



This is a repository copy of *Tin(IV) dopant removal through anti-solvent engineering enabling tin based perovskite solar cells with high charge carrier mobilities*.

White Rose Research Online URL for this paper:
<http://eprints.whiterose.ac.uk/150331/>

Version: Published Version

Article:

Bandara, R.M.I., Jayawardena, K.D.G.I., Adeyemo, S.O. et al. (11 more authors) (2019) Tin(IV) dopant removal through anti-solvent engineering enabling tin based perovskite solar cells with high charge carrier mobilities. *Journal of Materials Chemistry C*, 7 (27). pp. 8389-8397. ISSN 2050-7526

<https://doi.org/10.1039/c9tc02003a>

Reuse

This article is distributed under the terms of the Creative Commons Attribution-NonCommercial (CC BY-NC) licence. This licence allows you to remix, tweak, and build upon this work non-commercially, and any new works must also acknowledge the authors and be non-commercial. You don't have to license any derivative works on the same terms. More information and the full terms of the licence here:
<https://creativecommons.org/licenses/>

Takedown

If you consider content in White Rose Research Online to be in breach of UK law, please notify us by emailing eprints@whiterose.ac.uk including the URL of the record and the reason for the withdrawal request.



eprints@whiterose.ac.uk
<https://eprints.whiterose.ac.uk/>



Cite this: DOI: 10.1039/c9tc02003a

Tin(IV) dopant removal through anti-solvent engineering enabling tin based perovskite solar cells with high charge carrier mobilities†

R. M. I. Bandara,^a K. D. G. I. Jayawardena,^a S. O. Adeyemo,^b S. J. Hinder,^c J. A. Smith,^d H. M. Thirimanne,^a N. C. Wong,^a F. M. Amin,^a B. G. Freestone,^d A. J. Parnell,^d D. G. Lidzey,^d H. J. Joyce,^d R. A. Sporea^a and S. R. P. Silva^{*a}

We report the need for careful selection of anti-solvents for Sn-based perovskite solar cells fabricated through the commonly used anti-solvent method, compared to their Pb-based counterparts. This, in combination with the film processing conditions used, enables the complete removal of unwanted Sn⁴⁺ dopants, through engineering the anti-solvent method for Sn-based perovskites. Using a Cs_{0.05}(FA_{0.83}MA_{0.17})_{0.95}Pb_{0.5}Sn_{0.5}I₃ perovskite, charge carrier mobilities of $32 \pm 3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (the highest reported for such systems through the optical-pump terahertz probe technique) together with $\sim 28 \text{ mA cm}^{-2}$ short circuit current densities are achieved. A champion efficiency of 11.6% was obtained for solvent extraction using toluene (an 80% enhancement in efficiency compared to the other anti-solvents) which is further improved to 12.04% following optimised anti-solvent wash and thermal treatment. Our work highlights the importance of anti-solvents in managing defects for high efficiency low bandgap perovskite materials and develops the potential for all-perovskite tandem solar cells.

Received 14th April 2019,
Accepted 20th May 2019

DOI: 10.1039/c9tc02003a

rsc.li/materials-c

1. Introduction

Inorganic–organic hybrid metal halide perovskites have shown great promise in realising new materials for high performing photovoltaic (PV) devices with low fabrication cost.^{1–4} These perovskites possess a number of unique optoelectronic properties such as high light absorption, low Urbach energies, high charge carrier mobilities and high lifetimes, possessing high crystallinity even with low temperature processing, *etc.*^{2,5–7} allowing for the realisation of solution processed high performing optoelectronic devices. Emerging as strong competitors for current commercial silicon solar cells, single junction perovskite solar cells (PSC) have already achieved power conversion efficiencies (PCE) as high as 24.2%.⁸

Recently, Lead-Tin mixed (LTM) perovskites have gained increasing attention due to the ability of tuning the bandgap to lower energies than those possible with Pb-only perovskites, which also enables tailored absorber layers to be fabricated for use as low bandgap cells in all-perovskite tandem solar cells.^{9–11} The first incorporation of Sn²⁺ in Pb²⁺ perovskites to fabricate a LTM planar PSC device was reported in 2014 by Jen and co-workers which yielded a PCE of 10.1%.¹² By 2018, the efficiency of single junction LTM PSCs have reached 17.6% for MA_{0.4}FA_{0.6}Pb_{0.4}Sn_{0.6}I₃¹³ and MA_{0.5}FA_{0.5}Pb_{0.5}Sn_{0.5}I₃¹⁴ perovskite compositions.

One persistent issue limiting LTM PSCs performing to their maximum potential is the instability of Sn²⁺^{15,16} since Sn, is susceptible to oxidize into its more stable ionic form of Sn⁴⁺. Hence, Sn-based perovskites are highly sensitive to air and moisture. When in the perovskite matrix, this oxidation process is known to produce Sn vacancies which break the charge neutrality, thereby deteriorating the perovskite lattice.¹⁷ Furthermore, the Sn⁴⁺ ions act as p-type dopants in the perovskite film resulting in a “self-doping process”.^{18–20} These dopants play a major role in non-radiative recombination in Sn-based perovskites by acting as trap sites for mobile photo-generated charge carriers.^{15,21} Hence, reducing this oxidation process to enhance efficiencies in Sn-based PSCs has received much attention from the perovskite community.¹⁸ Generally, the incorporation of $\geq 50\%$ of Pb²⁺,¹⁶ as well as a number of additives are used in the

^a Advanced Technology Institute, Department of Electrical and Electronic Engineering, University of Surrey, Guildford, Surrey, GU2 7XH, UK.
E-mail: s.silva@surrey.ac.uk

^b Division of Electrical Engineering, Department of Engineering, University of Cambridge, 9 JJ Thomson Ave, Cambridge, CB3 0FA, UK

^c The Surface Analysis Laboratory, Faculty of Engineering and Physical Sciences, University of Surrey, Guildford GU2 7XH, UK

^d Department of Physics and Astronomy, University of Sheffield, Hicks Building, Hounsfield Road, Sheffield S3 7RH, UK

