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1 **First-principles DFT Insights into the Stabilization of Zinc Diphosphide (ZnP₂) Nanocrystals**
2 **via Surface Functionalisation by 4-Aminothiophenol for Photovoltaic Applications**

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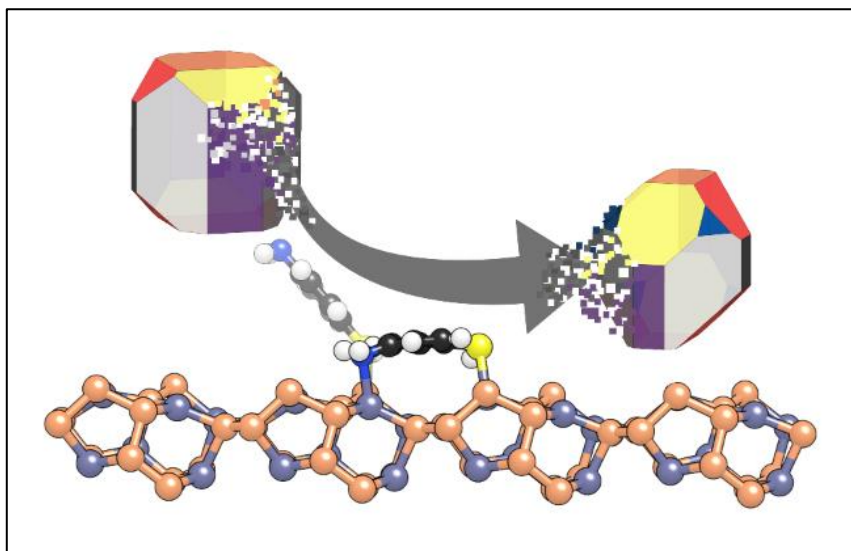
14 **ABSTRACT**

15 The resurgence of interest in zinc phosphide compounds as efficient solar absorbers has initiated
16 increasing efforts to improve their stability under humid and oxygen-rich conditions. Although
17 organic functionalisation has been suggested as a promising strategy to passivate zinc phosphide
18 nanoparticles, fundamental atomic-level insights into the adsorption processes and structures at zinc
19 diphosphide (ZnP₂) surfaces is still lacking. In this study, the interactions between 4-aminothiophenol
20 and the low Miller index surfaces of monoclinic ZnP₂ have been investigated by means of density
21 functional theory (DFT) calculations. A bidentate adsorption mode, in which the 4-aminothiophenol
22 binds through both its functional groups via Zn-N and Zn-S bonds, was predicted to be the strongest
23 form of interaction and monolayer-functionalised ZnP₂ surfaces were found to be highly stable under
24 adsorbate-rich conditions. Changes in the equilibrium morphology of ZnP₂ nanocrystallites upon
25 functionalisation and effects of humidity are also discussed. The results are expected to contribute
26 towards the rational design of ZnP₂-based materials for photovoltaic devices.

27 **KEYWORDS:** zinc diphosphide, 4-aminothiophenol, DFT, functionalisation, photovoltaics.

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29

30 **GRAPHICAL ABSTRACT**

31

32

33 **1 INTRODUCTION**

34 Earth-abundant semiconducting materials are attractive solutions in large-scale energy conversion
35 and storage deployment ^{1,2}. There is resurgence of interest in Zn-P compounds as promising materials
36 for cost-effective and scalable thin-film PV applications ³⁻⁶, because their constituent elements are
37 non-toxic, widely available, and more affordable alternatives to currently employed PV materials ⁷.
38 Among various Zn-P compounds, the tetragonal Zn₃P₂ has received particular attention owing to its
39 suitable direct band gap of 1.5 eV ⁸, high visible-light absorption and extinction coefficient (>10⁴
40 cm⁻¹) ^{9,10}, and long minority-carrier diffusion length (~10 μm) ¹¹. Although the band gap energies of
41 the monoclinic ZnP₂ member, reported in the range of 1.33-1.60 eV ¹²⁻¹⁴, also fall within the values
42 of the maximum absorption efficiency described by the Shockley-Queisser limit ¹⁵⁻¹⁷, it has been
43 largely overlooked as a candidate for PV applications. It has, however, been investigated and used as
44 anode material in Li-ion and Na-ion batteries, together with its tetragonal polymorph and Zn₃P₂
45 counterparts ¹⁸⁻²⁰, where ZnP₂ has shown to outperform Zn₃P₂ in electrochemical performance for it
46 has higher reversible capacity. In a recent DFT study, we have systematically characterised the
47 mechanical, structural, and electronic properties of both tetragonal and monoclinic ZnP₂ and
48 discussed their implications for PV applications ²¹. The spectroscopic limited maximum efficiency

(SLME) of monoclinic ZnP_2 was computed to be 10% for a reasonable reference thickness of 0.5 μm , demonstrating its promising potential as a photovoltaic material.

However, an inherent drawback of zinc-based semiconducting materials is their low surface stability in the presence of moisture and oxygen^{22–26}, and efforts are now directed towards surface modification to aid their integration into photovoltaic devices and exploit their suitable optoelectronic properties. An important part of these investigations has focused on the modification of known inorganic semiconductor surfaces with organic molecules. Organic functionalisation can enhance the surface stability against temperature and possible oxidation in the presence of oxygen and moisture that could result in material degradation. Protection strategies of zinc-based semiconductors (e.g. Zn-O^{27–29}, Zn-S^{30–32}, Zn_3P_2 ^{8,33}, Zn_3N_2 ^{34,35}, ZnSe^{36,37}) for photovoltaic applications, employing different surfactant molecules ranging from amines³⁸, over acids³⁹ and thiols⁴⁰, to polymers^{41–43}, have been developed and successfully applied in the laboratory. In addition, hybrid organic-inorganic nanostructures are of great interest because their optical and electrical properties can be efficiently tailored by surface functionalisation. For example, surface modification of ZnO nanoparticles with pyrene-1-carboxylic acid showed an increase in the short circuit current densities with lower open circuit voltage, which was assigned to better carrier collection³⁹. Furthermore, large scale functionalisation of Zn_3P_2 nanowires was achieved experimentally with two different ligand molecules, 4-aminothiophenol and 3-propanedithiol. Both systems were found to be stable over 120 days without any agglomeration or degradation, thus providing a promising method for Zn_3P_2 passivation⁴⁰. Based on these results, theoretical insight was reported on the adsorption strength and charge transfer between the 4-aminothiophenol and 3-propanedithiol and the Zn_3P_2 nanowires, which were represented by a (100) slab⁴⁴. A more recent study focused on the functionalisation of Zn_3P_2 nanocrystals with 4-aminothiophenol and its influence on the electronic properties of varying surfaces of the material⁴⁵. Different coupling schemes of available functional groups have been investigated and inequivalent interacting modes were assigned as the most favourable for distinct Zn_3P_2 surfaces.

74 Despite extensive investigations on the functionalisation of Zn_3P_2 , no studies have been dedicated to
75 the characterisation of the interface chemistry between ZnP_2 and organic molecules, which makes
76 this work timely. In the present study, a series of comprehensive DFT calculations were carried out,
77 aimed at unravelling the mechanism of adsorption of 4-aminothiophenol on the low Miller index
78 surfaces of monoclinic ZnP_2 and depict the properties of the resulting hybrid structures. Atomic-level
79 calculations based on DFT techniques have become powerful tools to describe organic-inorganic
80 interfaces because they are capable of accurately predicting the lowest-energy adsorption geometries
81 and identifying the charge transfer and other electronic effects^{46,47}. As the 4-aminothiophenol has
82 two functional groups (thiol ($-\text{SH}$) and amine ($-\text{NH}_2$) end group), different binding modes that
83 involve one or both functional groups and structures have been considered and the energetics of the
84 obtained adsorption configurations have been analysed. Using predicted surface energies, the
85 equilibrium morphologies of ZnP_2 nanocrystals before and after functionalisation with 4-
86 aminothiophenol were constructed according to the Wulff theorem. The results are expected to
87 provide useful information about the interactions between ZnP_2 and organic molecules for future
88 functionalisation strategies and shape modulation of ZnP_2 nanoparticles.

