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# Synergistic performance of Ni-Ca based dual functional materials under the coexistence of moisture and oxygen in CO<sub>2</sub> source for integrated carbon capture and utilisation

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#### ABSTRACT

Global warming and climate change require urgent reduction of CO<sub>2</sub> emissions. Integrated carbon capture and utilisation combined with dry reforming of methane (ICCU-DRM) has the potential to mitigate greenhouse gas emissions by capturing and converting  $CO_2$  and  $CH_4$  into valuable syngas. However, the performance and stability of dual-functional materials (DFMs) are unclear under real-world flue gas conditions, particularly in the presence of steam and O2. In this study, we elucidated the mechanisms of the performance of a Ni<sub>0.05</sub>/CaO<sub>0.95</sub> DFM under various flue gas conditions and investigated the combined effects of steam and O<sub>2</sub> on CO<sub>2</sub> uptake, CO and H<sub>2</sub> yields and cyclic stability. Our approach includes the synthesis and evaluation of the DFM using advanced characterisation techniques, such as XRD, in-situ infrared spectroscopy, CH<sub>4</sub>-TPR, SEM, EXD, FIB-SEM, BET, and XPS, to gain insights into the properties and behaviour of the DFM under different flue gas conditions. The findings show that under the simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>), there was a significant deterioration in performance, with CO<sub>2</sub> uptake of only 5.7 mmol.g-1, average H<sub>2</sub> yield of 11 mmol.g<sup>-1</sup> and average CO yield of 6.2 mmol.g<sup>-1</sup>. While under the ideal condition (10.0 %  $CO_2/N_2$ ), the CO<sub>2</sub> uptake, average H<sub>2</sub> yield and average CO yield were 10.7 mmol.g<sup>-1</sup>, 16.9 mmol.g<sup>-1</sup> and 23.1 mmol.g<sup>-1</sup>, respectively. FIB-SEM and XPS analysis highlighted the combined effect of steam and O<sub>2</sub> on the performance of Ni<sub>0.05</sub>/CaO<sub>0.95</sub> DFM during ICCU-DRM. Notably, the presence of O<sub>2</sub> promoted the oxidation of Ni, resulting in decreased catalytic activity and volumetric expansion of Ni particles. Furthermore, the presence of steam promoted CO<sub>2</sub> adsorption, inducing additional volumetric expansion of the CaO component of DFM. These synergistic expansions form a dense shell on the surface, hindering the reduction of internal NiO, thereby diminishing DFM performance.

#### 1. Introduction

Global warming causes severe weather conditions bringing significant challenges to social and economic development [1]. The large emissions of CO<sub>2</sub> are responsible for global warming. Technologies such as carbon capture and utilisation (CCU) [2,3] and carbon capture and storage (CCS) [4–7] have been intensively developed to reduce carbon emissions. In particular, integrated carbon capture and utilisation (ICCU) [8–11] has been gaining more attention. ICCU combines carbon capture and CO<sub>2</sub> conversion in a single reaction system isothermally by dual functional materials (DFMs), which contain both sorbents for  $CO_2$  capture and catalysts for  $CO_2$  conversion. This technology can directly convert the captured  $CO_2$  into chemicals without transportation and storage of the captured  $CO_2$ , as is the case with conventional  $CO_2$  adsorption. In addition, ICCU can occur in one single reactor. Compared with traditional CCU and CCS technologies, ICCU shows the advantages of low energy consumption and low capital investment.

DFMs, acting as combined  $CO_2$  sorbents and catalysts, are fundamental to ICCU technology. Today, extensive research efforts are being poured into enhancing DFM performance. This typically involves

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Fig. 1. ICCU-DRM evaluation of cycle 1, cycle 2 and cycle 14 using Ni $_{0.05}$ /Cao $_{0.95}$  under (a) ideal condition (10.0 % CO $_2$ /N $_2$ ); (b) O $_2$ -containing condition (10.0 % CO $_2$  + 6.0 % H $_2$ O/N $_2$ ); (c) Steam-containing condition (10.0 % CO $_2$  + 6.7 % O $_2$ /N $_2$ ) and (d) Simulated flue gas condition (10.0 % CO $_2$  + 6.0 % H $_2$ O + 6.7 % O $_2$ /N $_2$ ).



Fig. 2. XRD spectra of adsorbed samples under different flue gas conditions.

incorporating metal additives, as well as fine-tuning the distribution of catalysts and adsorbents [12]. For instance, a study demonstrated that a Ni/CaO DFM, when infused with metal oxides, exhibited impressive

performance for CO<sub>2</sub> capture and conversion. Of the variations tested, the one doped with ZrO<sub>2</sub> delivered the top performance [13]. Moreover, research has shown that introducing CeO2 into CaO-based DFMs can substantially boost their cycle stabilities [14,15]. It has also been observed that doping with CeO<sub>2</sub> can even improve CO<sub>2</sub> conversion rates [16]. Recently, a Ca-Fe bifunctional material was reported, which uses CaO for CO<sub>2</sub> capture and Fe as an oxygen carrier for conversion. By optimising the composition, a Ca-Fe DFM containing 27% Iron exhibited robust stability and CO yield over 10 cycles, without deactivation [17]. Apart from utilising additives and optimising proportions, scientists are also investigating new preparation methods to enhance DFM performance [18,19]. Techniques like co-precipitation [20] and sol-gel [21] can help fine-tune the morphology of the DFMs, porosity, and particle size, thereby augmenting their CO<sub>2</sub> adsorption capabilities. There are also several innovative research projects aimed at creating novel DFMs that are different from conventional CaO-based DFMs. For example, some researchers used DRIFTS-MS to study the capture and conversion mechanisms of a NiRu-Na<sub>2</sub>O DFM in CO<sub>2</sub> methanation [22]. These studies offered insight into adjusting catalyst-adsorbent interfaces, adsorption strength, and site distributions for DFMs optimisation. On another front, a Ru/K2CO3-MgO DFM was engineered for lowtemperature  $CO_2$  capture and methanation, and the optimised



Fig. 3. Average Pre-reaction duration of 14 cycles under the four different flue gas conditions (e), TG curve (f) and TPR rate by 10.0 %  $CH_4/N_2$  of  $Ni_{0.05}CaO_{0.95}$  with the temperature kept at 650 °C isothermally under atmospheric pressure and flow rate of 100 mL min<sup>-1</sup>.



Fig. 4. The coke on DFM after cycling under (a) ideal condition (10.0 %  $CO_2/N_2$ ); (b)  $O_2$ -containing condition (10.0 %  $CO_2 + 6.0$  %  $H_2O/N_2$ ); (c) Steam-containing condition (10.0 %  $CO_2 + 6.7$  %  $O_2/N_2$ ) and (d) Simulated flue gas condition (10.0 %  $CO_2 + 6.0$  %  $H_2O + 6.7$  %  $O_2/N_2$ ).

material demonstrated stable Methane yields over multiple cycles [23]. Some scientists also used PEI to capture CO<sub>2</sub>, forming carbamic acid salts. Under the influence of a Rh catalyst, these salts were hydrogenated to generate formate esters [24]. In essence, these pioneering studies offer various preparation strategies for DFMs that can guide their design for more effective ICCU.

In the realm of  $CO_2$  conversion techniques, dry reforming of methane (DRM) has emerged as a promising approach to utilise  $CO_2$  for syngas production (Eq. (1). DRM can aid in reducing CH<sub>4</sub>, a potent greenhouse gas with a global warming potential per molecule around 30 times that of  $CO_2$  [25]. Consequently, integrating ICCU and DRM into a single process (ICCU-DRM) is an attractive proposition to limit  $CO_2$  emissions from flue gas. However, catalysts employed in DRM frequently face the issue of coke deposition, which tends to diminish their catalytic performance [26]. Multiple studies are, therefore, focused on augmenting the resistance to coking and enhancing catalyst stability [27]. For instance, one research identified a Rh-Ni bimetallic catalyst that improves coking resistance by balancing methane dissociation and  $CO_2$  activation [28]. Other researchers synthesised a Ru/CeO<sub>2</sub> catalyst as a part of DRM, demonstrating that Ru and CeO<sub>2</sub> worked in synergy to

promote coking resistance and catalytic performance [29,30]. Interestingly, the ICCU-DRM process offers a unique solution to the coking issue during the CO<sub>2</sub> capture phase. The coke produced by DRM can be consumed through a reverse Boudouard reaction when it reacts with CO<sub>2</sub> [31,32] (Eq. (2). Therefore, the challenges associated with developing and modifying DFMs for ICCU-DRM might be distinct from traditional DRM. Despite its potential, very few studies target ICCU-DRM in their DFM research. As ICCU is a relatively nascent technology, the bulk of research has been concentrated on reverse water-gas shift (RWGS) and methanation. The complexity of ICCU-DRM also implies that related research is not yet comprehensive. However, recent studies are beginning to address this knowledge gap. One study used limestone-derived CaO as the CO2 sorbent and Ni supported on MgO-Al<sub>2</sub>O<sub>3</sub>, derived from a hydrotalcite precursor, as the DRM catalyst [33]. To enhance DFM performance in ICCU-DRM, a study employed N<sub>2</sub>doped porous carbon materials derived from biomass, revealing key insights into material properties necessary for effective CO2 adsorption and DRM catalysis [34]. Another study explored the synergistic effects of Ni-CaO DFMs in ICCU. By optimising the dispersion of Ni nanoparticles on a CaO carrier, impressive conversion efficiencies of 96.5 %