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9tc02003a



LTM perovskite precursors to slow down the oxidation process (which can still occur even under controlled conditions) with SnF₂ being the most commonly used additive.^{16,18,22} Addition of SnF₂-pyrazine complexes²³ and ascorbic acid²⁴ as antioxidants have also been reported. Interestingly, it has been reported that the extraction of Sn⁴⁺ from ionic mixtures for the determination of Sn⁴⁺ content in ores, alloys and food, is most effective when benzene, xylene or toluene is used as the diluent, used along with organic ligands.²⁵ Incidentally, toluene is one of the widely used anti-solvents^{26,27} that enables the formation of pin-hole free absorber layers for high performing LTM perovskite PSCs,^{27,28} with other common anti-solvents being anisole,^{9,16} chlorobenzene,²⁹ diethyl ether,²² etc. Despite their widespread use, the impact of these anti-solvents is less well understood, especially with regards to their impact on the Sn-based systems as means to remove undesired dopants. Based on the above insights, we report efficient removal of Sn⁴⁺ from LTM perovskite absorber layers by careful selection of anti-solvents and optimal thermal treatment, to further enhance the performance of LTM perovskite devices using a triple-cation LTM perovskite with the composition of Cs_{0.05}(FA_{0.83}MA_{0.17})_{0.95}Pb_{0.5}Sn_{0.5}I₃ system as the absorber layer. Unlike their Pb-based counterparts, the anti-solvents show a significant impact on performance in Sn-based perovskite PVs. For the LTM perovskite PVs studied here, we show that the use of toluene enables larger grain sizes, but more importantly, better removal of Sn⁴⁺ resulting in enhancement in all the device parameters with a champion PCE of 11.62% in comparison to chlorobenzene (CB) and anisole (which are two other commonly used anti-solvents). The average PCE for toluene treated devices is improved by >80% compared to anisole treated devices, and improved by >60% in average PCE compared to CB treated devices. This is further supported by the observation of significantly lower Urbach tails and trap density when toluene is used as the anti-solvent. The charge carrier mobility of these perovskites is substantially increased by toluene treatment, reaching $32 \pm 3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, the highest reported so far with measurements based on Optical Pump – Terahertz Probe (OPTP) spectroscopy for Sn-based PSCs. Furthermore, the values achieved herein also strongly compete with those obtained for Pb only perovskites based on the OPTP technique. Following these findings, the fabrication conditions were further tuned enabling the complete removal of Sn⁴⁺ dopants, which to the best of our knowledge, is the first report of complete dopant removal in Sn based PSCs after fabrication of the absorber layer.

2. Results and discussion

2.1. Surface morphological properties of LTM perovskite films

For fabrication of the perovskite films, the LTM perovskite solution with an overall concentration of 1 M was prepared (SI-1, ESI† for details). Thin films of this solution were cast *via* spin coating, inside a N₂ filled glove box. 75 μl of the anti-solvent (CB, anisole or toluene) was dropped 5 s prior to the end of the spinning cycle, and the coated substrates were annealed

at 70 °C for 20 min inside the glove box. Upon anti-solvent dropping, the perovskite layer changes colour from yellow to red-brown signifying the formation of a rich perovskite phase (Fig. 1(a)).

The surface of perovskite thin films formed using the different anti-solvents were investigated through Scanning Electron Microscopy (SEM), and are depicted in Fig. 1(b)–(d). The average grain sizes (Fig. S1, ESI†) as given by the distributions are $332.5 \pm 21.1 \text{ nm}$ for CB, $342.5 \pm 29.4 \text{ nm}$ for anisole and $387.5 \pm 13.4 \text{ nm}$ for toluene, indicating that the largest grains are formed by toluene treatment with the smallest being produced by CB. We note that these grain sizes are comparable with Pb–Sn mixed perovskites reported in literature with a similar Pb:Sn ratio.^{22,30} In optoelectronic materials (including PSCs), the grain boundaries are one source of non-radiative recombination sites that can limit device performance.³¹ Therefore, the formation of larger grains through the use of toluene as an anti-solvent will decrease the grain boundary trap density. Furthermore, films formed using toluene as the anti-solvent demonstrate a more uniform distribution of grain sizes with the lowest spread of grain size data (FWHM of distribution $26.8 \pm 4.3 \text{ nm}$) compared to the other two anti-solvents (FWHM_{CB} = $42.3 \pm 26.6 \text{ nm}$, FWHM_{ANI} = $58.9 \pm 36.5 \text{ nm}$) which suggests a more homogenous nucleation process. To investigate the effect of different anti-solvents on the bulk crystal structure of LTM, grazing incidence wide angle X-ray scattering (GIWAXS) experiments were performed on the samples. The resulting 2D scattering patterns for annealed films quenched with different anti-solvents are depicted in Fig. 1(e)–(g). The corresponding diffraction peak positions (Fig. S2a, ESI†) observed for (100), (110), (111) and (200) perovskite scattering planes are identified. These are in good agreement with those reported in literature for triple cation perovskites³² and show no dependence on the anti-solvent used. *In situ* GIWAXS measurements were taken on as deposited films tracking the intensity of the (100) scattering peak during annealing and show that the perovskite phase is well formed prior to annealing (Fig. S2b, ESI†). In particular, the toluene quenched sample exhibits the highest proportion of post-annealing crystallinity before annealing, indicating the perovskite phase is initially more crystalline than for either anisole or CB treatment. Orientational analysis was performed on the GIWAXS data (Fig. S3, ESI†) and shows comparable texture and crystallinity for all treatments with the toluene sample showing some preferential out-of-plane orientation (see SI-3, ESI† for details).

2.2. Effect of anti-solvents in Sn⁴⁺ removal

To identify the impact of the anti-solvent used on the removal of Sn⁴⁺, X-ray photoelectron spectroscopy (XPS) was carried out. The Sn²⁺ metal cations are observed at binding energies $486.40 \pm 0.05 \text{ eV}$ for Sn(3d_{5/2}), $494.83 \pm 0.06 \text{ eV}$ for Sn(3d_{3/2}) which are in good agreement with values reported in literature.^{33–35} Closer inspection of the aforementioned Sn peaks are observed to consist of shoulders at slightly higher binding energies (Fig. 2(a)) which, upon fitting with two Gaussians, were distinguished as Sn⁴⁺ peaks with binding energies of $487.57 \pm 0.03 \text{ eV}$ for Sn(3d_{5/2})