89 2 COMPUTATION DETAILS

90 The spin-polarised Kohn-Sham density functional theory (KS-DFT)^{48,49} calculations were carried
91 out within the Vienna Ab-initio Simulation Package (VASP)⁵⁰. The exchange-correlation functional
92 was approximated using the Perdew-Burke-Ernzerhof (PBE) parametrisation of the generalized
93 gradient approximation (GGA) for structural relaxations, whereas the hybrid HSE06 functional was
94 employed for the calculation of electronic properties.⁵¹ The projector augmented wave (PAW)^{52,53}
95 pseudo-potential method was imposed and core electrons up to and including the 3p for Zn, 2p for P
96 and S, and 1s levels for N and O, respectively, were kept frozen. The long-range dispersion
97 interactions were added through the DFT-D3 method, as developed by Grimme *et al*⁵⁴.

98 The kinetic energy cut-off of the wave functions was set to 500 eV, with bulk calculations carried out
99 in the reciprocal space with a $5 \times 5 \times 5$ k-point mesh. The METADISE code⁵⁵ was employed to

construct the surfaces and final surface models were built as slabs of material with periodic boundary conditions and a 20 Å vacuum was added in the direction orthogonal to the surface to prevent interaction between vertical images. Each surface slab is built of twelve atomic layers, comprising four ZnP-P-ZnP trilayers that each contain four ZnP₂ units, schematically represented for the (001) surface in Figure S1 of supplementary information, SI. All atoms in the surface slab were allowed to relax and a k-point mesh of 5 × 5 × 1 was used for the surface relaxations. The conjugate gradient technique, with total energy and force convergence criteria set at 10⁻⁶ eV and 0.01 eV/Å, respectively, has been used to perform structural optimisations.

Surface energies of relaxed bare slabs (γ) have been calculated as:

$$\gamma = \frac{E_{\text{slab}}^{\text{DFT}} - n \times E_{\text{bulk}}^{\text{DFT}}}{2A_{\text{slab}}}, \quad (1)$$

where $E_{\text{slab}}^{\text{DFT}}$ and $E_{\text{bulk}}^{\text{DFT}}$ represent the DFT energies of relaxed slab and bulk, respectively, A_{slab} is the surface area, and n is the ratio between the number of atoms in the slab and the number of atoms in the bulk. The electronic work function (ϕ), i.e. the energy required to withdraw an electron from the solid to the vacuum, was calculated as the difference of the vacuum electrostatic potential energy and the Fermi energy.

Adsorption of 4-aminothiophenol was carried out on the 2 × 1 ((011), (101), and (111)) and 2 × 2 ((001), (010), (100), and (110)) supercells of the ZnP₂ surfaces to minimise the lateral interactions between the adsorbate molecules. The k-point mesh was subsequently reduced to a Gamma point only as the cell vectors reached lengths over 30 Å. Three initial adsorption modes were chosen, a single horizontal and two vertical geometries, one with the -SH and the other with the -NH₂ group facing the surface. No constraints on surface atoms or the adsorbing molecule were imposed during the optimisation. In particular, the 4-aminothiophenol molecule was free to move laterally and vertically from its initial binding site or reorient itself to find the lowest-energy adsorption geometry.

123 The strength of interaction between 4-aminothiophenol and ZnP_2 surfaces was characterized by the
 124 adsorption energy (E_{ads}), calculated as:

$$125 \quad E_{\text{ads}} = E_{\text{surface+adsorbate}}^{\text{DFT}} - (E_{\text{surface}}^{\text{DFT}} + E_{\text{adsorbate}}^{\text{DFT}}) \quad (2)$$

126 with $E_{\text{surface+adsorbate}}^{\text{DFT}}$, $E_{\text{surface}}^{\text{DFT}}$, and $E_{\text{adsorbate}}^{\text{DFT}}$ representing calculated energies of the surface with adsorbed
 127 4-aminothiophenol, bare surface, and a 4-aminothiophenol molecule in a box, respectively. Negative
 128 E_{ads} indicates spontaneous adsorption of the molecule onto the surface with the most stable
 129 adsorption mode characterised by the most negative adsorption energy. The reference energy of a
 130 single adsorbate molecule was determined by optimising it in a cubic box of 15 Å a side and sampling
 131 only over a Gamma point.

132 Full coverage functionalisation was carried out only for the most favourable adsorption mode, where
 133 the surface is considered to be fully functionalised once all the zinc atoms on the surface are
 134 unavailable for further adsorption. Surface energies after functionalisation (γ_f) were determined as:

$$135 \quad \gamma_f = \frac{E_{\text{func surface}}^{\text{DFT}} - (n \times E_{\text{bulk}}^{\text{DFT}} + m \times E_{\text{adsorbate}}^{\text{DFT}})}{2A_{\text{slab}}} \quad (3)$$

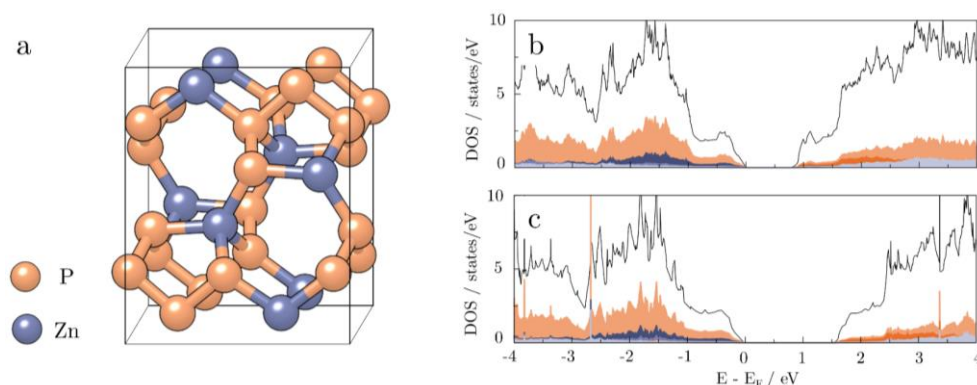
136 where $E_{\text{func surface}}^{\text{DFT}}$ is the calculated energy of the surface functionalised by a full monolayer, and m is
 137 the number of adsorbed 4-aminothiophenol molecules. Calculated surface energies were
 138 subsequently employed to obtain Wulff morphologies⁵⁶, as implemented in the Wulffmaker⁵⁷
 139 software.

140 **3 RESULTS AND DISCUSSION**

141 **3.1 Bulk and surface characterisation of ZnP_2**

142 The monoclinic ZnP_2 bulk structure (Figure 1a), optimised at the GGA level, has lattice parameters
 143 of $a=8.611$ Å, $b=7.239$ Å, and $c=7.530$ Å, which agree well with the available experimental data¹²
 144 of $a=8.863$ Å, $b=7.288$ Å, $c=7.560$ Å. In our previous study²¹, we found that standard GGA

145 significantly underestimated the electronic band gap of monoclinic ZnP_2 , with the calculated value
 146 of 0.76 eV being only half the experimental band gap, reported in the range of 1.33-1.60 eV^{13,58-60}.
 147 To resolve this deficiency, the hybrid HSE06 functional was employed, yielding an electronic band
 148 gap of 1.46 eV, in good agreement with experimental data. The orbital-decomposed densities of state
 149 computed with the GGA (PBE) and HSE06 functional are shown in Figure 1b and c, respectively.



150

151 **Figure 1:** Optimised structure of bulk monoclinic ZnP_2 (a) and its orbital decomposed densities of state
 152 computed with the GGA (PBE) (b) and HSE06 (c) functionals.

153 From the optimised bulk structure, the seven low Miller index surfaces were cut and relaxed by GGA
 154 in order to determine their optimal arrangements and stabilities. Shown in Figure 2 are the optimised
 155 structures of the most stable terminations of the seven surfaces, with the calculated unrelaxed and
 156 relaxed surface energies reported in Table 1. According to the calculated percentage relaxations, all
 157 surfaces undergo significant adjustments to reach the most stable configuration. The relaxed surfaces
 158 exhibit similar stabilities, as reflected in the small differences in their final surface energies. The (001)
 159 and (011) surfaces are predicted as the most stable with surface energies of 0.728 and 0.739 Jm^{-2} ,
 160 respectively. They are followed by the (100) and (101) surfaces whose calculated surface energies
 161 are 0.889 and 0.899 Jm^{-2} , respectively. With surface energies of 0.949, 1.028 and 0.985 Jm^{-2} , the
 162 (010), (110) and (111) surfaces are predicted to be less favourable, suggesting that the (110) surface
 163 is the least stable low Miller index surface of monoclinic ZnP_2 . Considering that all surfaces have
 164 similar stabilities and are hence likely to be expressed in the nanocrystal structure of ZnP_2 , the
 165 adsorption of 4-aminotiphenol (Figure 2, leftmost) was investigated on each surface to compare the
 166 nanocrystal morphologies before and after functionalisation.

