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## Accepted Manuscript

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# Systematic model identification and optimization-based active polymorphic control of crystallization processes

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## Abstract

Polymorphism is an important issue in industrial crystallization, since polymorphs of the same compound can present very different properties, such as solubility, melting point or density, influencing considerably the manufacturability and bioavailability of the final product.

This work proposes a model-based active polymorphic control strategy that allows obtaining large crystals of the stable polymorph at the end of a batch crystallization process, even in the case of erroneous seeding or in situ nucleation of a mixture of both the stable and metastable forms. A novel systematic experimental design was applied to estimate the kinetic parameters of dissolution, growth and secondary nucleation of the stable and metastable polymorphs of the model compound (ortho-aminobenzoic acid, OABA). Such experimental approach allows the determination of the studied kinetics without any correlation between parameters during the estimation, and without the need of off-line measurements of the crystal size distribution during the experiments.

The estimated kinetic parameters were used to build a population balance model for the calculation of the optimal temperature profile needed, during a batch cooling crystallization process, for the (i) elimination of the metastable form crystals nucleated in situ or erroneously seeded and the (ii) maximisation of the size of the crystals of the stable polymorph obtained at the end of the batch process.

**Keywords:** polymorphic control, population balance equations, batch crystallization optimization.

## 1. Introduction

Polymorphs of the same compound can have different physical characteristics such as solubility, stability, melting point and, most importantly, bioavailability. For this reason both discovering new polymorphs and designing new control strategies to tailor the polymorphic purity of the final product crystallized in industrial processes is very important. The choice of solvent, supersaturation conditions, temperature, pH and the use of additives can determine the polymorphic outcome of a cooling crystallization, while PAT tools can be used to check the purity of the final product and control its growth. ATR-FTIR, ATR-UV/Vis, in situ Raman and FBRM have been frequently used to control the growth of both stable and metastable polymorphs through different control approaches. Recently a feedback control technique, the active polymorphic feedback control (APFC), was developed to select and grow the desired polymorphic form of the crystallized compound (Simone et al. 2014). In this strategy both Raman and ATR-UV/Vis spectroscopy are used: the Raman probe can detect the nucleation or seeding of a polymorphic mixture and it eliminates the metastable form by triggering a controlled dissolution cycle.

**Table 1: Control strategies used by researchers to control the crystallization of polymorphic compounds (Simone et al. 2014).**

<b>Control approach</b>	<b>Reference</b>
Seeding the desired form between the solubility curve and the metastable limit line in order to avoid the nucleation of the other form and keep cooling until the supersaturation is consumed	(Threlfall 2000; Beckmann 2000)
Finding the correct amount of seed above which secondary nucleation of the metastable form is suppressed and solution-mediated transformation is avoided in a cooling crystallization	(Doki et al. 2003)
Seeding during a cooling crystallization and using focused beam reflectance measurement (FBRM) in combination with ATR-FTIR to check the total counts and the supersaturation in order to reach the desired size of the crystals and eliminate the fines via dissolution	(Doki et al. 2004)
Temperature control and concentration control for the conversion of the metastable form of a polymorph to its stable form (simulation and experimental work)	(Hermanto et al. 2007; Kee et al. 2009; Kee, et al. 2009; Hermanto et al. 2009)
Seeding and growth of the metastable form during a cooling crystallization performing supersaturation control	(Kee et al. 2009; Chew et al. 2007)
Combination of anti-solvent and cooling crystallization was performed to obtain the desired form of indomethacin in acetone	(Minamisono and Takiyama 2013)
Feedback control of the reactive crystallization of L-glutamic acid in a semi-batch precipitation was conducted using MID-IR or Raman, ATR-FTIR and a pH-meter	(Qu et al. 2009; Alatalo et al. 2010)
Control of Polymorphism in Continuous Crystallization via Mixed Suspension Mixed Product Removal Systems Cascade Design: estimation of the optimal operating conditions to crystallize one specific polymorph	(Lai et al. 2015)

ATR-UV/Vis is instead used to control the crystallization under conditions and allow only the growth of the stable form using supersaturation control. Other control approaches proposed in the literature either use the Raman system only to detect the formation of the unwanted polymorph as a trigger to restart the crystallization with a different cooling rate, or use only supersaturation control in conjunction with the suitable seeding procedure to drive the system in the phase diagram to obtain the desired polymorphic form. A summary of recent research works on polymorphic control is shown in Table 1.

The APFC strategy is a model-free approach, which was evaluated for the cooling crystallization of ortho-aminobenzoic acid, and led to pure polymorphic forms in the case of unseeded crystallization processes where nucleation of polymorph mixtures occurred, or for seeded crystallization with contaminated seed crystals containing an unwanted polymorph impurity (Simone et al. 2014). During the experiments performed, a partial dissolution of the desired form together with the elimination of the undesired form was observed. However, it is not clear whether such partial dissolution favours the attainment of larger crystals of the stable form at the end of the batch or not. A model based approach can help understanding if the initial dissolution cycle improve or worsen the final size distribution of the crystals of the stable form and how the temperature profile could be optimized in order to maximize such distribution. The aim of this work is to develop a model-based active polymorphic control by determining the kinetic parameters of the growth and polymorphic transformation of ortho-aminobenzoic acid through properly designed experiments, and then by simulating and optimizing the batch crystallization process in order to control both size and polymorphic purity of the final crystals.

Parameter estimation and modelling of polymorphic transformation has rarely been performed because of the complexity of the phenomenon, which involves two steps: dissolution of the metastable form and nucleation and growth of the stable one (Cardew and Davey, 1985).