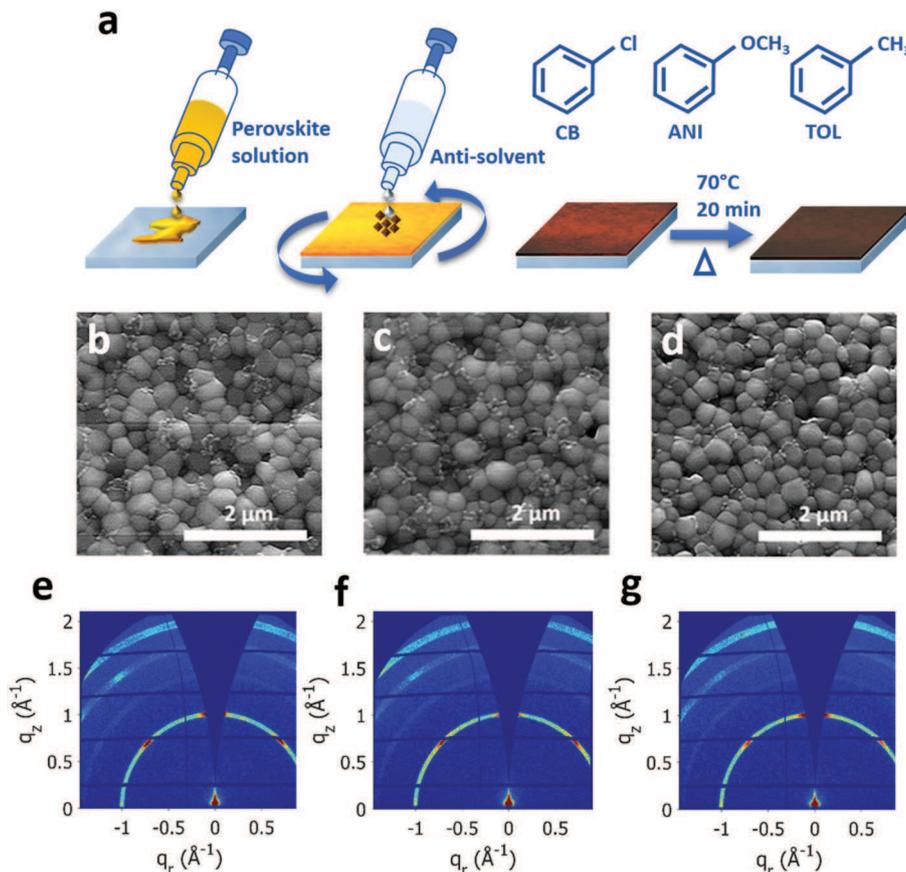


Fig. 1 (a) The deposition and solvent engineering of the perovskite thin film. Chlorobenzene (CB), anisole (ANI) or toluene (TOL) as the anti-solvent (chemical structures shown) is dropped 5 s before the spin coating process of the perovskite solution is finished, aiding nucleation of perovskite phase. The red-brown perovskite thin film is then annealed at 70 °C for 20 min to complete film formation and remove solvent residue. Scanning electron microscopic images of as cast films treated with (b) chlorobenzene, (c) anisole and (d) toluene showing average grain sizes of 332.5 ± 21.14 nm for CB, 342.5 ± 29.43 nm for anisole and 387.5 ± 13.39 nm for toluene. 2D GIWAXS scattering patterns for (e) chlorobenzene, (f) anisole and (g) toluene treated LTM perovskites.

and 496.29 ± 0.04 eV for $\text{Sn}(3d_{3/2})$. The results extracted from the fittings for different anti-solvents are given in Table S1 (ESI[†]). When considering the surface metal ion composition, the atomic ratio of $\text{Pb}^{2+}_{\text{A}(\%)} / \text{Sn}^{2+}_{\text{A}(\%)}$ gives 1.08, 0.79 and 0.77 for anisole (see SI-4, ESI[†] for details), CB and toluene respectively (see Table S1, ESI[†]). This undoubtedly shows that while CB and toluene form nearly identical, Sn rich surface compositions, anisole treatment has formed a Pb^{2+} rich surface, as depicted in Fig. 2(b)(black). We have calculated the $\text{Sn}^{4+}_{\text{A}(\%)} / \text{Sn}^{2+}_{\text{A}(\%)}$ based on the XPS measurements for different anti-solvents which is depicted in Fig. 2(b) (blue). It is clear that the lowest Sn^{4+} content is obtained from the films formed using toluene as the anti-solvent as opposed to when anisole and CB are used. This gives rise to a $\text{Sn}^{4+}_{\text{A}(\%)} / \text{Sn}^{2+}_{\text{A}(\%)}$ of 0.36, 0.31 and 0.27 for anisole, CB and toluene respectively, decreasing the Sn^{4+} fraction.

Next, we proceeded to study the effect of Sn^{4+} extraction on the charge carrier dynamics of the films by performing OPTP spectroscopy. The presence of Sn^{4+} traps in the perovskite film will act as recombination centres and in effect acts as an efficient quencher of the photo-generated electrons moving towards the ETL, through non-radiative recombination.^{37,38}

When a single electron is trapped in a Sn^{4+} trap the charge neutrality is not achieved as it leaves a residual positive charge on the trap. The still prevailing positive charge in the trap Coulombically attracts more electrons leading to faster trap assisted recombination.³⁸ The LTM perovskites, were photo-excited with a pump pulse wavelength of 950 nm with a duration of 35 fs and pump fluences between 82 and $320 \mu\text{J cm}^{-2}$. The photoconductivity decays of all three solvent-treated samples were similar (see SI-5, ESI[†] for details). However, the highest photoconductivity was observed for toluene treated LTMs followed by CB then anisole at all measured pump fluences (Fig. S5(a), ESI[†]). Analysis of the scattering times (τ) (Fig. 2(c)) indicates significantly longer scattering times for toluene treated LTMs compared to the other two, especially at lower fluences. The scattering time, we believe, is a direct correlation of the Sn^{4+} dopant concentration in Sn based perovskites,³⁶ which act as scattering sites.³⁶ Incidentally, these dopants are one of the major factors that hinder the charge carrier mobilities of Sn based perovskites.¹⁷ The larger grain sizes associated with the toluene treated LTMs would also reduce scattering at interfaces, further improving the mobility. The carrier mobility (μ)



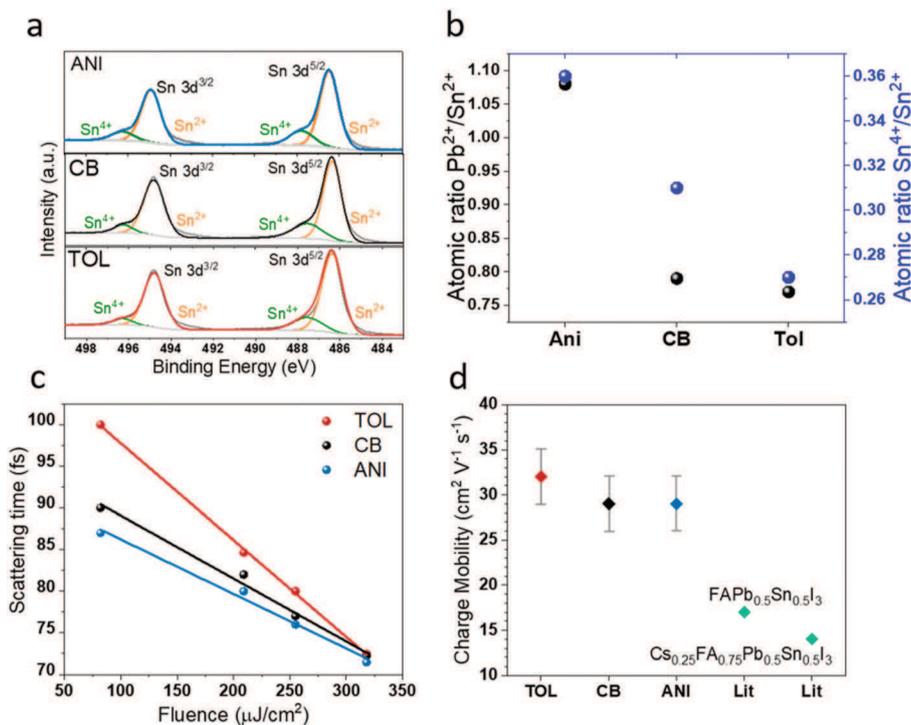


Fig. 2 (a) The X-ray photoelectron spectra (XPS) for Sn species on the LTM perovskite surface for anisole (top), chlorobenzene (middle) and toluene (bottom) treatments. The Sn⁴⁺ peaks (green) are seen as shoulders at slightly higher binding energies with each of the main Sn²⁺ (3d) peaks (orange). (b) Atomic ratio of surface Sn⁴⁺/Sn²⁺ (blue) and Pb²⁺/Sn²⁺ (black) as derived from XPS measurements for different anti-solvent treatments. The lowest Sn⁴⁺/Sn²⁺ is seen for toluene treatment. (c) The scattering time (τ) plotted as a function of fluence for LTM layers with different anti-solvent treatments. Toluene treated perovskites show the highest τ at all fluences (consistent with the decrease in Sn⁴⁺ dopant scattering sites and the reduction in grain boundaries), while τ decreases with increasing fluence (consistent with increased carrier-carrier scattering under high injection conditions). Lowest τ is observed for anisole treatment in all cases. The lines are guides for the eye. (d) The charge carrier mobilities for LTM layers with different anti-solvents, where toluene shows the highest values and anisole the lowest, in comparison with the literature values determined by OPTP,³⁶ for Pb-Sn mixed perovskite PSCs (the literature samples were treated with anisole).

values (Fig. 2(d)), extracted from the transient THz conductivity measurement for toluene treatment was found to be on average $32 \pm 3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ and the high intra-grain mobility was found to be marginally improved by toluene treatment. The average μ values of CB and anisole treated LTMs were $29 \pm 3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. All mobility values have a standard deviation of less than 10%. These μ values, to the best of our knowledge are the highest achieved so far for Pb-Sn mixed PSCs and strongly compete with Pb only perovskites, as evident from values reported for Pb-Sn and Pb perovskites based on the OPTP technique (see Fig. S5(b), ESI†).