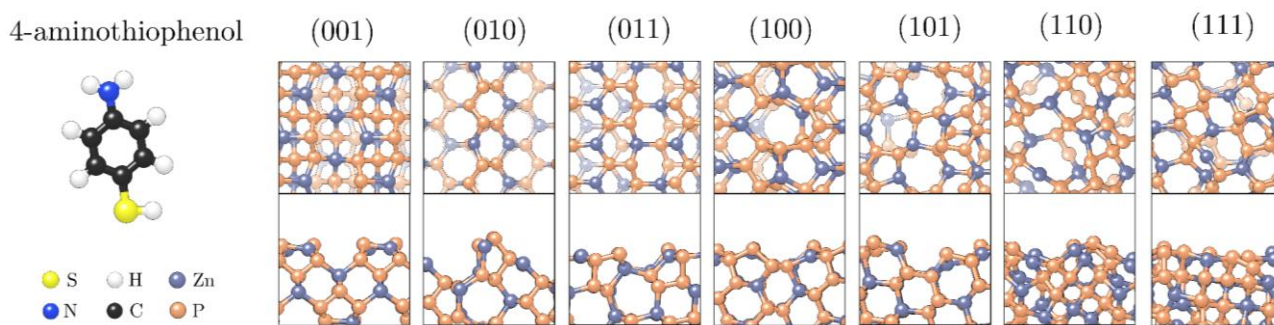


Figure 2: Optimised structures of 4-aminothiophenol (leftmost) and seven low Miller index surfaces of monoclinic ZnP_2 in top and side views. To create a depth perception, the top-most surface atoms have been represented with more vivid colors.

Table 1 Calculated surface energies (unrelaxed (γ_u) and relaxed (γ_r)), percentage relaxation, and work functions (ϕ) of most stable terminations of the seven low Miller index ZnP_2 surfaces.

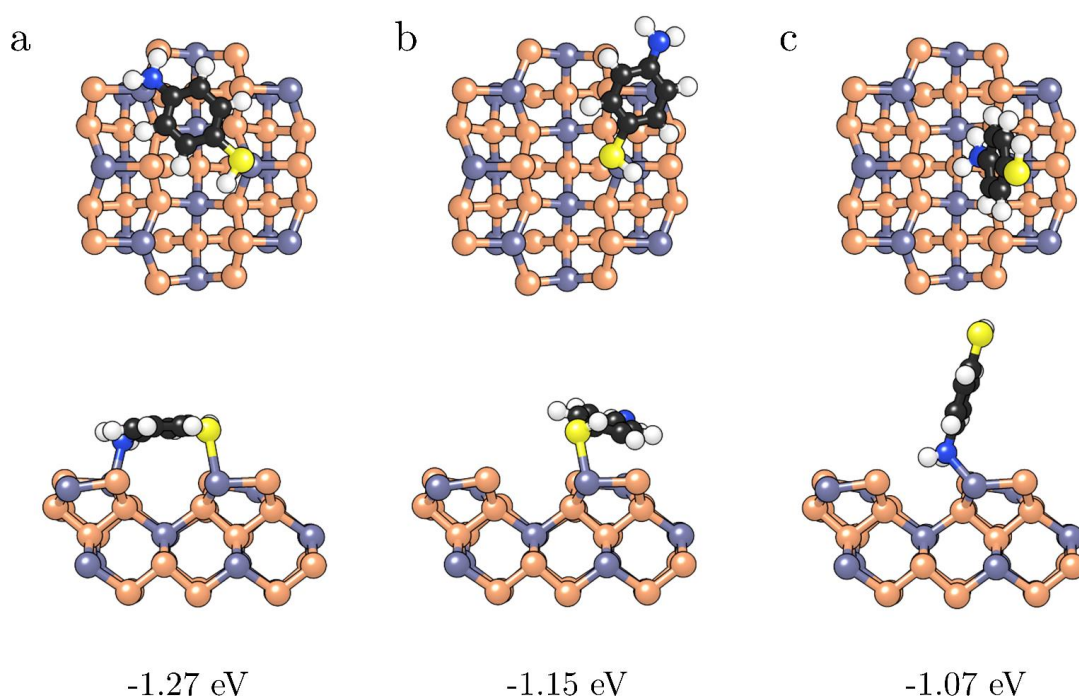
surface	termination	$\gamma_u / \text{Jm}^{-2}$	$\gamma_r / \text{Jm}^{-2}$	relaxation / %	ϕ / eV
(001)	-Zn-Zn-	1.006	0.728	-27.63	4.614
(010)	-P-P-	1.278	0.949	-25.74	4.561
(011)	-Zn-	1.001	0.739	-26.17	4.245
(100)	-P-Zn-	1.514	0.889	-41.28	4.429
(101)	-P-P-	1.165	0.899	-22.83	4.382
(110)	-P-	1.699	1.028	-39.49	4.402
(111)	-Zn-	1.364	0.985	-27.79	4.139

3.1 Single 4-aminothiophenol adsorption on ZnP_2 surfaces

To determine the preferred binding modes of the 4-aminothiophenol molecule on the various low Miller index surfaces of ZnP_2 , several different initial adsorption orientations were optimised without any symmetry constraints.

ZnP_2 (001) surface: At the ZnP_2 (001) surface, the 4-aminothiophenol binds in three possible modes as shown in Figure 3. The calculated adsorption energies, interatomic bond distances, and charge transfer are summarised in Table 2. The most stable binding mode is predicted to be a bidentate geometry (Figure 3a), wherein the 4-aminothiophenol molecule adsorbed parallel to the surface forming both N-Zn (2.26 Å) and S-Zn (2.64 Å) bond. This structure released an adsorption energy of -1.27 eV. Two vertical monodentate configurations, one binding through the sulphur of the thiol group (Figure 3b) and the other through the nitrogen of the amino group (Figure 3c), relaxed to differently slanted structures. Binding of 4-aminothiophenol through the thiol group resulted in repositioning horizontally towards the surface, with an angle between the N-S vector and a plane of

188 the surface (Figure S2 of SI) of 7.30° and a shorter S-Zn bond of 2.57 \AA compared to the same bond
 189 of the bidentate interaction mode. When interacting solely through the $-\text{NH}_2$ group, the 4-
 190 aminothiophenol molecule remains vertically oriented with an angle of 73.89° and a N-Zn distance
 191 of 2.17 \AA , which is also shorter than when adsorbed as a bidentate. However, the adsorption energies
 192 of both monodentate modes show weaker interaction and are calculated to be $E_{\text{ads}} = -1.15$ and $E_{\text{ads}} =$
 193 -1.07 eV for $-\text{SH}$ and $-\text{NH}_2$ interacting groups, respectively.



