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Supplementary Information

A Spheres-in-Tube Carbonaceous Nanostructure for High-Capacity and High-Rate Lithium-Sulfur Batteries

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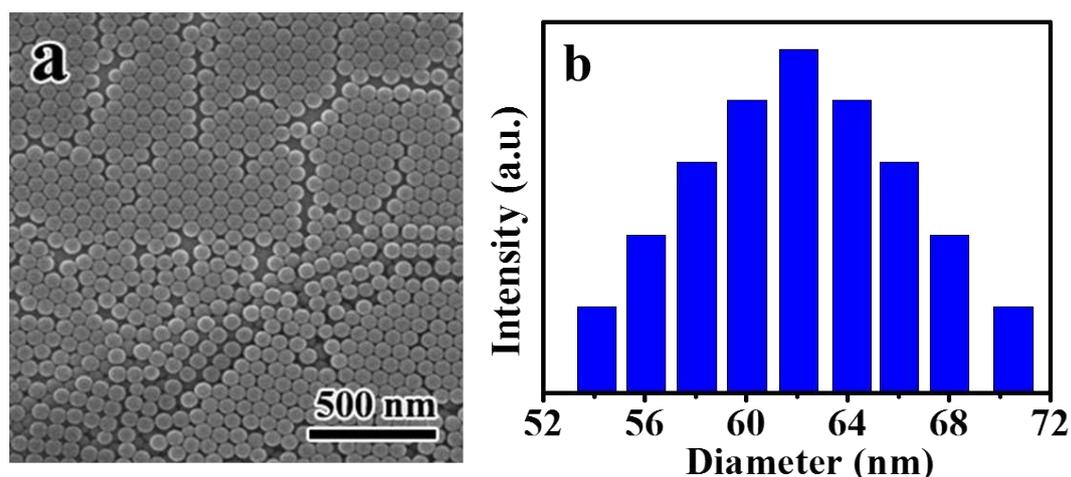


Fig. S1. SEM image and corresponding Dynamic Light Scattering result of SiO₂ nanoparticles prepared by traditional Stöber-Fink-Bohn method.

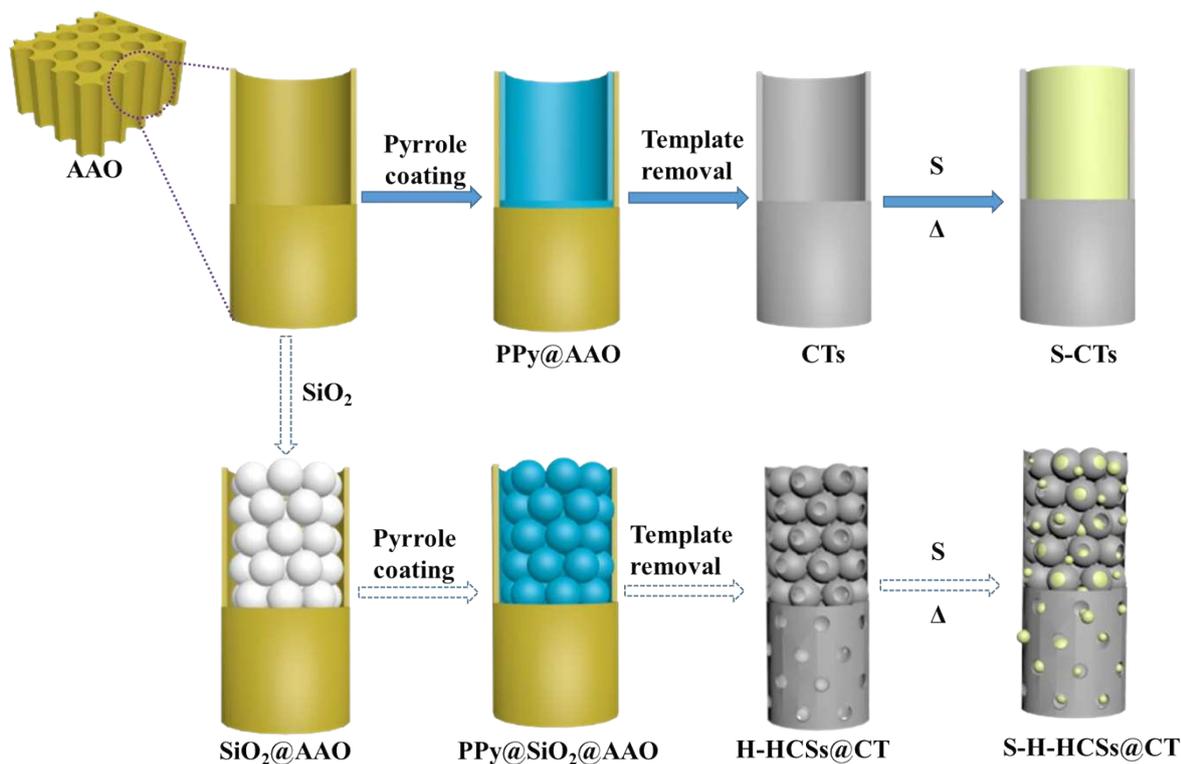


Fig. S2. Schematic illustration of the production of S-CTs and S-H-HCSs@CT.