2.3. Impact of Sn⁴⁺ removal in PSC device performance

To test the impact of Sn⁴⁺ extraction, LTM PSCs were fabricated using the inverted (p-i-n) perovskite device architecture. The active area of these devices was 0.68 cm^2 and illuminated under AM 1.5G solar spectrum through a large aperture area of 0.433 cm^2 . The current density-voltage relationship (J - V curves) for the champion device performances are depicted in Fig. 3(a), where it is clearly evident that LTM films formed using toluene as the anti-solvent yield higher values for V_{oc} , J_{sc} and FF leading to higher PCE compared to the other two anti-solvents (see Table 1). The champion PCE of 11.62% is achieved for toluene, which is a significant improvement compared to the champion

PCEs of 6.98% and 6.38% achieved for CB and anisole respectively. We note that in determining the J_{sc} values given here for our LTM PSCs we have used the integrated current from the EQE spectra in (Fig. 3(b)) to give a more accurate reading than those measured using halogen lamp based solar simulators.⁹ Here we obtain a very high >85% EQE for toluene treated LTMs. The distribution for the calculated J_{sc} values of LTM PSCs are given in Fig. 3(c) along with the other common device parameters, V_{oc} (Fig. 3(d)), FF (Fig. 3(e)) and %PCE (Fig. 3(f)). The results clearly indicate that toluene treatment significantly improves all device parameters (while the lowest performance is seen in anisole treated LTM PSCs). By comparison, the Pb only analogue of this perovskite does not show a considerable change in device performance depending on the anti-solvent used as described in SI-6 (ESI†). We especially note the significant improvement for the J_{sc} , where the highest value observed is 28.05 mA cm^{-2} as opposed to the highest values of 22.10 mA cm^{-2} and 21.48 mA cm^{-2} for anisole and CB treated PSCs respectively. The average J_{sc} for toluene devices; $27.59 \pm 0.32 \text{ mA cm}^{-2}$, is a 32.96% improvement over anisole devices ($20.75 \pm 0.93 \text{ mA cm}^{-2}$) and 30.63% improvement over CB devices ($21.12 \pm 0.56 \text{ mA cm}^{-2}$). Similar trends were observed in V_{oc} and FF, with the average V_{oc} of toluene devices ($0.62 \pm 0.02 \text{ V}$) being



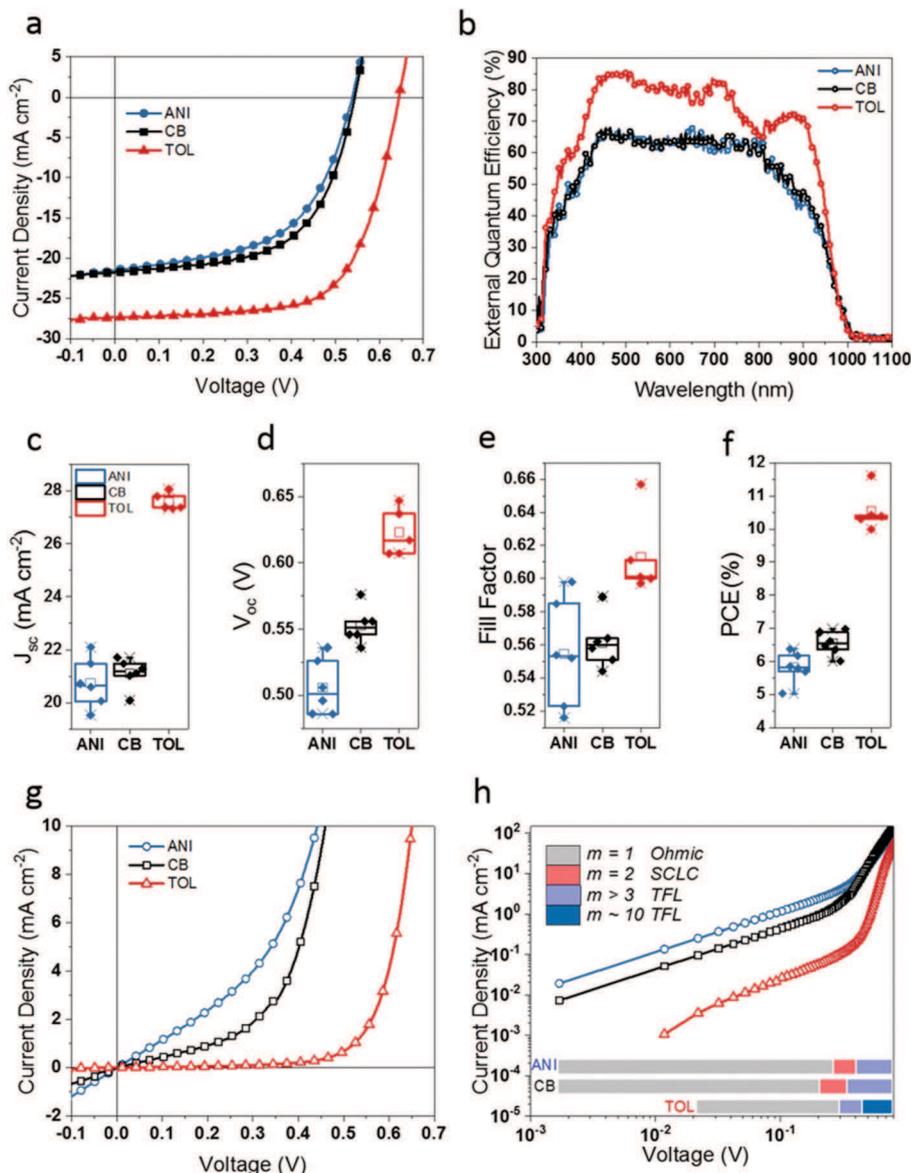


Fig. 3 (a) The current density–voltage (J – V) behaviour of the champion LTM PSC devices for different anti-solvent treatments where toluene treatment clearly shows highest values for V_{oc} , J_{sc} , FF (hence highest PCE) compared to the other two anti-solvents which give comparable device performances. (b) External quantum efficiencies (EQE) for the champion devices showing highest absorption for toluene treated devices. Box plots for (c) J_{sc} , (d) V_{oc} , (e) FF, and (f) %PCE of the LTM PSCs ($n = 6$), where the overall performance of toluene treated devices show nearly 70% average %PCE increase over the other two anti-solvent treated devices, and narrowest data spread. (g) The dark J – V characteristics of the above devices showing the least dark leakage current for a toluene treated device. (h) Plot of $\log(J)$ vs. $\log(V)$ for dark diode curves in (g), identifying the different charge transport regions.