194
 195 **Figure 3:** Top (top panel) and side views (bottom panel) of the horizontal (a), $-\text{SH}$ vertical (b), and $-\text{NH}_2$
 196 vertical (c) adsorption modes of 4-aminothiophenol on the $\text{ZnP}_2(001)$ surface. For clarity, only the top most
 197 layers of the surface are shown. Calculated adsorption energies are listed below each structure. For colour
 198 legend refer to Figure 2.
 199

200 Regardless of the interaction mode of the adsorption, a molecule of 4-aminothiophenol gained
 201 electronic charge upon binding to the $\text{ZnP}_2(001)$ surface. The amount of charge gained through a
 202 single Zn-N bond in the $-\text{NH}_2$ vertical binding mode, namely $-0.06 e^-$, is only half of what is
 203 transferred in the horizontal or $-\text{SH}$ vertical mode (-0.14 and $-0.13 e^-$, respectively). However, the
 204 zinc atom directly involved in the adsorption of the molecule through the $-\text{NH}_2$ group loses more than
 205 double the charge contained within the adsorbate. The remaining negative charge is traced to the top-
 206 most neighbouring phosphorus atom ($-0.06 e^-$) connected to the interacting zinc. All structural and

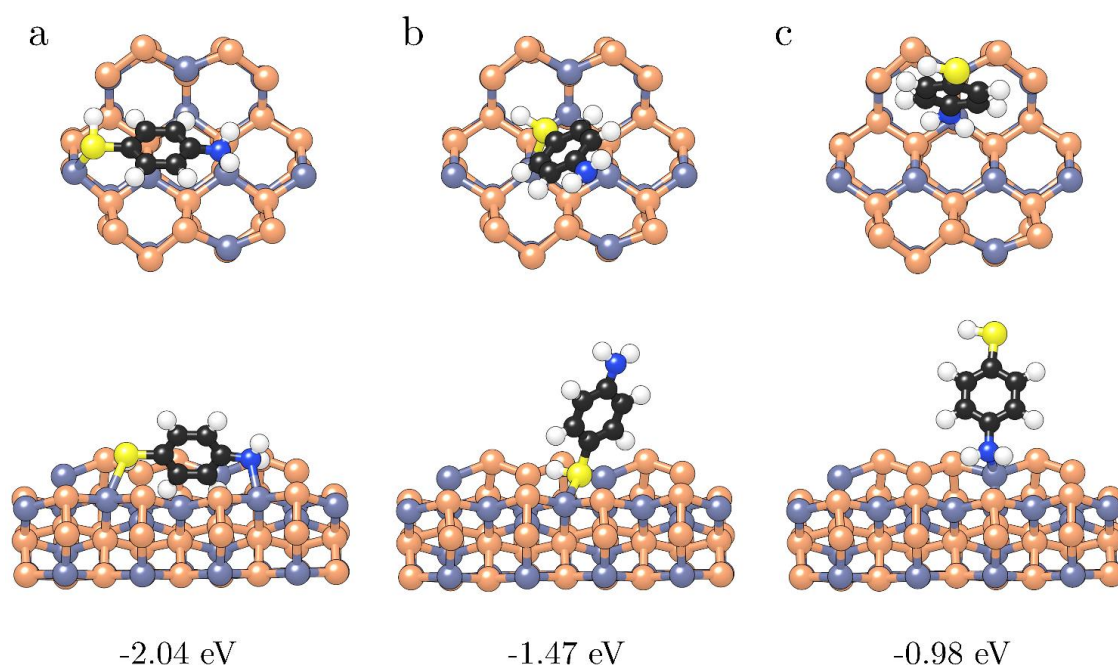
electronic parameters are listed in Table 2, while distribution of gained charge over atoms of the 4-aminothiophenol molecule can be found in Table S1, SI.

Table 2: Adsorption energies (E_{ads}), bond lengths (d), angles between the N-S vector and a surface plane (θ), and changes in the Bader charge (Δq) upon the adsorption of a single 4-aminothiophenol molecule on low Miller index surfaces of monoclinic ZnP_2 . Structural parameters are obtained through the relaxation with GGA (PBE) exchange functional, while HSE06 was used to calculate Bader charges. Negative changes in Bader charge represent gain in the electronic charge, while positive changes represent loss of electrons.

surface	binding mode	E_{ads} / eV	$d(\text{S-Zn})$ / Å	$d(\text{N-Zn})$ / Å	θ / °	$\Delta q(\text{adsorbate})$ / e^-	$\Delta q(\text{Zn}_\text{S})$ / e^-	$\Delta q(\text{Zn}_\text{N})$ / e^-
(001)	horizontal	-1.27	2.64	2.26		-0.14	0.02	0.06
	-SH vertical	-1.15	2.57		7.30	-0.13	0.01	
	-NH ₂ vertical	-1.07		2.17	73.89	-0.06		0.10
(010)	horizontal	-2.04	2.70	2.26		-0.06	0.01	0.08
	-SH vertical	-1.47	2.52		74.75	-0.07	0.02	
	-NH ₂ vertical	-0.98	2.18		69.82	-0.09		0.11
(011)	horizontal	-1.65	2.83	2.31		-0.09	0.02	0.06
	-SH vertical	-1.08	2.47		88.19	-0.11	0.03	
	-NH ₂ vertical	-0.92		2.18	28.30	-0.08		0.09
(100)	horizontal	-1.47	2.56	2.36		-0.11	0.00	0.04
	-SH vertical	-1.33	2.50		10.00	-0.12	0.02	
	-NH ₂ vertical	-0.78		2.19	80.05	-0.08		0.09
(101)	horizontal	-1.91	2.54	2.17		-0.16	0.03	0.09
	-SH vertical	-0.79	2.53		86.90	-0.08	0.03	
	-NH ₂ vertical	-1.17		2.17	70.51	-0.08		0.07
(110)	horizontal	-1.38	2.45	(4.31)		-0.06	0.02	
(111)	horizontal	-1.41	2.52	(3.89)		-0.13	0.06	0.01
	-SH vertical	-0.82	2.58		32.72	-0.11	0.07	
	-NH ₂ vertical	-1.23	2.15	2.15	60.48	-0.09		0.13

ZnP_2 (010) surface: The optimised binding modes of 4-aminothiophenol on the ZnP_2 (010) surface are presented in Figure 4. The channel-like structure of this surface, as indicated in Figure 2, can accommodate the 4-aminothiophenol molecule horizontally inside the channel, allowing for a relatively strong bidentate interaction (Figure 4a) with adsorption energy of $E_{\text{ads}} = -2.04$ eV. Established S-Zn and N-Zn bonds have lengths of 2.70 and 2.26 Å, respectively, and the whole molecule gained a charge of $-0.056 e^-$, mostly from the zinc atom interacting with the -NH₂ functional group. When adsorbed vertically through the -SH functional group (Figure 4b), sulphur binds to the Zn atom situated inside the surface channel (2.52 Å Zn-S bond), releasing an adsorption energy of -1.47 eV, with $-0.068 e^-$ of charge transferred to the adsorbate. When adsorbed via the amine nitrogen in a vertical monodentate geometry (Figure 4c), the bulkiness of the -NH₂ functional group prevents adsorption directly inside the channel and the N atom binds to an outer zinc atom instead (2.18 Å Zn-

226 N bond). Without further interactions between the rest of the molecule and the ZnP_2 (010) surface,
 227 the $-\text{NH}_2$ vertical binding mode released an adsorption energy of -0.98 eV, which is only half as
 228 favourable as the bidentate interaction. However, in this mode 4-aminothiophenol acquires the
 229 highest amount of charge from the surface, $-0.086 e^-$.

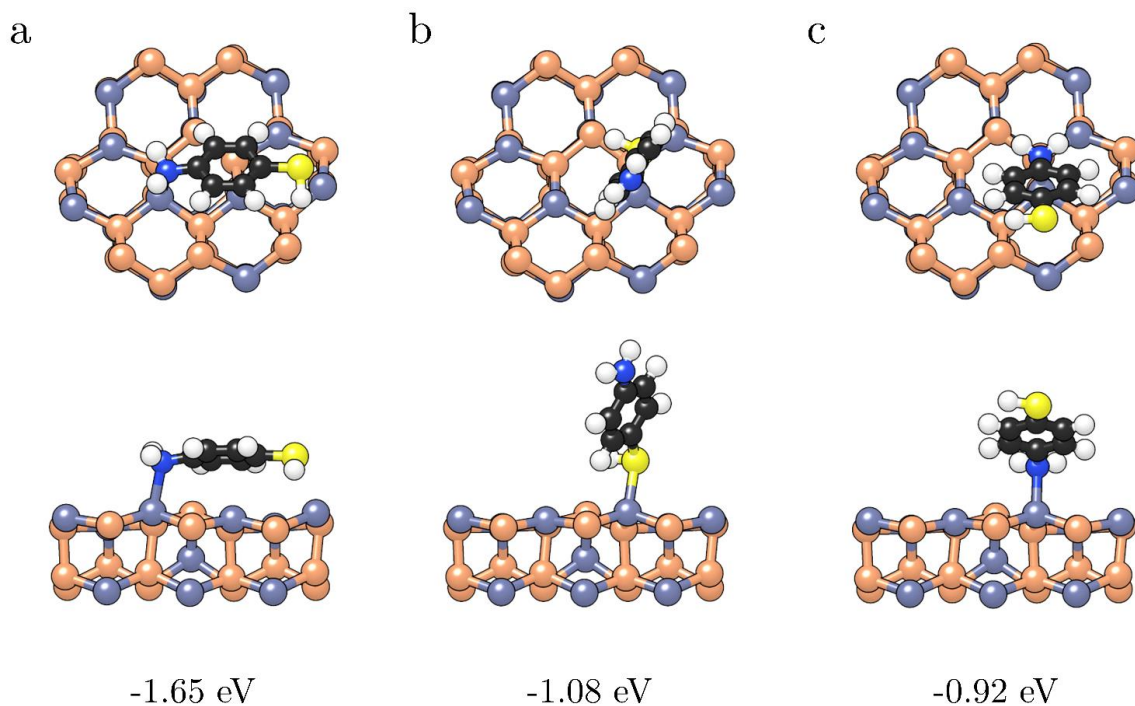


230

231 *Figure 4 Top (top panel) and side views (bottom panle) of horizontal (a), -SH vertical (b), and -NH₂ vertical*
 232 *(c) adsorption modes of 4-aminothiophenol on the ZnP_2 (010). For clarity, only the top most layers of the*
 233 *surface are shown. Calculated adsorption energies are listed below each structure. For colour legend refer to*
 234 *Figure 2.*