**Fig. 5.** Raman spectrums of Ni<sub>0.05</sub>CaO<sub>0.95</sub> cycled under ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>), O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>), steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>), simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and cycled then recovered by CO<sub>2</sub> under under ideal condition.

for CO<sub>2</sub> and 96.0 % for CH<sub>4</sub> were achieved [35].

$$CH_4 + CO_2 = 2CO + 2H_2 \tag{1}$$

$$C + CO_2 = 2CO \tag{2}$$

However, most of the research conducted so far has tested flue gas conditions containing only CO<sub>2</sub> and inert gas, typically N<sub>2</sub>. This does not align with real-world scenarios. Flue gases from power or cement plants often contain O<sub>2</sub> and steam. Therefore, a comprehensive understanding of the roles of O<sub>2</sub> and steam in this procedure is urgently required. Recently, a study by Alejandro and his team highlighted the adverse impact of oxidising species on the performance of a DFM composed of 4% Ru-8% Na<sub>2</sub>CO<sub>3</sub>-8% CaO/Al<sub>2</sub>O<sub>3</sub> [36]. Simultaneously, steam has been shown to boost the CO<sub>2</sub> capacity of CaO-based DFMs [37]. However, the overall impact of O2 and steam on the industrial ICCU-DRM process remains unclear. From a practical and industrial viewpoint, it would be advantageous to sidestep the additional costs tied to the removal of steam and O<sub>2</sub> from the flue gas. Therefore, our approach should lean towards the development of DFMs that can maintain superior performance and stability in conditions containing steam and O<sub>2</sub>. The first step in this journey is to discern the impact of steam and O<sub>2</sub> on DFM performance and stability. A critical area of focus should be the cyclical stability of DFMs under flue gas conditions that contain O2 and steam. Addressing this pressing issue calls for extensive experimental investigations to thoroughly understand these effects. The knowledge gleaned from such explorations will then serve as a foundation for suggesting modifications and optimising the performance of DFMs in flue gas conditions that include O<sub>2</sub> and steam.

Ni-CaO based DFMs have significant potential for widespread use in large-scale industrial applications [38] to explore the impact of steam

and  $O_2$  in flue gas on its performance. Therefore, we developed a  $Ni_{0.05}/CaO_{0.95}$  DFM and assessed its efficiency in the ICCU-DRM process under flue gas conditions. These conditions encompassed an ideal condition (10% CO<sub>2</sub>/N<sub>2</sub>), a steam-containing condition (10.0% CO<sub>2</sub> + 6.0% H<sub>2</sub>O/N<sub>2</sub>), an O<sub>2</sub>-containing condition (10.0% CO<sub>2</sub> + 6.7% O<sub>2</sub>/N<sub>2</sub>), and a simulated flue gas condition (10.0% CO<sub>2</sub> + 6.0% H<sub>2</sub>O + 6.7% O<sub>2</sub>/N<sub>2</sub>). Our in-depth analysis aimed to understand the effects of steam and O<sub>2</sub> on the stability of DFMs during ICCU-DRM operations. A special emphasis is placed on how O<sub>2</sub> and steam jointly impact CO<sub>2</sub> absorption, CO and H<sub>2</sub> yields, and cyclic stability.

#### 2. Experimental section

#### 2.1. Synthesis of dual functional material (DFM)

Choosing the appropriate DFMs for ICCU-DRM requires a balance between performance and economic feasibility. While certain noble metal catalysts such as Ru, Rh, and Pd showcase exceptional catalytic performance, their high cost necessitates a more economical alternative for large-scale applications. Ni, due to its affordability, has emerged as the most economically viable choice, making it the front-runner for widespread adoption [39,40]. Hence, Ni-CaO DFMs have promising industrial applicability, thereby Ni<sub>0.05</sub>/Cao<sub>0.95</sub> was synthesised as the DFM by the wet impregnation method. Specifically, Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (Sigma-Aldrich, >99.0 %) and Ca(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (Sigma-Aldrich, >99.0 %) were dissolved into distilled water with constant stirring at room temperature for 1 h. The mole ratio was:  $Ni_2^+$ :  $Ca_2^+$ :  $H_2O = 0.05$ : 0.95: 400. Subsequently, the solution was evaporated with stirring at 95 °C and then dried in a Muffle furnace at 130 °C for 12 h. After drying, the material was then ground into powder and calcined in still air in the Muffle furnace with the heating program of 5 °C min<sup>-1</sup> to 750 °C for 3 h to obtain the original Ni\_{0.05}/Cao\_{0.95}. Then it was reduced at 700  $^\circ\!C$  for 1 h in 100 mL min<sup>-1</sup> 5.0 % H<sub>2</sub>/Ar to obtain reduced Ni<sub>0.05</sub>/Cao<sub>0.95</sub> DFM.

#### 2.2. Material characterisation

The phase information of the adsorbed samples under the four different flue gas conditions was determined using powder X-ray diffraction (XRD) on a PANalytical Empyrean series two diffractometer system equipped with a Cu K $\alpha$  X-ray source (2 Theta = 5–80°) The in-situ infrared spectroscopy was used to investigate the decomposition of CaCO<sub>3</sub> under CH<sub>4</sub> and its subsequent CO<sub>2</sub> re-adsorption. Initially, the DFM sample, fully adsorbed under ideal conditions ( $10\% CO_2/N_2$ ), was placed in the sample holder. The sample was then purged with N<sub>2</sub> for 10 min, followed by the introduction of a 10% CH<sub>4</sub>/N<sub>2</sub> mixture for 30 min. Afterwards, the sample was purged again with N2 for 10 min, and finally, a gas mixture containing 10% CO2/N2 was introduced for adsorption. Throughout the experiment, the temperature was maintained at 650  $^{\circ}$ C, and the gas flow rate was set at 100 mL.min<sup>-1</sup>. The CH<sub>4</sub> temperature programmed reduction (CH<sub>4</sub>-TPR) curve was obtained using a Hi-Res TGA 2950 thermogravimetric analyser. The  $\mathrm{Ni}_{0.05}/$ CaO<sub>0.95</sub> DFM was pretreated at 650  $^{\circ}$ C for 10 min in an N<sub>2</sub> atmosphere. Subsequently, a 10.0% CH<sub>4</sub>/N<sub>2</sub> mixture was introduced, and the process continued until no further decrease in weight was observed, indicating equilibrium had been reached. A FEI Helios G4 CX Dual Beam high resolution monochromate field emission gun (FEG) scanning electron microscope (SEM) with a precise focused ion beam (FIB) was employed to obtain cross-sectional images of the samples. Energy-dispersive X-ray spectroscopy (EDX) was used to analyse the elemental distribution within the samples. The surface area and pore diameter were calculated using the Brunauer-Emmett-Teller (BET) method and the desorption isotherm branch, respectively. Lastly, the X-ray photoelectron spectrum (XPS) analysis was conducted on a Thermo Fisher Scientific NEXSA spectrometer to further characterise the samples.



Fig. 6. XPS analysis of C 1 s region of Ni<sub>0.05</sub>/CaO<sub>0.95</sub> after adsorption under (a) ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>); (b) O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>); (c) Steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and (d) Simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

#### 2.3. ICCU-DRM procedure

The experimental setup is shown in Fig. S1, which is similar to our previous report [40]. The ICCU-DRM evaluations of Ni<sub>0.05</sub>/Cao<sub>0.95</sub> DFM were conducted in a quartz reaction tube (OD: 12.0 mm; ID: 10.5 mm; L: 65.0 cm). The quartz tube was fixed in the middle of a tube furnace (Elite TSH 12/50/300-2416CG) for heating. The 0.30 g of Ni<sub>0.05</sub>/Cao<sub>0.95</sub> DFM which was fixed by quartz wool, was placed in the centre of the quartz tube. A thermocouple was placed in the same position as the sample to monitor the temperature. The inlet gases (CO<sub>2</sub>, N<sub>2</sub>, air) were controlled by mass flowmeters (OMEGA FMA-A2306). The mass flowmeters were calibrated by a bubble flowmeter. Steam was generated by introducing water using a water pump. The outlet gases were analysed by a flue gas analyser system, which included a flue gas analyser (Enerac 700 AV; NDIR detector) and an H<sub>2</sub> analyser (CX-02; TCD detector).