A first example of population balance applied to a polymorphic transformation was the study on the conversion of citric acid from the anhydrate to its monohydrate form (Fevotte et al. 2007). Seeded isothermal experiments were conducted to estimate the kinetic parameters for the growth and nucleation of the monohydrate form as well as for the dissolution of the anhydrate form. Power-law relationships were used to express dissolution and growth as a function of supersaturation, while secondary nucleation was expressed as a function of supersaturation as well as of the concentration of crystals of the stable form present in suspension. Raman and image analysis were used to measure solute concentration, crystal size distribution and polymorphic ratio; a finite elements method was used to solve the population balance equation, PBE (using the software FEMLAB). Many different solution techniques were used to solve the PBE for the polymorphic transformation of L-glutamic acid: moving pivot technique (Cornel et al. 2009), finite volume method in gPROMS (Ono et al., 2004) and the method of moments (Hermanto et al. 2007; Hermanto et al. 2009; Hermanto et al. 2011; Sheikholeslamzadeh and Rohani 2013). Despite working with the same system the authors of the mentioned studies used different types of equations to express the kinetics of the phenomena involved in the polymorphic transformation of L-glutamic acid. All the authors found a good agreement between simulated and experimental data, even when semi-empirical, simplified functions were used.

Ono et al. included in the model only dissolution of the metastable form (Sherwood correlation), size-dependent growth and secondary nucleation (semi-empirical function of supersaturation and mass of crystals of the stable form in slurry) of the stable form. More

phenomena were included in the models described by Hermanto et al. (2011), Cornel et al. (2009) and Sheikoleslamzadeh and Rohani (2013): primary nucleation and dissolution of the metastable form, secondary nucleation and growth of the stable form. Besides, less empirical correlations were used in such studies compared to the model described by Ono et al. (2004). In fact, the growth kinetics of the stable and metastable polymorphs of L-glutamic acid were found to be integration controlled and of the birth-and-spread type, with the exception of the studies performed by Hermanto and co-workers, where a power-law function was used to express the growth rate of the metastable form. A Sherwood correlation was used to estimate the dissolution of the metastable form in all the referenced works. The functions used to express the nucleation rates for both the stable and metastable forms of L-glutamic acid were different in the mentioned studies: Cornel et al. (2009) and Sheikoleslamzadeh and Rohani (2013) used primary nucleation exponential functions to describe the primary nucleation of the metastable form, while Hermanto et al. (2011) employed a simpler equation as a function of supersaturation and the third moment calculated for the metastable form. The kinetic of secondary nucleation of the stable form of L-glutamic acid was expressed with a semi-empirical function only of the mass of metastable crystals by Cornel et al. (2009) and of the mass of both the stable and metastable crystals by Hermanto et al. (2011) Sheikoleslamzadeh and Rohani (2013) instead employed a two-terms expression to estimate both the heterogeneous (exponential nucleation function) and surface secondary nucleation (as a function of the second moment of the metastable crystals) of stable L-glutamic acid.

The solution-mediated transformation of DL-methionine polymorphs was modelled by Wantha and Flood (2013) using the method of moments to solve the PBEs. In this work semi-empirical functions of the supersaturation were used to express the growth kinetics of both

forms and the dissolution kinetics of the metastable polymorph; a primary nucleation exponential function was used to estimate nucleation of the stable polymorph.

Schöll et al. (2006) solved the PBEs using the commercial software PARCIVAL for the parameter estimation of the kinetics of transformation of L-glutamic acid (Schöll et al. 2006). The model used included the kinetics of heterogeneous nucleation of the metastable form (exponential primary nucleation type function), size-independent growth rates of both the stable and metastable forms (integration controlled and birth and spread type of functions), dissolution of the metastable form (Sherwood correlation) and heterogeneous and surface nucleation of the stable form. A similar model was used to describe the polymorphic transformation of Bupirone hydrochloride from the metastable form II to stable form I (Trifkovic et al. 2012). Such model was solved using the methods of moments.

More recently, the methods of characteristics was used to describe the behaviour of the  $\alpha$  and  $\beta$  forms of para-aminobenzoic acid in a two stages MSMR reactor (Lai et al. 2015). The authors included in the model the growth of both stable and metastable forms (size-independent and surface integration controlled) and their secondary nucleation (semi-empirical equation as a function of the mass of crystals in suspension).

As explained in the previous paragraph, population balance models in the literature can include or not primary nucleation of both the stable and the metastable form but all of them include secondary nucleation of the stable form, expressed with semi-empirical functions, primary heterogeneous nucleation exponentials or with two-terms functions including both heterogeneous primary nucleation and surface secondary nucleation. In fact, secondary nucleation of the stable form and dissolution of the metastable polymorph are the key mechanisms happening during a polymorphic transformation (Cardew and Davey, 1985).

Only one theoretical study considered the presence of secondary nucleation of the metastable form and analysed its effect on the transformation time and the concentration profile (Kobari et al. 2014). However, the presence of secondary nucleation of the metastable form can be neglected if, during the crystallization process, the solute concentration is very close, or below the solubility of the metastable form. Thus, the supersaturation is too low to allow secondary nucleation.

The parameters necessary to define and model the active polymorphic control of ortho-aminobenzoic acid are: (i) dissolution kinetics for both forms, (ii) growth kinetics for both forms, (iii) secondary nucleation of the stable form (during transformation), and (iv) primary nucleation of the stable form (during transformation and after seeding far from the solubility curve). The estimated parameters will be then validated and applied to an optimization problem in order to design batch cooling crystallization processes that allow the growth of large crystals of the stable polymorph even in case of erroneous seeding or in situ nucleation of a mixture of the stable and metastable forms. In conclusion, the model-based active polymorphic control (mbAPC) proposed in this work represents a useful approach for the correct design of batch crystallization processes for polymorphic systems.