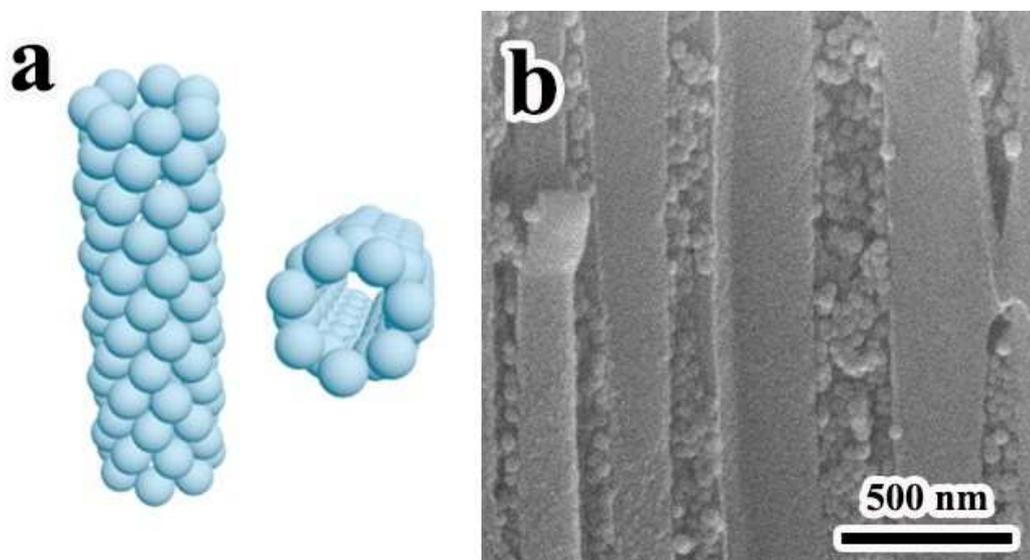


Fig. S3. Scheme models and related cross-sectional SEM images of confined assembled structure of SiO₂ in AAO a) $\alpha = 3.33$ ($\alpha = dc/ds$, dc means the AAO channel diameter, ds means the diameter of the SiO₂ NPs.)

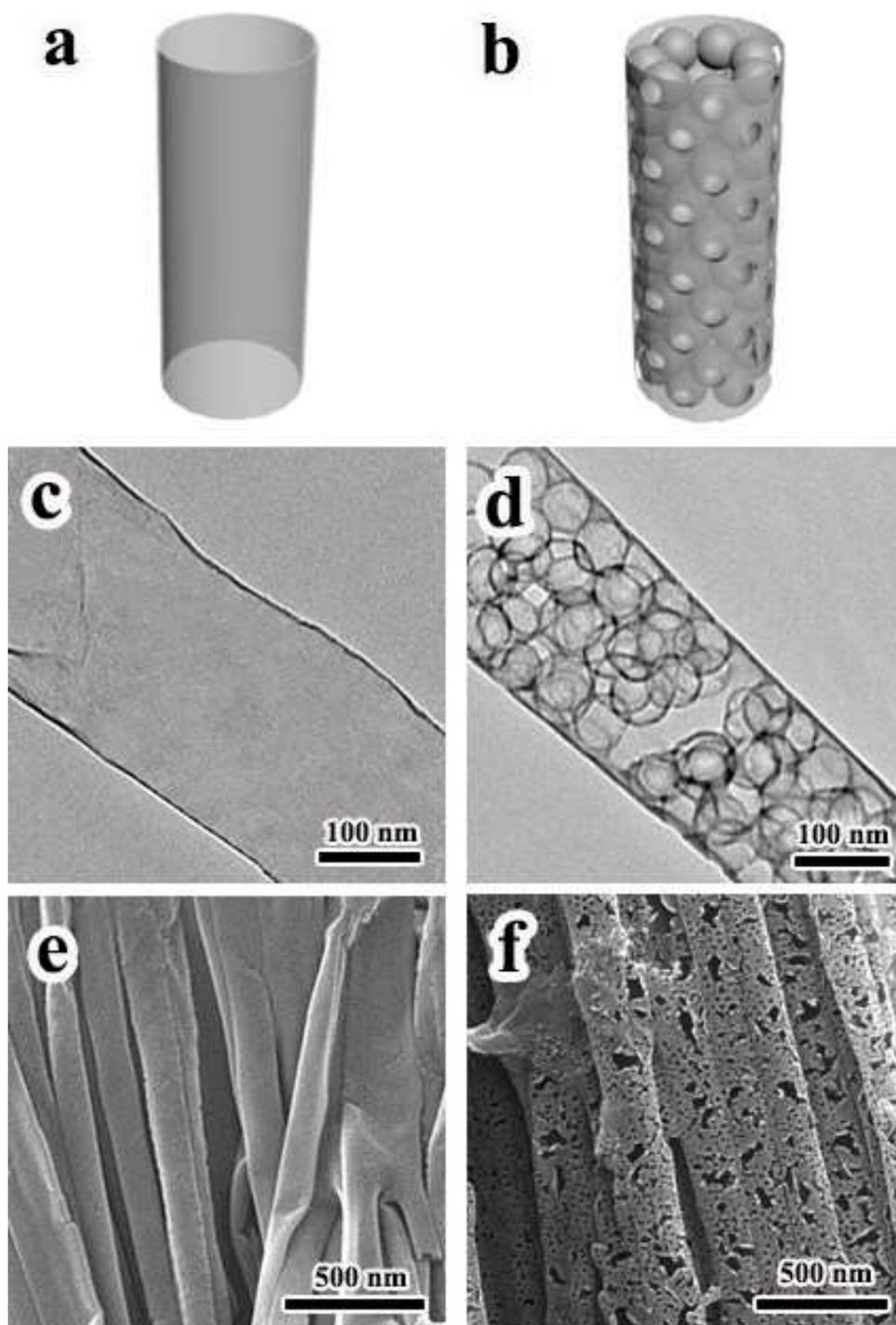


Fig. S4. Schematics of a) CTs and b) H-HCSs@CT. TEM and SEM images c) and e) show bare CTs. TEM and SEM images d) and f) show H-HCSs@CT with mesopores on the CT surface.

