

This is a repository copy of Suitability of alkali activated slag/fly ash (AA-GGBS/FA) concretes for chloride environments: Characterisation based on mix design and compliance testing.

White Rose Research Online URL for this paper: http://eprints.whiterose.ac.uk/148814/

Version: Accepted Version

Article:

Bondar, D, Basheer, M orcid.org/0000-0002-0835-8029 and Nanukuttan, S (2019) Suitability of alkali activated slag/fly ash (AA-GGBS/FA) concretes for chloride environments: Characterisation based on mix design and compliance testing. Construction and Building Materials, 216. pp. 612-621. ISSN 0950-0618

https://doi.org/10.1016/j.conbuildmat.2019.05.043

(c) 2019, Elsevier Ltd. This manuscript version is made available under the CC-BY-NC-ND 4.0 license https://creativecommons.org/licenses/by-nc-nd/4.0/

Reuse

This article is distributed under the terms of the Creative Commons Attribution-NonCommercial-NoDerivs (CC BY-NC-ND) licence. This licence only allows you to download this work and share it with others as long as you credit the authors, but you can't change the article in any way or use it commercially. More information and the full terms of the licence here: https://creativecommons.org/licenses/

Takedown

If you consider content in White Rose Research Online to be in breach of UK law, please notify us by emailing eprints@whiterose.ac.uk including the URL of the record and the reason for the withdrawal request.



eprints@whiterose.ac.uk https://eprints.whiterose.ac.uk/

Suitability of alkali activated slag/fly ash (AA-GGBS/FA) concretes for chloride environments: Characterisation based on mix design and compliance testing Dali Bondar¹, Marios Soutsos¹, Muhammed Basheer², Sreejith Nanukuttan¹ ¹School of Natural and Built Environment, Queen's University Belfast, BT9 5AG, UK ²School of Civil Engineering, University of Leeds, Leeds, LS2 9 JT, UK Email: D.Bondar@qub.ac.uk, m.soutsos@qub.ac.uk, p.a.m.basheer@leeds.ac.uk,

s.nanukuttan@qub.ac.uk

8 Abstract

7

9 Alkali Activated Materials (AAM's) exhibit superior performance in terms of abrasion and acid resistance and fire protection. Ground granulated blast furnace slag (GGBS) and fly ash are the two 10 most common industrial by-products used as precursors in alkali activated material. Alkali-activated 11 systems based on fly ash often require elevated temperature curing, while GGBS alone can lead to 12 flash setting, low workability and high degree of drying shrinkage. This article presents data from a 13 14 range of GGBS/fly ash blends designed to avoid elevated temperature curing, with desirable fresh 15 properties. Further the suitability of such blended concrete for their use in chloride environments is outlined with the help of diffusivity testing and binding assessment. The results show that while 16 attractive engineering properties can be achieved, the chloride ingress in the AA-GGBS/FA concretes 17 studied here with $\geq 20\%$ fly ash is moderate to high and this will make them unsuitable for protecting 18 steel reinforced in concrete exposed to chloride environments. 19

20 <u>Keywords:</u> alkali activated slag/fly ash (AA-GGBS/FA) concretes, workability, compressive
 21 strength, bulk resistivity, chloride diffusion coefficient, chloride binding

22 **1. Introduction**

Alkali Activated Materials (AAM's) have been under consideration as an alternative binder system 23 24 since 1895 [1]. The main limitations facing the commercial adoption of these binders are the 25 challenges in controlling engineering properties when using intrinsically variable waste-derived 26 precursors, and the lack of field performance data, especially in extreme environments. GGBS and fly ash are the major by-products used as precursors for making these types of binders. Previous 27 28 investigations show that alkali-activated high calcium systems such as AAS concretes usually set 29 rapidly, can have low workability, and sometimes also a high degree of drying shrinkage compared to Portland cement (PC) based systems, and all of these are known to affect durability properties [2-6]. 30 Low calcium alkali activated systems, on the other hand, can exhibit high workability and a lower 31 32 shrinkage than PC systems [7], but need elevated temperatures for curing and hardening. Chi and 33 Huang [8] studied the behaviour of the binder products and properties of AA-GGBS/FA blended 34 mortars and concluded that better properties, compared to PC, were obtained in terms of compressive 35 strength, flexural strength and water absorption. Abdalqader et al. [9] found that the highest strength 36 of AA-GGBS/FA obtained was with a paste mix consisting of GGBS: fly ash in a 3:1 ratio. Gao et al. 37 reported that concrete with a higher GGBS/fly ash ratio exhibit lower porosity [9-10]. However, despite having been aware of this materials potential for over a century, there is still relatively little 38 39 known about the durability of AAM's, especially the resistance to chloride ingress and the binding 40 behaviour. It was reported by Roy et al. that the advantageous properties of AA-GGBS/FA are their high early strength, rapid hardening and high ultimate strength, low heat of hydration, and resistance 41 to effects resulting from chemical attack. Diffusion coefficients of 50×10^{-15} and 5×10^{-15} m²/s were 42 43 reported for the blended OPC and GGBS paste activated with NaOH, with 0 and 100% GGBS 44 replacement [11].