The ICCU-DRM evaluations were conducted at ideal condition (10 %  $CO_2/N_2$ ), steam-containing condition (10.0%  $CO_2 + 6.0\% H_2O/N_2$ ), O<sub>2</sub>-containing condition (10.0%  $CO_2 + 6.7\% O_2/N_2$ ) and simulated power plant flue gas condition (10.0%  $CO_2 + 6.0\% H_2O + 6.7\% O_2/N_2$ ) respectively. For each condition, 14 ICCU-DRM cycles were conducted to observe the stability and performance of Ni<sub>0.05</sub>/Cao<sub>0.95</sub> under different flue gas conditions. The flow rates of inlet gas were all kept at

100 mL.min<sup>-1</sup> while assuming that the flow rates of outlet gas were the same. For a typical cycle, during the CO<sub>2</sub> capture stage, the simulated flue gases were introduced for 30 mins, and then the reaction tube was purged by N<sub>2</sub> for 5 mins. For the DRM stage, the 10.0 % CH<sub>4</sub>/N<sub>2</sub> was introduced until the CO concentration shown on the analyser was below 0.05 %. Then the simulated gases were introduced again to conduct the next cycle after purging by N<sub>2</sub> for 5 mins. The temperature was kept at 650 °C before finishing the 14th cycle. The ICCU-DRM performance was evaluated by the real-time signals of CH<sub>4</sub>, H<sub>2</sub>, CO and CO<sub>2</sub>.

#### 3. Results and discussions

3.1.  $Ni_{0.05}/Cao_{0.95}$  performance for ICCU-DRM under different flue gas conditions

## 3.1.1. Influence of $H_2O$ and $O_2$ presence on $CO_2$ capture performance during ICCU-DRM

Fig. 1 presents the real-time assessment of ICCU-DRM over cycles 1, 2, and 14, employing Ni<sub>0.05</sub>/Cao<sub>0.95</sub> under four distinct flue gas conditions, including ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>), O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0% H<sub>2</sub>O/N<sub>2</sub>), steam-containing condition (10.0 % CO<sub>2</sub> + 6.7% O<sub>2</sub>/N<sub>2</sub>), and simulated flue gas condition (10.0 % CO<sub>2</sub> +



Fig. 7. XPS analysis of C 1 s region of  $Ni_{0.05}$ /CaO<sub>0.95</sub> after 14 cycles under (a) ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>); (b) O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>); (c) Steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and (d) Simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

 $6.0 \% H_2O + 6.7 \% O_2/N_2$ ). As demonstrated in Fig. 1 (a), during the initial cycle, the gradual increase in  $CO_2$  concentration from 0% to 10% signifies the adsorption of CO<sub>2</sub> by CaO. Following a 5-minute purge with N<sub>2</sub>, 10% of CH<sub>4</sub> was introduced to commence the DRM reaction. As shown in Fig. 1 (a), starting from the second cycle, CO is generated during the CO<sub>2</sub> capture stage, with the concentration rapidly rising to 5.5% and then gradually decreasing to 0 % over the next approximately 600 s. Considering that the concentration of CO in the CO<sub>2</sub> capture stage of the first cycle is only 0.01 %, the generation of CO during the carbon capture stage is neglected. Therefore, it is plausible to exclude the possibility of CO<sub>2</sub> decomposition generating CO. In fact, the amount of CO generated from the thermal decomposition of CO<sub>2</sub> at lower concentrations using Ni-based catalysts is quite limited [41]. It often requires the aid of means such as plasma to achieve a considerable yield of CO by CO<sub>2</sub> pyrolysis [42]. Therefore, the main reason for the generation of CO should be related to the coke formed on the DFM surface during the previous cycle, which reacted with CO<sub>2</sub> through the reverse Boudouard reaction (Eq. (2)) [31].

Under  $O_2$ -containing conditions, as depicted in Fig. 1 (b), due to the reaction between coke and  $O_2$  [40,43], the duration of the appearance of CO was only about 300 s which was much shorter compared to that under ideal conditions (600 s). Meanwhile, the  $O_2$  generation was

delayed by about 200 s during the second cycle, suggesting that O<sub>2</sub> was involved in the reaction with coke. Concurrently, the CO2 curve exhibited a period of fluctuation. This could be attributed to the additional generation of CO<sub>2</sub> resulting from the reaction between O<sub>2</sub> and coke [44], causing a rapid rise of concentration from 0 % to 9.8 % in the CO<sub>2</sub> curve, while the continuous adsorption of CO<sub>2</sub> by CaO led to a decrease in its concentration to about 5.0 % as the coke was consumed by the reverse Boudouard reaction (Eq. (2)) [40]. As shown in Fig. 1 (c), under the steam-containing condition, H<sub>2</sub> production was observed during the CO<sub>2</sub> capture stage. The highest concentration was at around 4.0 %. This could be ascribed to the carbon steam gasification (CSG) reaction between coke and steam (Eq. (3)) [45]. Under simulated flue gas conditions containing both  $O_2$  and steam, like Fig. 1 (b), the duration of CO presence significantly decreased, only lasting for approximately 300 s. Meanwhile, the duration of H<sub>2</sub> appearance also noticeably reduced, lasting for about 50 s and with the highest concentration reaching only 2.5%. This indicates that the reaction between O2 and coke predominantly occurs during the CO<sub>2</sub> capture stage, thereby reducing the reverse Boudouard reaction and CSG reaction.

$$C + H_2 O = CO + H_2 \tag{3}$$

	Summary	y of atomic 1	ratio of C	1 s by	XPS analys	is.
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Sample	Name	Peak BE	Atomic %
A1	C-C/C=C	284.8	53.06
	C-0	286.38	8.52
	C=O	287.86	3.67
	0–C=O	289.37	34.75
A2	C–C/C=C	284.8	43.00
	C-O	286.27	13.25
	C=O	288.53	9.38
	0–C=O	289.51	34.37
A3	C-C/C=C	284.8	43.97
	C-O	286.23	8.81
	C=O	287.91	4.80
	0–C=O	289.34	42.42
A4	C-C/C=C	284.8	59.27
	C-0	286.33	10.78
	C=O	287.94	5.29
	0-C=0	289.42	24.66
C1	C-C/C=C	284.8	64.67
	C-0	286	16.54
	C=O	287.71	6.54
	0–C=O	290.6	12.25
C2	C-C/C=C	284.8	52.88
	C-O	286.13	27.46
	C=O	288.13	6.11
	0–C=O	290.83	13.55
C3	C-C/C=C	284.8	49.03
	C-0	285.79	24.15
	C=O	287.27	12.45
	0–C=O	290.35	14.38
C4	C–C/C=C	284.8	39.95
	C-0	285.79	42.45
	C=O	287.92	7.49
	0–C=0	290.39	10.11

A1:  $Ni_{0.05}/CaO_{0.95}$  adsorbed under ideal condition (10%  $CO_2/N_2$ ).

A2: Ni\_{0.05}/CaO\_{0.95} adsorbed under O2-containing condition (10.0 % CO2 + 6.0 % H2O/N2).

A3: Ni\_{0.05}/CaO\_{0.95} adsorbed under steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

A4: Ni\_{0.05}/CaO\_{0.95} adsorbed under simulated flue gas condition (10.0 % CO $_2$  + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

C1:  $Ni_{0.05}$ /CaO<sub>0.95</sub> regenerated after 14 cycles under ideal condition.

C2: Ni\_{0.05}/CaO\_{0.95} regenerated after 14 cycles under O<sub>2</sub>-containing condition. C3: Ni\_{0.05}/CaO\_{0.95} regenerated after 14 cycles under steam-containing condition.

C4: Ni\_{0.05}/CaO\_{0.95} regenerated after 14 cycles under simulated flue gas condition.

## 3.1.2. Influence of $H_2O$ and $O_2$ presence on $CO_2$ conversion during ICCU-DRM

In the DRM process, the equality of  $H_2$  and CO concentrations indicates the equilibrium of the DRM reactions [46]. For the ideal conditions of the 1st cycle, the concentrations of CO and  $H_2$  were nearly equal, at around 5.0 %-6.0 % from 2180 s to 3200 s. Subsequently, the concentration of  $H_2$  surpassed that of CO at around 3200 s. The DRM process comprises two simplified parallel reactions: the decomposition of CH<sub>4</sub> (Eq. (4)), and the reverse Boudouard reaction (Eq. (2)) [47]. The observed increase in  $H_2$  concentration relative to CO indicates an imbalance between the cracking of CH<sub>4</sub> and the consumption of coke by CO<sub>2</sub> [48], leading to coke deposition. As CaCO<sub>3</sub> continuously decomposes, the remaining amount of CaCO<sub>3</sub> decreases, leading to reduced production of CO<sub>2</sub>. This reduced CO<sub>2</sub> availability, in turn, limits the oxidation of coke formed from CH<sub>4</sub> cracking via reverse Boudouard reaction (Eq. (2)). Consequently, coke accumulated on the Ni surface [49,50]. Under the ideal condition, the CO:H<sub>2</sub> ratio for each cycle can be maintained between 1.0 and 1.3, indicating that such conditions favor the balanced progression of the DRM reaction for Ni<sub>0.05</sub>CaO<sub>0.95</sub>. However, under the O<sub>2</sub>-containing condition, as depicted in Fig. 1 (b), the two concentration curves diverge from the very first cycle. Specifically, the highest H<sub>2</sub> concentration reached 7.5%, whereas the peak CO concentration was only 5.0%. This discrepancy is due to the un-reduced state of NiO after O<sub>2</sub> oxidation, which impedes the catalytic activity for the reverse Boudouard reaction of Ni [51–53]. As a result, this oxidation of Ni leads to a decline in the reverse Boudouard reaction during the hydrogenation processes.