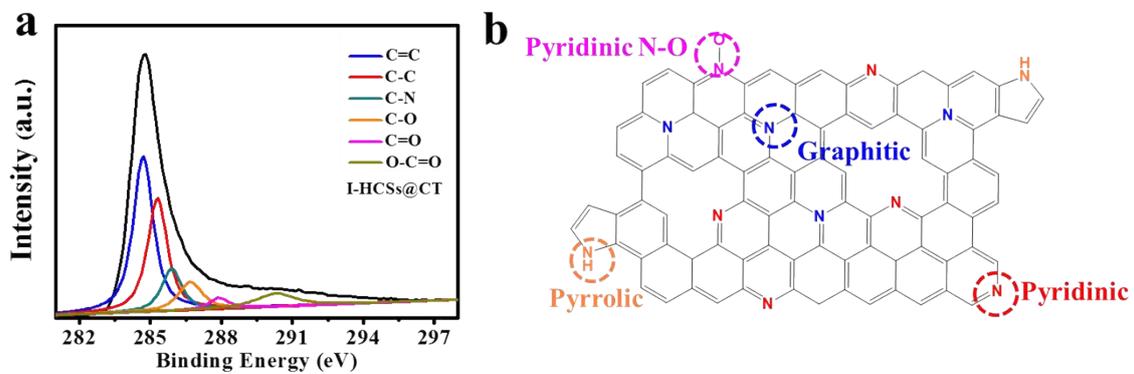


Fig. S5. a) High-resolution C1s XPS spectra of I-HCSs@CT; b) Schematic model of functional groups in I-HCSs@CT.

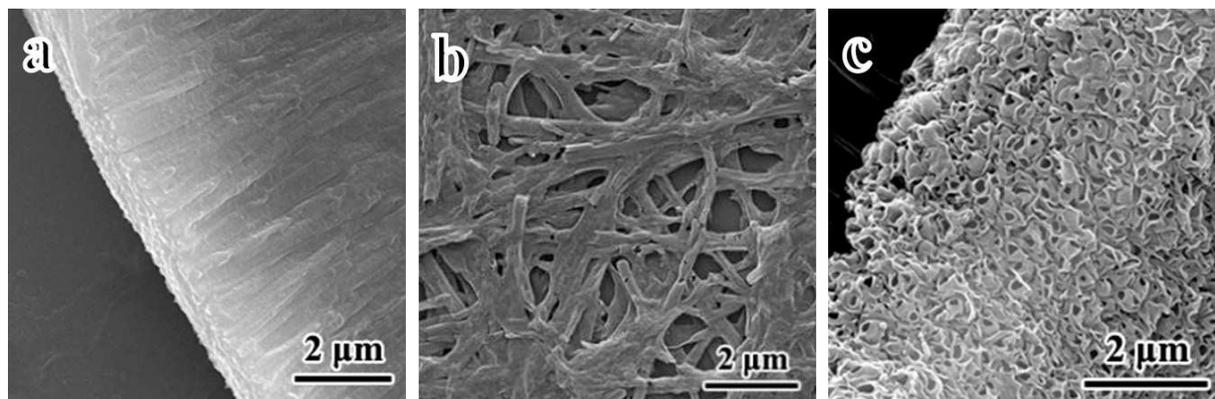


Fig. S6. SEM images of various products: a) S-I-HCSs@CT, b) S-H-HCSs@CT and c) S-CTs.

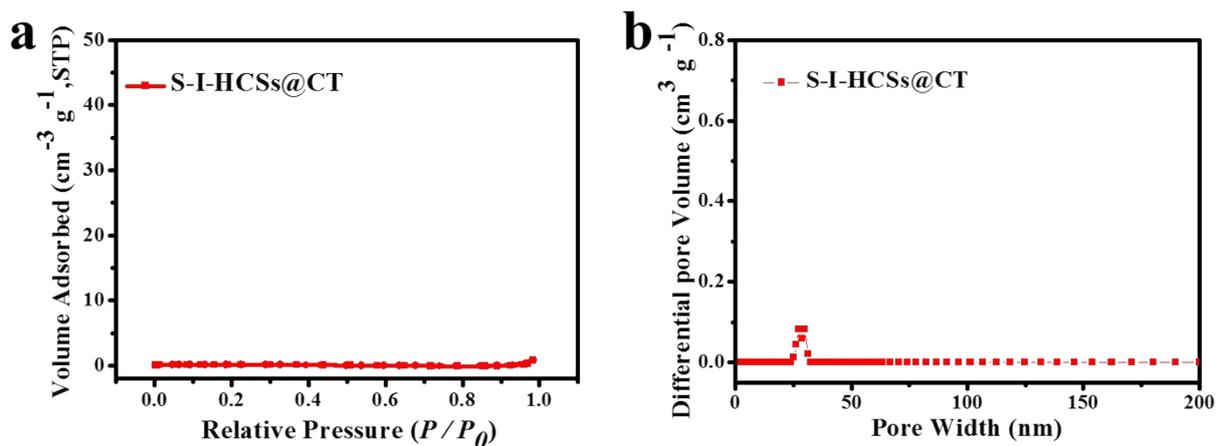


Fig. S7. a) Nitrogen adsorption/desorption isotherms of S-I-HCSs@CT; d) Pore size distribution of S-I-HCSs@CT.