235 **ZnP_2 (011) surface:** The optimised structures of the three binding modes of 4-aminothiophenol on
 236 the ZnP_2 (011) surface and their adsorption energies are shown in Figure 5. Horizontal positioning of
 237 the 4-aminothiophenol resulted in an interaction very close to that of a bidentate biding mode (Figure
 238 5a), with a Zn-N bond of 2.31 \AA and a distance between the -SH sulphur atom and the closest surface
 239 zinc atom of 2.83 \AA . Despite the prolonged Zn-S distance, this type of interaction is characterised by
 240 an adsorption energy of $E_{\text{ads}} = -1.65$ eV, which is more favourable by 0.57 and 0.73 eV, respectively,
 241 than the -SH (Figure 5b) and -NH₂ (Figure 5c) monodentate vertical binding modes. When interacting
 242 only through the -SH functional group, the adsorbate binds almost perpendicularly to the surface with
 243 the Zn-S bond optimised at 2.47 \AA . In the -NH₂ binding mode (Zn-N bond of 2.17 \AA), the molecule
 244 is tilted much closer to the surface at an angle of 28.30° . Transfer of negative charge to the 4-

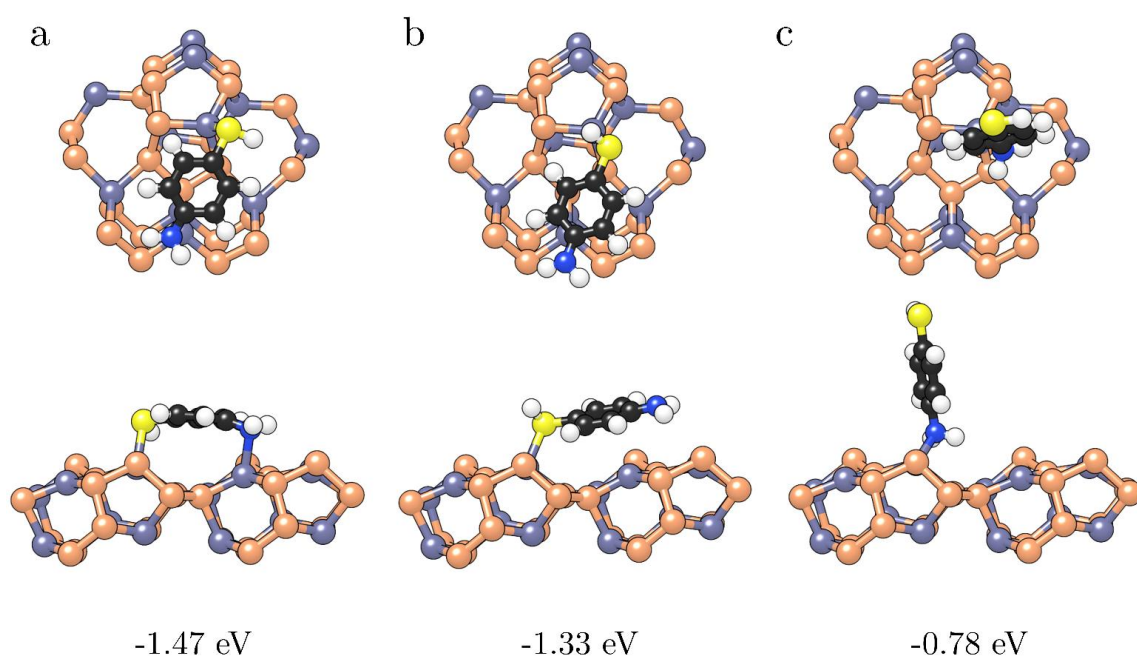
245 aminothiophenol is highest in the case of monodentate interaction through the sulphur atom, -0.11 e^-
 246 , and lowest for the opposite case of interacting solely through the amino group, -0.08 e^- .



247
 248 *Figure 5 Top (top panel) and side views (bottom panle) of horizontal (a), -SH vertical (b), and -NH₂ vertical*
 249 *(c) adsorption modes of 4-aminothiophenol on the ZnP₂ (011) surface. For clarity, only the top most layers of*
 250 *the surface are shown. Calculated adsorption energies are listed below each structure. For colour legend refer*
 251 *to Figure 2.*

252 **ZnP₂ (100) surface:** The three binding modes of 4-aminothiophenol on the ZnP₂ (100) surface,
 253 together with their corresponding adsorption energies, are presented in Figure 6. Zn-S and Zn-N
 254 bonds formed in the horizontal bidentate mode (Figure 6a) have lengths of 2.56 and 2.36 Å,
 255 respectively, and release an adsorption energy of -1.47 eV. For the -SH monodentate binding
 256 geometry (Figure 6b), the 4-aminothiophenol molecule is tilted towards the surface at an angle of
 257 10.00°, forming a Zn-S bond of 2.50 Å with the thiol hydrogen pointing away from the surface. This
 258 adsorption geometry is only 0.15 eV less favourable than the bidentate binding mode. In contrast,
 259 when the molecule is adsorbed in a monodentate mode through the amino group with a rather shorter
 260 2.17 Å Zn-N bond, it remains almost perfectly perpendicular (Figure 6c). However, this binding mode
 261 is also significantly less favourable with an adsorption energy of $E_{\text{ads}} = -0.78\text{ eV}$.

262 The charges gained through the adsorption process are similar for horizontal and -SH vertical
 263 interaction modes, -0.11 and -0.12 e^- , respectively. However, the amount of negative charge
 264 transferred from the zinc atom bound to sulphur is 0.02 e^- for the -SH monodentate binding mode,
 265 which is ten times that of the bidentate interaction, where the charge gained (0.002 e^-) is almost
 266 negligible, and the majority of the charge transfer comes from the Zn interacting with the -NH₂ group,
 267 0.04 e^- . Correspondingly, a significant change of 0.088 e^- is observed in the charge of the zinc atom
 268 interacting to form an amine monodentate, leading to a net charge gain of -0.08 e^- by the 4-
 269 aminothiophenol.



270

271 **Figure 6:** Top (top panel) and side views (bottom panle) of horizontal (a), -SH vertical (b), and -NH₂ vertical
 272 (c) adsorption modes of 4-aminothiophenol on the ZnP₂ (100) surface. For clarity, only the top most layers of
 273 the surface are shown. Calculated adsorption energies are listed below each structure. For colour legend refer
 274 to Figure 2.

275 **ZnP₂ (101) surface:** The structures and adsorption energies of the distinct adsorption modes of 4-
 276 aminothiophenol on the ZnP₂ (101) surface are presented in Figure 7. The (101) surface has a
 277 naturally stepped geometry which allows for efficient adsorption in the bidentate binding mode
 278 (Figure 7a), as seen in its adsorption energy of $E_{\text{ads}} = -1.91$ eV, the most favourable of the three
 279 adsorption modes. This is also reflected in the lengths of formed S-Zn and N-Zn bonds of 2.54 and
 280 2.17 Å. Compared to the bidentate binding, the monodentate -SH (Figure 7b) and -NH₂ (Figure 7c)

geometries released adsorption energies of -0.79 and -1.17 eV, respectively, with tilting angles of 86.90 and 70.51°, and Zn-S and Zn-N bonds converged at 2.53 and 2.17 Å, respectively. Charge gained in the most stable horizontal bidentate adsorption geometry on the (101) surface is -0.15 e⁻ with the zinc atom binding the nitrogen contributing three times more to the charge transfer compared to the zinc atom connected to the -SH sulphur. The 4-aminothiophenol gained charge of -0.08 e⁻ when adsorbed in both -SH and -NH₂ monodentate geometries.