As shown in Fig. 2, the peak at around  $38^\circ$ ,  $43^\circ$  and  $63^\circ$  correspond to NiO [54]. Ni is highly susceptible to oxidation, even in the absence of O<sub>2</sub> in the flue gas; almost all Ni species are oxidised due to the exposure to air, making Ni difficult to detect by XRD (Fig. 2). Therefore, during the CO<sub>2</sub> capture stage in the presence of O<sub>2</sub>, Ni particles undergo oxidation and lose catalytic activity [55]. As a result, NiO needs to be reduced by CH<sub>4</sub> during the CO<sub>2</sub> conversion stage of ICCU. Therefore, a certain period for pre-reduction of NiO is needed during the DRM stage [40], as shown in both Fig. 1 (b) and (d), where the concentration of  $CH_4$ remained at 10.0% for a specific period. As shown in Fig. 1 (c), in the case of steam-containing condition, in the first cycle, the highest concentration of H<sub>2</sub> was 8.0 %, while the highest concentration of CO was 6.0 %. Since this was the initial cycle, the reason for the higher  $H_2$ concentration than CO concentration was not due to catalyst performance degradation. Instead, it could be attributed to the steam adsorbed by the DFM during the  $CO_2$  capture stage, which led to additional  $H_2$ production through a side reaction with CH<sub>4</sub> via the steam methane reforming (SMR) reaction (Eq. (5)) [56]. Fig. 1 (d) demonstrates that under the simulated-flue gas condition, the concentration gap between H<sub>2</sub> and CO has widened further. In the first cycle, the highest H<sub>2</sub> concentration reached 9.6%, while the highest CO concentration was 4.9%. The possible reason could be attributed to the combined effects of catalyst performance deterioration and the occurrence of a side reaction of SMR.

$$CH_4 = C + 2H_2 \tag{4}$$

$$CH_4 + H_2O = CO + 3H_2 \tag{5}$$

Fig. 3 (a) illustrates the average pre-reduction duration of 14 cycles under four distinct types of flue gas conditions. Under the  $O_2$ -containing condition, the average pre-reaction duration for 14 cycles was 1100 s, while in the steam-containing condition, the average pre-reaction duration was 2500 s. The coexistence of steam and  $O_2$  significantly prolonged the pre-reduction duration in comparison to the scenario with  $O_2$  only. The prolonged pre-reduction duration implies that steam can function as a hindrance to the reduction of Ni. The prolonging effect of steam on the pre-reaction duration is directly related to the influence of steam on the catalyst performance. We will elucidate their connection and explain the underlying mechanism in the subsequent text.

To investigate the effects of  $O_2$  and steam in flue gas on the DRM process, CH<sub>4</sub>-TPR tests were performed on fully adsorbed samples under four different flue gas conditions. As depicted in Fig. 3 (b), the weight of the fully adsorbed samples under ideal conditions decreased from 100% to 82.5% after complete reduction with CH<sub>4</sub>. In contrast, under  $O_2$ -containing conditions, the weight percentage of the fully adsorbed sample ultimately dropped to 71%, attributable to the reduction of NiO causing greater weight loss. Under steam-containing conditions, the sample weight declined to 74%, significantly lower than under ideal conditions. The additional weight loss can be attributed to the decomposition of CaCO<sub>3</sub>, indicating a higher CaCO<sub>3</sub> percentage in samples under this condition. This observation emphasises the positive impact of steam on the adsorption performance of DFMs [57]. Under the ideal



Fig. 8. Yield of CO (a) and H<sub>2</sub> (b) during the CO<sub>2</sub> Capture stage and the CO<sub>2</sub> consumption during each cycle (c).

condition, the final weight percentage was 72%, similar to the O<sub>2</sub>-containing condition but lower than the steam-containing condition. Fig. 3 (c) displays the desorption rates of the samples. For conditions containing steam (steam-containing and simulated flue gas conditions), the initial desorption rate was around 2 %.min<sup>-1</sup>. In conditions without steam (ideal and O<sub>2</sub>-containing conditions), the initial desorption rate was only around 1.25 %.min<sup>-1</sup>. The higher desorption rate suggests that steam can affect the CaCO<sub>3</sub> structure [58], facilitating its decomposition. In the first 5 min, the desorption rate of the samples adsorbed under conditions without O<sub>2</sub> (ideal and steam-containing conditions) was slightly higher than those with O<sub>2</sub> (O<sub>2</sub>-containing and simulated flue gas conditions). The lower catalytic performance of DFM under O<sub>2</sub>-containing conditions might be due to the oxidation of Ni to NiO. Notably, the catalytic activity of NiO was poor for the DRM process [40]. Therefore, it illustrates that elemental Ni plays a key role in this process.

3.1.3. Influence of  $H_2O$  and  $O_2$  presence on coke formation in ICCU-DRM During the catalytic DRM reactions, coking is nearly inevitable [49,59]. As illustrated in Fig. 4, after the DRM reaction, the coke formed on the surface of the Ni can be visualised by SEM. Fig. 5 shows the Raman spectrums of the DFMs cycled under the four flue gas conditions. Three key bands differentiate these species based on their varying degrees of order: the D band, G band, and G' band. The D band, which is

approximately at 1350 cm<sup>-1</sup>, corresponds to amorphous or disordered carbon [60]. The G band, around 1586 cm<sup>-1</sup>, is related to the sp2 stretching vibration of graphitic carbon [61]. The G' band which is at around 2679 cm<sup>-1</sup> is linked to the second-order scattering of two phonons [62]. Under the ideal conditions, the intensity ratio I<sub>D</sub>:I<sub>G</sub> is 1.41 and the I<sub>G</sub>:I<sub>G'</sub> is 1.21. A I<sub>G</sub>:I<sub>G'</sub> ratio >1 typically suggests the formation of a single-layer coke structure. However, under O<sub>2</sub>-containing conditions, the  $I_D:I_G$  ratio decreases to 0.92, and  $I_G:I_G$ , reduces to 0.69, which is less than 1, implying a multi-layered coke structure [63]. With the addition of steam only in flue gas (steam-containing condition), the I<sub>D</sub>:I<sub>G</sub> ratio further decreases to 0.69, while the  $I_G:I_G$ , ratio is 0.66. Under the simulated flue gas conditions, the I<sub>D</sub>:I<sub>G</sub> ratio is the lowest among all groups at 0.32, with the  $I_G:I_{G'}$  ratio maintained at 0.69. This suggests that the DFM, during the CO<sub>2</sub> capture stage, when adsorbing flue gas containing steam or O<sub>2</sub>, tends to facilitate the formation of coke with a higher degree of structural order. Moreover, the coke formed under these conditions typically exhibits a multi-layered structure [64].

As shown in Fig. 5, no carbon-related peaks are detected in the samples post  $CO_2$  reduction, suggesting that the majority of the accumulated carbon is consumed during the  $CO_2$  capture stage. This observation aligns well with findings from previous research [31,40,54]. Consequently, in ICCU-DRM, the catalyst performance degradation issue caused by coke deposition can be addressed. However, with the increase

Summary of CO<sub>2</sub> consumptions, CO yields and H<sub>2</sub> yields during the CO<sub>2</sub> capture stage under ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>), O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>), steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>) of 14 cycles.

$\mathrm{CO}_2 + \mathrm{N}_2$				$CO_2 + H_2$	$20 + N_2$		
Cycle No.	$CO_2$ consumption (mmol. $g^{-1}$ )	CO yield (mmol. $g^{-1}$ )	$H_2$ yield (mmol. $g^{-1}$ )	Cycle No.	$CO_2$ consumption (mmol. $g^{-1}$ )	CO yield (mmol. $g^{-1}$ )	$H_2$ yield (mmol. $g^{-1}$ )
1	10.63	0.00	0.00	1	11.71	0.00	0.00
2	11.81	4.92	0.00	2	12.11	9.70	5.58
3	13.92	7.33	0.00	3	10.83	6.51	4.12
4	13.10	7.71	0.00	4	10.40	5.33	3.52
5	12.95	7.76	0.00	5	9.51	4.68	3.21
6	13.11	7.19	0.00	6	8.77	4.20	2.87
7	11.95	6.83	0.00	7	8.75	3.71	2.41
8	8.82	6.55	0.00	8	7.76	3.52	2.26
9	10.62	6.43	0.00	9	7.82	3.23	2.06
10	10.31	5.97	0.00	10	6.90	3.13	2.00
11	10.00	5.73	0.00	11	6.21	3.02	1.88
12	9.82	5.39	0.00	12	7.21	2.89	1.72
13	9.06	5.29	0.00	13	7.20	2.95	1.76
14	9.89	5.13	0.00	14	6.81	2.74	1.59