- In this investigation, blended binder systems were used as they provide control of properties such as setting time, workability, shrinkage, strength and durability compared to those based on 100% fly ash or GGBS. An experimental programme was developed to study the workability, strength development and resistance to chloride ingress for a range of AA-GGBS/FA concrete mixes that do not require high temperature curing. Fresh properties, compressive strength at different ages, bulk resistivity, long
- 50 term chloride diffusion and binding behaviour were determined and reported for various AA-
- 51 GGBS/FA concretes.

52 2. Experimental details

53 2.1 Materials

The primary raw materials used in this study were GGBS and low-CaO fly ash, which were provided by ECOCEM – Ireland and Power Minerals Ltd, UK, respectively. The chemical and physical properties are presented in Tables 1 and 2. GGBS and fly ash were blended in 80/20, 60/40, 40/60 and 20/80 mass proportions, to be activated and used as binder.

Sodium hydroxide (NaOH) powder or pellets were dissolved in water to produce the alkaline solutions. The chemical composition of the sodium silicate solution was 15.5% sodium oxide (Na₂O), 30.5% silicon oxide (SiO₂) and 54% water. In this study, the percentage Na₂O% in the paste (Na₂O%) was selected to be 6 and 8% by powder (GGBS and fly ash) weight, and the silicate modulus (molar

ratio SiO_2/Na_2O) was varied from 0.45, 1.0 and 1.25, by blending sodium silicate and sodium

63 hydroxide.

64 The aggregates used in this study were crushed basalt from local sources in Northern Ireland and 65 comprised of 16.5 mm and 10 mm crushed coarse aggregates, and 4 mm sand. These were combined

66 in a ratio of 48:12:40 to get the maximum packing density in (AA-GGBS/FA) concrete mixes. The

bulk specific gravity and water absorption of these materials were measured based on BS EN 1097-1

and are presented in Table 3. Potable tap water (i.e. drinking water quality) was used to make the

69 concrete mixes.

70 2.2 Mix details, casting procedure and test specimens

Sixteen (AA-GGBS/FA) concrete mixes were studied with different GGBS/fly ash content, water-to binder, percentage of alkali and the SiO₂/Na₂O ratio (silica modulus, Ms). The details of the different

73 mixes and their initial properties are presented in Table 4. The total binder content, which is the sum

of GGBS, fly ash and solid component of the WG, was kept constant at 425 kg/m^3 for all mixes and

75 the water content in the WG was taken into account while determining the mixing water. For

76 increasing proportions of the fly ash, the W/B for mixtures was reduced while the paste content of the

77 mixtures was kept constant.

78 Mixing

Blended cements were prepared by dry-mixing GGBS and fly ash in required proportions in a benchtop food mixer to get a homogeneous blended powder. Crushed basalt aggregates and sand were dry

81 mixed together in a laboratory pan-mixer for one minute. After adding the blended GGBS and fly ash

powder, mixing continued for another 2 minutes. The sodium hydroxide solution was then added, and

after 2 minutes of further mixing, sodium silicate solution was added and mixing continued for a

further minute. The details of the different mixes and their essential properties are presented in Table4.

86 Casting and curing of the specimens

Fresh properties of concrete were measured according to BS EN 12350 [12]. From each concrete mix,
nine 100 mm cubes, and three 100x200 mm cylinders, were cast for the determination of compressive

89 strength in accordance with BS EN 12390 [13], bulk resistivity and chloride diffusion coefficient 90 according to Nordtest NT Build 443 [14]. The concrete specimens were cast in three layers and 91 compacted on a vibrating table. After casting, all the specimens (still in the mould) were covered with 92 plastic sheets and left in the casting room for 24 h. They were demoulded and kept in sealed plastic 93 zip bags until the test date. The storage room was maintained at 23°C & 65% RH.