21.57% higher than that observed for anisole (0.51 ± 0.02 V) and 12.73% higher than that observed for CB (0.55 ± 0.01 V). The average FF of toluene (0.61 ± 0.03) is 10.91% and 8.93% improved over the devices fabricated using anisole (0.55 ± 0.03) and CB (0.56 ± 0.02) respectively. The resulting average PCEs depicted in Fig. 3(f) follow the trend, toluene ($10.54 \pm 0.63\%$) > CB ($6.55 \pm 0.36\%$) > anisole ($5.82 \pm 0.46\%$). The % improvement in average PCE of toluene treated devices is 81.10% and 60.92% compared to anisole and CB devices respectively. Furthermore, toluene treatment showed lower hysteresis in devices compared to the other two anti-solvents, as discussed in SI-7 (ESI[†]). These results are in good agreement with the

previously discussed OPTP and XPS measurements, where toluene as anti-solvent results in more efficient removal of Sn^{4+} from the surface of the perovskite film than the other two anti-solvents leading to the LTM PSCs with the highest charge carrier mobilities and highest efficiencies. In comparison, CB treated perovskites layers containing more Sn^{4+} dopants have given rise to lower performing PSCs, and anisole treatment has resulted in the lowest device performance due to even higher Sn^{4+} content. However, even though anisole treatment is clearly seen to be unfavourable for our LTM PSCs, the devices have somewhat comparable performances with CB treated ones. Considering the increased Sn^{4+} dopants alone,



Table 1 Device and optical parameters for LTM perovskites treated with 3 different anti-solvents

Anti-solvent	Champion PCE (%)	PCE (%)	J_{sc} (mA cm ⁻²)	V_{oc} (V)	Fill factor (FF)	Bandgap energy (eV)	Urbach energy (meV)
ANI	6.38	5.82 ± 0.46	20.75 ± 0.93	0.51 ± 0.02	0.55 ± 0.03	1.269 ± 0.020	24.58 ± 0.04
CB	6.98	6.55 ± 0.36	21.12 ± 0.56	0.55 ± 0.01	0.56 ± 0.02	1.262 ± 0.013	25.13 ± 0.04
TOL	11.62	10.54 ± 0.63	27.59 ± 0.32	0.62 ± 0.02	0.61 ± 0.03	1.265 ± 0.020	21.21 ± 0.03

anisole and CB devices could be expected to show a more noticeable difference in device performance, given that the Sn⁴⁺/Sn²⁺ consistently decreases from anisole to CB to toluene. A possible cause for the somewhat similar behaviour for devices fabricated using CB and anisole is the relatively high quantities of Pb²⁺ seen on the surface of anisole treated perovskite layers in the XPS study which may enable passivation of trap states.^{39,40} However, more specialised characterisation is required in order to identify how the increased surface Pb²⁺ impacts the interface and consequently overall device performance.

To elucidate the effect of anti-solvent on the band tails (occurring as a result of localised energy states in semiconducting materials⁴¹), the Urbach energies (E_U), were calculated using the EQE spectra (see SI-8, ESI† for detailed description). The lowest Urbach energy values were observed for toluene ($E_U = 21.21 ± 0.03$ meV) treated samples (Table 1), attributed to the lowest band-tail defect states and higher crystalline order throughout the thickness of the sample. Anisole and CB treated LTMs gave comparable E_U values of $24.58 ± 0.04$ meV and $25.13 ± 0.04$ meV respectively, resulting from a higher density of band tail defect states (due to the higher Sn⁴⁺) in these films. We note that the absorber layers processed using the different anti-solvents exhibit a similar bandgap of 1.26 eV (see SI-9, ESI†).

In order to further evaluate the impact of the Sn⁴⁺ removal process on the solar cell performance, we analysed the dark J - V curves of LTM PSCs. Careful inspection of the dark J - V curves (Fig. 3(g)) at negative voltages indicates that the dark leakage current of devices fabricated using CB and anisole are higher compared to the devices fabricated with toluene (0.64 mA cm⁻², 1.32 mA cm⁻² and ~80 μA cm⁻² at -0.1 V for CB, anisole and toluene respectively). Considering the devices' dark characteristics (Fig. 3(h)), different charge transport regions can be identified, namely; Ohmic transport, space charge limited current (SCLC) and trap filled limit (TFL) (see SI-10, ESI† for full analysis). In anisole and CB treated devices, the Ohmic regime is extended, as seen by the long initial region of constant slope (m) showing the quasi-Ohmic J - V behaviour ($m = 1$),⁴² which then gradually increases non-linearly through a region of $m = 2$, before reaching higher slopes expected for the TFL. Given that $m = 2$ corresponds to $J_{SCLC, shallow traps}$, it can be concluded that CB and anisole treatments have resulted in shallow traps in the LTM PSC. Considering the voltage at which anisole and CB devices transfer from the Ohmic region to SCLC region (CB = 0.21 V, anisole = 0.27 V), the more prolonged quasi-Ohmic region of anisole devices suggest a higher trap density in these devices than that of CB ones. In the case of toluene treated devices, the Ohmic region is shorter than that of the other two anti-solvent treatments, and the slope quickly transfers to a region of $m > 3$ around 0.30 V followed by a

prolonged region over which $m \approx 10$ (beyond 0.46 V bias). This marks the TFL region where $l > 2$ of eqn (S10) (ESI†), and the trap dependent SCLC region ($m \approx 2$) is hardly noticeable in this system. This rapid transition from Ohmic to TFL regime for toluene indicates that there are a reduced number of traps in this system, which is in agreement with the previous observations for the lower Sn⁴⁺ content (from the THz, XPS and photovoltaic device measurements).⁴³ V_{TFL} increases in the order of toluene (0.30 V) < CB (0.35 V) < anisole (0.41 V) in the order of increasing trap density.⁴⁴ In the TFL region, the slopes of CB and anisole dark curves reach $m \approx 4$, which is in contrast to $m \approx 10$ of toluene. We note that the trap free region (governed by Child's law) is not reached for any of the devices within the measured electric field range. The conclusion that this behaviour is caused by the higher number of parasitic charges brought upon by Sn⁴⁺ contamination for CB and anisole is further supported by recent literature where increased Sn vacancies were observed to give rise to a metal-like conductivity.⁴²

2.4. Proposed hypotheses for Sn⁴⁺ removal mechanisms

Based on the results discussed above, it is evident that the removal of Sn⁴⁺ dopants from the LTM films during the fabrication process has the most significant impact on the performance of the LTM PSCs. The Sn⁴⁺ ions produced during fabrication and originally present in the SnI₂ precursor are expected to exist in the system as SnI₄, and we propose several hypotheses for the removal mechanisms of this compound which is detrimental to device performance. Firstly, SnI₄ is a tetrahedral molecule (Fig. 4(a)) which, owing to its high symmetry, is non-polar in nature. Considering the relative polarities of the anti-solvents discussed (anisole = 0.198, CB = 0.188, toluene = 0.099) it can be seen that toluene is the least polar solvent. Given the non-polar nature of SnI₄, it should therefore be most soluble in toluene. It is also the lightest of the three anti-solvents, with a molecular weight of 92.14 g mol⁻¹ and has the lowest boiling point of 110.6 °C compared to 108.14 g mol⁻¹ and 155.5 °C of anisole and 112.56 g mol⁻¹ and 131.7 °C of CB.⁴⁵ Hence, during the annealing process of the perovskite layer when the prevailing solvent is evaporated, the removal of SnI₄ dissolved in the solvent can generally be considered higher with toluene.