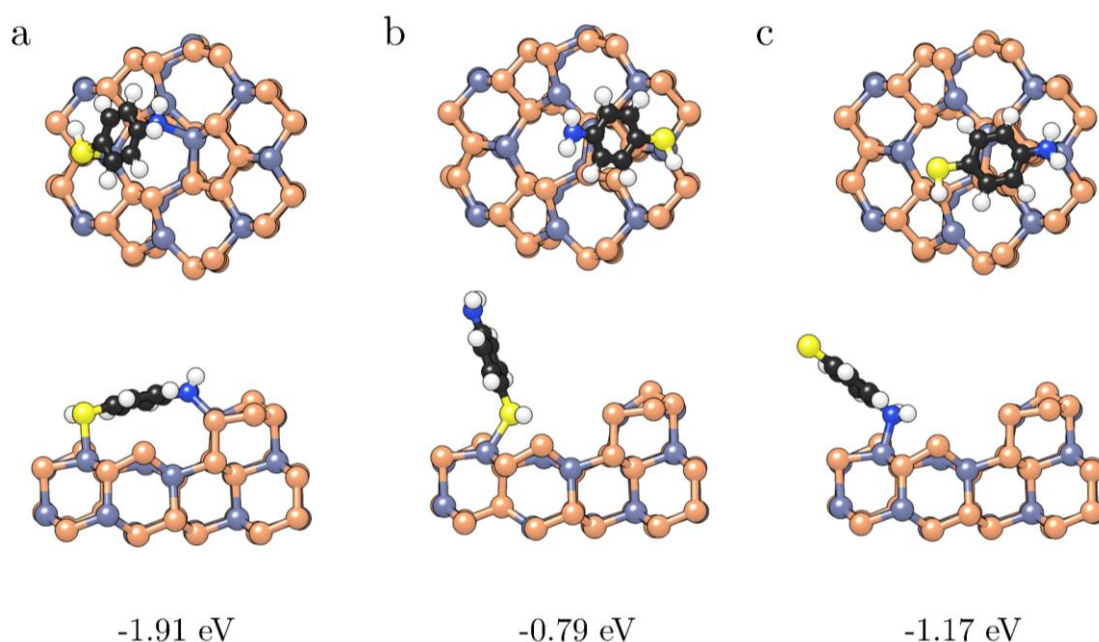
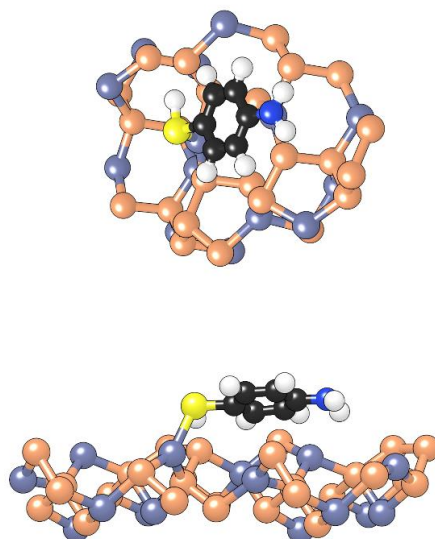


Figure 7: Top (top panel) and side views (bottom panel) of horizontal (a), -SH vertical (b), and -NH₂ vertical (c) adsorption modes of 4-aminothiophenol on the ZnP₂ (101) surface. For clarity, only the top most layers of the surface are shown. Calculated adsorption energies are listed below each structure. For colour legend refer to Figure 2.

ZnP₂ (110) surface: Only one binding mode was successfully optimised on the ZnP₂ (110) surface and its structure is depicted in Figure 8. The zigzag arrangement of surface zinc atoms and their interatomic distance hamper the 4-aminothiophenol molecule from forming a bidentate geometry with one S-Zn and one N-Zn bond. The molecule hence binds to the surface through a single S-Zn bond instead, with the adsorption energy of $E_{\text{ads}} = -1.38$ eV and a charge transfer of -0.06 e⁻.



-1.38 eV

297

298 **Figure 8:** Top (top panel) and side views (bottom panle) of horizontal adsorption mode of 4-aminothiophenol
 299 on the ZnP_2 (110) surface. For clarity, only the top most layers of the surface are shown. Calculated adsorption
 300 energy is also given. For colour legend refer to Figure 2.

301

302 **ZnP_2 (111) surface:** The 4-aminothiophenol binds to the (111) surface in two monodentate -SH
 303 geometries, horizontal (Figure 9a) and tilted (Figure 9b). The horizontal binding mode is
 304 characterised by the stronger adsorption energy of $E_{\text{ads}} = -1.41$ eV, compared to $E_{\text{ads}} = -0.82$ eV for
 305 the tilted mode. A true bidentate mode could not be established, even when positioning the molecule
 306 parallel to the surface, due to the inconvenient arrangement of the topmost Zn atoms. In the horizontal
 307 -SH geometry, the Zn-S bond has a length of 2.52 Å, while the distance between the nitrogen and the
 308 nearest zinc surface atom is 3.89 Å. The Zn-S bond distance in the tilted -SH monodentate geometry
 309 is predicted at 2.58 Å with a tilt angle of 32.72°. The monodentate -NH₂ vertical binding mode (Figure
 310 9c), established through the 2.15 Å Zn-N bond, is only 0.18 eV less favourable than when 4-
 311 aminothiophenol is adsorbed in the horizontal -SH geometry, and it releases an adsorption energy of
 312 -1.23 eV. Overall, the 4-amiothiophenol gained a charge of $-0.13 e^-$ and $-0.11 e^-$ when adsorbed in the
 313 horizontal and tilted -SH geometries, respectively, and $-0.09 e^-$ when binding through the -NH₂ group.

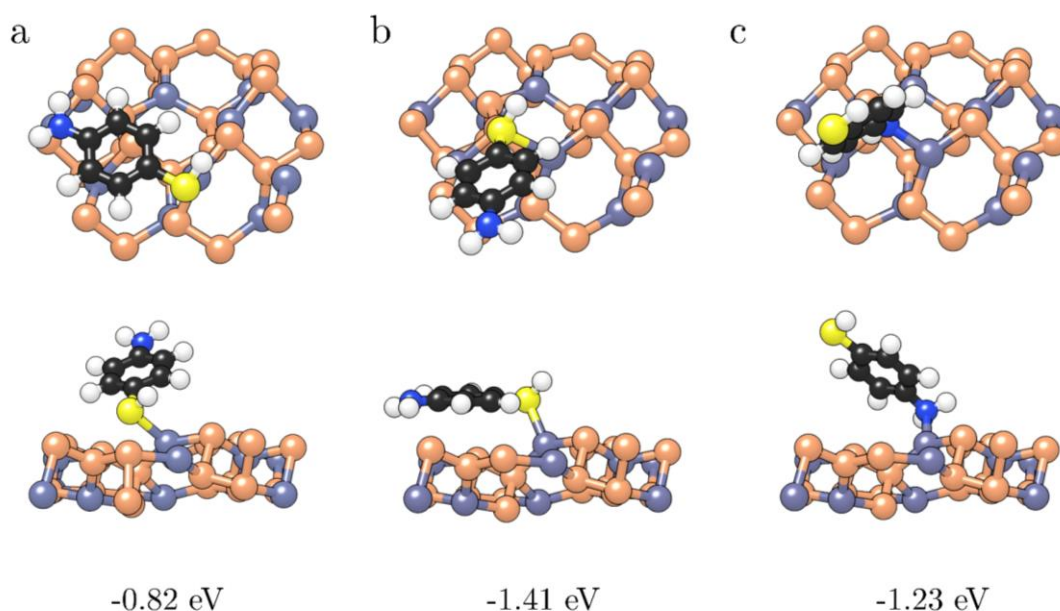


Figure 9 Top (top panel) and side views (bottom panel) of -SH horizontal (a), -SH vertical (b), and -NH₂ vertical (c) adsorption modes of 4-aminothiophenol on the ZnP₂ (111) surface. For clarity, only the top most layers of the surface are shown. Calculated adsorption energies are listed below each structure. For colour legend refer to Figure 2.