94 2.3 Testing procedures

95 Chloride transport through AA-GGBS/FA concretes was assessed using a non-steady state chloride diffusion test, Nordtest NT Build 443 [14]. One day before the test age of 91 days, the 100x200 mm 96 concrete cylinders were cut to four slices of diameter 100 mm with a thickness of 50 mm per mix. A 97 slice with a thickness of 50 mm from the cast surface (trowel finished face) was considered for 98 99 measuring bulk resistivity, and the rest were kept for carrying out the chloride diffusion test. The 100 vacuum saturation regime specified in the standard was used to precondition the slices so that the chloride flow is predominantly diffusive, and initial sorption or capillary forces are negligible. The 101 vacuum was applied to remove air for three hours and released afterwards. Samples were wrapped in 102 hessian saturated in deionised water to prevent leaching of ions, and placed in the container. The 103 weight of the sample was noted after an hour (W_1) and then vacuum was applied, followed by further 104 saturation. Weight was checked again (W₂). Usually after 6 hours, when W_i-W_(i-1) was less than 0.1%, 105 106 the samples were considered fully saturated; if not, saturation was continued until this criterion was met. After conditioning to a surface-dry condition, an epoxy resin was applied onto the surfaces of the 107 specimens in three layers except the exposure face (saw cut face). When the epoxy coating was dry, 108 109 the cores were immersed in a NaCl solution of concentration 165 g/L (~2.82 M) for six months. After immersion, two cores for each mix were profile ground to obtain concrete dust from different depths 110 up to a depth was determined for each sample by spraying a 0.1 N AgNO3 solution on the third 111 112 freshly broken concrete sample. These depths were measured from the exposed surface. The total chloride content of the dust samples was determined in accordance with the recommendations of 113 RILEM TC 178-TMC [15] using a pre-calibrated potentiometric titration method. The concrete dust 114 was dissolved in deionised water in accordance with RILEM TC 178-TMC recommendations, 115 considering 3-5 min digestion [16] to measure the pH value of the suspension, and for the 116 117 determination of water soluble chlorides. Chloride diffusivity and the surface chloride content were determined by using curve fitting to the error function solution of Fick's second law of diffusion, as 118 described in NT BUILD 443 [14]. 119

120 **3. Results and discussions**

121 The following sections discuss the slump, flow and air content, compressive strength, chloride 122 diffusivity, and chloride binding capacity of AA-GGBS/FA concretes. The histograms categorize the 123 results for groups of concretes with different ratios of GGBS/FA and the contours graph are a good 124 map for designing AA-GGBS/FA concretes mixes based on GGBS/FA ratio and activator parameters 125 (Na₂O%*Ms).

126 3.1 Slump, flow and air content

The purpose of this testing programme was to demonstrate the range of slump values that AA-127 GGBS/FA concretes is capable of producing, and the changes to the governing variables that are 128 necessary to achieve high slump with low air content of the GGBS/FA mixes to have better mixes for 129 130 durability. The slump and flow results of the concretes are presented in Figures 1 and 2. The slump values of the concretes show that all the mixes had a slump value greater than 120 mm. It can be 131 observed that although W/B was adjusted based on the proportion of the fly ash in the system, 132 133 increasing the fly ash in the blend increased the workability, and the maximum workability was achieved for mix with the maximum fly ash content due to its bubble structure which increase the 134 lubricity. The literature shows that increasing the sodium oxide content results in increased 135

136 workability, reduced setting times and higher compressive strength. However, high concentrations 137 resulted in loose structure in sodium hydroxide activated samples because of the excess water is being retained in the system before forming a monolithic hydrated structure [18, 19]. For mixes with the 138 same GGBS to fly ash content and water to binder ratios, comparing mixes 1, 5, 9 and 13 and mixes 139 4, 8, 12 and 16 respectively, the slump values of the AA-GGBS/FA concretes with the same Ms ratio 140 141 increased with an increase of $Na_2O\%$. The slump values also increased with the increase of Ms which 142 can be seen by comparing odd and even mixes (e.g. mix no. 1 against 2, 3 against 4) in the two groups with two different Na₂O% and the same ratios of GGBS to fly ash. 143

As is evident from Fig. 1, the AA-GGBS/FA concretes can be designed for S3, S4 and S5 slump ranges. Thus, all the mixes meeting the minimum slump requirement for their use in chloride environments, such as S2 specified in BS8500-1:2015 for different exposure of marine environments [17].

The contour map graphs in Figure 3 (a) and (b) shows the results of slump and flow test against 148 GGBS/FA ratio and a function of activator parameters or (Na₂O%*Ms) values for different mixes 149 with same paste content. ($Na_2O\%^*Ms$) is the ratio of the SiO₂ content of the activator to the total 150 cementitious materials content for mixes. It is depicted from both figures that increasing the 151 152 GGBS/FA ratio, decreases the workability of mixes for GGBS/FA<0.67 and GGBS/FA>1.5 while 153 increases that for 0.67<GGBS/FA<1.5. The maximum workability of mixes happens when 1.5 < GGBS/FA < 2.5 when (Na₂O%*Ms) values is less than 7.5. It is also evident from these figures 154 that workability increased when (Na₂O%*Ms) was increased which can be due to disparity of silica in 155 156 solution form.