 $\mathrm{CO}_2 + \mathrm{O}_2 + \mathrm{N}_2$ 

$CO_2 + H$	$_{2}O + O_{2} + N_{2}$		
Cycle	CO <sub>2</sub> consumption		

Cycle No.	$CO_2$ consumption (mmol.g <sup>-1</sup> )	CO yield (mmol.g <sup>-1</sup> )	H <sub>2</sub> yield (mmol.g <sup>-1</sup> )	Cycle No.	$CO_2$ consumption (mmol.g <sup>-1</sup> )	CO yield (mmol.g <sup>-1</sup> )	H <sub>2</sub> yield (mmol.g <sup>-1</sup> )	
1	9.96	0.00	0.00	1	5.96	0.00	0.00	
2	4.34	2.35	0.00	2	5.41	3.94	0.62	
3	4.67	2.39	0.00	3	4.56	2.64	0.32	
4	4.05	2.08	0.00	4	4.76	1.92	0.08	
5	4.30	1.81	0.00	5	4.62	1.44	0.07	
6	5.97	1.21	0.00	6	4.15	1.45	0.04	
7	4.35	1.22	0.00	7	4.08	1.08	0.05	
8	3.93	1.14	0.00	8	4.17	1.02	0.06	
9	3.69	1.03	0.00	9	4.06	0.87	0.04	
10	3.75	0.88	0.00	10	3.41	0.70	0.02	
11	3.73	0.84	0.00	11	3.75	0.74	0.02	
12	3.74	0.94	0.00	12	3.32	1.01	0.01	
13	3.93	0.85	0.00	13	3.18	0.99	0.02	
14	4.00	0.77	0.00	14	2.94	0.89	0.05	

in cycle numbers, we still observe a decline in the  $CO_2$  adsorption and conversion performance of DFM. This decrease is likely not associated with carbon deposition, but rather attributable to sintering or other changes within the DFMs. As CO is a toxic and harmful gas, the amount of CO generated during the  $CO_2$  capture stage should be minimised as much as possible.

Figs. 6 and 7, respectively, present the XPS spectra and peak fitting results for the C 1s region of the DFM samples following adsorption under four different flue gas conditions and after 14 cycles. Semiquantitative analysis was performed on different types of carbon based on the fitting results. The specific Peak Binding Energies (BE) for individual C-C/C=C, C-O, C=O, and O-C=O [65-67] species can be found in Table 1. For the samples that underwent adsorption under ideal conditions, O2-containing conditions, steam-containing conditions, and simulated flue gas conditions, the atom percentages of C-C/C=C were 53.06%, 43.00%, 43.97%, and 59.27%, respectively. After cyclic processes, these percentages changed to 64.67%, 52.88%, 49.03%, and 39.95%. For the first three conditions, an increase in the percentage of C-C/C=C after cycling indicates the presence of C-C/C=C chemical bonds in the coke. However, under simulated flue gas conditions, the percentage of C-C/C=C decreased after cycling. In contrast, when comparing with the post-adsorption samples, the content of C-O in the cyclic samples increased from 10.78% to 42.45%. This suggests that for the samples that adsorbed under simulated flue gas conditions, the proportion of C-O in coke is higher than C-C/C=C. As for the O-C=O species, it is suggested that the percentage of these species in the cyclic samples is lower than in the post-adsorption samples, implying that O-C=O is the primary form in CaCO<sub>3</sub>.

#### 3.2. Product yields and cycle stability in the presence of $O_2$ and $H_2O$

Fig. 8 and Table 2 show the CO<sub>2</sub> consumptions, CO yields and H<sub>2</sub> vields during the CO<sub>2</sub> capture stage under the four different flue gas conditions. As illustrated in Fig. 8 (a), under the ideal condition, the CO yield gradually increased from 4.8 mmol $\cdot$ g<sup>-1</sup> to 7.8 mmol.g<sup>-1</sup> between the 2nd and 4th cycles, and subsequently decreased from 7.8 mmol.g<sup>-1</sup> to 5.8 mmol.g<sup>-1</sup> from the 5th to the final cycle. The decrease in CO yield can be attributed to the reduced amount of coke due to catalyst performance deterioration as the cycle numbers increase [31,40]. Under the steam-containing condition, the CO yield peaked in the second cycle at 9.9 mmol.g<sup>-1</sup> but gradually declined in the following cycles, with the CO yield in the final cycle dropping to 3 mmol. $g^{-1}$ . Fig. 8 (b) reveals a decrease in H<sub>2</sub> yield from 5.5 mmol.g<sup>-1</sup> to 1.5 mmol.g<sup>-1</sup>. Under the O<sub>2</sub>containing condition, the CO yield was significantly lower, decreasing from 4 mmol.g<sup>-1</sup> to 1.1 mmol.g<sup>-1</sup>. As shown in Fig. 8 (b), the H<sub>2</sub> yield under simulated flue gas conditions was much lower, with the highest value in the second cycle at 0.73 mmol.g<sup>-1</sup>, while the H<sub>2</sub> yield in the subsequent cycles became almost negligible.

As shown in Fig. 8 (c), under the simulated flue gas condition, the  $CO_2$  consumption during the  $CO_2$  capture stage was actually the lowest during the 14 cycles, at 10.7 mmol.g<sup>-1</sup>. From the 2nd to the 3rd cycle,  $CO_2$  consumption increased to 13.2 mmol.g<sup>-1</sup>, and then slightly decreased to 10.0 mmol.g<sup>-1</sup>. The increase in  $CO_2$  consumption from the 2nd cycle onwards is due to the side reaction between  $CO_2$  and coke, while the subsequent decrease should be caused by the reduced  $CO_2$  uptake due to the sintering of CaO [12]. In the presence of steam, the  $CO_2$  consumption in the 1st cycle was 11.5 mmol.g<sup>-1</sup>, slightly increased to 12.1 mmol.g<sup>-1</sup> in the 2nd cycle, and gradually decreased to 7.2 mmol. g<sup>-1</sup> as the number of cycles increased. Since steam can also react with



**Fig. 9.** Yield of CO and H<sub>2</sub> during DRM stage for Ni<sub>0.05</sub>/Cao<sub>0.95</sub> under (a) ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>); (b) O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>); (c) Steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and (d) Simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

coke and compete with the reaction between  $CO_2$  and coke [68],  $CO_2$ consumption only shows a slight increase. For the O2-containing condition, the  $CO_2$  consumption in the 1st cycle was 10.0 mmol.g<sup>-1</sup>, which was similar compared to ideal and steam-containing conditions. As there was no coke formed on the DFM during the 1st cycle, this cycle could reflect the intrinsic CO<sub>2</sub> capacity of the DFM, indicating that O<sub>2</sub> has little effect on the  $CO_2$  capacity of the DFM [69]. In the subsequent cycles, the  $CO_2$  consumption significantly decreased to 4.3 mmol.g<sup>-1</sup>, due to the additional CO<sub>2</sub> generation by the reaction between O<sub>2</sub> and coke. However, it did not show a further noticeable reduction with increasing cycle numbers, remaining at around 4.3 mmol.g<sup>-1</sup>. A possible reason for the stability of CO<sub>2</sub> consumption could be that the volume expansion of Ni during the oxidation process acted as a physical barrier to prevent the sintering of CaO [70]. However, under the simulated flue gas conditions, the CO<sub>2</sub> consumption in the first cycle was significantly lower than the other three conditions, only 5.7 mmol.g<sup>-1</sup>, as shown in Fig. 8 (d). This suggests that the intrinsic CO2 adsorption performance of the DFM is substantially reduced due to the combined effects of steam and O<sub>2</sub>. The  $CO_2$  consumption in the final cycle was only 2.6 mmol.g<sup>-1</sup>.

The yields of CO, H<sub>2</sub>, and CO<sub>2</sub> during the DRM stage are shown in Fig. 9 and Table 3. As depicted in Fig. 9 (a), under the ideal condition, the CO yield gradually decreased from 22.1 mmol.g<sup>-1</sup> to 14.5 mmol.g<sup>-1</sup> over 14 cycles, while H<sub>2</sub> decreased from 26.0 mmol.g<sup>-1</sup> to 18.9 mmol. g<sup>-1</sup>. The CO<sub>2</sub> yield, however, was almost negligible at less than 0.8 mmol.g<sup>-1</sup>. As shown in Fig. 9 (b), under the O<sub>2</sub>-containing condition, the yields of CO and H<sub>2</sub> were significantly lower compared to the ideal condition, with the highest CO yield being 18.9 mmol.g<sup>-1</sup> and the lowest 11.6 mmol.g<sup>-1</sup> during the 14 cycles. Meanwhile, the highest H<sub>2</sub> yield was 29.7 mmol.g<sup>-1</sup>, and the lowest was 15.1 mmol.g<sup>-1</sup>. The CO<sub>2</sub> yield was significantly more than the ideal condition, with the highest