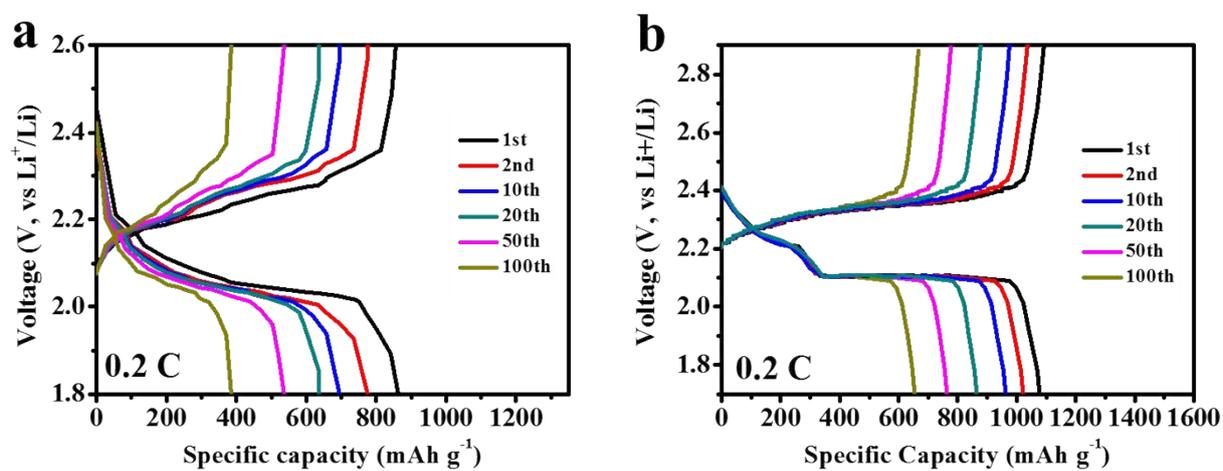


Fig. S8. Discharge/charge voltage profiles of a) S-CTs and b) S-H-HCSs@CT at 0.2C.

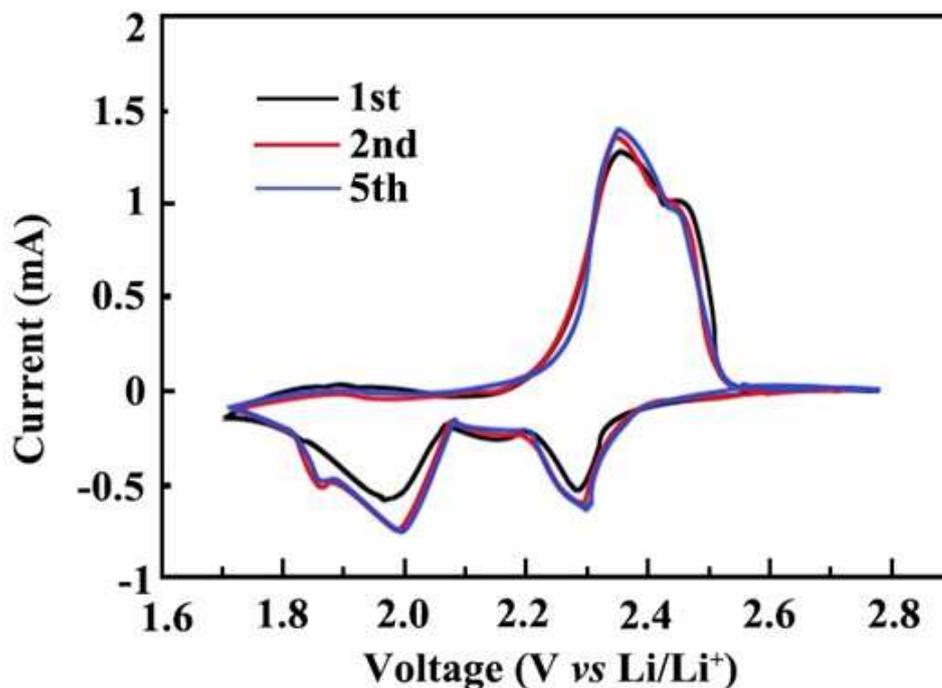


Fig. S9. CV profiles of S-I-HCSs@CT at a scan rate of 0.5 mV s⁻¹.

In the first cathodic scan, there are two well-defined reduction peaks at 2.28 and 2.01 V, corresponding to the multistep reduction mechanism of elemental sulfur. The first peak is the transition from elemental S to long-chain polysulfides (Li_2S_X , $4 < X \leq 8$); the latter is associated with further reduction of the higher polysulfide species (Li_2S_X , $4 < X \leq 8$) to the lower polysulfide species (Li_2S_X , $X \leq 8$). In the anodic scan, there is a shoulder peak at ≈ 2.39 V and a strong, broader peak at 2.43 V. They are associated with the reverse reactions in the charging stage. In the second anodic scan, the two anodic peaks shift to lower potentials at ≈ 2.36 and ≈ 2.41 V.