The air content of the GGBS/FA mixes are presented in Figure 4 and Figure 5 and they show the 157 contour graph for air content in fresh mixes of AA-GGBS/FA concretes against GGBS/FA ratio and 158 159 (Na₂O%*Ms) values. The lowest air content was measured for mixes with a GGBS/fly ash ratio of 60/40 while the highest was for 20/80 (except mix 15) while considering the map analysis there is two 160 ideal island which shows the minimum air content in the fresh AA-GGBS/FA concretes. These two 161 162 islands happen when 0.6<GGBS/FA<1.0 or 1.0<GGBS/FA<2.25 and the (Na₂O%*Ms) for mixes is more than 6.75% and less than 4.25%, respectively which are recommendable to have workable AA-163 GGBS/FA concrete with low air content. Figure 4 also shows in the mixes made with a higher alkali 164 percentage, the air content is lower (comparing Mixes 1, 5 and 13 to Mixes 4, 8 and 16, respectively) 165 and higher Ms for mixes has had no major effect, especially for mixes with higher alkaline content 166 (comparing Mixes 3 to 4 or 7 to 8). However higher silica content in the activator was more effective 167 168 when the fly ash proportion increased in the blend, and this causes higher air content when using higher silica content (comparing Mixes 15 to 16) which means silica do not let air to be pushed out 169 170 from fresh concrete.

171 3.2 Compressive strength

Figure 6 shows the compressive strengths of AA-GGBS/FA concrete mixes at 2, 28 and 90 days. 172 From the results, it can be stated that: (1) the first three groups had more than 20 MPa after 2 days and 173 174 more than 50 MPa after 28 days curing and 28 day strength is 80–90% of the 91 day strength, offering insight into the short and long term microstructural development in such binders, (2) as is expected, 175 the more fly ash in the system, the lower is the early age and long term strength, (3) an increase in 176 Na₂O% and Ms generally increases the compressive strength of AA-GGBS/FA concretes, which 177 agrees with the results reported by others. The last one is due to more N-A-S-H (sodium alumina 178 179 silicate hydrate) reaction product being generated, based on the increase of Na₂O%, and more C-A-S-H (calcium alumina silicate hydrate) reaction product being produced, based on the increase of SiO₂ 180 181 content. Silica modulus of 0.45 will drop down the strength for Na₂O% of 8% and this can be because of low silica content in the activator to make the gel formation. The activated GGBS/fly ash concretes 182 with Ms of 1.0 up to 1.25 generally obtained the highest compressive strength (see Figure 6), which 183

- 184 also has a good agreement with the literature and shows that the activated GGBS/fly ash with Ms of around 1.0 obtains the best hydration. From the 28day compressive strengths, Table 4 and Figure 6 185 both show that except mixes no. 7(1.5-8%-0.45), 11(0.67-8%-0.45) and 13 to 16 with 186 GGBS/FA=0.25, all achieved the required strength for the EN206 exposure classes XS3 and XD3 187 which is equivalent to an average strength \geq 56.99 MPa (calculated as 45 +1.48(8.1) MPa; where 8.1 188 189 is the standard deviation of the test results). However, mixes 7(1.5-8%-0.45), 11(0.67-8%-0.45) and 16(0.25-8%-1.0) met the strength requirement of 48.99 (=37+1.48(8.1)) MPa for the exposure classes 190 XS1, XD1 and XD2. Although it was evident that the strength requirement and w/b will not comply 191 192 with BS 8500-1:2015 norms, the intention of including the mixes were to assess their performance 193 against Cl⁻ ingress.
- Figure 7 (a) to 7 (c) presents the contour map of 2, 28 and 90 days compressive strength against GGBS/FA ratio and (Na₂O%*Ms) values. In general, there is an increase in the 2, 28-days compressive strength values with increase in the SiO₂ content of the activator to the total cementitious materials content. However, when the SiO₂ content of the activator reach to 6.5% and 7.5% of the total cementitious materials content respectively the reverse trend is started. 90-days compressive strength does not change significantly for GGBS/FA ratio >1.5 and (Na₂O%*Ms) >6.
- 200 3.3 Bulk electrical resistivity
- Electrical resistivity can evaluate microstructure of concrete and be related to the pore structure and 201 conductivity of the pore solution to predict the diffusion coefficients of chloride ions. The test set-up 202 for measuring bulk resistivity is shown in Figure 8. Figure 9 depicts the average bulk electrical 203 resistivity for different AA-GGBS/FA concrete mixes. Higher resistivity could be attributed to denser 204 structure with low connected porosity and/or less conductivity of the binder matrix and such mixes 205 206 will be expected to be superior in resisting ionic (chloride) flow. Higher GGBS content in AA-207 GGBS/FA concretes resulted in higher bulk resistivity, while mixes with higher alkali content show higher bulk resistivity when GGBS dominates the blend. Furthermore, bulk electrical resistivity 208 decreases for mixes made with higher silica modulus (except Mixes 3 and 4), whereas the bulk 209 210 resistivity seems to be a maximum at Na₂O=8% and Ms=1.0 when GGBS/FA=80/20. This may be related to the alkali content which is required to produce maximum reaction products and gives the 211 confidence for designing the concretes for strength and durability. 212
- Figure 10 shows contour graph of bulk electrical resistivity for different mixes against GGBS/FA 213 ratio and $(Na_2O\%*Ms)$ values. It is depicted from Figure 10 that for mixes with GGBS/FA < 0.67 the 214 SiO₂ content of the activator has minimal influence on bulk resistivity and the resistivity does not 215 216 change for $(Na_2O\% *Ms) < 6.25\%$. This can show that for GGBS to fly ash less than 0.67 and $(Na_2O\%^*Ms) < 6.25\%$ pore structure dominates the conductivity of the system which is not 217 dependent to the activator specification. Whereas for $(Na_2O\% * Ms) > 6.25\%$ the resistivity decreases 218 with increasing of the (Na₂O%*Ms) which shows that pore solution domination and higher 219 220 conductivity because of free ions in the binder matrix reduce the resistivity. For mixes with same (Na₂O%*Ms) the bulk resistivity increases with increasing of GGBS to fly ash ratio for GGBS/FA 221 <0.67 and GGBS/FA >1.5. This shows the effect having denser structure and is because of higher 222 GGBS content in the binder matrix of mixes. While for 0.67<GGBS/FA <1.5 and (Na₂O%*Ms) <7.5 223 pore solution domination causes the reduction of resistivity and higher conductivity because of free 224 ions in the pore solution. This can show the range of (Na₂O%*Ms) which is suitable for activation 225 corresponding to different GGBS to fly ash ratio and should be more than 7.5 for 226 227 0.67<GGBS/FA<1.5.
- 228 3.5 Chloride diffusion