3.2 Functionalisation at full coverage of the ZnP₂ surfaces

Full coverage functionalisation of ZnP₂ surfaces was investigated by considering multiple 4-aminothiophenol molecules adsorbed in the horizontal binding mode, which was shown to be the most favourable type of interaction on all surfaces. Each surface was functionalised with a full monolayer of 4-aminothiophenol, where monolayer coverage had been established when no combination of uncoordinated zinc atoms remained available to accommodate another adsorbate molecule in a flat bidentate binding mode. The calculated adsorption energies at full coverage are reported in **Error! Reference source not found.**, with the strength of adsorption predicted to follow the trend (010) > (101) > (011) > (110) > (100) > (111) > (001). Using the full coverage energetics, the surface energies and work functions of the functionalised surfaces were also calculated and are listed in **Error! Reference source not found.**

To determine the effect of the functionalisation on the electronic properties of ZnP₂ surfaces, changes in the decomposed densities of state (DOS) and band gap energies upon adsorption have been examined. Orbital decomposed DOS of the (001) surface before and after full monolayer

functionalisation are shown in Figure 10. Similar to the densities of state of the bulk material, the top of the valence band of the (001) surface is dominated by p states of phosphorus and p and d states of zinc, while the bottom of the conduction band is composed mainly of the s states of P atoms and p states of Zn atoms.

Table 3: Adsorption energy per molecule for horizontal binding mode (\bar{E}_{ads}), surface energy (γ_f), percentage stabilisation, and work function (ϕ_f) for seven low Miller index surfaces of monoclinic ZnP_2 upon a full monolayer functionalisation with 4-aminothiophenol.

surface	\bar{E}_{ads} / eV	γ_f / Jm $^{-2}$	stabilisation / %	ϕ_f / eV
(001)	-1.13	0.678	-6.8	4.207
(010)	-2.10	0.881	-7.1	3.816
(011)	-1.65	0.722	-2.3	3.927
(100)	-1.44	0.853	-4.0	4.924
(101)	-1.90	0.770	-14.4	4.269
(110)	-1.51	0.898	-12.6	4.837
(111)	-1.40	0.883	-10.3	3.057

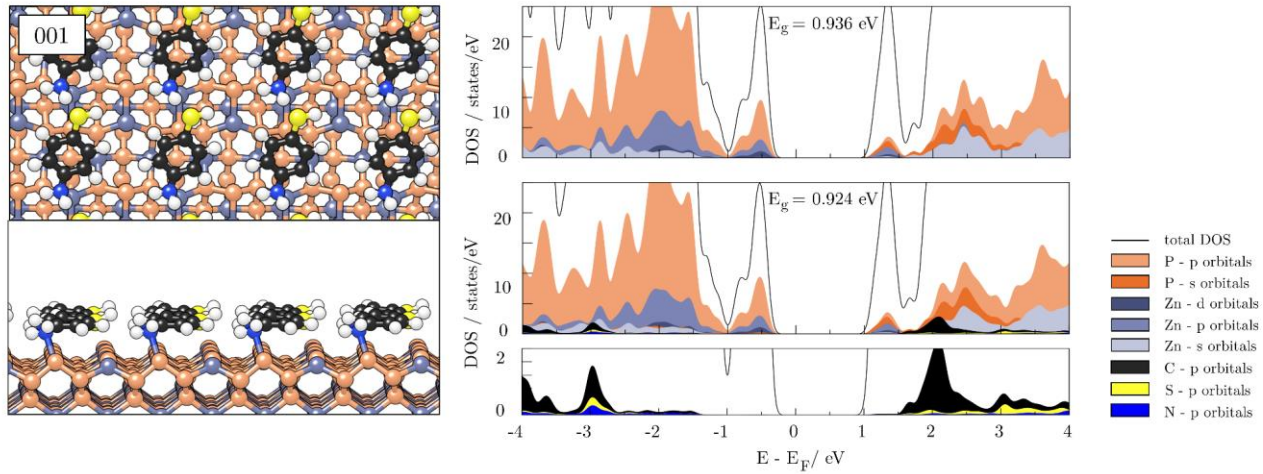


Figure 10: Left: (001) surface of monoclinic ZnP_2 functionalised with a full monolayer of 4-aminothiophenol in horizontal binding mode in top view (top panel) and side view (bottom panel). Right: orbital decomposed DOS of bare 001 surface of monoclinic ZnP_2 with accompanying band gap energy, E_g (top panel); orbital decomposed DOS of functionalised 001 surface of monoclinic ZnP_2 presented on the left with accompanying band gap energy, E_g (middle panel); enlarged DOS of the atoms of 4-aminothiophenol after monolayer functionalisation of the (001) surface of monoclinic ZnP_2 (bottom panel).

The adsorption of 4-aminothiophenol is shown to be characterised by the Zn-S and Zn-N bonding states localised at around -3.0 eV below the Fermi level. The electronic band gaps of the naked and functionalised (001) surfaces are calculated at 0.936 and 0.924 eV, respectively, indicating that the functionalisation with 4-aminothiophenol has only a negligible effect on the electronic structure. The composition of the valence and conduction band edges of the functionalised surface remained

unchanged and no additional states were introduced in the close proximity or within the energy gap. Corresponding figures of the orbital decomposed DOS for the remaining six surfaces can be found in the supplementary information, Figures S3 and S4, where no significant effects of the adsorbate monolayers on the electronic properties was observed, and band gap energies of the functionalised surfaces remained within 0.005-0.100 eV of their naked counterparts.

3.3 Morphology of ZnP_2 nanocrystal before and after 4-aminothiophenol functionalisation

Modifications in the shape of the ZnP_2 nanocrystal after 4-aminothiophenol functionalisation were ascertained through the Wulff construction [56], based on the surface energies of the clean and monolayer-protected surfaces, Figure 11. Under thermodynamic conditions, the equilibrium morphology of a crystal possesses minimal total surface free energy for a given volume based on the Gibbs formulation. Functionalisation is demonstrated to have a stabilising effect on all ZnP_2 surfaces, as reflected in the lower surface energies, with the extent of the stabilisation varying from 2.3% for the (011) surface up to 14.4% for the (101) surface.

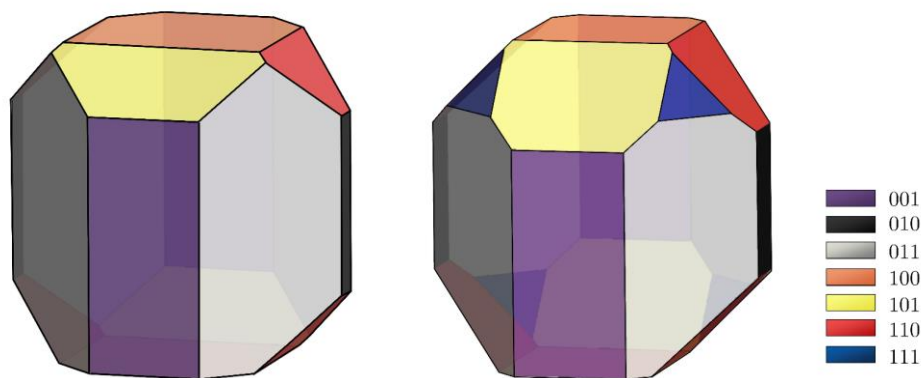


Figure 11: Wulff morphology of bare (left) and 4-aminothiophenol functionalised (right) monoclinic ZnP_2 crystal.

Adsorption induced changes in the surface energies are also visible in the final morphology of the ZnP_2 nanoparticle. Prior to the functionalisation, the equilibrium shape of the bare crystal contains six out of the seven surfaces studied, with the (111) surface not appearing in the morphology. The (001) and (011) surfaces account for the greatest share of the surface area, in agreement with their

376 favourable surface stabilities among the seven surfaces investigated. Differences in the adsorption
 377 strength of 4-aminothiophenol on the seven surfaces influences their stabilities and, hence, their
 378 expression in the morphology of the functionalised nanoparticle. The weaker adsorption on the (001)
 379 and (100) surfaces compared to the others resulted in a reduction of their surface shares, whereas the
 380 (101) and (110) surfaces experience an increase in their share of the available surface area, resulting
 381 from the higher stabilisation from the 4-aminothiophenol adsorption. Owing to the stabilisation of the
 382 (111) surface after functionalisation, it now appears in the final equilibrium morphology.