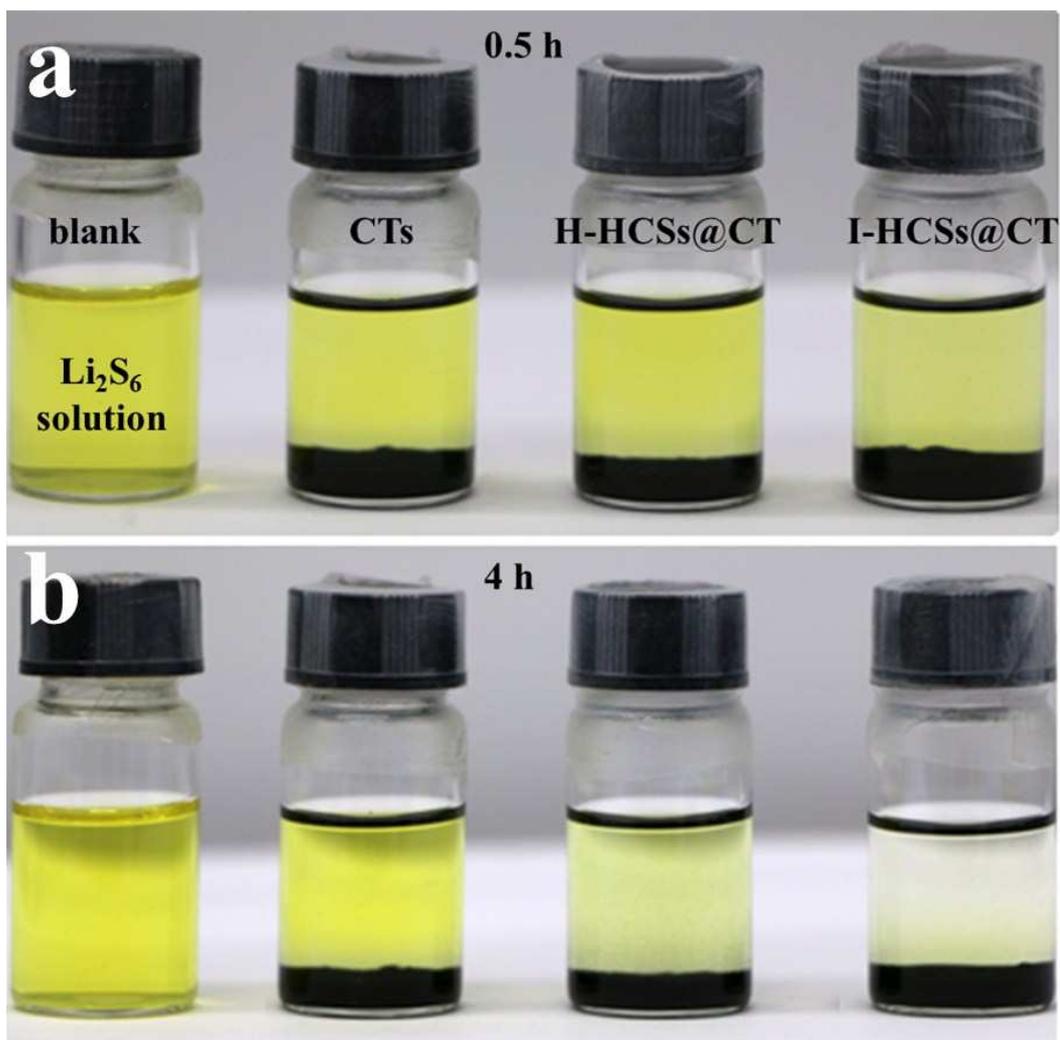


Fig. S10. Digital images of sealed vials with a lithium polysulfides solution (Li_2S_6 dissolved in DOL/DME solvents): a) 0.5 h and b) 4h, after the addition of CTs, H-HCSs@CT and I-HCSs@CT powders.

In Fig. S10a and b, the color of Li_2S_6 solution changed to light yellow after mixing with I-HCSs@CT for 0.5 h, and the solution became almost colorless after 4 h, indicating the strong interaction between I-HCSs@CT and polysulfides.