## 2. Materials and Methods

The model compound used for the experiments is ortho-aminobenzoic acid (OABA), which has three known different polymorphic forms (Jiang et al. 2010b; Jiang et al. 2010a; Jiang et al. 2008). The transformation studied in this work is the one from the metastable form II to the stable form I in a solution of 90% water and 10% IPA, below 50°C (see Figure 1a and b). OABA (>98% Form I, Sigma-Aldrich), isopropyl alcohol (99.97%, Fisher Scientific) and ultrapure water obtained via a Millipore ultra-pure water system, were used. Experiments

were performed in a 400 ml jacketed vessel; a PT-100 temperature probe connected to a Huber Ministat 230 thermoregulator was used to control the temperature. A RXN1 Raman analyser with immersion probe and 785 nm laser (Kaiser with iC Raman 4.1 software) was used to capture detect different polymorphs in suspension, while a MSC621 Carl Zeiss UV/Vis (in-house LabView software) with Hellma ATR (type 661.822-UV) probe was used to determine the solute concentration. A Malvern Mastersizer 2000 was used to determine the crystal size distribution at the beginning and during the experiments.

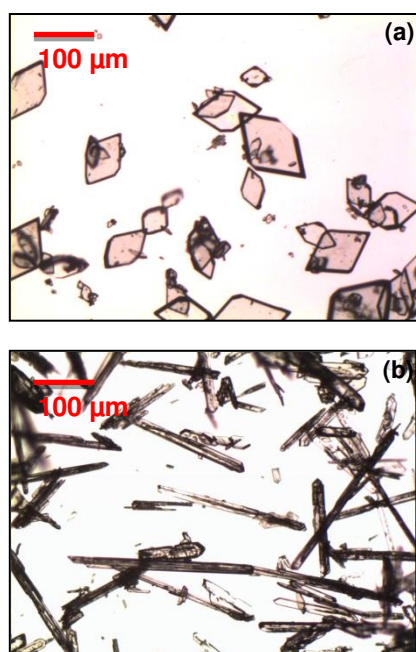


Figure 1: Micrographs of OABA crystals: (a) prismatic form I, and (b) needle-like form II.

The mean and the standard deviation of the crystal size distributions measured using the Mastersizer, were used to calculate a Gaussian curve that approximate the experimental data. This approximation was necessary to avoid the overestimation of fine particles in the measured samples, due to the non-spherical shape of the OABA crystals of both polymorphic forms. In fact, the volume and number distributions of needles and flat crystals measured by laser diffraction can show a large number of fine crystals (or even a bimodal shape) simply

because of the orientation of elongated particles on their shortest side during the measurement (Su et al. 2017).

A previously developed calibration model (Simone et al. 2014) was used to determine solute concentration from ATR-UV/Vis while specific Raman peaks for form I and II were tracked during the experiments to estimate the rate of transformation and check the composition of the slurry during the experiments performed with the metastable polymorph. Furthermore, the initial seeds were analysed with a Raman microscope (DXR Raman, Thermofisher) in order to check their purity.

The solubility curves for both form I and II between 10 and 40 °C were estimated using an the ATR-UV/Vis probe (interpolating data from a slow heating profile). Despite the system being enantiotropic (Jiang et al. 2010b; Jiang et al. 2010a; Jiang et al. 2008), in the used temperature interval the two OABA polymorphs can be considered monotropically related.

The formulas used for the solubility (Simone et al. 2014) are:

$$S_I = 1.267 \cdot 10^{-5} T^2 - 2.283 \cdot 10^{-4} T + 4.105 \cdot 10^{-3} \quad (1)$$

$$S_{II} = 1.299 \cdot 10^{-5} T^2 - 2.082 \cdot 10^{-4} T + 4.808 \cdot 10^{-3} \quad (2)$$

with the temperature  $T$  expressed in °C and the solubility  $S_I$  and  $S_{II}$  calculated in g/g solvent.

The solubility of form II and form I have been interpolated with polynomial functions and not with a Van't Hoff type equation to keep consistency with our previous experimental APFC study (Simone et al. 2014). In such paper polynomials were used as this is the only type of equation that can be currently input in the in-house software (CryPRINS) to perform supersaturation control during batch crystallization experiments.

## 2.1 Population balance model and solution

For the description of a particle population, let us introduce the monovariate number density function  $f(L, t)dL$ , which expresses the number of crystals within the  $L, L + dL$  crystal size domain ( $L$  expressed in  $\mu\text{m}$ ) in unit volume of suspension in the  $t$  time moment (expressed in seconds). Then, population balance equations can be used to predict and simulate polymorphic transformations considering one equation for each polymorph. Three main mechanisms must be considered during a transformation: nucleation and growth of the more stable form and the dissolution of the less stable polymorph. In the mbAPC also dissolution of form I must be considered and estimated. Indicating with the index  $II$  the parameters of the metastable form of OABA, and with  $I$  the ones of the stable one, the PBE for the studied system, using the simplified  $f(L, t) \rightarrow f$  notations, are:

$$\frac{\partial f_{II}}{\partial t} = \frac{\partial(D_{II}f_{II})}{\partial L} \quad (3)$$

for dissolution of form II,

$$\frac{\partial f_{II}}{\partial t} + \frac{\partial(G_{II}f_{II})}{\partial L} = 0 \quad (4)$$

for growth of form II,

$$\frac{\partial f_I}{\partial t} = \frac{\partial(D_I f_I)}{\partial L} \quad (5)$$

for dissolution of form I, and

$$\frac{\partial f_I}{\partial t} + \frac{\partial(G_I f_I)}{\partial L} = B_I \delta(L - L_0) \quad (6)$$

for growth and nucleation of form I: where  $f_{II}$  and  $f_I$  are the average number density functions of the metastable and stable form of OABA and  $\delta(L - L_0)$  is the Dirac delta function ( $\delta = \infty$  if  $L = L_0 = 0$  and  $\delta = 0$  if  $L \neq L_0$  with  $\int_{-\infty}^{+\infty} \delta(x) dx = 1$ ).