reaching 4.1 mmol.g<sup>-1</sup> and the lowest 1.5 mmol.g<sup>-1</sup>. This suggests that Ni, after being oxidised during the CO<sub>2</sub> capture stage, may not be fully reduced during the DRM stage [36], resulting in lower catalytic performance compared to the ideal condition [40]. Under steam-containing conditions, as shown in Fig. 9 (c) and Table 3, the H<sub>2</sub> yield increased significantly, reaching a maximum of 36.3 mmol. $g^{-1}$  and dropping to 18.1 mmol.g<sup>-1</sup> after 14 cycles. At the same time, the CO<sub>2</sub> yield also increased, reaching a maximum of 4.3 mmol.g<sup>-1</sup>, indicating that CH<sub>4</sub> and H<sub>2</sub>O underwent the SMR reaction [71] (Eq. (5)). The CO yield was not significantly affected, with the highest being 22.2 mmol.g<sup>-1</sup> and the lowest 12.9 mmol.g<sup>-1</sup>. Notably, under the simulated flue gas condition, both CO and H<sub>2</sub> yields were both significantly lower, with the highest CO yield at only 10.6 mmol.g<sup>-1</sup> and the lowest at 3.9 mmol.g<sup>-1</sup>. However, the H<sub>2</sub> yield reached 23.2 mmol.g<sup>-1</sup> during the first two cycles, it was only 8.6 mmol. $g^{-1}$  by the 14th cycle. As shown in Fig. S2, the average yields of CO and H<sub>2</sub> were also the lowest under simulated flue gas conditions. At the same time, the CO2 yield was obviously higher than the other three conditions, with a maximum of 5.5  $\mathrm{mmol.g}^{-1}$  and a minimum of 3.9 mmol.g<sup>-1</sup>. The lowest CO and H<sub>2</sub> yields among the four conditions suggest that steam can promote the deteriorating effect of O<sub>2</sub> on DFM performance.

#### 3.3. Discussion of the role of steam and $O_2$ for ICCU-DRM

In this section, we aim to elucidate the impacts of steam and  $O_2$  on the reaction mechanism, focusing primarily on the  $CO_2$  capture stage and the associated changes in the  $CaCO_3$  and Ni. The mechanism investigation incorporates a range of characterisation techniques, including in-situ infrared spectroscopy, FIB-SEM, BET surface area analysis, and XPS test.

14

1.53

containing condition (10.0 %  $CO_2 + 6.7$  %  $O_2/N_2$ ) and simulated flue gas condition (10.0 %  $CO_2 + 6.0$  %  $H_2O + 6.7$  %  $O_2/N_2$ ) of 14 cycles.  $\rm CO_2 + N_2$  $\mathrm{CO}_2 + \mathrm{H}_2\mathrm{O} + \mathrm{N}_2$ Cycle No. CO<sub>2</sub> yield (mmol.g<sup>-1</sup>) CO vield (mmol.g<sup>-1</sup>)  $H_2$  yield (mmol.g<sup>-1</sup>) Cycle No.  $CO_2$  yield (mmol.g<sup>-1</sup>) CO yield (mmol.g<sup>-1</sup>) H<sub>2</sub> yield (mmol.g<sup>-1</sup>) 0.68 22.10 26.04 4.12 18.71 29.06 1 1 2 0.65 20.47 27.532 4.16 22.20 36.33 3 0.55 19.01 25.19 3 20.58 32.44 0.65 4 0.48 19.00 26.48 4 0.58 19.30 29.16 25.28 18.39 26.81 5 0.45 18.22 5 0.56 6 0.41 17.46 23.99 6 0.70 17.01 25.317 0.38 16.91 23.02 7 0.50 15.56 24.95 8 0.40 16.79 23.49 8 0.43 15.92 23.68 9 0.40 16.12 22.05 9 0.42 15.18 22.34 10 0.42 15.62 20.99 10 0.45 14.62 21.52 11 0.37 15.35 20.44 11 0.41 14.13 20.28 12 0.35 15.00 19.81 12 0.43 13.73 20.32 19.36 13 0.36 14.90 19.62 13 0.43 13.41 0.36 14.51 12 91 14 18 92 14 0 44 18.14  $CO_2 + O_2 + N_2$  $CO_2 + H_2O + O_2 + N_2$ H<sub>2</sub> yield (mmol.g<sup>-1</sup>) CO yield (mmol.g<sup>-1</sup>) CO<sub>2</sub> yield (mmol.g<sup>-1</sup>) CO yield (mmol.g<sup>-1</sup>) Cycle No. CO<sub>2</sub> yield (mmol.g<sup>-1</sup>) H<sub>2</sub> yield (mmol.g<sup>-1</sup>) Cycle No. 15.14 23.34 10.59 23.19 1 4.09 5.45 1 18.94 29.75 13.32 23.28 2 1.62 2 2.67 3 18.63 27.91 3 1.53 2.31 12.36 19.62 4 1 4 9 17 84 26 34 4 2 39 9.80 16.68 5 1.48 17.19 24.81 5 2.82 7.73 14.28 6 2.53 14.59 21.84 6 2.88 5.85 10.97 7 2.13 14.07 7 4.01 2.90 8.21 20.06 8 19.59 8 1.88 13.82 3.85 2.52 7.52 9 1.58 13.49 19.32 9 3.86 1.65 5.13 10 10 1.44 13.00 17.57 3.53 1.70 5.05 11 11 2.04 1.60 12.50 16.29 3.19 6.03 12.25 2.36 6.22 12 1.59 15.92 12 3.05 13 1.60 11.96 15.56 13 2.323.72 8.86

Summary of CO<sub>2</sub>, H<sub>2</sub>, and CO yields during the DRM stage under ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>), O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>), steam-



14

2.17

Fig. 10. (a): In-situ Infrared Spectroscopy of CaCO<sub>3</sub> decomposition under CH<sub>4</sub>; (b): In-situ Infrared Spectroscopy of CO<sub>2</sub> Re-adsorption after CaCO<sub>3</sub> Decomposition under CH<sub>4</sub>.

Fig. 10 (a) displays the in-situ infrared spectroscopy of CaCO<sub>3</sub> decomposition under CH<sub>4</sub>, while Fig. 10(b) shows the re-adsorption of CO<sub>2</sub> after this process. As shown in Fig. 10 (a), peaks at 3016 cm<sup>-1</sup> and 1304 cm<sup>-1</sup> could be attributed to the C–H vibrations of CH<sub>4</sub> [72]. The 1342 cm<sup>-1</sup> peak was related to carbonate ions, possibly an intermediate in the CaCO<sub>3</sub> and CH<sub>4</sub> reaction [73]. Peaks at 2475 cm<sup>-1</sup> and 1777 cm<sup>-1</sup> correspond to CO<sub>3</sub><sup>2–</sup> [74] and C=O [75,76], while the 2104 cm<sup>-1</sup> peak was due to C-O bonding [77,78]. It shows that the C=O and carbonate should be the key intermediates of the reaction between CH<sub>4</sub> and CaCO<sub>3</sub> during the ICCU-DRM stage. During the re-adsorption of CO<sub>2</sub>, as shown

11.62

15.06

in Fig. 10 (b), a split peak at 2326 cm<sup>-1</sup> resulted from CO<sub>2</sub> gas interference [72]. The 2104 cm<sup>-1</sup> peak should represent the C-O bond, as it also appeared during the CO<sub>2</sub> capture stage, possibly due to the overlap of CO and C-O in CaCO<sub>3</sub>. Fig. S3 further support the conclusion, as it demonstrates a stronger 2104 cm<sup>-1</sup> peak during the DRM stage and a weaker one during CO<sub>2</sub> capture. It is indicated that the decomposition of CO<sub>2</sub> is a critical step in the DRM, further highlighting the vital role of Ni catalysts in the DRM process. This is because Ni provides a surface for reactants to adsorb onto, thereby encouraging collisions between reactants and thus enhancing the reaction rate of the reverse Boudouard

3.94

8.58



Fig. 11. FIB-SEM/EDX images and EDX elemental mappings of the adsorbed sample under O<sub>2</sub>-containing condition (a nd b) and under simulated flue gas condition (c and d).

Summary of pore size and surface area by BET analysis.

Sample	BET Surface Area $(m^2.g^{-1})$	Pore volume (cm <sup>2</sup> .g <sup>-1</sup> )	Average pore diameter (Å)
A1	1.45	0.00187	57.11
A2	1.04	0.00128	53.92
A3	4.81	0.01341	129.63
A4	0.48	0.00011	14.93

A1: Ni<sub>0.05</sub>/CaO<sub>0.95</sub> adsorbed under ideal condition (10% CO<sub>2</sub>/N<sub>2</sub>).

A2: Ni\_{0.05}/CaO\_{0.95} adsorbed under O2-containing condition (10.0 % CO2 + 6.0 % H2O/N2).

A3: Ni<sub>0.05</sub>/CaO<sub>0.95</sub> adsorbed under steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

A4: Ni\_{0.05}/CaO\_{0.95} adsorbed under simulated flue gas condition (10.0 % CO\_2 + 6.0 % H\_2O + 6.7 % O\_2/N\_2).