Furthermore, recent reports in the literature have described 6-coordinated complex formation of SnI₄ with organic ligands resulting in complexes such as [SnI₄(C₅H₅NO)₂] and [SnI₄{(C₆H₅)₂SO}₂].⁴⁶ It has been shown in these studies that bonding occurs through coordination of electrons on the O atom of the ligand directly to the Sn centre of SnI₄. Incidentally, the solvents used for the LTM precursor solution; DMF (C₃H₇NO) and DMSO (C₂H₆SO), which are known to act as common ligands for metal centres,⁴⁷ are of similar nature to the ligands involved



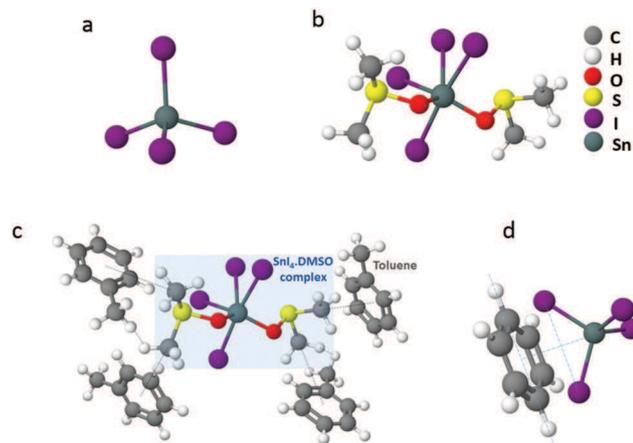


Fig. 4 (a) Tetrahedral structure of SnI_4 . (b) Possible 6-coordinate complex formation of SnI_4 with DMSO, where DMSO molecules act as ligands for the Sn^{4+} metal centre. (c) Possible interaction of $\text{SnI}_4(\text{DMSO})_2$ complex with non-polar toluene molecules. (The figures are not drawn to scale.) (d) The complex formation of SnI_4 with non-polar aromatic molecules (benzene shown here as the aromatic molecule for the ease of the reader) as proposed by J. F. Murphy and D. E. Baker,⁴⁸ where two of the iodine atoms of SnI_4 are placed on the benzene ring, with the benzene plane lying parallel to a line drawn joining the iodine atoms.

in the aforementioned complexes. Therefore, it can be hypothesised that similar co-ordination is viable in the LTM perovskite with DMF and DMSO. Fig. 4(b) depicts one possible complex that can be formed *via* DMSO ligand. If such bonding occurs, the more polar $-\text{CO}$ and $-\text{SO}$ groups would participate in bonding with SnI_4 , exposing the more non-polar C-H groups away from the SnI_4 centre. It is then possible for these complexes to be solubilised in the non-polar toluene (Fig. 4(c)) and hence be removed from the perovskite crystal, *via* thermal annealing.

In addition to the above, early reports in chemical research suggest the dissolution of SnI_4 *via* complex formation in various organic solvents.⁴⁸ Interestingly, it has been shown that with non-polar aromatic solvents such as benzene, toluene or xylene, there is a tendency for SnI_4 to form a 1:1 (mole ratio) complex, through the occurrence of an acid–base like interaction, where SnI_4 acts as a Lewis acid and the aromatic ring acts as the Lewis base.⁴⁸ When the basicity of the aromatic ring increases, such as through the addition of methyl groups to the ring (*i.e.*, benzene to toluene to xylene) the strength of the complex was found to be strengthened, hence increasing the solubility of SnI_4 in the order of benzene < toluene < xylene. Considering the anti-solvents used herein, the low polarity of toluene solubilises SnI_4 more easily enabling the formation of a complex in comparison to CB and anisole. Previous work indicates that two of the SnI_4 iodine atoms participate in bonding with the aromatic group to make the $\text{SnI}_4\text{-A}$ (A = aromatic solvent molecule) complex where a line conjoining the iodine atoms lie in parallel to the plane of the ring (Fig. 4(d)).⁴⁸ Due to the tetrahedral structure of SnI_4 , the Sn atom in the complex should be positioned directly over the six-fold centre of the aromatic group. Moreover, considering the previous reports, the formation of a $\text{SnI}_4\text{-TOL}$ complex with

toluene can be suggested here, however it should be noted that the formation of this complex seems much less likely compared to the $\text{SnI}_4\text{-DMSO}$ complex hypothesised above. It is reasonable to speculate that complexing is initiated during the anti-solvent dripping stage, which is partially removed due to the high spin speed employed during the spin coating procedure. The remaining $\text{SnI}_4\text{-TOL}$ is easily removed during the annealing step, owing to the higher evaporation rate of toluene compared to the other two anti-solvents.

The removal of Sn^{4+} could happen *via* either of the hypothesised mechanisms, or a combination of them. Further in-depth experimental and simulation studies are currently in progress for the purpose of verifying the exact mechanism for Sn^{4+} removal and will be reported in a subsequent manuscript.

2.5. Solvent engineering and optimised thermal treatment for efficiency improvement

Following the understanding of this novel aspect of the anti-solvent engineering process, the fabrication process was further modified to promote complete removal of Sn^{4+} using toluene. This was carried out by changing physical parameters such as temperature and fine tuning the anti-solvent dripping procedure. Following recent reports on using higher annealing temperatures for FA and MA containing Pb–Sn mixed perovskites,^{22,27} we have used 120 °C as the annealing temperature for the modified system. Here we successfully facilitated the complete removal of Sn^{4+} as seen in the XPS spectra depicted in Fig. 5(a). The corresponding champion devices (Fig. 5(b)) showed ~14% improvement in V_{oc} , and 6% improvement on FF compared to the toluene treated champion reference device (Fig. 3) leading to higher PCE of 12.04%. The current density was calculated using the EQE profile for the device, shown in Fig. 5(c). In contrast to the other device parameters, the J_{sc} of these devices

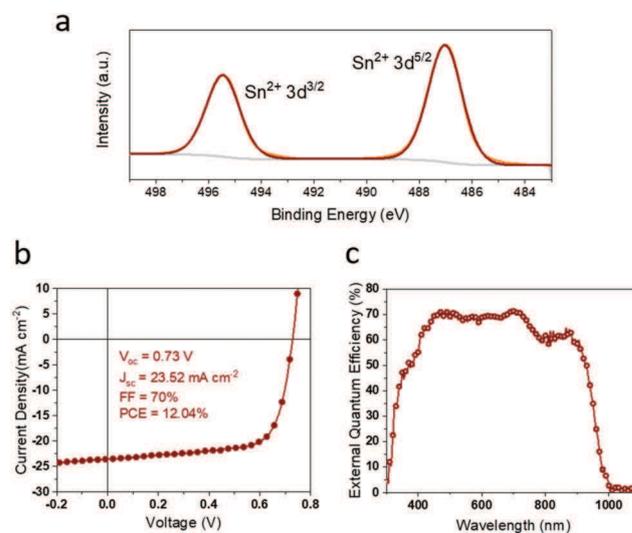


Fig. 5 (a) The X-ray photoelectron spectra (XPS) for Sn species on the LTM perovskite surface after the modified solvent engineering process, showing Sn^{2+} peaks. No shoulders are seen in the peaks signifying the complete removal of Sn^{4+} . The corresponding (b) champion J - V curve and device parameters, and (c) EQE resulted by the modified process.



was observed to decrease. This can be attributed to the thickness variation of the LTM layer resulting from the post-treatments and Sn⁴⁺ removal. The thickness of the absorber layer is directly related to the quantity of light absorbed by the PSC, which in turn is related to the current output from the device. Hence, further improvements in J_{sc} are expected *via* optimisation of the absorber layer thickness.

