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Abstract: Amorphous calcium carbonate (ACC) plays a key role in biomineralisation processes in sea organisms. Neutron and x-ray diffraction have been performed for a sample of magnesium-stabilised ACC, which was prepared with a Mg:Ca ratio of 0.05:1 and 0.25 H₂O molecules per molecule of CO₃. The Empirical Potential Structure Refinement method has been used to make a model of magnesium-stabilised ACC and the results revealed a fair agreement with the experimental diffraction data. The model has well-defined CO₃ and H₂O molecules. The average coordination number of Ca is 7.4 and is composed of 6.8 oxygen atoms from CO₃ molecules and 0.6 oxygen atoms from H₂O molecules. The average Ca-O bond length is 2.40 Å. The distribution of Ca in the model is homogeneous with a uniformly distributed Ca-rich network and no evidence of the Ca-poor channels as previously reported for a Reverse Monte Carlo model of ACC.

Journal of
Non-Crystalline Solids

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As corresponding author, I Dr. Gavin Mountjoy, hereby confirm on behalf of all authors that:

1. This manuscript has not been published, was not, and is not being submitted to any other journal. If presented at a conference, the conference is identified. If published in conference proceedings, substantial justification for re-publication must be presented.
2. All necessary permissions for publication were secured prior to submission of the manuscript.
3. All authors listed have made a significant contribution to the research reported and have read and approved the submitted manuscript, and furthermore, all those who made substantive contributions to this work have been included in the author list.

Response to reviewers

Reviewer 1:

Overall I believe that the work is sufficiently interesting to warrant publication, albeit with the addition of an extra figure to show the $g(r)$ comparison between model and data.

We have added this as Figure 1b. Also the following sentence at the start of Section 4: "In terms of the total pair distribution function $g(r)$, there is fair quantitative agreement in short range order up to 3 Å."

Finally, I did pick up a few typographic errors in this manuscript that should be corrected: Page 3, Line 3: Apparent lost fragment in the text ".the relation between ACC and the structures of crystalline calcium carbonate phases." I believe that the full stop should be replaced with "as well as".

Page 3, Line 8: Unnecessary capital "I" on "It"

Page 4, second line from bottom of page: missing "ions" in the phrase "Mg ion compare to other ions and thus the"

Page 10, fourth line from bottom of page: spelling mistake "expected" should be "expected"

We have corrected these typographical errors.

Reviewer 2:

It is interesting that substitution of deuterium for hydrogen during the synthesis procedure was apparently not considered. Some comment on this might be appropriate, specifically the loss of information associated with the incoherent contribution. I suspect that the authors are already aware that further gains can be had by combining separate results from hydrogen and deuterium prepared samples.

A comment that substitution of H for D will be attempted in future work has been added to the end of Section 4:" In the present study, the proton content of the sample was reduced by using heat treatment. An alternative approach is to deuterate the sample by substituting protons for deuteriums, D, which has the benefits of reducing incoherent and inelastic scattering, and providing scattering contrast due to the different neutron scattering length for D compared to protons. In the present study, an unsuccessful attempt was made to produce a deuterated ACC sample, but the authors plan to attempt this again in future work".

A controversial feature in the Goodwin X-ray study was the speculation regarding the Ca-poor channel network. This likely resulted from an incorrect density used in the RMC modeling. How sensitive is the EPSR result to density, and were different densities considered? Also, were different H₂O contents considered in the EPSR modeling?

A comment has been added that different densities and H₂O contents were not considered. Also that the values used by Goodwin et al were chosen to be equal to those of monohydrocalcite.

Section 2.1: “Note that the above parameters differ from the parameters $n=1.0$ and density of 2.43 g/cm^3 reported for ACC in [16], which were taken to be equal to those in crystalline monohydrocalcite.”

Section 2.4: “The density and H₂O content was based on experimental data as discussed in Section 2.1, and the effect on EPSR results of varying these parameters was not explored.”

Minor comments: During reading, I noted the occasional grammar issue and some sentences seemingly not completed. For example, on page 6 (last sentence before section 2.2), there appears to be something missing from this sentence, such as . to differences in sample heating or treatment.

This sentence has been corrected.

Re-reading of the manuscript has lead us to make some corrections to grammar. These are indicated in the copy of the manuscript below with “revisions highlighted”.

Additional changes:

It was noticed that on Figure 3 the horizontal axis scale values were wrong due to being rounded to 0 decimal places. This has been corrected by showing values to 1 decimal place.

Neutron and X-ray Diffraction and Empirical Potential Structure Refinement

Modelling of Magnesium Stabilised Amorphous Calcium Carbonate

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Abstract

Amorphous calcium carbonate (ACC) plays a key role in biomineralisation processes in sea organisms. Neutron and x-ray diffraction have been performed for a sample of magnesium-stabilised ACC, which was prepared with a Mg:Ca ratio of 0.05:1 and 0.25 H₂O molecules per molecule of CO₃. The Empirical Potential Structure Refinement method has been used to make a model of magnesium-stabilised ACC ~~having and the results revealed a~~ fair agreement with the experimental diffraction data. The model has well-defined CO₃ and H₂O molecules. The average coordination number of Ca is ~~found to be~~ 7.4 ~~which and is~~ composed of 6.8 oxygen atoms from CO₃ molecules and 0.6 oxygen atoms from H₂O molecules, ~~and t.~~ The average Ca-O bond length is 2.40 Å. The distribution of Ca in the model is homogeneous with a uniformly distributed Ca-rich network and no evidence of the Ca-poor channels ~~which~~ ~~were~~ as previously reported for a Reverse Monte Carlo model of ACC.

1. Introduction

Amorphous calcium carbonate (ACC) is believed to play a key role in biomineralisation processes within organisms, including sea urchins, crustaceans and sea shells (bivalvia) [1], which use the crystalline calcium carbonate(s) as structural building blocks. Most notably, ACC has been proposed to be the form of calcium carbonate prior to its crystallisation to aragonite or calcite to form new structural elements of the organism [1]. ACC may thus have an analogous role to the mammalian equivalent, amorphous calcium phosphate (ACP), which is involved in biomineralisation of crystalline calcium phosphates within human bones [2].

Greater understanding of biomineralisation processes should include detailed knowledge of the atomic structure of all calcium carbonate phases, including ACC, and of the mechanisms and pathways of crystallisation of ACC in biomineralisation processes, as well as the relation between ACC and the structures of crystalline calcium carbonate phases. The composition of ACC has been reported as $\text{CaCO}_3 \cdot n\text{H}_2\text{O}$ where $n = 1-1.6$ [3-6]. The variation in water content, n , is related to the process by which ACC is obtained, from a calcium source and two common reactants for carbonate ions being CO_2 gas molecules and carbonate ions in solution. Furthermore, ACC water content depends of the presence of impurities and on aging, and. It is known that ACC dehydrates with time before transforming to crystalline CaCO_3 , and that this dehydration being is faster with increasing temperature and slower with increasing Mg content [4,7-9]. The composition of ACC phases is thus similar to that of monohydrocalcite $\text{CaCO}_3 \cdot \text{H}_2\text{O}$. At ambient temperature, pure ACC is less stable than any of the crystalline polymorphs of calcium carbonate CaCO_3 (i.e. calcite, aragonite, vaterite, monohydrocalcite and ikaite) and rapidly crystallises upon dehydration, typically within 2 minutes in aqueous solution and in less than 1 day at room temperature in air, to either vaterite or calcite [8-10] (the crystallisation to aragonite does not occur at ambient temperatures in a pure system).

Although the term amorphous calcium carbonate has been used for approximately a century [11] it remains to be presented a comprehensive set of structural data and corresponding structural model for an ACC phase. Structural characterisation of ACC using infrared and Raman spectroscopy are frequently reported [e.g. 12-13] but are difficult to interpret in terms of disordered atomic structures. X-ray absorption spectroscopy has provided the most direct probe of Ca in ACC phases, and a Ca-O bond length of 2.40 Å and a Ca-O coordination number of approximately 6 is typically reported [14]. A key experimental technique for

studying the atomic structure of amorphous materials is diffraction over a large angular range of scattering sufficient to obtain information about the pair distribution function [15].

However, such diffraction studies of ACC have been limited due to its inherent instability.

Notably there was a high energy x-ray diffraction study of two types of ACC reported by Michel et al in 2008 [3] that was used for Reverse Monte Carlo (RMC) modelling by Goodwin et al in 2010 [16]. The RMC study reported that the same results were obtained for two different starting configurations: a disordered monohydrocalcite model (i.e. $n=1$), and a random model containing the same molecular units. The RMC model obtained for ACC was reported to have a calcium rich network (referring to Ca-O-Ca linkages) and Ca-poor channels (regions more than 4 Å away from a calcium atom). In addition, a recent study of the structure of ACC [17] ~~by combining of ed~~ ⁴³Ca NMR measurements and molecular dynamics (MD) modelling and used the Goodwin et al (2010) model as a starting point, finding and found that the Ca-rich and Ca-poor channels proposed by Goodwin et al (2010) undergo “significant structural reorganization” and “shrink in size” during the 2 ns MD simulation.

One way to facilitate diffraction studies is to work with more stable forms of ACC, which allow more time consuming experiments to be performed, ~~such as particularly~~ neutron diffraction. The use of both neutron and x-ray diffraction provides important complementary information due to the different scattering powers of elements for neutrons and x-rays. It has been reported that ACC can be stabilised with various additives such as magnesium [18-20,25], phosphate [19], and poly(aspartic acid) [19]. The present study is focused on Mg-stabilised ACC. The addition of a very small amount of magnesium greatly increases the lifetime of ACC. The most likely cause is the high dehydration energy of the Mg^{2+} -aquo ion compared to the Ca^{2+} ion [21] that would slow down the dehydration dynamics of ACC [20].

Another benefit of adding magnesium is the smaller size of the Mg ion compared to other ions and thus the Mg ions should have a minor effect on the structure of “pure” ACC. The use of Mg stabilised ACC has made it possible to measure neutron and x-ray diffraction data which is-are reported here for the first time.