In order to close the system of equations that characterize the presented model the liquid phase mass balance is required (temperature is the controlled variable, therefore, the energy balance is not necessary):

$$\frac{dc}{dt} = -3k_v\rho_c \left[ \int_0^{\infty} G_I L^2 f_I dL + \int_0^{\infty} G_{II} L^2 f_{II} dL \right] \quad (7)$$

Where  $k_v$  is the volume shape factor and  $\rho_c$  stands for the crystal density. The mass balance Eq.(7) refers to the case when both populations are growing and it considers 0 nucleon size.

The equation remains valid for dissolution by applying simply  $D_I / D_{II}$  dissolution rates.

$D_I$  and  $D_{II}$  are the dissolution rates of the two forms and  $G_I$  and  $G_{II}$  the growth rates of form I and II defined as:

$$D_I \left[ \frac{\mu m}{s} \right] = k_{dI} (1 - S_I)^{d_I} \exp\left(\frac{E_{dI}}{RT}\right) \quad (8)$$

$$D_{II} \left[ \frac{\mu m}{s} \right] = k_{dII} (1 - S_{II})^{d_{II}} \exp\left(\frac{E_{dII}}{RT}\right) (1 + L_{II}) \quad (9)$$

$$G_I \left[ \frac{\mu m}{s} \right] = k_{gI} (S_I - 1)^{g_I} \exp\left(\frac{E_{gI}}{RT}\right) \quad (10)$$

$$G_{II} \left[ \frac{\mu m}{s} \right] = k_{gII} (S_{II} - 1)^{g_{II}} \exp\left(\frac{E_{gII}}{RT}\right) (1 + L_{II}) \quad (11)$$

with the supersaturation defined as  $S_{I,II} = \frac{c}{c_{sat,I,II}}$ .

Two types of nucleation of the stable form were estimated: (i) primary nucleation after seeding of a mixture of polymorphs if  $c_l > c_{satll}$  (ii) secondary nucleation during polymorphic transformation (if  $c_l > c_{satl}$ ). The different types of nucleation can be described by the equation:

$$B_l \left[ \frac{\#}{m^3 \text{sec}} \right] = k_b \left[ \frac{b}{(\log S_l)^2} \right] \exp \left( -\frac{E_b}{RT} \right) + k_s (1 - S_l)^s \exp \left( -\frac{E_s}{RT} \right) \quad (12)$$

A high resolution finite volume method (HR-FVM) was used to solve the model-equations (Gunawan et al. 2004). The basic idea of HR-FVM is the discretization of the continuous population density function; denoting with  $h$  the size and  $k$  the time interval,  $f_l^m$  is the approximate (discrete) population density function defined as:

$$f_l^m \approx \frac{1}{h} \int_{(l-1)h}^{lh} f(L, m, k) dL \quad (13)$$

where  $m$  and  $l$  are integers such that  $m \geq 0$  and  $N \geq l \geq 1$  and  $N$  stands for the mesh size (i.e. the number of discretization points). Then, the population balance Eq. (6) reduces to a system of algebraic equations:

$$\begin{aligned} f_l^{m+1} = & f_l^m - \frac{k}{h} (G_l f_l^m - G_{l-1} f_{l-1}^m) \\ & - \left[ \frac{k G_l}{2h} \left( 1 - \frac{k G_l}{h} \right) (f_{l+1}^m - f_l^m) \phi_l \right. \\ & \left. - \frac{k G_{l-1}}{2h} \left( 1 - \frac{k G_{l-1}}{h} \right) (f_l^m - f_{l-1}^m) \phi_{l-1} \right] \\ & + \epsilon_b \frac{k}{h} B \end{aligned} \quad (14)$$

In Eq. (14)  $\epsilon_b$  is a binary existence variable with values  $\{0,1\}$  which controls the existence of nucleation. In this PBE formulation,  $\epsilon_b = 1$  if  $l = 1$  (nucleon size) and is 0 otherwise. It is

worth noticing that the same Eq. (14) equation is used for growth and dissolution stages, treating the dissolution as negative growth and keeping in mind that the nucleation rate is 0 if the solution is undersaturated.  $\phi_l = f(\theta_l)$  is the flux limiter function and  $\theta_l$  is the ratio of consecutive gradients:

$$\theta_l = \frac{f_l^m - f_{l-1}^m}{f_{l+1}^m - f_l^m} \quad (15)$$

The Van Leer flux limiter of Eq. (16) has been successfully applied in the solution of population balance equations thus is adopted in this study too (Gunawan et al. 2004).

$$\phi(\theta_l) = \frac{|\theta_l| + \theta_l}{1 + |\theta_l|} \quad (16)$$

Note that the numerical apparatus Eqs. (13) - (16) applies for both populations:  $f_l^m \rightarrow f_{I,l}^m$  (with  $G_{I,l}$ ,  $D_{I,l}$  and  $B_I$ ) and  $f_l^m \rightarrow f_{II,l}^m$  (with  $G_{II,l}$ ,  $D_{II,l}$  and  $B_{II} = 0$ ). The resulted algebraic equation systems are solved simultaneously.