3. Conclusion

We report a new insight on the importance of careful selection of the anti-solvents for Sn-based perovskite PSCs. Compared to their Pb-only counterparts, the qualities of the anti-solvents used are observed to be more vital for the Sn-based perovskite PSC performance due to their impact on managing the Sn⁴⁺ defects of the absorber layers. Based on this, an approach for the complete removal of parasitic Sn⁴⁺ in Sn-based PSCs through solvent engineering is identified. The efficient extraction of Sn⁴⁺ from the perovskites was tested using a triple cation Pb–Sn mixed perovskite with the formula Cs_{0.05}(FA_{0.83}MA_{0.17})_{0.95}Pb_{0.5}Sn_{0.5}I₃. The results indicate that toluene acts as an efficient extraction medium for Sn⁴⁺, when used as an anti-solvent in the PSC fabrication process. Toluene treated LTM films not only show a lower Sn⁴⁺ quantity on the surface of the LTM films, but also show drastically improved PSC device parameters (V_{oc} , J_{sc} , FF and PCE) compared to the other two anti-solvents examined; anisole and CB resulting in a champion PCE of 11.62%. The charge carrier mobilities reach $32 \pm 3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ with the toluene treatment which are the highest reported so far based on the OPTP technique for Sn based PSCs. The atomic percentage of Sn⁴⁺ decreases in the order of anisole > CB > toluene and the device performance increases in the order of anisole < CB < toluene which is further substantiated by the lower impact of parasitic charges for LTM PSCs under toluene treatment. Bandgap energies for all three LTM films were similar ($\sim 1.26 \text{ eV}$) and the perovskite crystal structure of the thin films was found to be of similar nature, indicating that the major cause for the better device performance of toluene treated PSCs originates from the ability of toluene to remove Sn⁴⁺ dopants from the LTM film surface. These devices were further improved by fine tuning the device fabrication process to aid in complete removal of Sn⁴⁺ from the system. The Sn⁴⁺ extraction process outlined herein points towards higher efficiencies in tin based PSCs by simple modifications to the anti-solvent engineering route enabling further improvements for single junction Pb–Sn and Sn-only PSCs, as well as all perovskite multi-junction PVs.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

R. M. I. B., K. D. I. J., H.M.T., R. A. S and S. R. P. S. gratefully acknowledge support from EPSRC *via* research grant EP/R025304/1

and the European Commission H2020 funded CORNET program (Grant agreement ID: 760949). R. M. I. B. acknowledges support from the University of Surrey Overseas Research Scholarship/ University Research Scholarship. A. J. P. is grateful to Xenocs for their ongoing help and support in the user program at the University of Sheffield. H. J. J. acknowledges funding from the European Research Council (grant no. 716471, ACrossWire). SRPS acknowledges funding support from the Marcus Lee Foundation. We thank the Warwick Centre for Ultrafast Spectroscopy for access to experimental facilities.

References

- 1 M. M. Lee, J. Teuscher, T. Miyasaka, T. N. Murakami and H. J. Snaith, Efficient hybrid solar cells based on meso-structured organometal halide perovskites, *Science*, 2012, **338**, 643–647.
- 2 M. A. Green, A. Ho-Baillie and H. J. Snaith, The emergence of perovskite solar cells, *Nat. Photonics*, 2014, **8**, 506–514.
- 3 M. A. Green and A. Ho-Baillie, Perovskite Solar Cells: The Birth of a New Era in Photovoltaics, *ACS Energy Lett.*, 2017, **2**, 822–830.
- 4 M. Saliba, J.-P. Correa-Baena, M. Grätzel, A. Hagfeldt and A. Abate, Perovskite Solar Cells: From the Atomic Level to Film Quality and Device Performance, *Angew. Chem., Int. Ed.*, 2018, **57**, 2554–2569; V. Ferguson, S. R. P. Silva and W. Zhang, Carbon Materials in Perovskite Solar Cells: Prospects and Future Challenges, *Energy Environ. Mater.*, 2019, DOI: 10.1002/eem2.12035.
- 5 C. Katan, A. D. Mohite and J. Even, Entropy in halide perovskites, *Nat. Mater.*, 2018, **17**, 377–379.
- 6 H. J. Snaith, Present status and future prospects of perovskite photovoltaics, *Nat. Mater.*, 2018, **17**, 372–376.
- 7 S. Shi, Y. Li, X. Li and H. Wang, Advancements in all-solid-state hybrid solar cells based on organometal halide perovskites, *Mater. Horiz.*, 2015, **2**, 378–405.
- 8 Best research-cell efficiencies, <https://www.nrel.gov/pv/assets/images/efficiency-chart-20180716.jpg>, accessed: April 2019.
- 9 G. E. Eperon, *et al.*, Perovskite-perovskite tandem photovoltaics with optimised band gaps, *Science*, 2016, **354**, 861–865.
- 10 B. Zhao, *et al.*, High Open-Circuit Voltages in Tin-Rich Low-Bandgap Perovskite-Based Planar Heterojunction Photovoltaics, *Adv. Mater.*, 2017, **29**, 1604744.
- 11 R. Prasanna, *et al.*, Band Gap Tuning via Lattice Contraction and Octahedral Tilting in Perovskite Materials for Photovoltaics, *J. Am. Chem. Soc.*, 2017, **139**, 11117–11124.
- 12 F. Zuo, *et al.*, Binary-Metal Perovskites Toward High-Performance Planar-Heterojunction Hybrid Solar Cells, *Adv. Mater.*, 2014, **26**, 6454–6460.
- 13 D. Zhao, *et al.*, Low-bandgap mixed tin–lead iodide perovskite absorbers with long carrier lifetimes for all-perovskite tandem solar cells, *Nat. Energy*, 2017, **2**, 17018.
- 14 G. Kapil, *et al.*, Highly Efficient 17.6% Tin–Lead Mixed Perovskite Solar Cells Realised through Spike Structure, *Nano Lett.*, 2018, **18**, 3600–3607.