Modelling of diffraction data is also a key technique in the study of amorphous materials. The Empirical Potential Structure Refinement (EPSR) method [22] was specifically developed for modelling the structure of molecular liquids [23], but it is also applicable to amorphous solids with molecular units, such as ACC. The EPSR method has some similar features to RMC [24]. EPSR is like-similar to RMC in that it using-uses a Monte Carlo method which samples different structures against an “acceptance” criterion, but unlike RMC, the acceptance criteria is the energy minimum based on the interatomic potentials. EPSR is like RMC in using the diffraction data to guide the modelling process, but unlike RMC, the difference between the simulated and experimental diffraction is used to adjust the interatomic potential. The present study reports a new EPSR model of Mg-stabilised ACC. Some features of this model are found to be different to the previously reported RMC model of ACC.

2. Method

2.1 Sample preparation

Magnesium stabilised ACC (Mg-ACC) was synthesised using the following method based on Rodriguez-Blanco et al [25] and Jiang et al [18]. ~~Two solutions of~~ 100mM solutions of CaCl₂ and ~~100mM of~~ Na₂CO₃ were prepared ~~and 5mL of each solution were transferred to separate~~

~~polypropylene tubes~~. 1 mL of a 500 mM solution of MgCl_2 was added to 5 mL of the CaCl_2 solution and this was then rapidly mixed with 5 mL of the Na_2CO_3 solution (to give an ion ratio of 1:1:1 Ca:Mg: CO_3). This mixing produced a white precipitate which was then rapidly filtered and washed with isopropanol and dried in flowing air [25].

The precipitate was found to have a Mg:Ca ratio of 0.05:0.95 as determined by x-ray fluorescence. The precipitate was further dried and heat-treated for 1 hour at 150 °C to reduce the water content. ~~TGA-DTA~~ Thermogravimetric and differential thermal analyses showed weight losses of 18% and 21% up to temperatures of 150 °C and 350 °C respectively, the latter being when calcite crystallises. These weight losses are consistent with a water content of $n=1.45$ before and $n=0.25$ after heat treatment at 150 °C. This reduction of water content is important for neutron diffraction ~~to~~ because it reduces the detrimental effects of inelastic and incoherent scattering from protons. The heat-treated sample, denoted Mg-ACC, was confirmed amorphous by lab-based powder x-ray diffraction and ~~the lifetime of sample was found to be~~ to remain amorphous for a period of at least two weeks. The density of Mg-ACC after heat treatment was measured to be 2.19 g/cm^3 using a He pycnometer. Note that the above parameters ~~This differs~~ from the ~~reported parameters~~ $n=1.0$ and density of 2.43 g/cm^3 reported for ACC in [316]; ~~and we attribute this to differences in sample, which were taken to be equal to those in crystalline monohydrocalcite.~~

2.2 Neutron diffraction

Neutron diffraction data were measured using the Small Angle Neutron Diffractometer for Liquids and Amorphous Samples (SANDALS) [26] at the ISIS pulsed neutron source, UK. The sample was produced 24 hours before measurement and sealed until it was loaded into a

flat plate sample holder with 25 micron thick vanadium plates that were 4mm apart. The sample holder was sealed and maintained at a temperature of 5 °C during the experiment. Time-of-flight data were collected over a wide range of momentum transfer, Q , with a useful maximum of 30 \AA^{-1} (where $Q = 4\pi \sin \theta/\lambda$). Experimental corrections of the data were performed using the Gudrun program [27]. The SANDALS diffractometer has a forward scattering geometry, which helps ~~to~~ reduce the severity of the inelasticity correction, which can be problematic for samples containing hydrogen [26]. There remained a background contribution due to inelastic scattering from H, and this is removed by EPSR (using a back Fourier transform of the part of the real space data in the region $r < 1.5 \text{ \AA}$). The structure factor $S(Q)$ from neutron diffraction is shown in Figure 1, including a factor of \bar{b}^2 (where \bar{b} is the average coherent neutron scattering length).

2.3 X-ray diffraction

X-ray diffraction data were collected at the MCX beamline [28] at the ELETTRA synchrotron, Italy, using a wavelength of $\lambda = 0.620 \text{ \AA}$. The sample was produced 2 days before travelling to the synchrotron and sealed until it was loaded into a 10-micron thick 1.0 mm diameter borosilicate glass capillary. The capillary was rotated at 180 rpm during the experiment. X-ray diffraction data were collected up to 2θ of 162° or $Q = 20 \text{ \AA}^{-1}$. The processing of the raw data included corrections for the background scattering, polarization, sample thickness, absorption, capillary, Compton scattering, and the sharpening function (equal to the square of the average form factor per atom) [15]. The structure factor $S(Q)$ from x-ray diffraction is shown in Figure 1.

2.4 Empirical Potential Structural Refinement Modelling

Empirical potential structure refinement (EPSR) [22,29] is a suitable method for generating and refining models of amorphous structures based on molecular units. The present study used the EPSR18 package. Structurally realistic molecular units were initially constructed using GHEMICAL4.0 software [30]. A cubic box with length $L=39.55 \text{ \AA}$ containing 950 Ca, 50 Mg, 1000 CO_3 and 250 H_2O units (total 5750 atoms) was assembled using the EPSR shell.

This model corresponds to the measured density for Mg-ACC of 2.19 g/cm^3 . The density and H_2O content was based on experimental data as discussed in Section 2.1, and the effect on EPSR results of varying these parameters was not explored. The molecules were randomly translated so they move apart throughout the box. To further separate the molecules, the molecules were randomly rotated and translated 10,000 times.

The EPSR procedure initially relaxed the random starting configuration to an energy minimised structure using a simulated annealing approach with a sequence of temperatures: 10000K, 5000K, 1000K, 500K and 300K. At each temperature EPSR carries out Monte Carlo sampling of structures until reaching an equilibrium minimum energy. Starting inter-molecular potentials for each inter-atomic pair correlation i - j were generated by EPSR using built-in parameters for Lennard-Jones potentials for each pair of elements. Coulomb interactions are included with formal charges for ions.

The potential refinement was then implemented within the EPSR. The difference between the simulated and experimental neutron and x-ray diffraction data are used to adjust the interatomic potentials. The comparison between simulation and experiment is carried out using the structure factors $S(Q)$ (see Figure 1). The adjusted potentials are then used for a

new Monte Carlo sample of structures until reaching equilibrium minimum energy. This process was iterated 20,000 times.

3. Results

The final level of agreement obtained between simulated and experimental diffraction data is shown in Figure 1. The model obtained is shown in Figure 2. The partial pair distribution functions (PDFs) $g_{ij}(r)$ have been calculated from the model. Figure 3 shows the PDFs involving CO_3 and H_2O molecules respectively. The continuous lines show intra-molecular correlations, and the dashed lines show inter-molecular correlations. The first four peaks in the intramolecular PDFs (solid lines) represent the internal structure of the molecules: O-H and H-H distances in H_2O , and C-O and O-O distances in CO_3 . ~~In addition, t~~The inter-molecular PDFs (dashed lines) show H...H and H...O correlations between H_2O (light dashed), and C...O and O...O correlations between all molecules (heavy dashed). These correlations and their peak distances are summarised in Table 1.

Figure 4 shows the PDFs involving Ca. Most prominent are the Ca-O distances due to Ca bonding to the oxygen atoms which are present in CO_3 and H_2O molecules, where the peak distance is 2.40 Å. The derived average coordination number of Ca is ~~found to be~~ 7.4 which is composed of 6.8 oxygen atoms from CO_3 molecules and 0.6 oxygen atoms from H_2O molecules. These features of the model are in good agreement with the EXAFS results of Michel et al [3] for ACC prepared by a similar method: Ca-O coordination number of 6.7 ± 1.7 and peak distance of 2.41 ± 0.02 Å. Due to the coordination of Ca to oxygen in CO_3 and H_2O molecules, there are consequently noticeable correlations of Ca...C and Ca...H. The PDFs for

Mg are similar to those for Ca, but they are not presented in detail because of the poor statistics due to the small number of Mg in the model.

The first peak at 3.3-4.5 Å in the Ca-Ca PDF (see Figure 4) corresponds to two calcium ions which share bonds with a common oxygen, i.e. Ca-O-Ca. Figure 5 ~~has been drawn to show~~ these Ca-O-Ca configurations as single blue lines ~~which originate~~ originating from and ~~terminate~~ terminating at the positions of Ca ions. This feature of the structure was previously referred to as the "Ca-rich network" [16], and Figure 5 indicates that ~~the~~ such a "Ca rich network" ~~which~~ is homogeneously distributed throughout the present model. ~~Figure 5 has also been drawn to show~~ The grey surfaces around regions ~~which are~~ more than 3.3 Å distant from Ca, ~~a~~ represent the structural feature which was previously referred to as "Ca-poor channels" [16]. Figure 5 shows that there are only a few marginal volumes fitting this criterion. In the previous study [16] a less stringent limit of 4 Å was used and the present model has no such volumes.

4. Discussion

This study uses both neutron and x-ray diffraction data as a more stringent test of simulation than previous studies (e.g. [16]). Note that the two techniques employed have different sensitivity to inter-atomic correlations, due to the different atomic scattering factors for neutrons and x-rays. In particular, neutrons are more sensitive to oxygen and hydrogen, whereas x-rays are more sensitive to calcium. The final level of agreement obtained is shown in Figure 1. In terms of the structure factor S(Q), ~~There~~ there is fair quantitative agreement for $Q > 4 \text{ \AA}^{-1}$ corresponding to short range order, and fair qualitative agreement $1 < Q < 4 \text{ \AA}^{-1}$

corresponding to medium range order. In terms of the total pair distribution function $g(r)$, there is fair quantitative agreement in short range order up to 3 Å.

The EPSR method has been successful at incorporating well-defined CO₃ and H₂O molecular units with the expected intra-molecular correlations. The inter-molecular correlations ~~in~~ (see Table 1) include H...O and O...O peaks due to hydrogen bonding of H₂O molecules, although the peak distances are slightly shorter than expected for typical H...O hydrogen bond lengths of 1.6-2.0 Å. H₂O molecules are outnumbered 4 to 1 by CO₃ molecules, so the presence of hydrogen bonding would suggest a tendency for H₂O molecules to be found near to one another.