The time step is recalculated in all iterations to satisfy the Courant-Friedrichs-Lewy (CFL) criterion and the numerical system is stable if  $CFL \leq 1$ .

$$CFL = \frac{k}{h} \max\{G_{I,l}, G_{II,l}, D_{I,l}, D_{II,l}\} \quad (17)$$

Practically the  $CFL$  is fixed and  $k$  is expressed from Eq. (17). Finally, the solute mass balance takes the form:

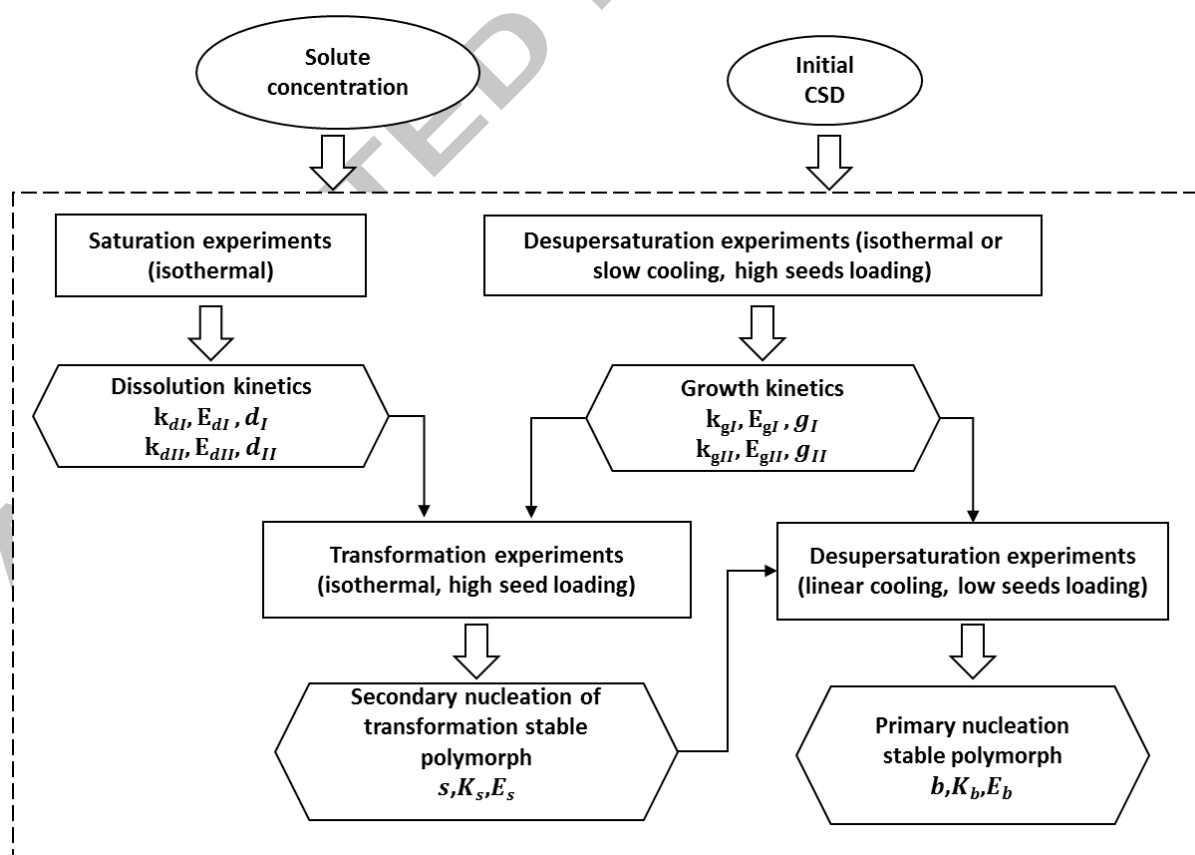
$$c^{m+1} = c^m - 3k_v \rho_C h^2 \sum_{l=1}^N L^2 [f_{I,l}^m G_{I,l} + f_{II,l}^m G_{II,l}] \quad (18)$$

Similarly to the Eq.(14), the mass balance Eq.(18) is applicable for dissolution as well, involving the dissolution rate for undersaturated conditions. An extended version of the *CrySiV* function (Szilagyi and Nagy, 2016) was used to efficiently solve the equation system

and a combination of Evolution Strategy with Covariance Matrix Adaptation (*ES-CMA*) global optimization algorithm (Hansen et al. 2003) and Matlab's *nlinfit* function (Levenberg-Marquardt algorithm) was employed to estimate the parameters and the confidence intervals.

## 2.2 Systematic experimental design for the model identification

Experiments were planned carefully in order to simplify the estimation of the kinetic parameters: the different phenomena were isolated as shown in Figure 2. Growth and dissolution for both forms were estimated through seeded saturation or desupersaturation experiments. The secondary nucleation of the stable form was estimated through isothermal transformation experiments and using the dissolution and growth kinetics already estimated. Finally secondary nucleation of form I after seeding was evaluated through desupersaturation experiments with low seeds loading at high supersaturation.