Concrete sample disks with all surface painted except the top face, placed in air tight container withsodium chloride solution for diffusion test as per NT Build 443 are shown in Figure 11.

- 231 Figure 12 presents average chloride diffusion depth and D_{nssd} for mixes. As shown in Figure 12 the
- chloride diffusion depth of AA-GGBS/FA concretes was from 13 mm for mixes 2(6%-1.25), 3(8%-0.45), 4(8%-1.0) with GGBS/FA=80/20 to 40 mm for mixes 13(6%-1.0) and 14(6%-1.25) with GGBS/FA=20/80.
- The non-steady state diffusion coefficient with AA-GGBS/FA concretes was resulted between 235 $1.05*10^{-12}m^2/s$ [mix3 (8%-0.45)] with GGBS/FA=80/20 and $126.51*10^{-12}m^2/s$ [mix14 (6%-1.25)] 236 with GGBS/FA=20/80. It can be seen that all the D_{nssd} for AA-GGBS/FA concretes was moderate to 237 high except for the mixes with GGBS to fly ash ratio of 80/20. This is in line with Mundra et al. 238 239 findings that states for fly ash blended AAMs concretes service life influenced by Cl⁻ transport. The non-steady diffusion coefficient, D_{nssd}, of the AA-GGBS/FA concretes was found to be influenced by 240 241 the fly ash content, Na₂O% and Ms (at mixes with low alkali content). All the D_{nssd} values for AA-242 GGBS/FA concretes were higher for higher fly ash content which is due to the influence of larger pore size and more pore connectivity of the concrete. 243
- The third group of data, with average X_d closer to 35 mm, the average D_{nssd} is 13.56. This is three times bigger than second group which has an X_d close to 30. This can be because of more chloride
- absorption in third group which has more fly ash in the system. Activated fly ash has been stated to
- have zeolite formation with better initial surface absorption (physical chloride binding) than activated
- slag [21]. Surface absorption is stated the main responsible for around 90% of the total chloride
- 249 uptake [22].
- Figure 13 shows the contour map of D_{nssd} of different AA-GGBS/FA concrete mixes against 250 GGBS/FA ratio and (Na₂O%*Ms) values. D_{nssd} of AA-GGBS/FA concrete mixes does not change 251 significantly with increase of the ratio of the SiO₂ content of the activator to the total cementitious 252 materials content for GGBS/FA ratio <1.0 and (Na₂O%*Ms) <6.0. Furthermore (Na₂O%*Ms) =6.0 is 253 254 an extreme point and at which D_{nssd} of AA-GGBS/FA concrete mixes for values less than 6.0 is decreasing while for values more than 6.0 is increasing. The plaid area in Figure 13 shows the limits 255 256 of designing AA-GGBS/FA concrete mixes suitable for chloride exposure environments, if 257 reinforcing steel is preferred.
- 258 3.4 Chloride binding in AA-GGBS/FA concrete
- 259 There is limited information available on chloride binding of AA-GGBS/FA concrete in the literature. The available CSH and aluminate phases (C-(N)-A-S-H or two layered double hydroxides) in 260 261 activated GGBS contribute to the physical and chemical binding reactions [22, 23] while there is no information on the conformed phases in activated fly ash systems to show contribution of that to 262 chloride binding reactions. Binding behavior of different alkali activated slag (AAS) concrete was 263 reported in previous article by the authors [24]. Following the same procedure, the total and water-264 soluble chloride concentration of the concrete dust removed for chloride diffusion test in section 3.3 265 266 were determined for mixes 13-16. This data is presented in Figures 14 a-d along with the corresponding pH values. It is evident from Figure 14 that for the mixes studied, the total chloride 267 concentration is almost entirely represented by water-soluble chlorides. It is known that pH reduction 268 269 can release the Cl⁻ otherwise bound to Friedel's salt [25] in PC based systems. In a typical PC based system, the total Cl will be composed of both bound and free chlorides, and free chloride is often 270 represented by the water-soluble fraction which may also contain a proportion of the adhered Cl⁻ 271 physically bound to the cement phases. Two aspects are noticeable from the figure, (1) the pH values 272 are ≤ 11 for most depth (or all depth), (2) the surface region is undergoing leaching-induced changes 273 274 during Cl⁻ transport that result in a reduction of pH; there is also a skin effect which causes a nearsurface dip in the total Cl⁻ content. While as resulted in previous study, for alkali activated slag (AAS) 275 276 concrete the pH values are >11 for most depth and the surface region occurs within the depth of 5-12 277 mm [24].