This is the first model to be reported for Mg-stabilised ACC, and the first model of any kind of ACC to be compared with experimental neutron diffraction data. The Ca-O coordination is composed of relative contributions of CO₃ and H₂O of 6.8 and 0.6 respectively. These are in the ratio of 12:1. In comparison the available ~~oxygens~~ oxygens are 3 per CO₃ molecule and 1 per H₂O molecule, and these molecules are in ~~the a~~ ratio of 4:1. ~~So~~ Thus, the available oxygen from CO₃ and H₂O molecules are in the ratio of 12:1, the same ratio as observed in the Ca-O coordination. The average Ca...C coordination number is 6.6 which is almost the same as the average number, 6.8, of Ca bonds to oxygen in CO₃ molecules. This means that on average each carbonate oxygen which is bonded to calcium is from a separate carbonate molecule. . The number of Ca bonds to carbonate oxygens is 6.8 times the number of Ca in the model. Since the ratio of Ca to carbonate ~~oxygen~~ oxygen in the model is 1:3, this means that the number of carbonate ~~oxygen~~ oxygen bonds to Ca is 6.8 divided by 3 or 2.2. So each O in a CO₃ unit is typically bonded to 2 different Ca neighbours, i.e. Ca-O-Ca linkage.

The present EPSR model of Mg-ACC has Mg:Ca ratio of 0.05:0.95 and a water content of $n=0.25$, which, This is a somewhat different composition to the previously reported RMC model of ACC₂ which had no Mg and $n=1.0$ [16]. Nevertheless, it is interesting to compare the Ca distribution in these two models. The previously reported RMC model of ACC had a "Ca-rich network" (based on Ca-O-Ca linkages)₂ which was inhomogeneously distributed, and interspersed with significant volumes of "Ca-poor channels" (based on the limit of volumes more than 4 \AA distant from Ca). The homogeneous Ca distribution in the present EPSR model of Mg-ACC fails to reproduce the inhomogeneous features in the previously reported RMC model of ACC.

In the present study, the proton content of the sample was reduced by using heat treatment. An alternative approach is to deuterate the sample by substituting protons for deuteriums, D, which has the benefits of reducing incoherent and inelastic scattering, and providing scattering contrast due to the different neutron scattering length for D compared to protons. In the present study, an unsuccessful attempt was made to produce a deuterated ACC sample, but the authors plan to attempt this again in future work.

5. Conclusions

Preparation of Mg-stabilised ACC has enabled the first neutron diffraction measurement of Mg-doped ACC. The EPSR modelling method has used this neutron diffraction data, together with x-ray diffraction data, to build a model of Mg-stabilised ACC₂ which is in fair agreement with the experimental diffraction data. The model contains the expected molecular units of CO_3 and H_2O with some hydrogen bonding of H_2O . Calcium has an average coordination of 6.8, with a peak distance of 2.40 \AA , in agreement with previously

reported EXAFS results [3]. The Ca coordination of approximately 6.7 is comprised of oxygens from separate CO₃ molecules, and Ca-O-Ca linkages are prevalent throughout the structure. The Ca distribution is homogeneous and does not show the inhomogeneous characteristics which were previously reported for an RMC model of ACC [16].

6. Acknowledgements

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Figure captions:

Figure 1: (a) Experimental structure factors $S(Q)$ of Mg-stabilised ACC (black lines) for neutron and x-ray diffraction together with the simulated functions (red lines) for the EPSR model. (Note $S(Q)$ for neutron diffraction includes the factor \bar{b}^{-2} in barns per steradian). (b)

Total pair distribution functions $g(r)$.

Figure 2: EPSR model of Mg-stabilised ACC containing 5750 atoms with box length of 39.55Å (oxygen is red, carbon is grey, hydrogen is white, calcium is blue, magnesium is green).

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Figure 5: The Ca-rich network in the EPSR model of Mg-stabilised ACC (see Fig.2), showing Ca-O-Ca linkages as blue sticks. Marginal volumes of Ca-poor channels (based on the criterion of regions more than 3.3 Å distant from Ca) are shown as grey surfaces.

Table 1: interatomic correlations involving CO₃ and H₂O molecular units. (* denotes correlations involving hydrogen bonding as discussed in the text. Distances shown in brackets are approximate, and the corresponding correlation is also shown in brackets.)

distance (Å)	atoms types i-j	type of correlation between molecular units
0.93 and (1.2)*	H-O	intra and inter* H ₂ O
1.32	C-O	intra CO ₃
1.56 and (1.9)*	H-H	intra and inter* H ₂ O
2.22	O-O	intra CO ₃
(2.4)	C-H	inter CO ₃ and H ₂ O
(2.8)	O-O	inter CO ₃ (and H ₂ O)
(3.3)	C-O	inter CO ₃ (and H ₂ O)
(3.5)	C-C	inter CO ₃

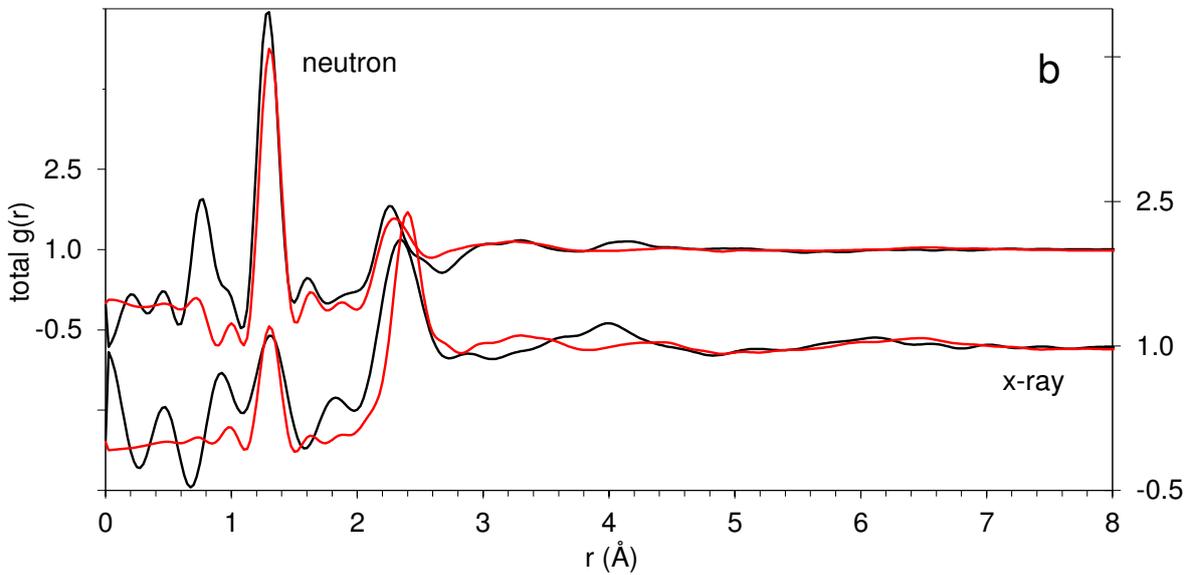
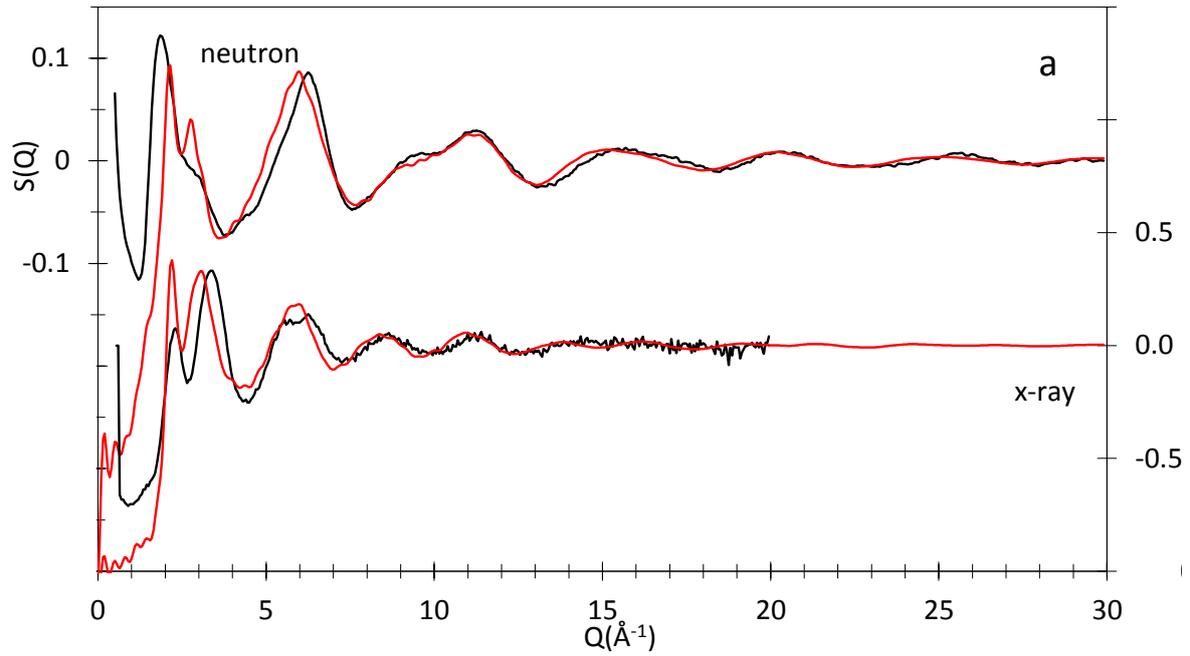