**Figure 2: Schematic of the experimental design used for the parameters estimation. Growth and dissolution for both forms were estimated through seeded saturation of desupersaturation experiments. The secondary nucleation of the stable form was estimated through isothermal transformation experiments and using the dissolution and growth kinetics already estimated. Finally secondary nucleation of form I after seeding was evaluated through desupersaturation experiments with low seeds loading at high supersaturation.**

This approach has two main advantages: (i) a correlation between the estimated parameters is avoided because only one phenomenon occurs in each set of experiments; (ii) only concentration data and the initial crystal size distribution (CSD) are needed for the estimation, sampling is not necessary to estimate the CSD during the crystallization process (Besenhard et al. 2015). Therefore, in this work the concentration profile is the only measured output used for the fitting procedure.

The present systematic approach allows the determination of the necessary kinetics parameters using only limited CSD data, which can often be unreliable. In fact, such data cannot be easily obtained online by standard process analytical technologies (e.g. FBRM) and is often estimated via off-line techniques such as optical or scanning electron microscopy or laser diffraction. The need of sampling and the off-line nature of the traditional CSD measurement techniques lower the accuracy and reliability of the collected data. A detailed list of the experiments performed and their conditions is shown in Table 2. Some of the experiments reported in Table 2 could be conducted consecutively in the same solution: growth of the metastable form can be estimated by a seeded experiment that can then be used to estimate secondary nucleation of the stable form by just letting the metastable form transform. In these cases, sampling at the beginning of the transformation is necessary to estimate the initial crystal size distribution to use in the parameter estimation. In particular, the kinetic parameters of growth of form II ( $k_{gII}$ ,  $E_{gII}$  and  $g_{II}$ ) and secondary nucleation of form I ( $s$ ,  $k_s$  and  $E_s$ ) were estimated from the same isothermal experiments, number 11 to 14,

each one conducted at a different temperature. All data points before the start of the nucleation of the stable form were used to estimate the growth of the metastable form II, while data collected after the appearance of the stable form were used for the estimation of the kinetics of secondary nucleation of such polymorph.

**Table 2: Description of the conditions used in the different experiments to determine the kinetics parameters of OABA.**

<b>Parameter estimated</b>	<b>Experimental conditions</b>	<b>Experiment number</b>
$k_{dI}$ , $E_{dI}$ and $d_I$	Four isothermal experiments (10, 15, 25 and 35 °C). Seeds were added to the solvent in the amount necessary to have a saturated solution after the complete dissolution	1-4
$k_{dII}$ , $E_{dII}$ and $d_{II}$	Four isothermal experiments (10, 15, 25 and 35 °C). Seeds were added to the solvent in the amount necessary to have a saturated solution. Raman spectroscopy was used to check the absence of polymorphic transformation during dissolution	5-8
$k_{gI}$ , $E_{gI}$ and $g_I$	Seeds (10% of the total solute) were added to a saturated solution at 40 °C. A slow linear cooling was then applied to avoid nucleation. Two experiments at different cooling rates were performed (−0.1 and −0.05 °C/min)	9-10
$k_{gII}$ , $E_{gII}$ and $g_{II}$	Four isothermal experiments (10, 15, 25 and 35 °C). Seeds of the metastable form (10% of the total solute) were added to supersaturated solutions (5 °C of supersaturation)	11-14 (Data points until the nucleation of the stable form started)
$s$ , $k_s$ and $E_s$	Isothermal seeded polymorphic transformation experiments at four different temperatures (10, 15, 25 and 35 °C)	11-14 (Data points after the nucleation of the stable form)

		started)
$b$ , $k_b$ and $E_b$	Seeding of a mixture of polymorphs at saturated condition for form II (about 40 °C) and cooling at $-1\text{ }^\circ\text{C}/\text{min}$ (three experiments, amount of seeds of 10% of the total solute)	15-17

The results of one of the combined experiments are shown in Figure 3. Growth of metastable form and secondary nucleation of the stable form at 10 °C are measured. The first 4000 s of the experiment were used, together with the other three isothermal growth experiments, to estimate growth of form II while the remaining time was used to estimate secondary nucleation of form I. Another important piece of information shown in Figure 3 is that the system can be considered neither growth nor dissolution controlled as in the case of previously studied compounds.

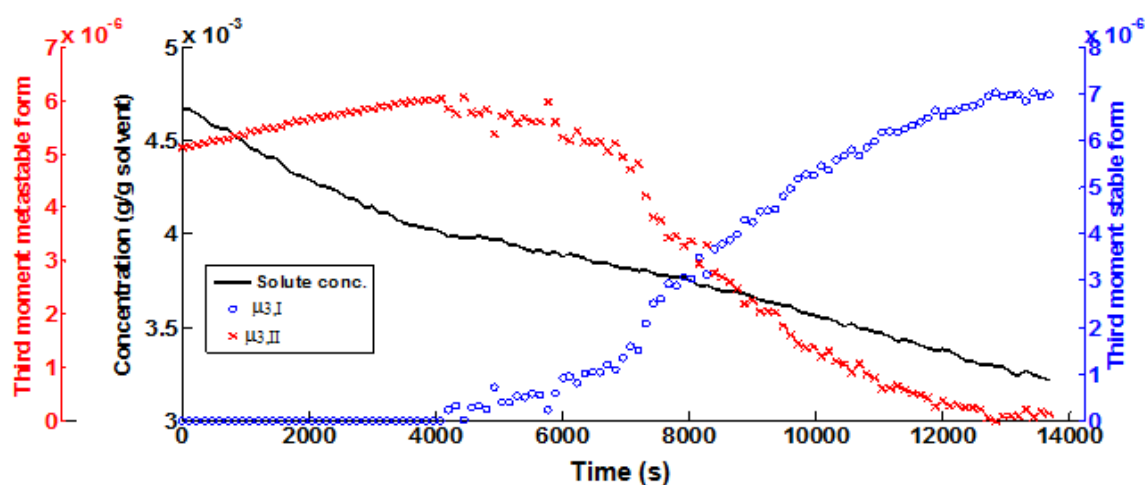


Figure 3: Results for growth of form II and transformation experiment at 10 °C.