- 278 The main reaction product in AA-GGBS/FA concrete is a Na-Al silicate hydrate (N-A-S-H) gel which
- it seems have no tendency to bind chlorides in the alkalinity range observed in mixes 13-16. The lower alkalinity could have affected the stability of the reaction products that bind chlorides. The least fluctuation in pH reduction is for the mix 13, that has 6% Na₂O and Ms = 1.
- Finite the maximum reduction is for the maximum rate max
- The results presented in this article shows that AA-GGBS/FA with GGBS/FA ratio less than 2.5 are not suitable for chloride exposure environments, if reinforcing steel is preferred. If the GGBS/FA ratio is between 2.5 and 4, silica content in the activators should be selected based on Figure 13 to have concrete with low diffusivity.
- 286

287 **4.** Conclusion

- Results show that it is possible to achieve desirable fresh properties and compressive strength for AA-GGBS/FA concrete when cured at 20°C. This means such concretes do not require elevated temperature curing for strength gain.
- 291 2. A higher proportion of GGBS results in high early and long-term strength.
- AA-GGBS/FA concretes with GGBS to fly ash ratio less than 4 have moderate to high
 chloride migration coefficients, making them non-viable for protection of reinforcing steel in
 chloride environments.
- 295 4. The blends can be categorised based on the GGBS to fly ash ratio (GGBS/FA) as low diffusivity (GGBS/FA = 80/20), high diffusivity (GGBS/FA = 60/20), and very high diffusivity (GGBS/FA = 40/60 & 20/80).
- 5. No bound chloride in the alumina silicate-dominated alkali-activated concretes, characteristic
 of a lack chemical binding in the fly ash dominated AA-GGBS/FA concretes and as a result,
 the chloride diffusivity in these binders depends on the role of the pore structure in physical
 chloride binding and encapsulation which highlights the importance of the pore structure
 studies in AA-GGBS/FA concretes.
- 303 6. Typically, workable AA-GGBS/FA concretes require binder content of 425 kg/m³
 304 (GGBS/FA=4) and a water/binder of 0.47, which is close to the maximum acceptable limit for
 305 XD3 and XS3 chloride exposure classes as per BS 8500.

306 Acknowledgements

The authors gratefully acknowledge the Engineering and Physical Sciences Research Council for providing the financial support for conducting this work [EP/M003272/1]. The authors would like to acknowledge the technical support provided by Dr. John Provis at The University of Sheffield as PI of the project.