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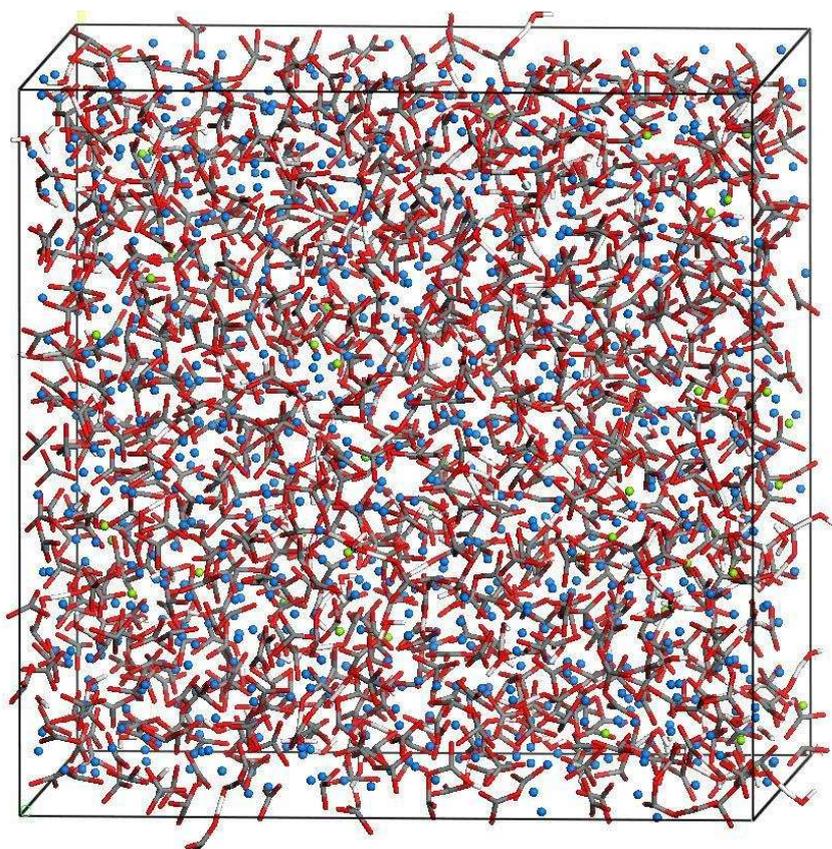


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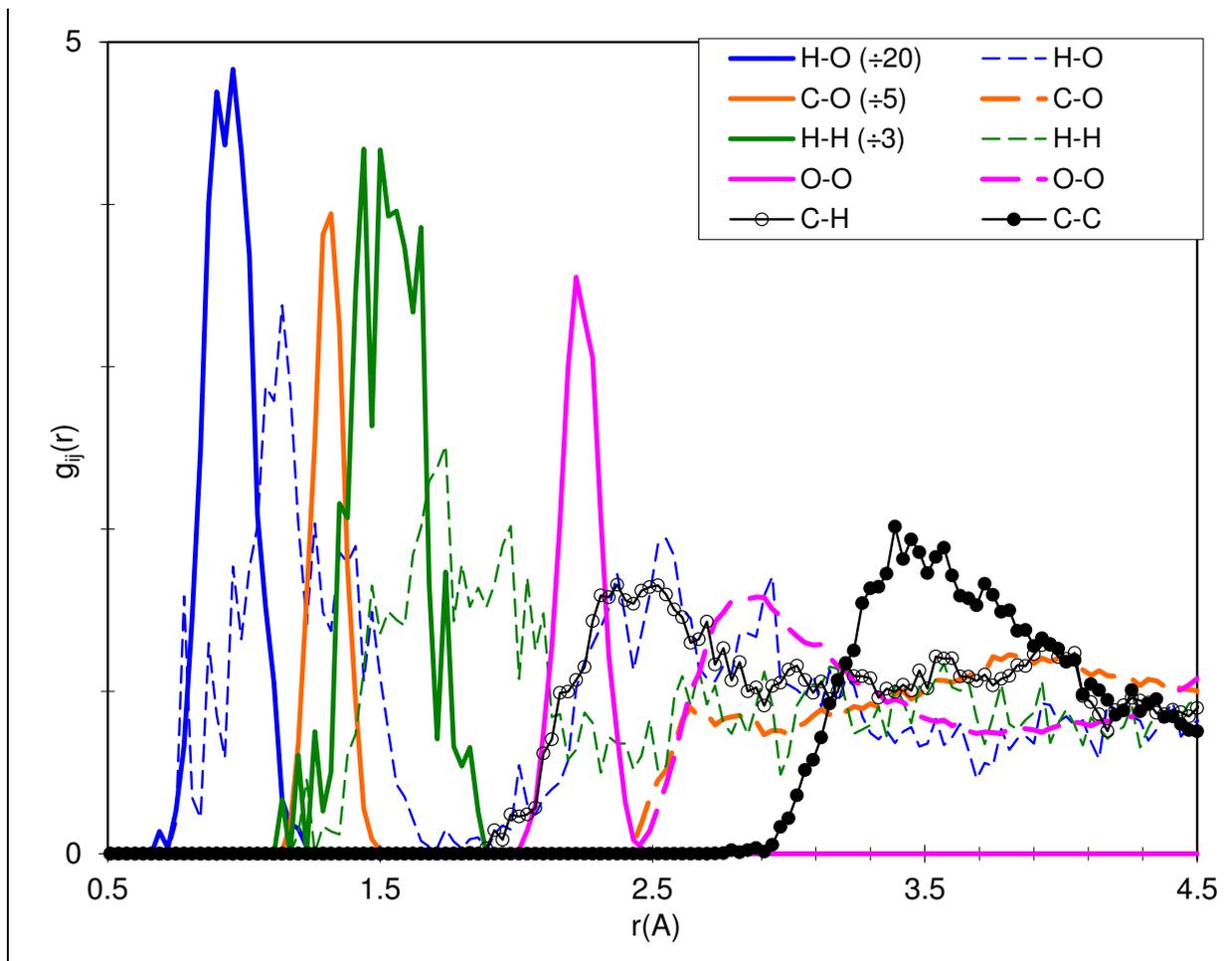


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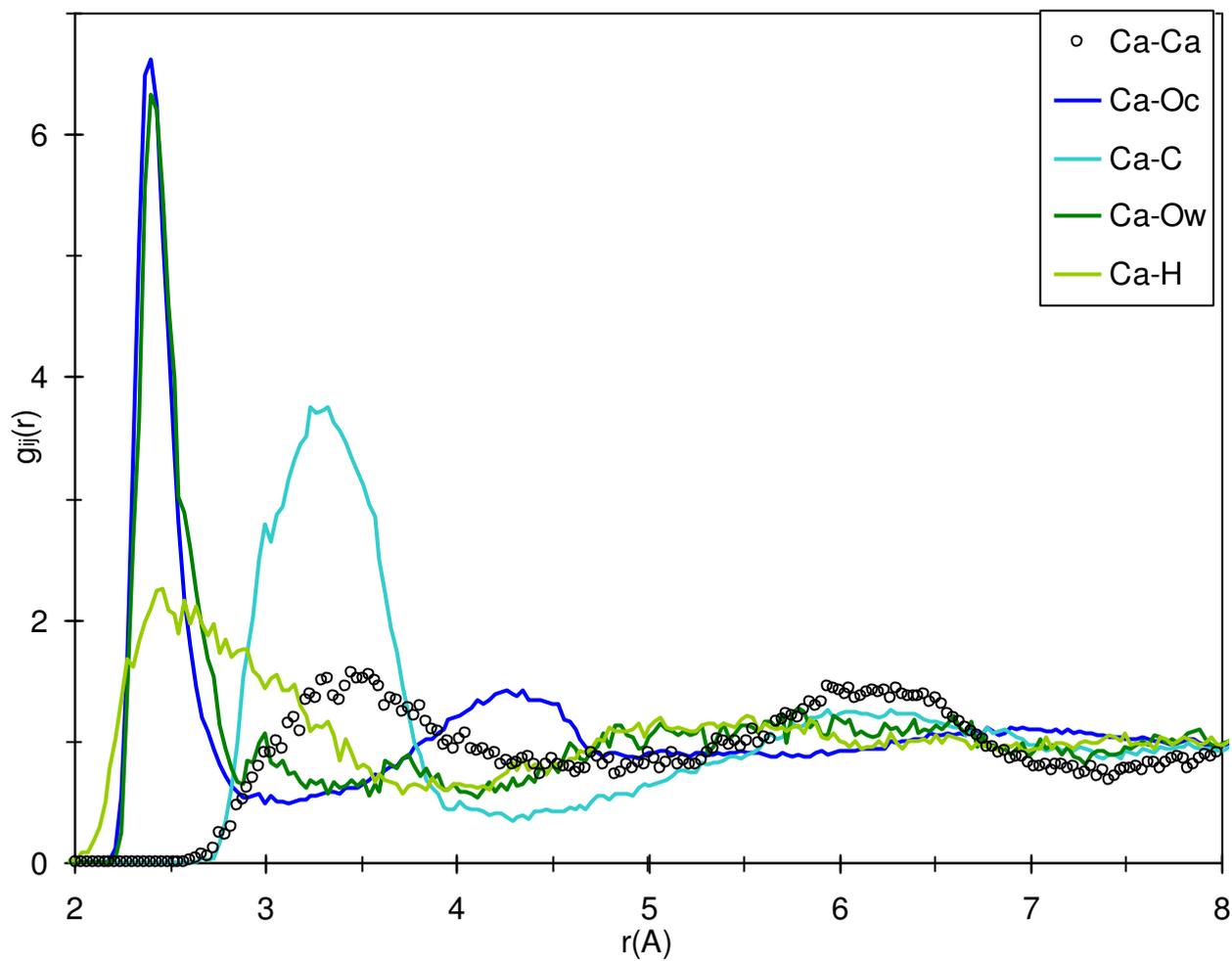