383 ***3.4 Thermodynamic stability of functionalised ZnP₂ surfaces***

384 Calculated adsorption energies are good descriptors of the strength of interaction between the 4-
 385 aminothiophenol and various ZnP₂ surfaces, but they do not give any indication of the effects of
 386 experimental conditions and chemical potential of the adsorbate on the stability of functionalised
 387 surfaces. Stabilities of the surfaces upon functionalisation were therefore estimated and compared
 388 based on the change in the free Gibbs surface energy with respect to the conditions of temperature
 389 and pressure ($\Delta\gamma_f(T, p)$) as:

$$390 \quad \Delta\gamma_f(T, p) = \frac{E_{\text{funct surface}}^{\text{DFT}} - E_{\text{surface}}^{\text{DFT}}}{2A} - \frac{m}{2A} \Delta\mu_a(T, p)$$

391 where $\Delta\mu_a(T, p)$ is the variation of the chemical potential of 4-aminothiophenol to account for
 392 experimental conditions. Assuming adsorbate-rich conditions (corresponding to $\Delta\mu_a(T, p) = 0$, when
 393 the chemical potential of 4-aminothiophenol equals the total energy of the molecule in the gas phase),
 394 calculated changes in the free Gibbs surface energies are represented in Figure 12a. All surfaces
 395 undergo full monolayer functionalisation under these conditions, with the most significant change in
 396 the Gibbs surface free energy noted for the (101) and (110) surfaces.

397 To fully capture the behaviour of the surfaces under various conditions of adsorbate concentrations,
 398 surface phase diagrams have been suggested based on the changes in the free Gibbs surface energies
 399 with chemical potential of the 4-aminothiophenol. The phase diagrams were plotted for the two most

stable functionalised surfaces, (101) and (110), as shown in Figure 12b and c, respectively. The two surfaces show similar trends, with monolayer coverage favoured up to the chemical potential of -1.50 and -2.00 eV. A transition to the lower coverage of 0.50 ML occurs at $\Delta\mu_a = -1.89$ eV for the (101) surface, while the appearance of the bare surface should occur shortly after, at $\Delta\mu_a = -1.91$ eV, which is shown in the enlarged area of the plot.

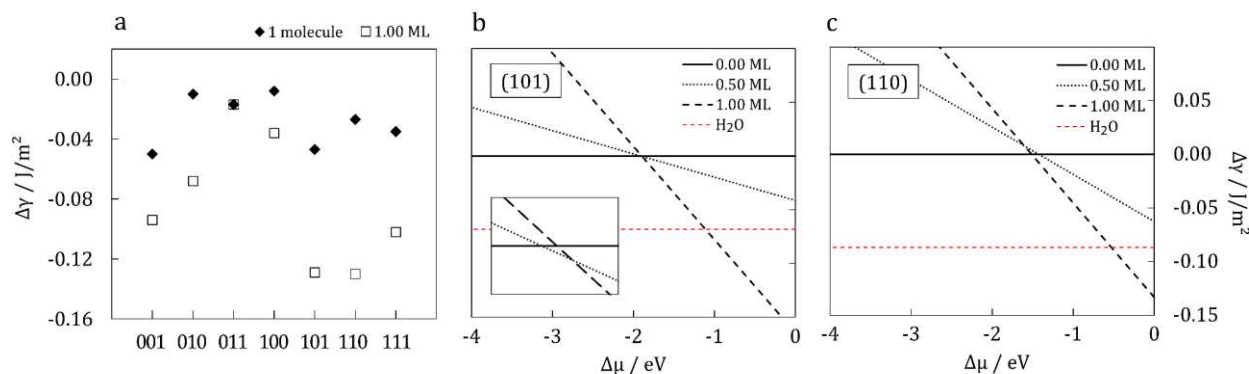


Figure 12: (a) Surface free-energy variations for seven low Miller index surfaces of monoclinic ZnP_2 functionalised with a single molecule of 4-aminothiophenol and a full monolayer under adsorbate-rich conditions ($\mu_a = 0.00$ eV). Surface phase diagrams of the (101) surface (b) with the enlarged area of the intersection, and of the (110) surface (c). y-axis is the same for the two surface phase diagrams.

Conversely, an immediate transition to a bare surface is observed at $\Delta\mu_a = -1.51$ eV for the (110) surface without the appearance of the surface with half a monolayer of 4-aminothiophenol. The crossovers between high and low coverages (or, in most cases, bare surfaces) of all seven surfaces occur in the range of -1.50 to -2.00 eV; the only exception is the (001) surface where the intersection happens at $\Delta\mu_a = -1.12$ eV. If standard conditions are considered ($p = 1$ atm, $T = 300$ K), those transitions can be assigned to environments with extremely low adsorbate concentrations, $p \sim 10^{-15} - 10^{-17}$ atm for the six surfaces or $p \sim 10^{-10}$ atm for the (001) surface (note: ultra-high vacuum chambers operate at about 10^{-12} atm).

The effect of humidity was introduced by adding the variation of the surface energies with adsorbed water. Previous study has shown spontaneous interaction of water molecules with all ZnP_2 surfaces⁶¹, and the calculated surface energies after full water monolayer adsorption under $p = 1$ atm

($p_{\text{H}_2\text{O}}=0.035$ atm) and $T = 300$ K have also been provided in the phase diagrams (red dashed lines). It has to be noted that only the lowest chemical potential of water needed for the formation of the full monolayer was considered and the true surface energy could hence be even lower. When the transition from 1.00 ML of 4-aminothiophenol to 1.00 ML of water will take place depends on the surface, but it always occurs at much higher chemical potentials of 4-aminothiophenol compared to the transitions towards the non-functionalised surfaces. For the surfaces whose phase diagrams are depicted in Figure 12, transitions are observed at $\Delta\mu_a = -0.82$ eV for the (101) surface and $\Delta\mu_a = -1.08$ eV for the (110) surface, which corresponds to the partial pressures of $p \sim 10^{-6} - 10^{-9}$ atm. Further estimation of the stability of the functionalised surfaces under more realistic aqueous media is not plausible solely from DFT calculations, but the provided results are suggestive of the possible protection of ZnP_2 nanoparticles under humid conditions by 4-aminothiophenol functionalisation.

4 CONCLUSIONS

Density functional theory calculations have been carried out to provide detailed atomic-level insights into the effects of the functionalisation by 4-aminothiophenol on the structural and electronic properties of monoclinic ZnP_2 nanoparticles. The strongest interaction between the 4-aminothiophenol and all the ZnP_2 surfaces was established when the molecule was located close to the surface in a horizontal position. When the arrangement of the topmost zinc atoms was favourable, this provided an adsorption site which allowed both the $-\text{NH}_2$ and $-\text{SH}$ functional groups to form covalent bonds with the surface. Analysis of electronic structures of the surfaces functionalised by a full monolayer revealed minimal changes in the band gap energies, without new states in the band gap region, although changes in the morphology of the nanocrystals were much more prominent. Thermodynamically, the surfaces were found to be significantly stabilised after functionalisation, with full monolayer coverages favoured under adsorbate-rich conditions and expected to persist even in the presence of water, although it remains to see if in aqueous media this improvement in the stability remains sufficient to overcome the formation of a water monolayer. However, minimal

447 modifications of the optically favourable band gap energies and adsorption strengths comparable to
448 those observed at Zn_3P_2 surfaces indicate that 4-aminothiophenol is a promising candidate for the
449 functionalisation of monoclinic ZnP_2 nanoparticles and thin films for photovoltaic applications.

450 **ASSOCIATED CONTENT**

451 *S Supporting Information

452 The Supporting Information contains adsorption structures, Bader charge analyses, and the orbital
453 decomposed DOS of the monoclinic ZnP_2 surfaces with and without a full monolayer
454 functionalisation with 4-aminothiophenol.

455

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463 Notes

464 The authors declare no competing financial interest.

465

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