311 **References**

- 312 [1] J. Whiting, "Manufacture of Cement", U.S. Patent 544,706, 1895.
- [2] A. M. Rashad, Properties of alkali-activated fly ash concrete blend with slag, Iran Journal of
 Material Science and Engineering, 10 (1), 2013, 57-64
- [3] S. Aydin, A ternary optimization of mineral additives of alkali activated cement mortars, Constr.
 Build. Mater. 43 (2013) 131-138
- [4] N.K. Lee, H.K. Lee, Setting and mechanical properties of alkali-activated fly ash/slag concrete
 manufactured at room temperature, Constr. Build. Mater. 47 (2013) 1201-1209

- 319 [5] F. Collins, J. Sanjayan, Effect of pore size distribution on drying shrinkage of alkali activated slag 320 concrete, Cem. Concr. Res., 30 (2000) 1401-1406
- 321 [6] A.A. Melo Neto, M.A. Cincotto, W. Repette, Drying and autogenous shrinkage of pastes and mortarwith activated slag cement, Cem. Concr. Res., 38 (2008) 565-574 322
- [7] Y. Ma, G. Ye, The shrinkage of alkali activated fly ash, Cem.Concr. Res., 368 (2015) 75-82 323
- [8] M. Chi, R. Huang, "Binding mechanism and properties of alkali-activated fly ash/slag mortars", 324 325 Constr. and Build. Mater. 40 (2013) 291-298
- [9] A.F. Abdalqader, F. Jin, A. Al-Tabbaa, Characterisation of reactive magnesia and sodium 326 carbonate-activated fly ash/slag paste blends, Constr.Build. Mater. 93 (2015) 506-513 327
- [10] X. Gao, Q.L. Yu, H.J.H. Brouwers, Assessing the porosity and shrinkage of alkali activated slag-328 fly ash composites designed applying a packing model, Constr. Build. Mater. 119 (2016)175-184 329
- [11] D. M. Roy, W. Jiang, M.R. Silsbee, Chloride diffusion in ordinary, blended, and alkali-activated 330 331 cement pastes and its relation to other properties, Cem. Concr. Res. 30 (2000) 1879-1884
- 332 [12] BS EN 12350-2, 5&7, Testing fresh concrete-Part 2, 5&7: Slump test and Flow test and Air content measurement, BSI, London, 2009 333
- 334 [13] BS EN 12390-3, Testing hardened concrete-Part 3: Compressive strength of test specimens, BSI, London, 2009 335
- 336 [14] NT BUILD 443, Concrete, hardened: accelerated chloride penetration, NORDTEST, Espoo, 337 1995
- [15] RILEM TC 178-TMC: Testing and modelling chloride penetration in concrete. Analysis of total 338 339 chloride content in concrete, recommendation, Mater. Struct. 35 (2002) 583-585
- 340 [16] RILEM TC 178-TMC: Testing and modelling chloride penetration in concrete. Analysis of water 341 soluble chloride content in concrete, recommendation, Mater. Struct. 35 (2002) 586-588
- [17] BS 8500-1, Concrete—complementary British Standard to BS EN 206-1—part 1: method of 342 specifying and guidance for the specifier, BSI, London, 2015 343
- [18] D. Bondar, C.J. Lynsdale, N.B. Milestone, N. Hassani, A.A. Ramezanianpour, Effect of type, 344
- form, and dosage of activators on strength of alkali-activated natural pozzolans, Cem. Concr. 345 346 Compos. 33 (2011) 251-260
- 347 [19] D. Bondar, S. Nanukuttan, M. Soutsos, P.A. Muhammed Basheer, J. Provis, Suitability of alkali
- activated GGBS/Fly ash concrete for chloride environments. in Tagnit-Hamou, A.(eds), The 10th 348 349 ACI/RILEM International Conference on Cementitious Materials and Alternative Binders for
- 350 Sustainable Concrete, 2-4 October 2017, Montreal, Canada, 2017, 35.1-35.14
- [20] S. Mundra, S.A. Bernal, M. Criado, P. Hlaváček, G. Ebell, S. Reinemann, G.J.G. Gluth, J.L. 351 352 Provis, Steel corrosion in reinforced alkali-activated materials, RILEM Technical Letters, 2 (2017) 353 33-39
- [21] M.W. Grutzeck, S.Kwan, M. DiCola, Zeolite formation in alkali-activated cementitious systems, 354 Cem. Concr. Res. 34(6) (2004) 949-955 · 355
- [22] X. Ke, S.A. Bernal, J.L. Provis, Uptake of chloride and carbonate by Mg-Al and Ca-Al layered 356
- 357 double hydroxides in simulated pore solutions of alkali activated slag cement, Cem. Concr. Res. 100
- (2017) 1–13. 358

- [23] X. Ke, S.A. Bernal, O.H. Hussein, J.L. Provis, Chloride binding and mobility in sodium
 carbonate-activated slag pastes and mortars, Mater. Struct. 50 (2017) #252.
- [24] D. Bondar, Q. Ma, M. Soutsos, M. Basheer, J.L. Provis, S. Nanukuttan, Alkali activated slag
 concretes designed for a desired slump, strength and chloride diffusivity, Constr. Build. Mater. 190
 (2018) 191–199
- 364 [25] M. Castellote, C. Andrade, C. Alonso, Chloride-binding isotherms in concrete submitted to non-
- 365 steady-state migration experiments, Cem. Concr. Res., 29 (1999) 1799-1806

