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#### **Supplementary Information**

# Dramatic impact of the TiO<sub>2</sub> polymorph on the electrical properties of 'stoichiometric' Na<sub>0.5</sub>Bi<sub>0.5</sub>TiO<sub>3</sub> ceramics prepared by solid-state reaction

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## 1. Comparison of the electrical conductivity of NBT ceramics sintered with and without binder

Here two methods were used to sinter dense NBT ceramics, including 1) uni-axial cold pressing followed by isostatic pressing at 200 MPa (denoted as "cip") and 2) uni-axial pressing with a 5 wt.% water solution of polyvinyl alcohol (PVA) as binder without isostatic pressing (denoted as "binder"). Fig.S1 compares the bulk conductivity ( $\sigma_b$ ) of Bi-deficient NBT (NB<sub>0.49</sub>T), nominally stoichiometric NBT (NB<sub>0.50</sub>T) and Bi-excess NBT (NB<sub>0.51</sub>T) prepared by the above two methods. These ceramics were all prepared using rutile TiO<sub>2</sub>. Fig.S1 shows use of binder causes a slight decrease of  $\sigma_b$  of NB<sub>0.50</sub>T without changing the  $\sigma_b$ -1000/T relationship. The two sintering methods do not cause the significantly different electrical conductivity of NB<sub>0.50</sub>T prepared using rutile and anatase TiO<sub>2</sub>, and therefore do not influence the major conclusion of this work.

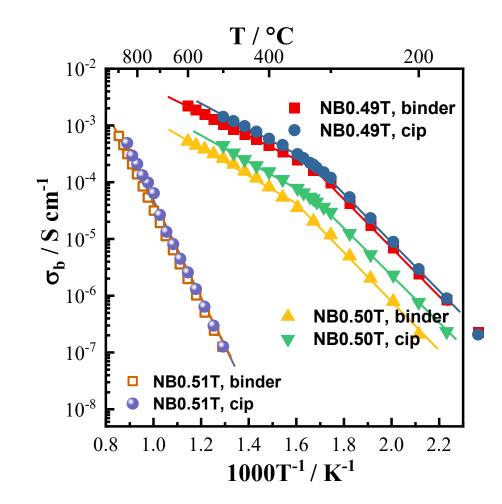


Figure S1. Arrhenius plots for σ<sub>b</sub> of Bi-deficient NBT (NB<sub>0.49</sub>T), nominally stoichiometric NBT (NB<sub>0.50</sub>T) and Bi-excess NBT (NB<sub>0.51</sub>T). All ceramics were prepared by rutile TiO<sub>2</sub>. "cip" represents uni-axial cold pressing followed by isostatic pressing. "binder" represents uni-axial cold pressing with PVA as a binder without isostatic pressing.

## 2. XRD patterns of NBT ceramics prepared from different reagents

Fig.S2 shows the XRD patterns of NBT ceramics prepared by different reagents. The numbers after NBT represent different combinations of Na<sub>2</sub>CO<sub>3</sub>, Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> reagents from Table 1 in the main text. All NBT ceramics are phase pure with a rhombohedral structure.

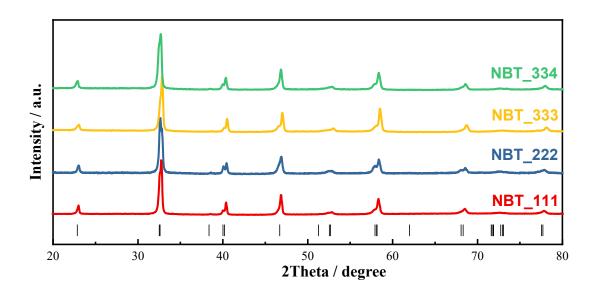


Figure S2. XRD patterns of sintered NBT ceramics prepared by different reagents. The vertical lines below the patterns represent the peak positions for NBT with a rhombohedral structure.

## 3. Full synchrotron XRD patterns in the $2\theta$ range between $15-85^{\circ}$

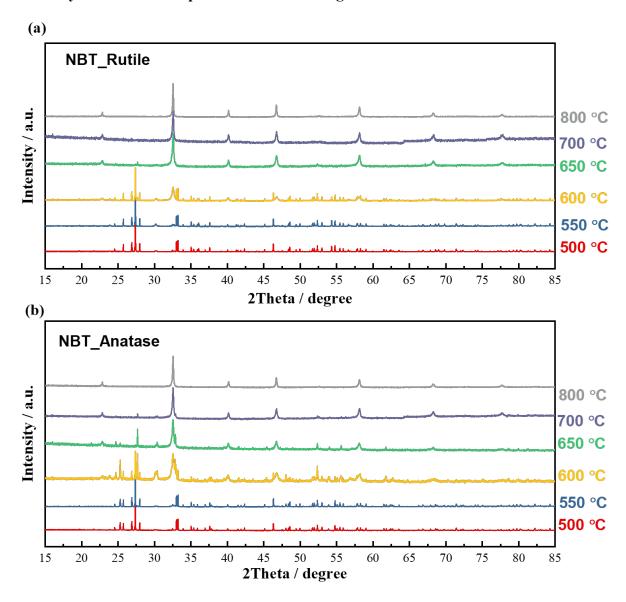


Figure S3. Evolution of the XRD patterns with calcination temperature in the  $2\theta$  range between 15-85. (a) Mixture with rutile  $TiO_2$  and (b) with anatase  $TiO_2$ .