### 3. Results and Discussion

#### 3.1 Parameters estimation and validation

The kinetic parameters estimated from all the experiments are shown in Table 3 while Figures 4 to 7 show the simulated and experimental data for dissolution and growth of form I and II at different conditions, as well as the two types of nucleation. In order to validate the parameters estimated a leave-one-out cross-validation was performed using all the available experiments.

Table 3: Kinetic parameters estimated from the performed experiments.

Parameter	Value	Upper limit of the 95% interval of confidence	Lower limit of the 95% interval of confidence
$k_{dI} \left[ \frac{\mu m}{s} \right]$	$-3.45 \cdot 10^8$	$-1.38 \cdot 10^8$	$-5.6 \cdot 10^8$
$E_{dI} \left[ \frac{kJ}{mol} \right]$	48091	49587	46595
$d_I [-]$	0.65	0.69	0.61
$k_{dII} \left[ \frac{\mu m}{s} \right]$	-204	-101	-307
$E_{dII} \left[ \frac{kJ}{mol} \right]$	24276	25545	23001
$d_{II} [-]$	0.87	0.91	0.83
$k_{gI} \left[ \frac{kJ}{mol} \right]$	$6.20 \cdot 10^{13}$	$1.39 \cdot 10^{14}$	$8.84 \cdot 10^{12}$
$E_{gI} \left[ \frac{kJ}{mol} \right]$	90628	93378	87877
$g_I [-]$	0.82	0.88	0.75
$k_{gII} \left[ \frac{kJ}{mol} \right]$	0.00232	0.00449	0.00014
$E_{gII} \left[ \frac{kJ}{mol} \right]$	9466	11388	7445
$g_{II} [-]$	0.41	0.47	0.36
$k_s \left[ \frac{\#}{m^3 sec} \right]$	$7.92 \cdot 10^4$	$4.84 \cdot 10^6$	$1.34 \cdot 10^2$

$E_s \left[ \frac{kJ}{mol} \right]$	-42323	-24254	-60300
$s [-]$	3.95	4.86	3.04
$k_b \left[ \frac{\#}{m^3 sec} \right]$	$6.02 \cdot 10^{42}$	$1.85 \cdot 10^{70}$	$2.69 \cdot 10^{14}$
$E_b \left[ \frac{kJ}{mol} \right]$	179480	348390	164390
$b [-]$	0.1*	0.0129	-0.054