Table 1: Oxide composition of GGBS & Fly ash

precursor	Component (mass% as oxide)						
	SiO ₂	Al_2O_3	CaO	Fe ₂ O ₃	MgO	others	LOI
GGBS	35.7	11.2	43.9	0.3	6.5	2.09	0.31
Fly ash	46.8	22.5	2.2	9.1	1.3	14.5	3.6

Table 2: Physical properties of GGBS & Fly ash

Material	GGBS	Fly ash
Fineness ≥45 µm	7.74%	18.39%
Particle density	2.86	2.21
Water absorption	35.14%	27%

Table 3: Physical properties of aggregates

Aggregates	Bulk specific gravity	Bulk SSD Specific gravity	Water Absorption (%)	
Sand (0-4mm)	2.72	2.73	0.75	
Fine Crushed Agg. (5-10mm)	2.67	2.75	3.14	
Coarse Crushed Agg. (10-16mm)	2.60	2.67	2.60	

Table 4: The details of the different mixes and their properties

Mix No.	GGBS/FA (kg/m ³)	GGBS/FA ratio	Ms (=SiO ₂ /Na ₂ O)	Na ₂ O (%)	W/B	Slump (mm)	2 days Comp. St. (MPa)	28 days Comp. St. (MPa)	Concrete Grades
1	340/85	4	1.0	6	0.47	170	54.6	81.1	C65/80
2	340/85	4	1.25	6	0.47	200	51.2	84.8	C65/80
3	340/85	4	0.45	8	0.47	200	35.2	60.2	C49/60
4	340/85	4	1.0	8	0.47	230	50.9	80.6	C65/80
5	255/170	1.5	1.0	6	0.46	230	42.1	73.4	C56/70
6	255/170	1.5	1.25	6	0.46	240	40.3	78.5	C60/75
7	255/170	1.5	0.45	8	0.46	205	28.8	52.4	C40/50
8	255/170	1.5	1.0	8	0.46	240	31.9	75.7	C60/75
9	170/255	0.67	1.0	6	0.44	170	28.2	64.8	C49/60
10	170/255	0.67	1.25	6	0.44	210	22.3	69.4	C52/65
11	170/255	0.67	0.45	8	0.44	120	22.2	54.0	C40/50
12	170/255	0.67	1.0	8	0.44	250	23.0	61.2	C49/60
13	85/340	0.25	1.0	6	0.43	225	13.8	42.6	C32/40
14	85/340	0.25	1.25	6	0.43	220	9.5	47.1	C35/45
15	85/340	0.25	0.45	8	0.43	200	9.7	28.4	C20/25
16	85/340	0.25	1.0	8	0.43	250	10.4	49.2	C35/45





Fig. 1-Slump results of AA-GGBS/FA concretes



Fig. 2-Flow results of AA-GGBS/FA concretes









410 Fig. 3-Contour graph for workability of AA-GGBS/FA concretes - (a) Slump test, (b) Flow test





Fig. 4-Air content in fresh mixes of AA-GGBS/FA concretes





Fig. 5-Contour graph for air content in fresh mixes of AA-GGBS/FA concretes





Fig. 6-Compressive strength of AA-GGBS/FA concretes mixes at 2, 28 and 90days









Fig. 7-Contour graph for compressive strength of AA-GGBS/FA concretes (a) 2days, (b) 28days, (c) 90days



Fig. 8-Bulk resistivity test equipment and sample







Fig. 10-Contour graph for bulk electrical resistivity (Ω .m) of AA-GGBS/FA concretes



443	
444	
445 446	Fig. 11-Concrete sample disks with all surface painted except the top face, placed in air tight container with sodium chloride solution for diffusion test as per NT Build 443
447	
448	





Fig. 12-Chloride penetration depths (X_d) and D_{nssd} of AA-GGBS/FA concretes



Fig. 13-Contour graph for $D_{nssd}\,(x10^{-12}\mbox{ m}^2/s)$ of AA-GGBS/FA concretes





457 Figure 14 Total and water-soluble chloride concentration and pH for concrete dust samples collected
458 for NT Build 443 test for mixes 13, 14, 15 and 16

- AA-GGBS/FA concrete cured at 20^oC results in desirable fresh properties and strength.
- A higher proportion of GGBS results in high early and long-term strength.
- AA-GGBS/FA concretes with GGBS/FA<4 are non-viable for protection of steel in chloride environments.
- There is no chloride binding capacity in the fly ash dominated AA-GGBS/FA concretes.
- The chloride diffusivity in AA-GGBS/FA concretes depends mostly on the pore structure.

Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 3(a).fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 3(b).fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 5.fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 7(a).fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 7(b).fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 7(c).fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 10.fig Supplementary MATLAB .fig files Click here to download Supplementary MATLAB .fig files: Fig 13.fig