#### reaction [79].

The key role of Ni in the catalytic activity of DFMs necessitates a thorough examination of the structural attributes, especially under the influences of steam and  $O_2$ , which induce a notable decline in their catalytic efficiency. This reduction can potentially be attributed to the properties of the elemental Ni, requiring further exploration. FIB-SEM was used to probe the cross-sectional morphologies and elemental dissemination within the samples exposed to  $O_2$ -containing conditions and simulated flue gas conditions, respectively. In Fig. 11, a comparative

analysis of the internal morphological differences of Ni<sub>0.05</sub>/Cao<sub>0.95</sub> DFMs subjected to these varying conditions is illustrated. When assessing Fig. 11 (b), a concentration surge of Ni near the surface of samples fully adsorbed under the O<sub>2</sub>-containing condition can be observed, with a prominent reduction at approximately 1.4  $\mu$ m from the surface, denoting a superficial predominance of Ni in the sample. Fig. 11 (a) further elucidates this trend, with EDX mappings of Ni revealing a primary distribution area within the external layer of the sample, while the existence of a void at the center suggests a significantly diminished Ni concentration within the interior of the samples.

On assessing the samples fully adsorbed under simulated flue gas conditions (Fig. 11 (c)), a characteristic of a white outer layer is observed. It is evident that steam significantly influences the morphology of the DFM. As presented in Fig. 11 (d), the examination of Ni elemental concentration reveals a lower concentration in the white area, with the peak value of Ni concentration observed at the intersection of the white outer shell and the interior darker region. This is further corroborated by the EDX mappings of Ni (Fig. 11 (c)), indicating a sparse distribution of Ni throughout the cross-section of the samples. It is suggested that a considerable proportion of Ni is deeply embedded within the sample. Therefore, steam could enhance the dispersion of Ni particles within the samples durin the carbon capture stage.

Table 4 presents the BET test results of the samples subjected to adsorption under four distinct flue gas conditions. The samples fully adsorbed under the steam-containing condition reached the maximum



**Fig. 12.** XPS analysis of Ca 2p region of Ni<sub>0.05</sub>/CaO<sub>0.95</sub> after adsorption under (a) ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>); (b) O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>); (c) Steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and (d) Simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).



Fig. 13. XPS analysis of Ca 2p region of Ni<sub>0.05</sub>/CaO<sub>0.95</sub> after 14 cycles under (a) ideal condition (10.0 % CO<sub>2</sub>/N<sub>2</sub>); (b) O<sub>2</sub>-containing condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O/N<sub>2</sub>); (c) Steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) and (d) Simulated flue gas condition (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

values of BET surface area, pore volume, and average pore diameter, which were 4.81 m<sup>2</sup>.g<sup>-1</sup>, 0.01341 cm<sup>2</sup>.g<sup>-1</sup>, and 129.63 Å, respectively. Conversely, under the simulated flue gas condition, these three values were the smallest, specifically 0.48 m<sup>2</sup>.g<sup>-1</sup>, 0.00011 cm<sup>2</sup>.g<sup>-1</sup>, and 14.93 Å, respectively. The observed decrease in BET surface area, pore volume, and average pore diameter under simulated flue gas conditions implies that the presence of steam and O<sub>2</sub> may negatively impact the structural properties of the samples.

XPS testing for valence states of Ca 2p in the samples offers a novel perspective for elucidating the underlying mechanisms, enhancing the depth and precision of our analysis. Full XPS spectra of adsorbed and cycled samples under different flue gas conditions are shown in Fig. S4. Figs. 12 and 13 illustrate the outcomes of peak fitting for the Ca 2p spectra acquired from XPS analysis of samples following adsorption under four disparate flue gas conditions and samples after 14 reaction cycles, respectively. Fig. S5 displays the semi-quantitative evaluation of the peak fitting, uncovering the atomic percentages of CaO 2p, CaCO<sub>3</sub> 2p, and Ca(OH)<sub>2</sub> 2p within the Ca 2p elements of each respective sample. The specific Peak Binding Energies (BE) of individual CaO 2p, CaCO<sub>3</sub> 2p and Ca(OH)<sub>2</sub> 2p [80–82] can be seen in Table 5. Through the comparison of various Ca 2p atom percentages, the adsorption capacity of DFM for CO<sub>2</sub> under diverse flue gas conditions and the regeneration capability of DFM can be assessed.

Taking the samples after adsorption and after cycling under ideal condition as benchmarks, it can be observed that the content of CaO 2p after adsorption is 14.38%, while that of CaCO<sub>3</sub> 2p is 85.62%. After cycling, not all CaCO<sub>3</sub> can be converted back to CaO; the content of CaO

2p is 83.5%, while that of CaCO<sub>3</sub> 2p is 16.5%. In comparison, it can be found that the presence of O<sub>2</sub> alone in the flue gas seems to have little impact on the carbonation of CaO, as CaO 2p is 13.76% and CaCO<sub>3</sub> 2p is 86.24%. However, due to the reduction in catalytic performance caused by Ni oxidation, the regeneration capacity of CaO 2p under O2-containing condition decreases. The content of CaO 2p after cycling is 66.65%, which is lower than that under ideal conditions, and the content of CaCO<sub>3</sub> is 16.5%, which is higher than that under ideal condition. Under the steam-containing condition, the content of CaO 2p is significantly reduced compared to the previous two conditions, accounting for only 6.66%, while 81.09% is CaCO3 2p, and 12.25% is Ca(OH)2 2p. After cycling, the content of CaO 2p in the sample remains lower than that under the ideal condition, at 68.4%, with CaCO<sub>3</sub> 2p content at 24.64%, and 6.96% of Ca(OH)<sub>2</sub> 2p. This indicates that although previous studies have suggested that steam is favorable for CO2 adsorption, the increased thickness of the formed CaCO<sub>3</sub> [83] layer could encapsulate the catalyst, thereby reducing catalytic performance and hindering the decomposition of CaCO<sub>3</sub> and the regeneration of CaO.

For the sample adsorbed under simulated flue gas conditions, the content of CaO 2p is 14.07%, that of CaCO<sub>3</sub> 2p is 74.97%, and Ca(OH)<sub>2</sub> 2p accounts for 10.96%. Upon comparison with the sample after adsorption under water-containing conditions, a decline in the contents of CaCO<sub>3</sub> 2p and Ca(OH)<sub>2</sub> 2p is evident. In conjunction with the prior analysis, this phenomenon may be attributed to the rapid oxidation of Ni and the formation of a CaCO<sub>3</sub> layer on the surface. Such processes obstruct the penetration of steam and CO<sub>2</sub> into the sample interior, consequently causing a certain level of blockage on the surface and

Summary	of atomic	ratio of	Ca 2p b	y XPS	analysis.
					-

Sample	Name	Peak BE	Atomic %
A1	CaO 2p <sub>3/2</sub>	346.29	14.38
	CaO 2p <sub>1/2</sub>	349.61	
	$CaCO_3 2p_{3/2}$	347.11	85.62
	CaCO <sub>3</sub> 2p <sub>3/2</sub>	350.64	
A2	CaO 2p <sub>3/2</sub>	346.23	13.76
	CaO $2p_{1/2}$	349.57	
	$CaCO_2 2D_2/2$	347.09	86.24
	$CaCO_3 2p_{1/2}$	350.66	
A3	CaO 2p <sub>3/2</sub>	346.28	6.66
	$CaO 2p_{1/2}$	349.69	
	$CaCO_2 2p_2/2$	347.52	81.09
	$CaCO_2 2p_1/2$	351.04	
	$Ca(OH)_{0} 2p_{1/2}$	348 56	12.25
	$Ca(OH)_2 2p_{3/2}$	352 15	12.25
	Ca(OII) <sub>2</sub> 2p <sub>1/2</sub>	332.13	
Δ <i>Δ</i>	(a) 2n	346.08	14.07
117	$CaO 2p_{3/2}$	340.00	14.07
	$CaO 2p_{1/2}$	246.02	74.07
		250.42	/4.5/
	$CaCO_3 2p_{3/2}$	330.42	10.06
	$Ca(OH)_2 2p_{3/2}$	347.99	10.96
	$Ca(OH)_2 2p_{1/2}$	351.42	
C1	$C_{2}O_{2}D_{2}$	346 27	83 50
01	CaO 2p <sub>3/2</sub>	240.27	05.50
	$CaO 2p_{1/2}$	247.22	16 50
	$CaCO_3 2p_{3/2}$	347.32	10.50
	CaCO <sub>3</sub> 2p <sub>3/2</sub>	350.73	
C2	$C_{2}O_{2}D_{2}$	347 74	66 65
02	$C_{2}O_{2}P_{3/2}$	351.34	00.00
	$CaO_2p_{1/2}$	240 54	22.2E
	$CaCO_3 2p_{3/2}$	251.04	33.33
	GaGO <sub>3</sub> 2p <sub>1/2</sub>	331.64	
C3	(a) 2n	347 29	68.40
60	$CaO 2p_{3/2}$	350.78	00.10
	$CaO 2p_{1/2}$	240 12	24.64
	$CaCO_3 2p_{3/2}$	340.13	24.04
	$CaCO_3 2p_{1/2}$	351.04	6.06
	$Ca(OH)_2 2p_{3/2}$	349.18	6.96
	$Ca(OH)_2 2p_{1/2}$	352.42	
C4	CaO 2n	346.01	10 / 2
64	$CaO 2p_{3/2}$	340.01	12.40
	$CaO 2p_{1/2}$	245.41	64 67
	CaCO <sub>3</sub> 2p <sub>3/2</sub>	346.99	64.67
	CaCO <sub>3</sub> 2p <sub>3/2</sub>	350.46	00.00
	Ca(OH) <sub>2</sub> 2p <sub>3/2</sub>	348.00	22.90
	Ca(OH) <sub>2</sub> 2p <sub>1/2</sub>	351.45	

A1: Ni<sub>0.05</sub>/CaO<sub>0.95</sub> adsorbed under ideal condition (10% CO<sub>2</sub>/N<sub>2</sub>).