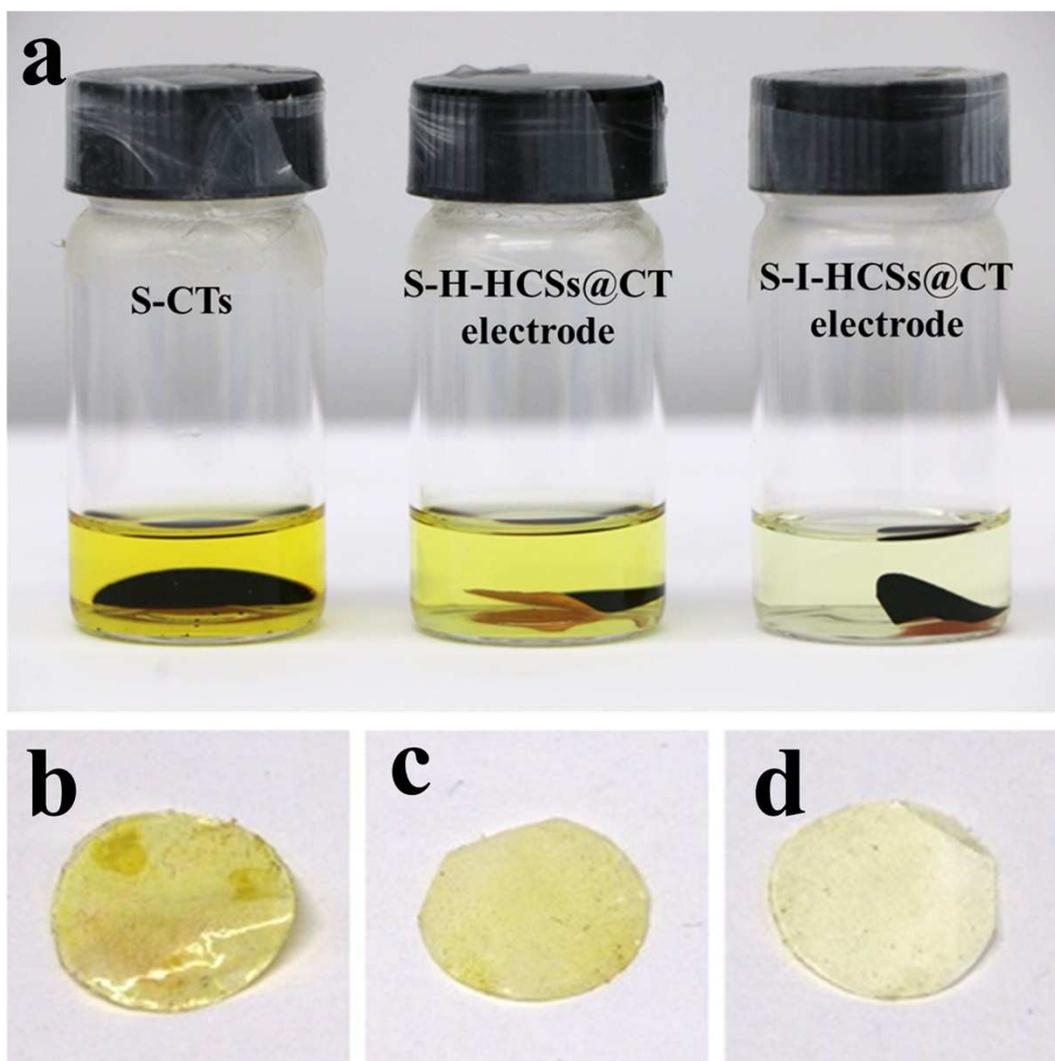


Fig. S11. a) The electrode films of S-CTs, S-H-HCSs@CT and S-I-HCSs@CT cells after 100 cycles, immersing in 5 mL DOL solvent after 1h. Separators from coin cells after 100 cycles of b) S-CTs, c) S-H-HCSs@CT and d) S-I-HCSs@CT.

In Fig. S11, the DOL solution contained the S-H-HCSs@CT and S-CTs electrode and the corresponding separator show much deeper “yellow” color than the S-I-HCSs@CT electrode, indicating the I-HCSs@CT host has much stronger interaction with polysulfides to restrict their dissolution into the organic electrolyte.

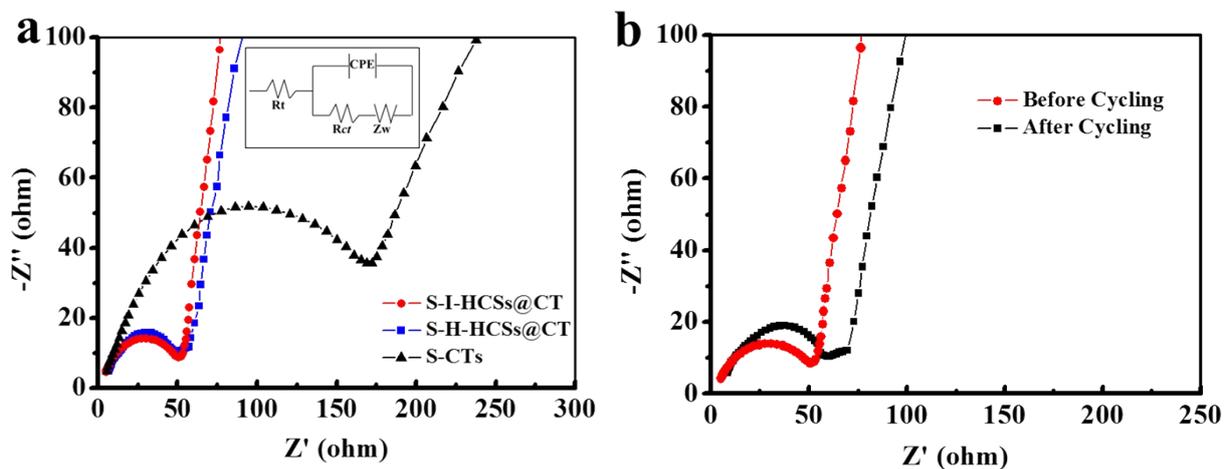


Fig. S12. a) Nyquist plots of S-I-HCSs@CT, S-H-HCSs@CT and S-CTs after the 3rd cycle. b) Nyquist plots of S-I-HCSs@CT before and after 500 cycles at 0.5 C.

Compared with S-CTs, S-I-HCSs@CT and S-H-HCSs@CT has a smaller semicircle, indicating a lower charge transfer resistance (R_{ct}) at the electrode interface. The positive effect of HCSs in S-I-HCSs@CT accounts for better high-rate electrochemical performances than S-CTs.