- 15 J. Fan, *et al.*, Molecular Self-Assembly Fabrication and Carrier Dynamics of Stable and Efficient $\text{CH}_3\text{NH}_3\text{Pb}_{(1-x)}\text{Sn}_x\text{I}_3$ Perovskite Solar Cells, *ChemSusChem*, 2017, **10**, 3839–3845.
- 16 T. Leijtens, R. Prasanna, A. Gold-Parker, M. F. Toney and M. D. McGehee, Mechanism of Tin Oxidation and Stabilisation by Lead Substitution in Tin Halide Perovskites, *ACS Energy Lett.*, 2017, **2**, 2159–2165.
- 17 N. K. Noel, *et al.*, Lead-free organic–inorganic tin halide perovskites for photovoltaic applications, *Energy Environ. Sci.*, 2014, **7**, 3061–3068.
- 18 S. Gupta, D. Cahen and G. Hodes, How SnF_2 Impacts the Material Properties of Lead-Free Tin Perovskites, *J. Phys. Chem. C*, 2018, **122**, 13926–13936.
- 19 N. K. Noel, *et al.*, Lead-free organic–inorganic tin halide perovskites for photovoltaic applications, *Energy Environ. Sci.*, 2014, **7**, 3061–3068.
- 20 Y. Takahashi, *et al.*, Charge-transport in tin-iodide perovskite $\text{CH}_3\text{NH}_3\text{SnI}_3$: origin of high conductivity, *Dalton Trans.*, 2011, **40**, 5563.
- 21 K. Yamada, *et al.*, Anomalous Dielectric Behavior of a Pb/Sn Perovskite: Effect of Trapped Charges on Complex Photoconductivity, *ACS Photonics*, 2018, **5**, 3189–3197.
- 22 W. Liao, *et al.*, Fabrication of Efficient Low-Bandgap Perovskite Solar Cells by Combining Formamidinium Tin Iodide with Methylammonium Lead Iodide, *J. Am. Chem. Soc.*, 2016, **138**, 12360–12363.
- 23 S. J. Lee, *et al.*, Fabrication of Efficient Formamidinium Tin Iodide Perovskite Solar Cells through SnF_2 –Pyrazine Complex, *J. Am. Chem. Soc.*, 2016, **138**, 3974–3977.
- 24 X. Xu, *et al.*, Ascorbic acid as an effective antioxidant additive to enhance the efficiency and stability of Pb/Sn-based binary perovskite solar cells, *Nano Energy*, 2017, **34**, 392–398.
- 25 B. M. Sargar and M. A. Anuse, Rapid Solvent Extraction of Tin(IV) with High Molecular Weight Amine from Hydrochloric Acid Solution, *J. Chin. Chem. Soc.*, 2003, **50**, 841–848.
- 26 Z. Yang, *et al.*, Stable Low-Bandgap Pb–Sn Binary Perovskites for Tandem Solar Cells, *Adv. Mater.*, 2016, **28**, 8990–8997.
- 27 A. Rajagopal, *et al.*, Highly Efficient Perovskite-Perovskite Tandem Solar Cells Reaching 80% of the Theoretical Limit in Photovoltage, *Adv. Mater.*, 2017, **29**, 1702140.
- 28 X. Liu, *et al.*, Improved efficiency and stability of Pb–Sn binary perovskite solar cells by Cs substitution, *J. Mater. Chem. A*, 2016, **4**, 17939–17945.
- 29 M. Lyu, *et al.*, Highly compact and uniform $\text{CH}_3\text{NH}_3\text{Sn}_0.5\text{Pb}_0.5\text{I}_3$ films for efficient panchromatic planar perovskite solar cells, *Sci. Bull.*, 2016, **61**, 1558–1562.
- 30 G. E. Eperon, *et al.*, Perovskite-perovskite tandem photovoltaics with optimised band gaps, *Science*, 2016, **354**, 861–865.
- 31 T. S. Sherkar, C. Momblona, L. Gil-Escrig, H. J. Bolink and L. J. A. Koster, Improving Perovskite Solar Cells: Insights From a Validated Device Model, *Adv. Energy Mater.*, 2017, **7**, 1602432.
- 32 H. Tsai, *et al.*, Light-induced lattice expansion leads to high-efficiency perovskite solar cells, *Science*, 2018, **360**, 67–70.
- 33 NIST XPS Database, Selected Element Search Menu. Available at: <https://srdata.nist.gov/xps/selEnergyType.aspx>, accessed: 1st August 2018.
- 34 T.-B. Song, *et al.*, Importance of Reducing Vapor Atmosphere in the Fabrication of Tin-Based Perovskite Solar Cells, *J. Am. Chem. Soc.*, 2017, **139**, 836–842.
- 35 M. A. Stranick and A. Moskwa, SnO by XPS, *Surf. Sci. Spectra*, 1993, **2**, 45–49.
- 36 L. M. Herz, Charge-Carrier Mobilities in Metal Halide Perovskites: Fundamental Mechanisms and Limits, *ACS Energy Lett.*, 2017, **2**, 1539–1548.
- 37 M. A. Green, *Solar cells: operating principles, technology, and system applications*, Prentice-Hall, 1982.
- 38 T. S. Sherkar, *et al.*, Recombination in Perovskite Solar Cells: Significance of Grain Boundaries, Interface Traps, and Defect Ions, *ACS Energy Lett.*, 2017, **2**, 1214–1222.
- 39 Q. Chen, *et al.*, Controllable Self-Induced Passivation of Hybrid Lead Iodide Perovskites toward High Performance Solar Cells, *Nano Lett.*, 2014, **14**, 4158–4163.
- 40 G. A. MacDonald, *et al.*, Methylammonium lead iodide grain boundaries exhibit depth-dependent electrical properties, *Energy Environ. Sci.*, 2016, **9**, 3642–3649.
- 41 A. S. Hassanien and A. A. Akl, Influence of composition on optical and dispersion parameters of thermally evaporated non-crystalline $\text{Cd}_{50}\text{S}_{50-x}\text{Se}_x$ thin films, *J. Alloys Compd.*, 2015, **648**, 280–290.
- 42 Y. Zong, Z. Zhou, M. Chen, N. P. Padture and Y. Zhou, Lewis-Adduct Mediated Grain-Boundary Functionalisation for Efficient Ideal-Bandgap Perovskite Solar Cells with Superior Stability, *Adv. Energy Mater.*, 2018, **8**, 1800997.
- 43 D. Shi, *et al.*, Solar cells. Low trap-state density and long carrier diffusion in organolead trihalide perovskite single crystals, *Science*, 2015, **347**, 519–522.
- 44 N. A. Nismy, K. D. G. I. Jayawardena, A. A. D. T. Adikaari and S. R. P. Silva, Nano-engineering of hybrid organic heterojunctions with carbon nanotubes to improve photovoltaic performance, *Org. Electron.*, 2015, **22**, 35–39.
- 45 The PubChem Project. Available at: <https://pubchem.ncbi.nlm.nih.gov/>, accessed: 26th September 2018.
- 46 E. Wlazlak, W. Macyk, W. Nitek and K. Szaciłowski, Influence of π -Iodide Intermolecular Interactions on Electronic Properties of Tin(IV) Iodide Semiconducting Complexes, *Inorg. Chem.*, 2016, **55**, 5935–5945.
- 47 M. Ozaki, *et al.*, Solvent-Coordinated Tin Halide Complexes as Purified Precursors for Tin-Based Perovskites, *ACS Omega*, 2017, **2**, 7016–7021.
- 48 J. F. Murphy and D. E. Baker, Complexes of Stannic Iodide with Aromatic Hydrocarbons, *Can. J. Chem.*, 1965, **43**, 1272–1278.