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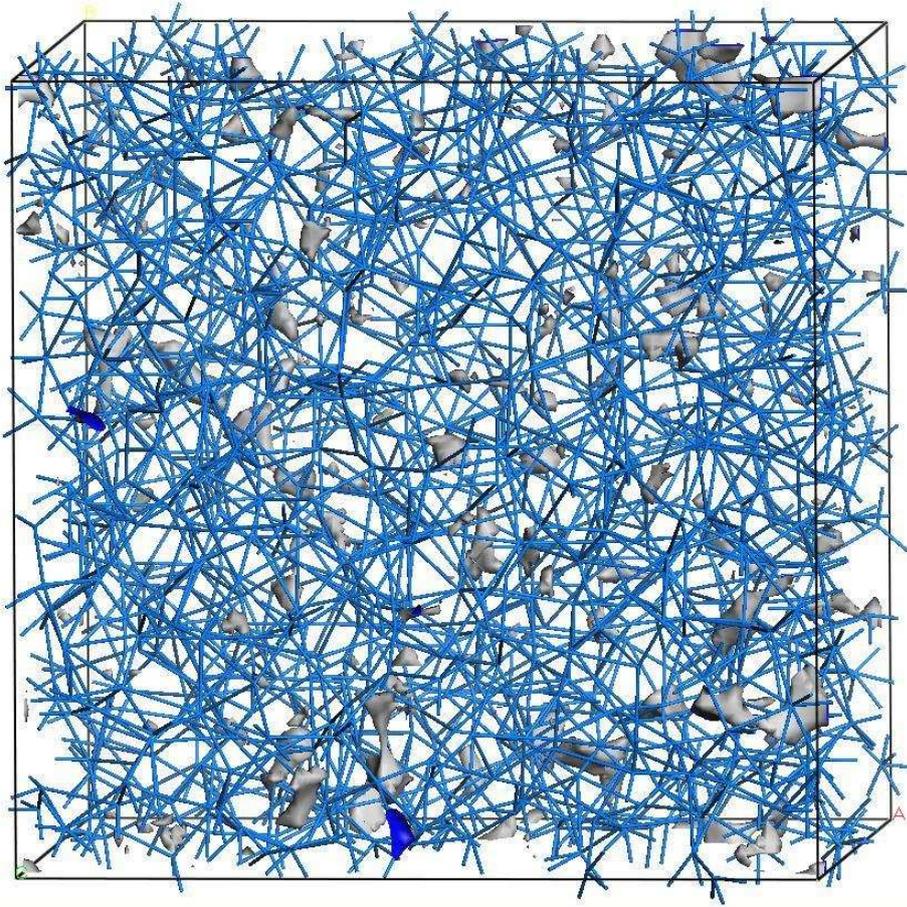


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The structure of Mg-stabilised amorphous calcium carbonate has been characterised.

The calcium oxygen bond length is 2.40 Å and the coordination number is 7.4.

Calcium is found to be homogeneously distributed throughout the structure.

Neutron and X-ray Diffraction and Empirical Potential Structure Refinement Modelling of Magnesium Stabilised Amorphous Calcium Carbonate

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Abstract

Amorphous calcium carbonate (ACC) plays a key role in biomineralisation processes in sea organisms. Neutron and x-ray diffraction have been performed for a sample of magnesium-stabilised ACC, which was prepared with a Mg:Ca ratio of 0.05:1 and 0.25 H₂O molecules per molecule of CO₃. The Empirical Potential Structure Refinement method has been used to make a model of magnesium-stabilised ACC and the results revealed a fair agreement with the experimental diffraction data. The model has well-defined CO₃ and H₂O molecules. The average coordination number of Ca is 7.4 and is composed of 6.8 oxygen atoms from CO₃ molecules and 0.6 oxygen atoms from H₂O molecules. The average Ca-O bond length is 2.40 Å. The distribution of Ca in the model is homogeneous with a uniformly distributed Ca-rich network and no evidence of the Ca-poor channels as previously reported for a Reverse Monte Carlo model of ACC.

1. Introduction

Amorphous calcium carbonate (ACC) is believed to play a key role in biomineralisation processes within organisms, including sea urchins, crustaceans and sea shells (bivalvia) [1], which use the crystalline calcium carbonate(s) as structural building blocks. Most notably, ACC has been proposed to be the form of calcium carbonate prior to its crystallisation to aragonite or calcite to form new structural elements of the organism [1]. ACC may thus have an analogous role to the mammalian equivalent, amorphous calcium phosphate (ACP), which is involved in biomineralisation of crystalline calcium phosphates within human bones [2].

Greater understanding of biomineralisation processes should include detailed knowledge of the atomic structure of all calcium carbonate phases, including ACC, and of the mechanisms

and pathways of crystallisation of ACC in biomineralisation processes, as well as the relation between ACC and the structures of crystalline calcium carbonate phases. The composition of ACC has been reported as $\text{CaCO}_3 \cdot n\text{H}_2\text{O}$ where $n = 1-1.6$ [3-6]. The variation in water content, n , is related to the process by which ACC is obtained from a calcium source and two common reactants for carbonate ions are CO_2 gas molecules and carbonate ions in solution. Furthermore, ACC water content depends of the presence of impurities and on aging. It is known that ACC dehydrates with time before transforming to crystalline CaCO_3 , and that this dehydration is faster with increasing temperature and slower with increasing Mg content [4,7-9]. The composition of ACC phases is thus similar to that of monohydrocalcite $\text{CaCO}_3 \cdot \text{H}_2\text{O}$. At ambient temperature, pure ACC is less stable than any of the crystalline polymorphs of calcium carbonate CaCO_3 (i.e. calcite, aragonite, vaterite, monohydrocalcite and ikaite) and rapidly crystallises upon dehydration, typically within 2 minutes in aqueous solution and in less than 1 day at room temperature in air, to either vaterite or calcite [8-10] (the crystallisation to aragonite does not occur at ambient temperatures in a pure system).

Although the term amorphous calcium carbonate has been used for approximately a century [11] it remains to be presented a comprehensive set of structural data and corresponding structural model for an ACC phase. Structural characterisation of ACC using infrared and Raman spectroscopy are frequently reported [e.g. 12-13] but are difficult to interpret in terms of disordered atomic structures. X-ray absorption spectroscopy has provided the most direct probe of Ca in ACC phases, and a Ca-O bond length of 2.40 \AA and a Ca-O coordination number of approximately 6 is typically reported [14]. A key experimental technique for studying the atomic structure of amorphous materials is diffraction over a large angular range of scattering sufficient to obtain information about the pair distribution function [15]. However, such diffraction studies of ACC have been limited due to its inherent instability.

Notably there was a high energy x-ray diffraction study of two types of ACC reported by Michel et al in 2008 [3] that was used for Reverse Monte Carlo (RMC) modelling by Goodwin et al in 2010 [16]. The RMC study reported that the same results were obtained for two different starting configurations: a disordered monohydrocalcite model (i.e. $n=1$), and a random model containing the same molecular units. The RMC model obtained for ACC was reported to have a calcium rich network (referring to Ca-O-Ca linkages) and Ca-poor channels (regions more than 4 Å away from a calcium atom). In addition, a recent study of the structure of ACC [17] combined ^{43}Ca NMR measurements and molecular dynamics (MD) modelling and used the Goodwin et al (2010) model as a starting point, and found that the Ca-rich and Ca-poor channels proposed by Goodwin et al (2010) undergo “significant structural reorganization” and “shrink in size” during the 2 ns MD simulation.

One way to facilitate diffraction studies is to work with more stable forms of ACC, which allow more time consuming experiments to be performed, particularly neutron diffraction. The use of both neutron and x-ray diffraction provides important complementary information due to the different scattering powers of elements for neutrons and x-rays. It has been reported that ACC can be stabilised with various additives such as magnesium [18-20,25], phosphate [19], and poly(aspartic acid) [19]. The present study is focused on Mg-stabilised ACC. The addition of a very small amount of magnesium greatly increases the lifetime of ACC. The most likely cause is the high dehydration energy of the Mg^{2+} -aquo ion compared to the Ca^{2+} ion [21] that would slow down the dehydration dynamics of ACC [20]. Another benefit of adding magnesium is the smaller size of the Mg ion compared to other ions and thus the Mg ions should have a minor effect on the structure of “pure” ACC. The use of Mg stabilised ACC has made it possible to measure neutron and x-ray diffraction data which are reported here for the first time.

Modelling of diffraction data is also a key technique in the study of amorphous materials. The Empirical Potential Structure Refinement (EPSR) method [22] was specifically developed for modelling the structure of molecular liquids [23], but it is also applicable to amorphous solids with molecular units, such as ACC. The EPSR method has some similar features to RMC [24]. EPSR is similar to RMC in that it uses a Monte Carlo method which samples different structures against an “acceptance” criterion, but unlike RMC, the acceptance criteria is the energy minimum based on the interatomic potentials. EPSR is like RMC in using the diffraction data to guide the modelling process, but unlike RMC, the difference between the simulated and experimental diffraction is used to adjust the interatomic potential. The present study reports a new EPSR model of Mg-stabilised ACC. Some features of this model are found to be different to the previously reported RMC model of ACC.

2. Method

2.1 Sample preparation

Magnesium stabilised ACC (Mg-ACC) was synthesised using the following method based on Rodriguez-Blanco et al [25] and Jiang et al [18]. 100mM solutions of CaCl_2 and Na_2CO_3 were prepared. 1mL of a 500 mM solution of MgCl_2 was added to 5mL of the CaCl_2 solution and this was then rapidly mixed with 5mL of the Na_2CO_3 solution (to give an ion ratio of 1:1:1 Ca:Mg: CO_3). This mixing produced a white precipitate which was then rapidly filtered and washed with isopropanol and dried in flowing air [25].

The precipitate was found to have a Mg:Ca ratio of 0.05:0.95 as determined by x-ray fluorescence. The precipitate was further dried and heat-treated for 1 hour at 150 °C to reduce the water content. Thermogravimetric and differential thermal analyses showed weight losses of 18% and 21% up to temperatures of 150 °C and 350 °C respectively, the latter being when calcite crystallises. These weight losses are consistent with a water content of $n=1.45$ before and $n=0.25$ after heat treatment at 150 °C. This reduction of water content is important for neutron diffraction because it reduces the detrimental effects of inelastic and incoherent scattering from protons. The heat-treated sample, denoted Mg-ACC, was confirmed amorphous by lab-based powder x-ray diffraction and to remain amorphous for a period of at least two weeks. The density of Mg-ACC after heat treatment was measured to be 2.19 g/cm^3 using a He pycnometer. Note that the above parameters differ from the parameters $n=1.0$ and density of 2.43 g/cm^3 reported for ACC in [16], which were taken to be equal to those in crystalline monohydrocalcite.