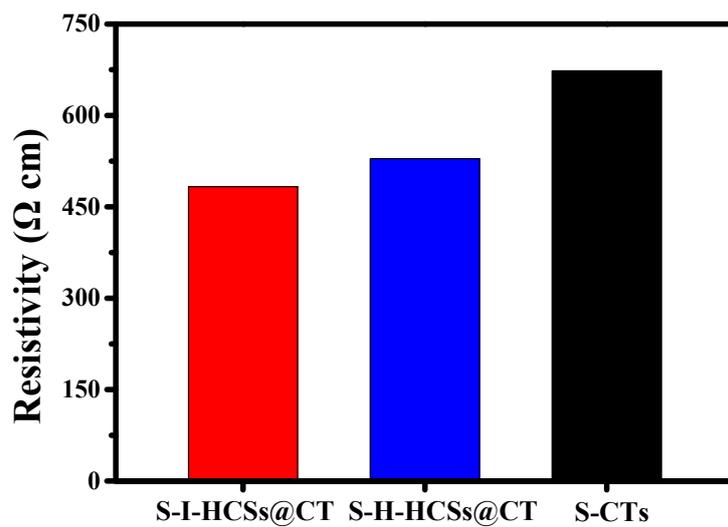


Fig. S13 Electrical resistivity of S-I-HCSs@CT, S-H-HCSs@CT and S-CTs.

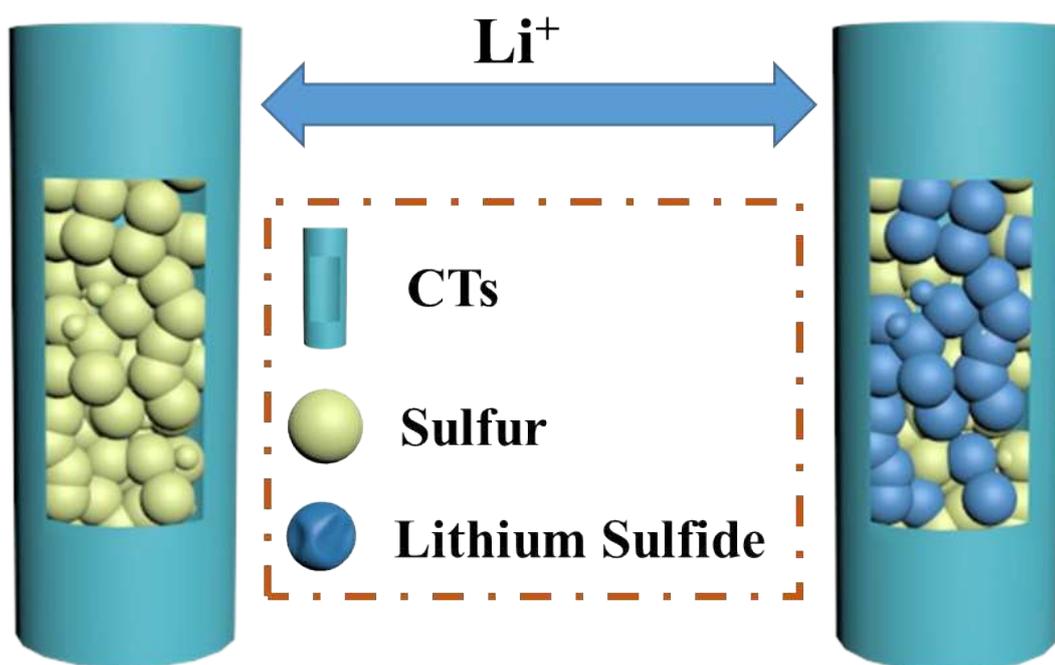


Fig. S14. Schematic diagram of the mechanism for reversible electrochemical reaction of S-CTs.

