	1 1 3	35.847	0.0217	31.8	2.5016	2.5030
39.613	2 0 3	39.605	0.0076	2.1	2.2733	2.2737
40.819	1 1 4	40.808	0.0103	2.9	2.2089	2.2094
43.638	0 0 6	43.622	0.0157	4.8	2.0725	2.0732
44.029	2 1 0	44.021	0.0085	6.2	2.0550	2.0554
44.647	2 1 1	44.649	-0.0018	8.4	2.0280	2.0279
46.549	1 1 5	46.538	0.0113	27.5	1.9494	1.9499
46.879	1 0 6	46.861	0.0185	3.1	1.9365	1.9372
49.459	2 1 3	49.453	0.0055	4.2	1.8413	1.8415
50.299	3 0 0	50.295	0.0034	32.8	1.8126	1.8127
52.549	3 0 2	52.544	0.0051	4.9	1.7401	1.7403
52.880	1 1 6	52.878	0.0015	31.0	1.7300	1.7301
53.389	2 1 4	53.389	-0.0000	6.3	1.7147	1.7147
55.728	2 0 6	55.710	0.0181	6.5	1.6481	1.6436
58.174	2 1 5	58.173	0.0016	4.0	1.5845	1.5846
58.938	3 0 4	58.928	0.0104	35.6	1.5658	1.5651
59.299	2 2 1	59.285	0.0140	11.6	1.5571	1.5575
59.749	1 1 7	59.747	0.0016	13.4	1.5455	1.5455
60.799	2 2 2	60.806	-0.0076	14.1	1.5223	1.5221
61.400	3 1 0	61.425	-0.0245	5.1	1.5038	1.5032
61.964	3 1 1	61.924	0.0399	4.1	1.4954	1.4973
63.289	2 2 3	63.294	-0.0046	7.9	1.4632	1.4631
63.709	2 1 6	63.705	0.0043	3.1	1.4595	1.4596
65.869	3 1 3	65.840	0.0294	2.1	1.4168	1.4174
68.749	3 0 6	68.732	0.0174	2.7	1.3643	1.3646
69.168	3 1 4	69.170	-0.0013	5.0	1.3571	1.3571
69.928	2 1 7	69.922	0.0060	2.8	1.3442	1.3443
70.932	4 0 2	70.898	0.0332	6.0	1.3276	1.3281
73.344	3 1 5	73.347	-0.0023	2.2	1.2898	1.2897
75.020	1 1 9	75.025	-0.0054	4.9	1.2651	1.2650
75.979	2 2 6	75.975	0.0039	8.8	1.2515	1.2515
76.369	4 0 4	76.396	-0.0269	2.6	1.2450	1.2457
76.790	2 1 8	76.806	-0.0153	2.8	1.2403	1.2400
78.050	3 2 2	78.062	-0.0120	4.4	1.2234	1.2232
78.889	1 0 10	78.876	0.0132	1.8	1.2124	1.2126

Crystal symmetry: hexagonal

Refined cell:  $a = 6.28319(6) \text{ \AA}$ ,  $\alpha = 90.0^\circ$ , ( $a = b$ )

$c = 12.4358(2) \text{ \AA}$ ,  $\gamma = 120.0^\circ$ ,

$V = 425.173(8) \text{ \AA}^3$

No.	Phases	Observed unit cell parameters (Å)		Volume (Å <sup>3</sup> )
		a	c	
1.	LaZrTa <sub>3</sub> O <sub>11</sub> (LZT)	6.28724(4)	12.4525(1)	426.296(5)
2.	LaHfNb <sub>3</sub> O <sub>11</sub> (LHN)	6.2986(2)	12.4189(5)	426.68(3)
3.	LaHfTa <sub>3</sub> O <sub>11</sub> (LHT)	6.28319(6)	12.4358(2)	425.173(8)
4.	PrHfTa <sub>3</sub> O <sub>11</sub> (PHT)	6.281145(8)	12.4412(2)	425.08(3)
5.	NdHfTa <sub>3</sub> O <sub>11</sub> (NHT)	6.2578(1)	12.3938(3)	420.32(2)
6.	SmHfTa <sub>3</sub> O <sub>11</sub> (SHT)	6.23885(8)	12.3569(2)	416.53(1)
7.	GdHfTa <sub>3</sub> O <sub>11</sub> (GHT)	6.2293(1)	12.3421(3)	414.77(2)

Table 3: Lattice parameters and unit cell volumes for ABC<sub>3</sub>O<sub>11</sub> phases