A2: Ni\_{0.05}/CaO\_{0.95} adsorbed under O2-containing condition (10.0 % CO2 + 6.0 % H2O/N2).

A3: Ni\_{0.05}/CaO\_{0.95} adsorbed under steam-containing condition (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>).

A4: Ni\_{0.05}/CaO\_{0.95} adsorbed under simulated flue gas condition (10.0 % CO\_2 + 6.0 % H\_2O + 6.7 % O\_2/N\_2).

C1:  $\mathrm{Ni}_{0.05}/\mathrm{CaO}_{0.95}$  regenerated after 14 cycles under ideal condition.

C2: Ni<sub>0.05</sub>/CaO<sub>0.95</sub> regenerated after 14 cycles under O<sub>2</sub>-containing condition. C3: Ni<sub>0.05</sub>/CaO<sub>0.95</sub> regenerated after 14 cycles under steam-containing condition.

C4:  $Ni_{0.05}/\text{CaO}_{0.95}$  regenerated after 14 cycles under simulated flue gas condition.

leading to a reduction in the amount of CaO subjected to carbonation and hydroxylation. The most crucial information pertains to the sample after cycling, in which the contents of CaO 2p, CaCO<sub>3</sub> 2p, and Ca(OH)<sub>2</sub> 2p are 12.43 %, 64.67 %, and 22.9 %, respectively. The sample appears nearly unchanged compared to its state after adsorption. This suggests that under simulated flue gas conditions, the capability of DFM regeneration is poor.

Based on the characterisation results of in-situ infrared, FIB-SEM, BET surface area analysis, and XPS analysis, we have analysed the influence mechanism of steam and O2 on DMF. Through the characterisation results of in-situ infrared and the analysis of the cause of the prereaction duration mentioned earlier, it is confirmed that Ni in the nonoxidised state plays a critical role in the progress of DRM. In other words, Ni in the oxidised state exhibits almost no catalytic activity and needs to be reduced by CH<sub>4</sub> first to initiate DRM. It is suggested that after the reduction of NiO in DMF, CaCO3 could be decomposed to generate CO<sub>2</sub>, reacting with CH<sub>4</sub>. Therefore, the internal structural differences of DMF after adsorption were compared under O2-containing conditions and simulated flue gas conditions using FIB-SEM. As the presence of steam accelerated the expansion of CaO [57,84-86], thereby encapsulating NiO (Fig. S6). Therefore, the NiO in DMF after adsorption under simulated flue gas conditions is more difficult to reduce compared to that under O<sub>2</sub>-containing conditions.

It was found that under simulated flue gas conditions, the DMF samples post-adsorption exhibited the lowest values for BET surface area, pore volume, and average pore diameter. This trend was not replicated in the samples adsorbed under steam-containing conditions. Based on these observations, it can be recognised as a connection to the specific oxidative processes of Ni. Due to the volumetric expansion that Ni [21,51,87] undergoes during the oxidation process, which is also a feature of CaO absorption of  $CO_2$ , the simultaneous action of both elements results in a swift formation of a dense shell on the surface of the DFM. This is particularly the case when steam is present, as it accelerates the absorption of  $CO_2$  and promotes this expansion. This phenomenon can be confirmed by the outer white section of the sample in Fig. 11 (c).

This shell formation leads to a decrease in the permeability of CH<sub>4</sub> into the sample, inhibiting its contact with NiO for the reduction process. Therefore, a significant portion of NiO was permanently embedded within the DFM, reducing its reducibility and causing a loss in catalytic activity. In the absence of Ni, it becomes difficult for CaCO<sub>3</sub> to decompose quickly. Therefore, the regenerative capability of DFMs under simulated flue gas conditions is significantly reduced, as revealed by the XPS testing results and leads to a significant decrease in the yields of CO and H<sub>2</sub> (Fig. 9).

#### 4. Conclusions

In conclusion, this study systematically investigated the ICCU-DRM process under various flue gas conditions, focusing on the effects of  $O_2$  and steam. These results offer valuable insights into the effects of  $O_2$  and steam on the ICCU-DRM process and emphasise the need to consider these factors in the design and operation of practical industrial applications. The key findings are as follows:

- (1) O<sub>2</sub> alone in the flue gas (10.0 % CO<sub>2</sub> + 6.7 % O<sub>2</sub>/N<sub>2</sub>) negatively affects the DRM performance of the DFM, reducing the highest CO yield and H<sub>2</sub> from 22.1 mmol.g<sup>-1</sup> to 18.9 mmol.g<sup>-1</sup> and from 18.9 mmol.g<sup>-1</sup> to 15.1 mmol.g<sup>-1</sup> compared to the ideal condition (10% CO<sub>2</sub>/N<sub>2</sub>), respectively, primarily due to the generation of extra CO<sub>2</sub> from O<sub>2</sub> reacting with coke and oxidising Ni.
- (2) FIB-SEM/EDX images and EDX elemental mappings show that the presence of steam in the flue gas (10.0 % CO<sub>2</sub> + 6.0 % H<sub>2</sub>O + 6.7 % O<sub>2</sub>/N<sub>2</sub>) results in NiO encapsulation by CaCO<sub>3</sub>, which hinders NiO reduction by CH<sub>4</sub> and leads to a decrease in catalytic performance with lowest CO yield of 3.9 mmol.g<sup>-1</sup> and H<sub>2</sub> yield of only 8.6 mmol.g<sup>-1</sup>.
- (3) The formation of a CaCO<sub>3</sub> shell layer due to rapid CO<sub>2</sub> adsorption under simulated flue gas conditions is the primary reason for the reduction in surface area and pore size, with surface area, pore volume, and average pore diameter of 0.48 m<sup>2</sup>.g<sup>-1</sup>, 0.00011 cm<sup>2</sup>. g<sup>-1</sup>, and 14.93 Å, respectively. XPS analysis confirms these findings, revealing that the presence of both O<sub>2</sub> and steam in the flue gas significantly reduces the regenerative capability of DFM.

As after adsorption, the content of CaO 2p was 14.07%, CaCO<sub>3</sub> 2p was 74.97%, and Ca(OH)<sub>2</sub> 2p was 10.96%. After 14 cycles the contents of CaO 2p, CaCO<sub>3</sub> 2p, and Ca(OH)<sub>2</sub> 2p were 12.43%, 64.67%, and 22.9%, nearly unchanged compared to the samples after adsorption.

With the current understanding of the interaction mechanism between steam and O<sub>2</sub>, there is a compelling need for the development of innovative strategies to prevent or mitigate the effect of their simultaneous presence. This has become more feasible and urgently necessary, given their observed impact on the ICCU-DRM process. Effective strategies might involve refining the structure of catalysts, particularly adjusting how catalysts are distributed on sorbents. The goal of these adjustments is to ensure the catalysts are positioned in a way that prevents them from being encapsulated by CaCO<sub>3</sub> after the CO<sub>2</sub> capture stage. It is vital to guarantee that the catalyst is reduced during the DRM stage, thereby ensuring the restoration of its catalytic performance. Moreover, system-level improvements could also be introduced to address this issue. For instance, incorporating metal elements at the front end of DFM could serve to reduce the O<sub>2</sub> content, while the addition of desiccants could minimise the level of steam, thereby diminishing the possibility of simultaneous existence of O<sub>2</sub> and steam. This approach presents another possible innovation for optimising the ICCU-DRM process. Through the implementation of these two measures, the practical applicability of the process in industrial scenarios could be enhanced, potentially leading to improved performance and increased system stability.

#### CRediT authorship contribution statement

Xiaotong Zhao: Investigation, Formal analysis, Data curation, Methodology, Writing - original draft. Shuzhuang Sun: Methodology, Formal analysis, Investigation. Yingrui Zhang: Investigation, Formal analysis. Yuanyuan Wang: Investigation, Formal analysis. Yuan Zhu: Investigation, Formal analysis. Paul Williams: FIB-SEM test, Review & editing. Shaoliang Guan: Investigation. Chunfei Wu: Conceptualization, Writing – review & editing, Supervision, Project administration, Funding acquisition.

#### **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data availability

The authors are unable or have chosen not to specify which data has been used.

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#### Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.seppur.2023.124866.

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