2.2 Neutron diffraction

Neutron diffraction data were measured using the Small Angle Neutron Diffractometer for Liquids and Amorphous Samples (SANDALS) [26] at the ISIS pulsed neutron source, UK. The sample was produced 24 hours before measurement and sealed until it was loaded into a flat plate sample holder with 25 micron thick vanadium plates that were 4mm apart. The sample holder was sealed and maintained at a temperature of 5 °C during the experiment. Time-of-flight data were collected over a wide range of momentum transfer, Q , with a useful maximum of 30 \AA^{-1} (where $Q = 4\pi \sin \theta/\lambda$). Experimental corrections of the data were performed using the Gudrun program [27]. The SANDALS diffractometer has a forward scattering geometry, which helps reduce the severity of the inelasticity correction, which can

be problematic for samples containing hydrogen [26]. There remained a background contribution due to inelastic scattering from H, and this is removed by EPSR (using a back Fourier transform of the part of the real space data in the region $r < 1.5 \text{ \AA}$). The structure factor $S(Q)$ from neutron diffraction is shown in Figure 1, including a factor of \bar{b}^2 (where \bar{b} is the average coherent neutron scattering length).

2.3 X-ray diffraction

X-ray diffraction data were collected at the MCX beamline [28] at the ELETTRA synchrotron, Italy, using a wavelength of $\lambda = 0.620 \text{ \AA}$. The sample was produced 2 days before travelling to the synchrotron and sealed until it was loaded into a 10-micron thick 1.0 mm diameter borosilicate glass capillary. The capillary was rotated at 180 rpm during the experiment. X-ray diffraction data were collected up to 2θ of 162° or $Q = 20 \text{ \AA}^{-1}$. The processing of the raw data included corrections for the background scattering, polarization, sample thickness, absorption, capillary, Compton scattering, and the sharpening function (equal to the square of the average form factor per atom) [15]. The structure factor $S(Q)$ from x-ray diffraction is shown in Figure 1.

2.4 Empirical Potential Structural Refinement Modelling

Empirical potential structure refinement (EPSR) [22,29] is a suitable method for generating and refining models of amorphous structures based on molecular units. The present study used the EPSR18 package. Structurally realistic molecular units were initially constructed using GHEMICAL4.0 software [30]. A cubic box with length $L = 39.55 \text{ \AA}$ containing 950 Ca, 50 Mg, 1000 CO_3 and 250 H_2O units (total 5750 atoms) was assembled using the EPSR shell.

This model corresponds to the measured density for Mg-ACC of 2.19 g/cm^3 . The density and H_2O content was based on experimental data as discussed in Section 2.1, and the effect on EPSR results of varying these parameters was not explored. The molecules were randomly translated so they move apart throughout the box. To further separate the molecules, the molecules were randomly rotated and translated 10,000 times.

The EPSR procedure initially relaxed the random starting configuration to an energy minimised structure using a simulated annealing approach with a sequence of temperatures: 10000K, 5000K, 1000K, 500K and 300K. At each temperature EPSR carries out Monte Carlo sampling of structures until reaching an equilibrium minimum energy. Starting inter-molecular potentials for each inter-atomic pair correlation i - j were generated by EPSR using built-in parameters for Lennard-Jones potentials for each pair of elements. Coulomb interactions are included with formal charges for ions.

The potential refinement was then implemented within the EPSR. The difference between the simulated and experimental neutron and x-ray diffraction data are used to adjust the interatomic potentials. The comparison between simulation and experiment is carried out using the structure factors $S(Q)$ (see Figure 1). The adjusted potentials are then used for a new Monte Carlo sample of structures until reaching equilibrium minimum energy. This process was iterated 20,000 times.

3. Results

The final level of agreement obtained between simulated and experimental diffraction data is shown in Figure 1. The model obtained is shown in Figure 2. The partial pair distribution functions (PDFs) $g_{ij}(r)$ have been calculated from the model. Figure 3 shows the PDFs involving CO_3 and H_2O molecules respectively. The continuous lines show intra-molecular correlations, and the dashed lines show inter-molecular correlations. The first four peaks in the intramolecular PDFs (solid lines) represent the internal structure of the molecules: O-H and H-H distances in H_2O , and C-O and O-O distances in CO_3 . The inter-molecular PDFs (dashed lines) show H...H and H...O correlations between H_2O (light dashed), and C...O and O...O correlations between all molecules (heavy dashed). These correlations and their peak distances are summarised in Table 1.

Figure 4 shows the PDFs involving Ca. Most prominent are the Ca-O distances due to Ca bonding to the oxygen atoms which are present in CO_3 and H_2O molecules, where the peak distance is 2.40 Å. The derived average coordination number of Ca is 7.4 which is composed of 6.8 oxygen atoms from CO_3 molecules and 0.6 oxygen atoms from H_2O molecules. These features of the model are in good agreement with the EXAFS results of Michel et al [3] for ACC prepared by a similar method: Ca-O coordination number of 6.7 ± 1.7 and peak distance of 2.41 ± 0.02 Å. Due to the coordination of Ca to oxygen in CO_3 and H_2O molecules, there are consequently noticeable correlations of Ca...C and Ca...H. The PDFs for Mg are similar to those for Ca, but they are not presented in detail because of the poor statistics due to the small number of Mg in the model.

The first peak at 3.3-4.5 Å in the Ca-Ca PDF (see Figure 4) corresponds to two calcium ions which share bonds with a common oxygen, i.e. Ca-O-Ca. Figure 5 shows these Ca-O-Ca

configurations as single blue lines originating from and terminating at the positions of Ca ions. This feature of the structure was previously referred to as the "Ca-rich network" [16], and Figure 5 indicates that such a "Ca rich network" is homogeneously distributed throughout the present model. The grey surfaces around regions more than 3.3 Å distant from Ca represent the structural feature which was previously referred to as "Ca-poor channels" [16]. Figure 5 shows that there are only a few marginal volumes fitting this criterion. In the previous study [16] a less stringent limit of 4 Å was used and the present model has no such volumes.

4. Discussion

This study uses both neutron and x-ray diffraction data as a more stringent test of simulation than previous studies (e.g. [16]). Note that the two techniques employed have different sensitivity to inter-atomic correlations, due to the different atomic scattering factors for neutrons and x-rays. In particular, neutrons are more sensitive to oxygen and hydrogen, whereas x-rays are more sensitive to calcium. The final level of agreement obtained is shown in Figure 1. In terms of the structure factor $S(Q)$, there is fair quantitative agreement for $Q > 4 \text{ \AA}^{-1}$ corresponding to short range order, and fair qualitative agreement $1 < Q < 4 \text{ \AA}^{-1}$ corresponding to medium range order. In terms of the total pair distribution function $g(r)$, there is fair quantitative agreement in short range order up to 3 Å.

The EPSR method has been successful at incorporating well-defined CO_3 and H_2O molecular units with the expected intra-molecular correlations. The inter-molecular correlations (see Table 1) include H...O and O...O peaks due to hydrogen bonding of H_2O molecules, although the peak distances are slightly shorter than expected for typical H...O hydrogen bond lengths

of 1.6-2.0 Å. H₂O molecules are outnumbered 4 to 1 by CO₃ molecules, so the presence of hydrogen bonding would suggest a tendency for H₂O molecules to be found near to one another.

This is the first model to be reported for Mg-stabilised ACC, and the first model of any kind of ACC to be compared with experimental neutron diffraction data. The Ca-O coordination is composed of relative contributions of CO₃ and H₂O of 6.8 and 0.6 respectively. These are in the ratio of 12:1. In comparison the available oxygens are 3 per CO₃ molecule and 1 per H₂O molecule, and these molecules are in a ratio of 4:1. Thus, the available oxygen from CO₃ and H₂O molecules are in the ratio of 12:1, the same ratio as observed in the Ca-O coordination. The average Ca...C coordination number is 6.6 which is almost the same as the average number, 6.8, of Ca bonds to oxygen in CO₃ molecules. This means that on average each carbonate oxygen which is bonded to calcium is from a separate carbonate molecule. . The number of Ca bonds to carbonate oxygens is 6.8 times the number of Ca in the model. Since the ratio of Ca to carbonate oxygen in the model is 1:3, this means that the number of carbonate oxygen bonds to Ca is 6.8 divided by 3 or 2.2. So each O in a CO₃ unit is typically bonded to 2 different Ca neighbours, i.e. Ca-O-Ca linkage.

The present EPSR model of Mg-ACC has Mg:Ca ratio of 0.05:0.95 and a water content of n=0.25. This is a somewhat different composition to the previously reported RMC model of ACC, which had no Mg and n=1.0 [16]. Nevertheless, it is interesting to compare the Ca distribution in these two models. The previously reported RMC model of ACC had a "Ca-rich network" (based on Ca-O-Ca linkages), which was inhomogeneously distributed, and interspersed with significant volumes of "Ca-poor channels" (based on the limit of volumes more than 4 Å distant from Ca). The homogeneous Ca distribution in the present EPSR

model of Mg-ACC fails to reproduce the inhomogeneous features in the previously reported RMC model of ACC.

In the present study, the proton content of the sample was reduced by using heat treatment.

An alternative approach is to deuterate the sample by substituting protons for deuteriums, D, which has the benefits of reducing incoherent and inelastic scattering, and providing scattering contrast due to the different neutron scattering length for D compared to protons.

In the present study, an unsuccessful attempt was made to produce a deuterated ACC sample, but the authors plan to attempt this again in future work.

5. Conclusions

Preparation of Mg-stabilised ACC has enabled the first neutron diffraction measurement of Mg-doped ACC. The EPSR modelling method has used this neutron diffraction data, together with x-ray diffraction data, to build a model of Mg-stabilised ACC, which is in fair agreement with the experimental diffraction data. The model contains the expected molecular units of CO₃ and H₂O with some hydrogen bonding of H₂O. Calcium has an average coordination of 6.8, with a peak distance of 2.40 Å, in agreement with previously reported EXAFS results [3]. The Ca coordination of approximately 6.7 is comprised of oxygens from separate CO₃ molecules, and Ca-O-Ca linkages are prevalent throughout the structure. The Ca distribution is homogeneous and does not show the inhomogeneous characteristics which were previously reported for an RMC model of ACC [16].