Sites	Properties	LaHfTa <sub>3</sub> O <sub>11</sub>	PrHfTa <sub>3</sub> O <sub>11</sub>	NdHfTa <sub>3</sub> O <sub>11</sub>	LaHfNb <sub>3</sub> O <sub>11</sub>
2c (RE Elements)	(x, y, z)	0.3333, 0.6667, 0.25	0.3333, 0.6667, 0.25	0.3333, 0.6667, 0.25	0.3333, 0.6667, 0.25
	Uiso	0.3(2)	0.7(2)	0.3(3)	0.50(4)
	Occupancy	1.0	1.0	1.0	1.0
6g (Ta, Hf / Nb)1	(x, y, z)	0.6418(2), 0.0, 0.0	0.6414(2), 0.0, 0.0	0.6413(2), 0.0, 0.0	0.6430(1), 0.0, 0.0
	Uiso	0.50(3)	0.61(5)	0.50(5)	0.15(2)
	Occupancy	0.75 / 0.25	0.75 / 0.25	0.75 / 0.25	0.75 / 0.25
2d (Ta, Hf / Nb)2	(x, y, z)	0.3333, 0.6667, 0.75	0.3333, 0.6667, 0.75	0.3333, 0.6667, 0.75	0.3333, 0.6667, 0.75
	Uiso	0.5(1)	0.2(2)	0.3(2)	0.1(4)
	Occupancy	0.75 / 0.25	0.75 / 0.25	0.75 / 0.25	0.75 / 0.25
4f (O1)	(x, y, z)	0.3333, 0.6667, 0.39(1)	0.3333, 0.6667, 0.0042(2)	0.3333, 0.6667, 0.043(2)	0.3333, 0.6667, 0.388(1)
	Uiso	0.5(2)	0.8(3)	1.7(3)	0.38(3)
	Occupancy	1.0	1.0	1.0	1.0
6g (O2)	(x, y, z)	0.244(2), 0.0, 0.0	0.241(2), 0.0, 0.0	0.244(2), 0.0, 0.0	0.2484(2), 0.0, 0.0
	Uiso	0.5(2)	0.8(3)	1.7(3)	0.88(3)
	Occupancy	1.0	1.0	1.0	1.0
12i (O3)	(x, y, z)	0.949(2), 0.377(2), 0.3415(6)	0.953(3), 0.382(3), 0.3430(8)	0.953(3), 0.385(3), 0.3420(8)	0.944(1), 0.373(2), 0.34257(6)
	Uiso	0.5(2)	0.8(3)	1.7(3)	0.46(2)
	Occupancy	1.0	1.0	1.0	1.0
a (Å)		6.28320(6)	6.281145(8)	6.2578(1)	6.2986(2)
c (Å)		12.4358(2)	12.4412(2)	12.3939(3)	12.4189(5)
Cell Volume, V (Å <sup>3</sup> )		425.175(8)	425.08(1)	420.32(1)	426.68(3)
R <sub>wp</sub> (%)		7.02	7.60	6.42	5.73
R <sub>p</sub> (%)		5.30	5.74	4.389	4.46
χ <sup>2</sup>		4.704	7.475	5.996	5.878

Table 4: Structural refinement data for LaHfTa<sub>3</sub>O<sub>11</sub>, PrHfTa<sub>3</sub>O<sub>11</sub>, NdHfTa<sub>3</sub>O<sub>11</sub> and LaHfNb<sub>3</sub>O<sub>11</sub>

<b>Bonding</b>		<b>No.</b>	<b>LaHfTa<sub>3</sub>O<sub>11</sub></b>	<b>PrHfTa<sub>3</sub>O<sub>11</sub></b>	<b>NdHfTa<sub>3</sub>O<sub>11</sub></b>	<b>LaHfNb<sub>3</sub>O<sub>11</sub></b>
(Ta, Hf / Nb) 1	O1	2×	2.080(4) Å	2.087(5) Å	2.082(5) Å	2.0858(5) Å
	O2	2×	1.992(2) Å	1.988(3) Å	1.986(3) Å	1.9963(7) Å
	O3	2×	2.009(7) Å	1.992(9) Å	1.998(9) Å	1.9979(8) Å
	O2	1×	2.494(9) Å	2.51(1) Å	2.48(1) Å	2.485(2) Å
(Ta, Hf / Nb) 2	O3	6×	2.012(9) Å	2.02(1) Å	2.01(1) Å	2.0014(7) Å
RE Elements	O1	6×	2.61(2) Å	2.44(2) Å	2.56(2) Å	2.622(1) Å
	O3	2×	2.45(1) Å	2.58(2) Å	2.42(1) Å	2.4921(8) Å

Table 5: Selected bond distances of LaHfTa<sub>3</sub>O<sub>11</sub>, PrHfTa<sub>3</sub>O<sub>11</sub>, NdHfTa<sub>3</sub>O<sub>11</sub> and LaHfNb<sub>3</sub>O<sub>11</sub>