#### 4. Particle size distribution

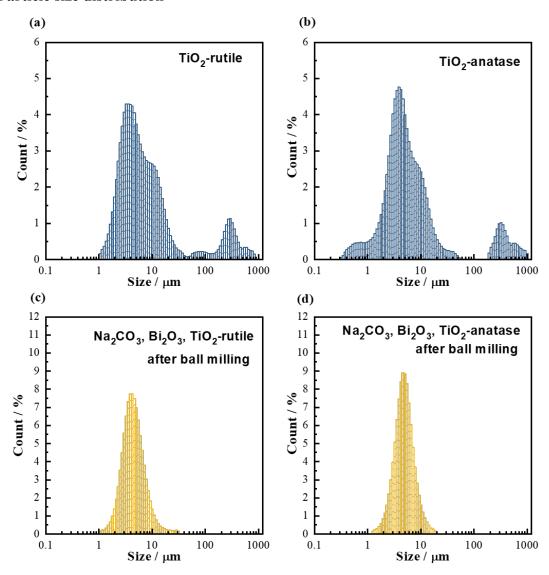


Figure S4. Particle size distribution of (a) raw rutile TiO<sub>2</sub>, (b) raw anatase TiO<sub>2</sub>, (c) ball-milled and sieved mixture of pre-dried Na<sub>2</sub>CO<sub>3</sub>, Bi<sub>2</sub>O<sub>3</sub> and rutile TiO<sub>2</sub>, and (d) ball-milled and sieved mixture of Na<sub>2</sub>CO<sub>3</sub>, Bi<sub>2</sub>O<sub>3</sub> and anatase TiO<sub>2</sub>.

## 5. Evolution of Bi<sub>2</sub>O<sub>3</sub>, Bi<sub>12</sub>TiO<sub>20</sub> and NBT fractions during solid-state reaction

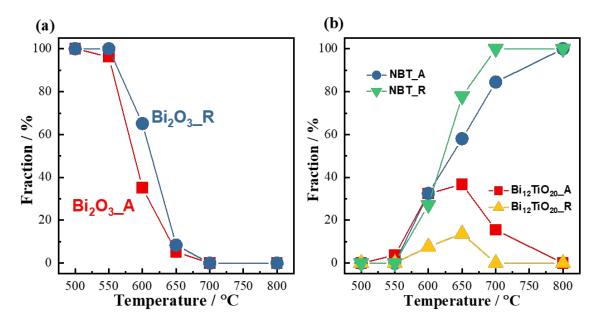


Figure S5. Evolution of the fraction parameter of Bi-containing phases during solid-state reaction when rutile and anatase TiO<sub>2</sub> are used as reagents. (a) Bi<sub>2</sub>O<sub>3</sub>, (b) Bi<sub>12</sub>TiO<sub>12</sub> and NBT.

#### 6. Impedance spectroscopy of NBT R<sub>0.5</sub>A<sub>0.5</sub>

Impedance data for NBT\_R<sub>0.5</sub>A<sub>0.5</sub> are presented in Nyquist (Fig.S6a) and Bode (M"-logf, Fig.S6b) plots. Z\* plots for NBT\_R<sub>0.5</sub>A<sub>0.5</sub> at 500 ° C show two poorly resolved arcs and a low-frequency spike corresponding to an electrode effect. The electrode spike is characteristic of ionic conduction behavior. With increasing temperature, the two arcs gradually merge into one suggesting a different activation energy for the two responses. M"-logf plots at lower temperatures, e.g., 300 ° C, show a broad peak, suggesting an inhomogeneous electrical microstructure for NBT\_R<sub>0.5</sub>A<sub>0.5</sub>. The M"-logf peak height increases with increasing temperature suggesting a reduced relative permittivity with increasing temperature between 300 – 700 ° C. The inhomogeneous electrical microstructure agrees with the random distribution of large and small grains, as shown in Fig.6c.

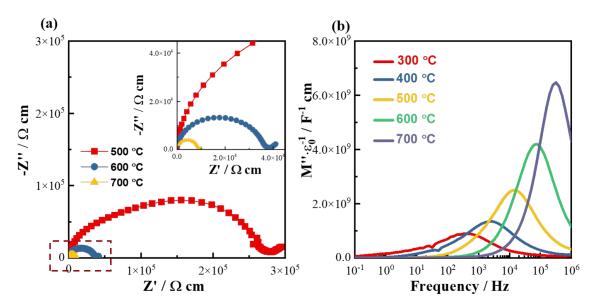


Figure S6. Impedance data for NBT\_R<sub>0.5</sub>A<sub>0.5</sub>. (a)  $Z^*$  plots at 500, 600 and 700 ° C and (b) M"-logf plots at 300-700 ° C at increments of 100 ° C. The inset figure in (a) is an expanded view of the rectangular region.