6. Acknowledgements

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Figure 3: pair distribution functions (PDFs) $g_{ij}(r)$ for Mg-stabilised ACC from the EPSR model. Intramolecular PDFs (solid lines) represent the internal structure of the molecules: O-H and H-H distances in H₂O, and C-O and O-O distances in CO₃. Inter-molecular PDFs (dashed lines) show H...H and H...O correlations between H₂O (light dashed), and C...O and O...O correlations between all molecules (heavy dashed).

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Figure 5: The Ca-rich network in the EPSR model of Mg-stabilised ACC (see Fig.2), showing Ca-O-Ca linkages as blue sticks. Marginal volumes of Ca-poor channels (based on the criterion of regions more than 3.3 Å distant from Ca) are shown as grey surfaces.

Table 1: interatomic correlations involving CO₃ and H₂O molecular units. (* denotes correlations involving hydrogen bonding as discussed in the text. Distances shown in brackets are approximate, and the corresponding correlation is also shown in brackets.)

distance (Å)	atoms types i-j	type of correlation between molecular units
0.93 and (1.2)*	H-O	intra and inter* H ₂ O
1.32	C-O	intra CO ₃
1.56 and (1.9)*	H-H	intra and inter* H ₂ O
2.22	O-O	intra CO ₃
(2.4)	C-H	inter CO ₃ and H ₂ O
(2.8)	O-O	inter CO ₃ (and H ₂ O)
(3.3)	C-O	inter CO ₃ (and H ₂ O)
(3.5)	C-C	inter CO ₃

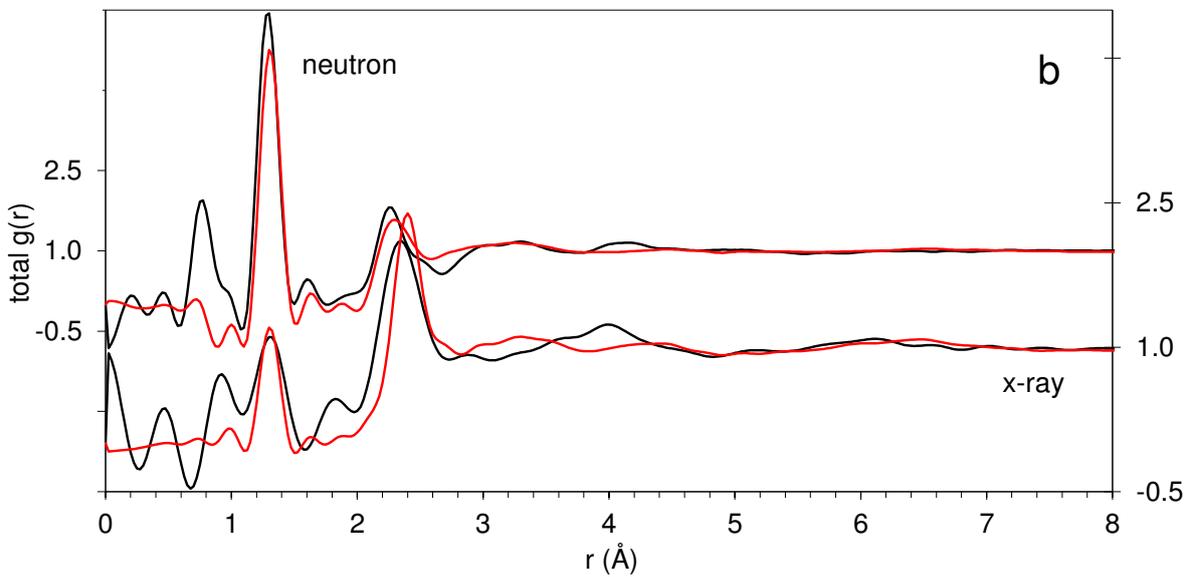
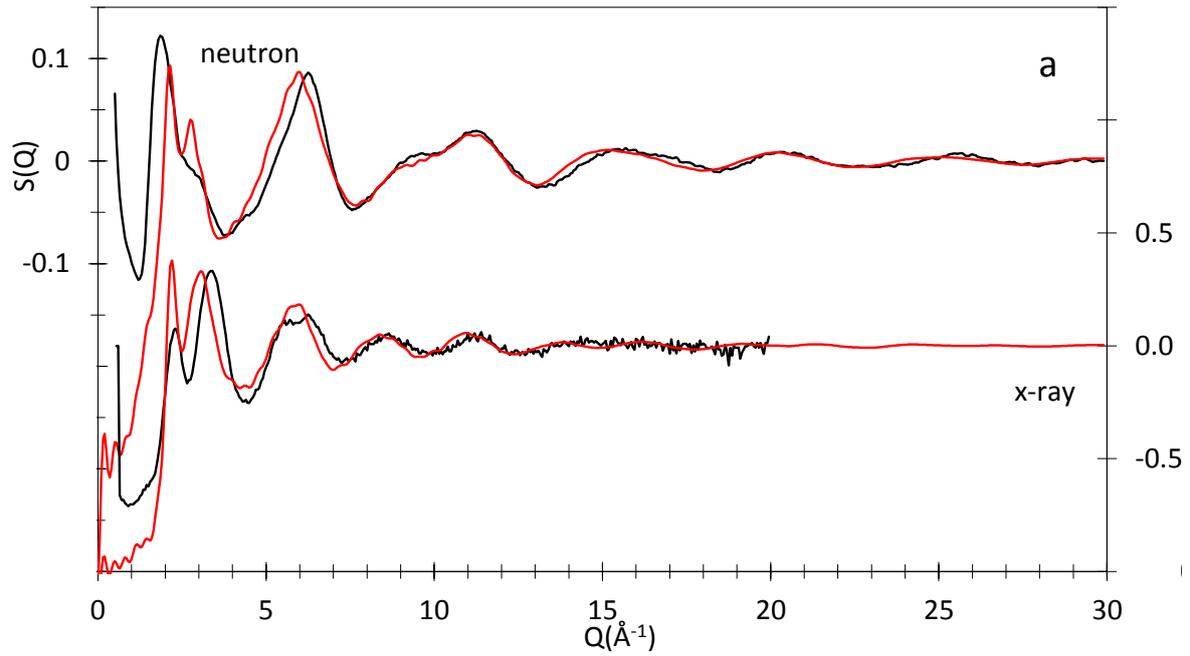


Figure 1: (a) Experimental structure factors $S(Q)$ of Mg-stabilised ACC (black lines) for neutron and x-ray diffraction together with the simulated functions (red lines) for the EPSR model. (Note $S(Q)$ for neutron diffraction includes the factor \bar{b}^2 in barns per steradian). (b) Total pair distribution functions $g(r)$.

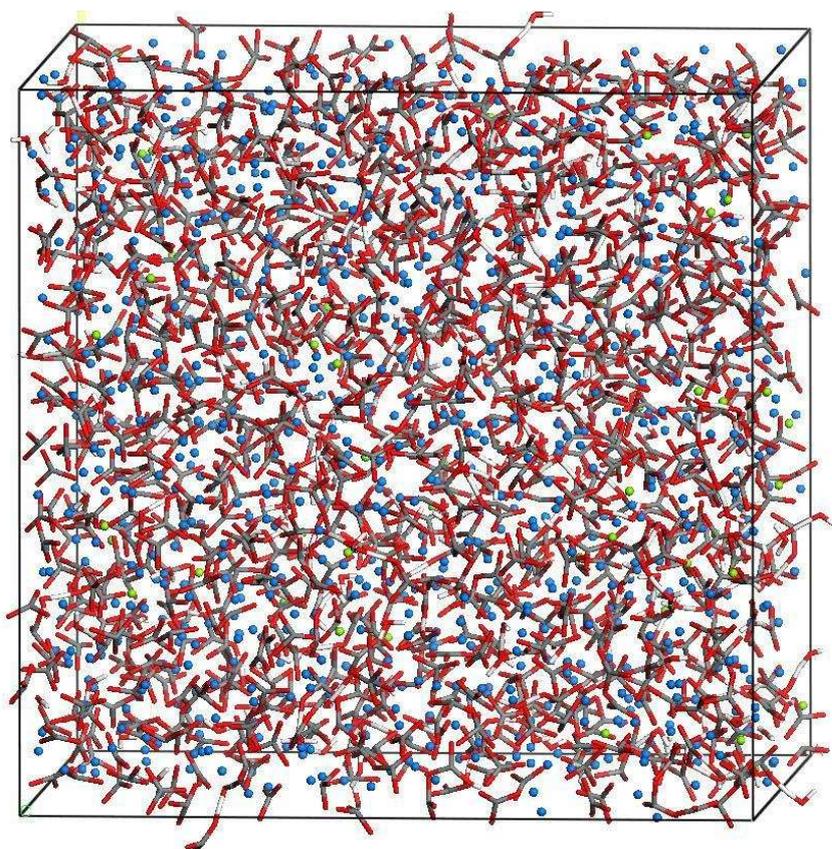


Figure 2: EPSR model of Mg-stabilised ACC containing 5750 atoms with box length of 39.55Å (oxygen is red, carbon is grey, hydrogen is white, calcium is blue, magnesium is green).