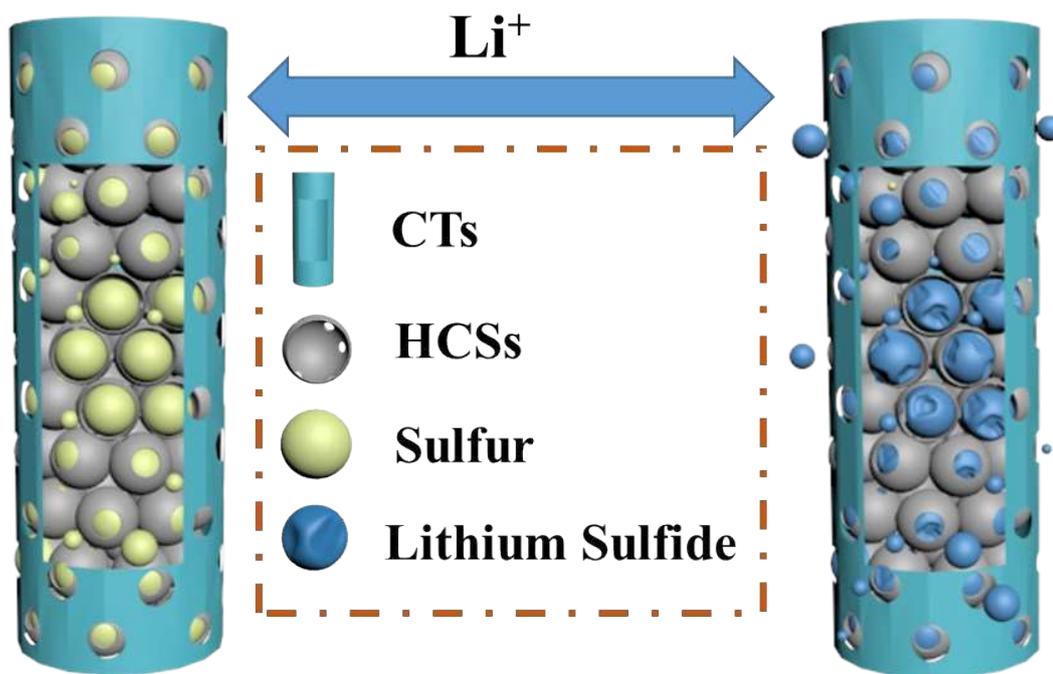


Fig. S15. Schematic diagram of the mechanism for reversible electrochemical reaction of S-HCSs@CT.

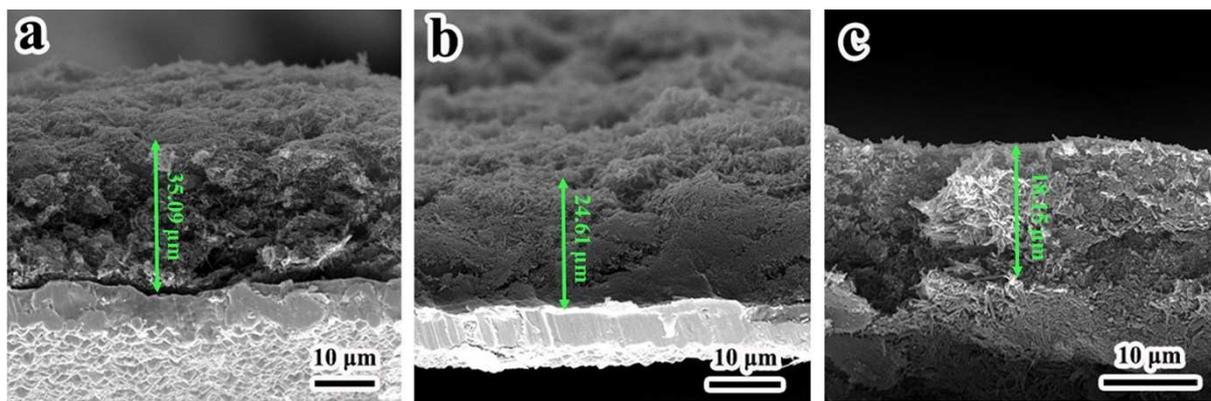


Fig. S16. Cross section SEM images of a) S-CTs, b) S-H-HCS@CT and c) S-I-HCSs@CT after 100 charge/discharge cycles. Compared with the volume change of S-CTs (35.09 μm) and S-H-HCS@CT (24.61 μm), the S-I-HCSs@CT show smaller volume change (18.15 μm).

Table S1. A comparison of various carbon-based sulfur host materials for lithium sulfur batteries.

Sulfur host materials	Sulfur loading mass	Capacity (mAh g⁻¹)	Cycling performance (mAh g⁻¹)	Retention	Ref.
Hollow polymer spheres	/	1179 (0.1 C)	573 (0.5 C, 1000 cycles)	63.7%	1
Hollow carbon spheres	68.1%	1198 (0.2 C)	628 (0.5 C, 200 cycles)	69.8%	2
Disordered carbon nanotubes	59.3%	1543 (0.1 C)	713 (0.1 C, 100 cycles)	46.3%	3
Hollow carbon nanofiber	78.6%	1403 (0.2 C)	730 (0.2 C, 150 cycles)	52.1%	4
N-doped Hollow carbon spheres	/	1126 (0.2 C)	1012 (0.2 C, 100 cycles)	89.9%	5
Double-Shelled Hollow Carbon Spheres	64.1%	1012 (0.1 C)	794 (0.1 C, 100 cycles)	78.5%	6
Holey carbon nanotube	47.4%	1183 (0.2 C)	943 (0.5 C, 200 cycles)	82.0%	7
Graphene oxide	/	1327 (0.02 C)	989 (0.1 C, 55 cycles)	67.1%	8
Reduced graphene oxide	/	1573 (0.1 C)	912 (1 C, 500 cycles)	61.9%	9
MnO ₂ filling hollow carbon nanofibers	71.2%	1214 (0.05 C)	736 (0.5 C, 300 cycles)	79.7%	10
Wrapped graphene	70.1%	700 (0.1 C)	600 (0.5 C, 140 cycles)	74.7%	11
Porous hollow carbon	/	1186	991	71.4%	12

		(0.1 C)	(0.5 C, 100 cycles)		
Ordered Mesoporous Carbon Nanoparticles	70.0%	1163	845	72.6%	13
		(1C)	(1 C, 100 cycles)		
Multichannel Carbon Nanofiber	80.0%	1385	1013	72.7%	14
		(0.2 C)	(0.2 C, 300 cycles)		
Tube-in-tube carbon nanotube	71.2%	1273	647	58.1%	15
		(0.5 A g ⁻¹)	(2 A g ⁻¹ , 200 cycles)		
Sphere-in-tube carbon nanostructures	72.1%	1426	746	80.4%	This work
		(0.1 C)	(0.5 C, 500 cycles)		

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