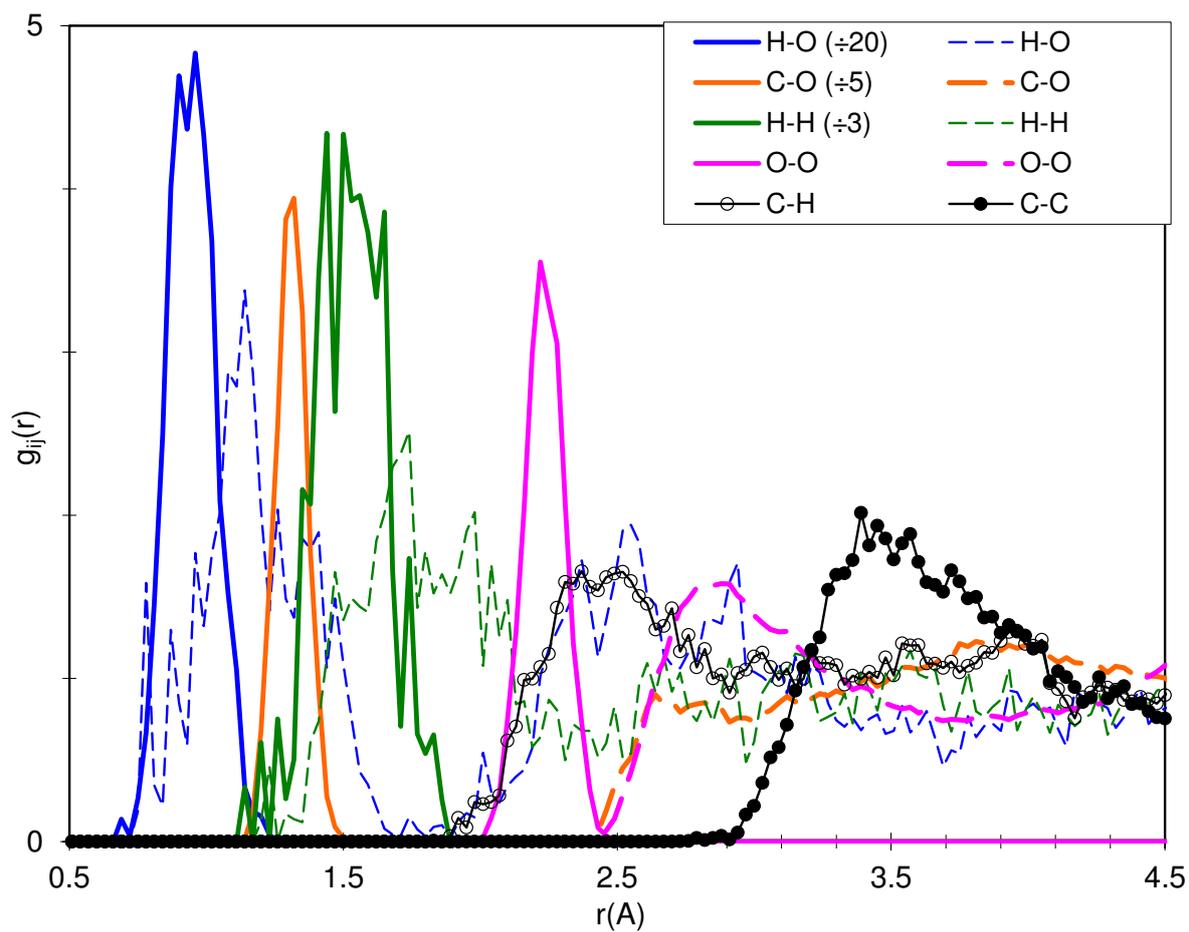


Figure 3: pair distribution functions (PDFs) $g_{ij}(r)$ for Mg-stabilised ACC from the EPSR model. Intramolecular PDFs (solid lines) represent the internal structure of the molecules: O-H and H-H distances in H_2O , and C-O and O-O distances in CO_3 . Inter-molecular PDFs (dashed lines) show H...H and H...O correlations between H_2O (light dashed), and C...O and O...O correlations between all molecules (heavy dashed).

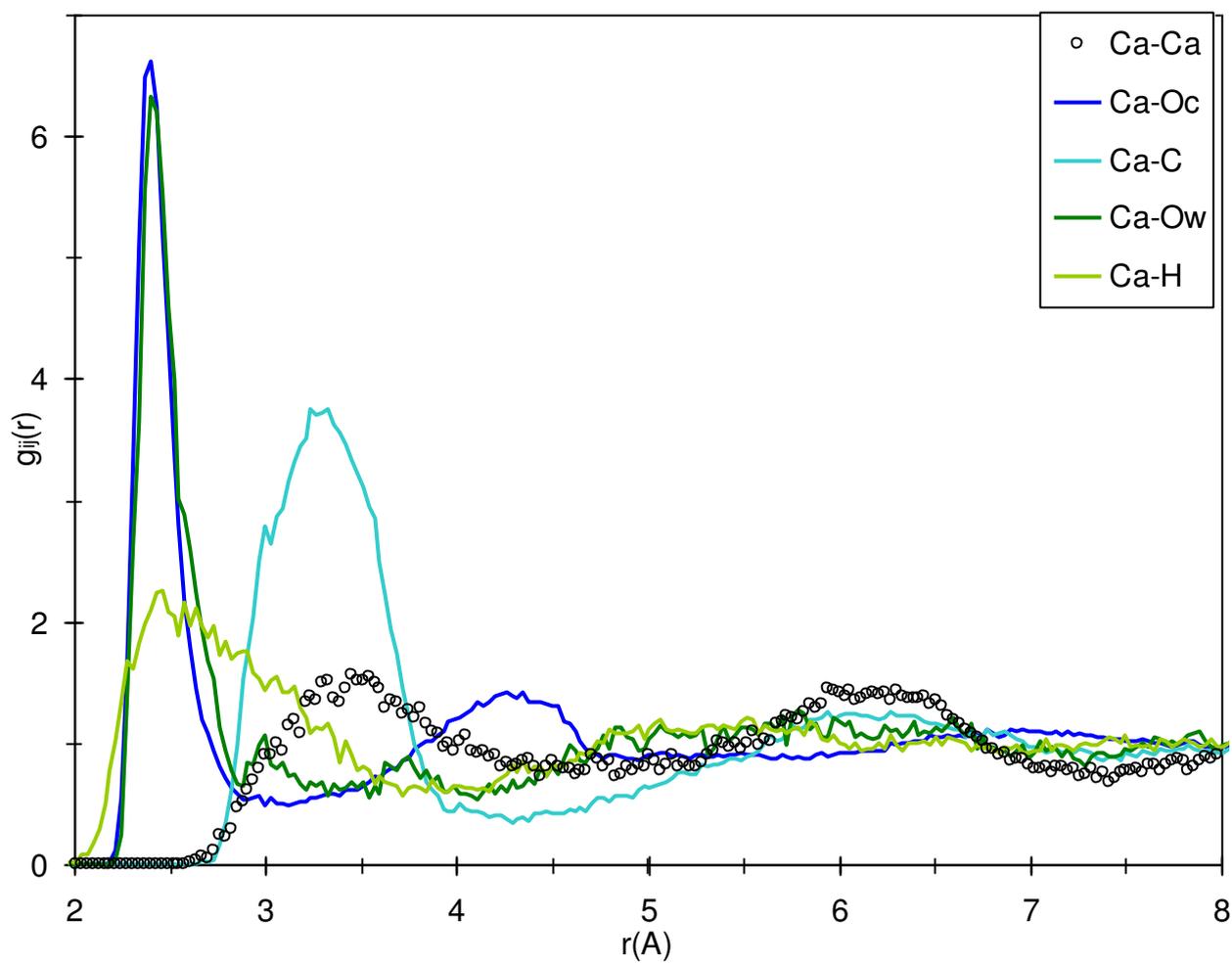


Figure 4: Pair distribution functions (PDFs) $g_{ij}(r)$ for Mg-stabilised ACC from the EPSR model. PDFs involving calcium are shown (Note Oc and Ow refer to oxygens in CO_3 and H_2O molecules respectively).

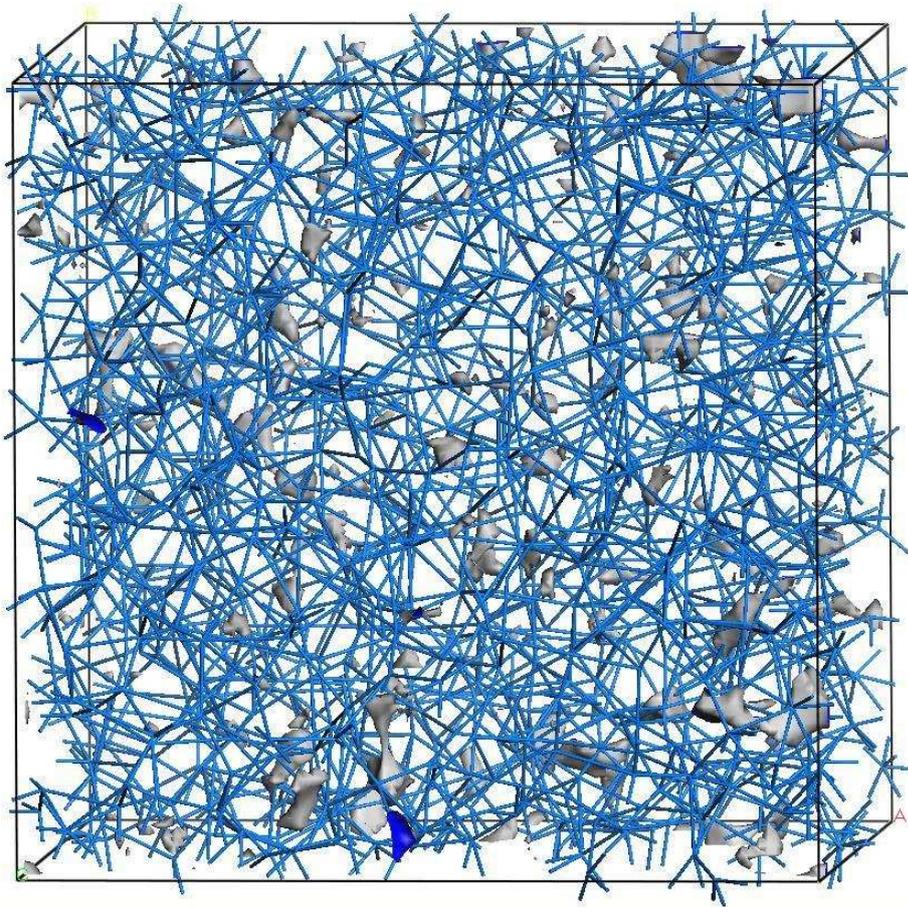


Figure 5: The Ca-rich network in the EPSR model of Mg-stabilised ACC (see Fig.2), showing Ca-O-Ca linkages as blue sticks. Marginal volumes of Ca-poor channels (based on the criterion of regions more than 3.3 Å distant from Ca) are shown as grey surfaces.