\* Lower bound of searching domain

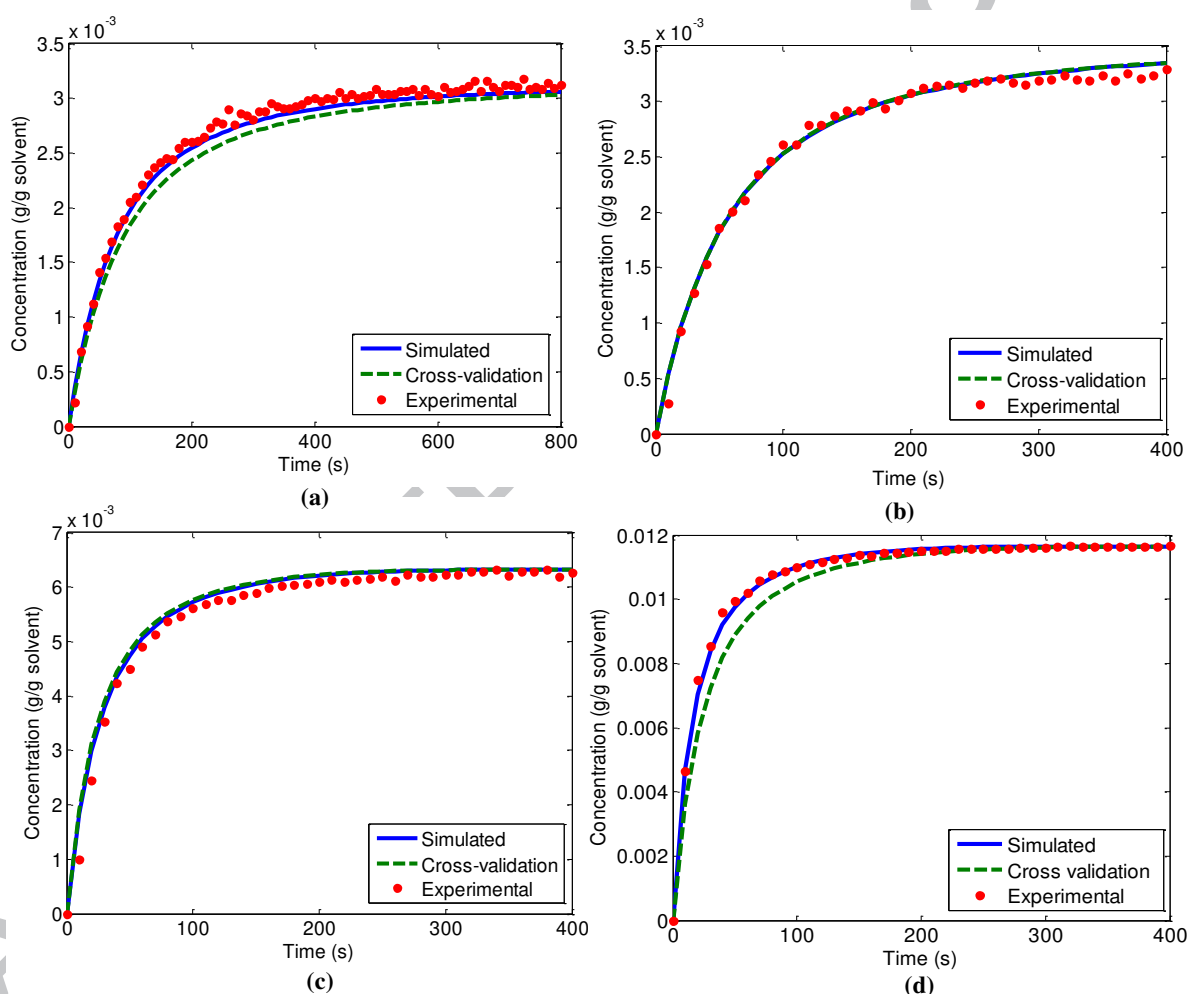
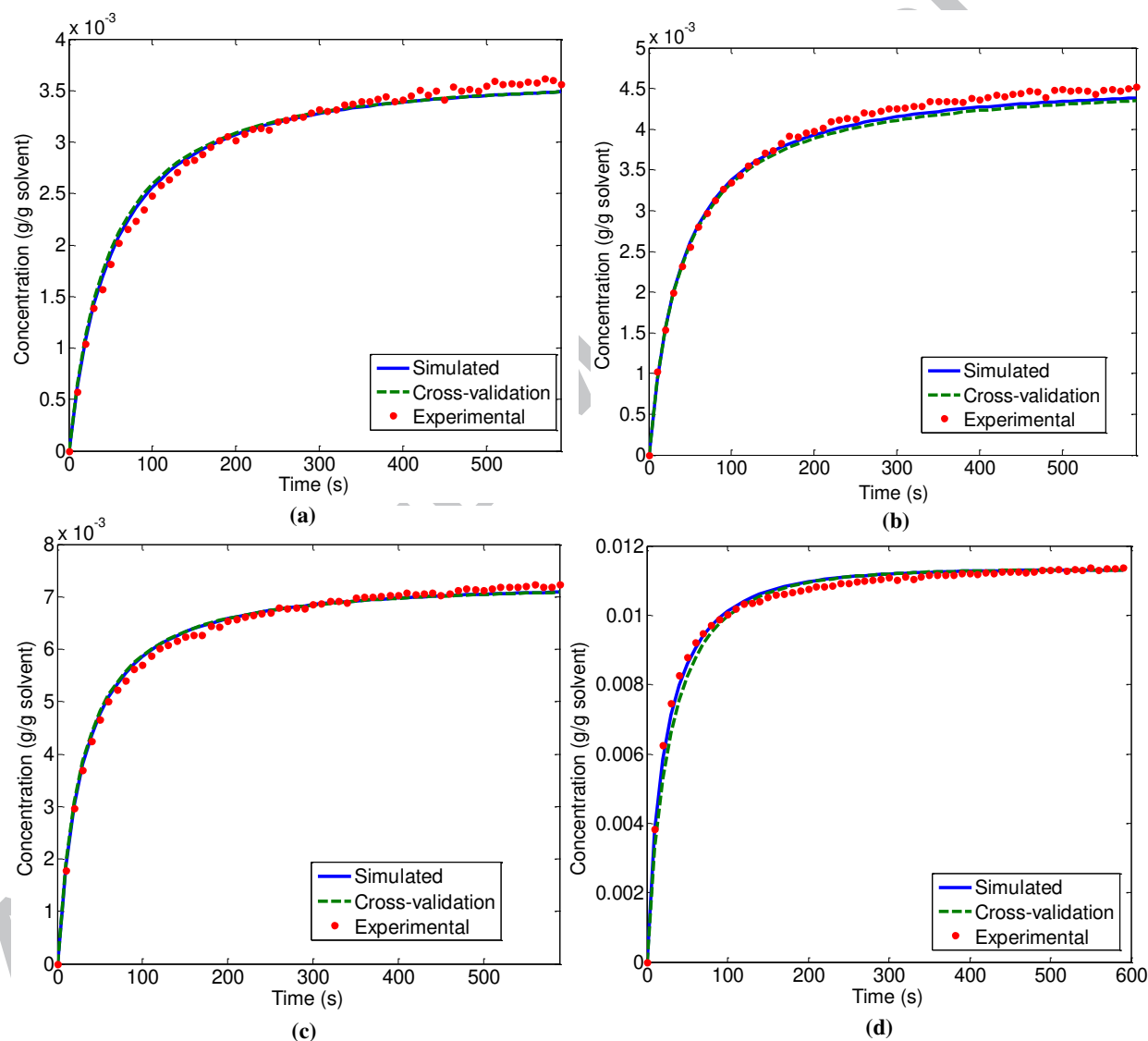


Figure 4: Dissolution experiments for form I: (a) Experimental, fitted and cross-validation concentration profile for the dissolution of form I at 10 °C (b) Experimental, fitted and cross-validation concentration profile for the dissolution of form I at 15 °C (c) Experimental, fitted and cross-validation concentration profile for the dissolution of form I at 25 °C (d) Experimental, fitted and cross-validation concentration profile for the dissolution of form I at 35 °C.

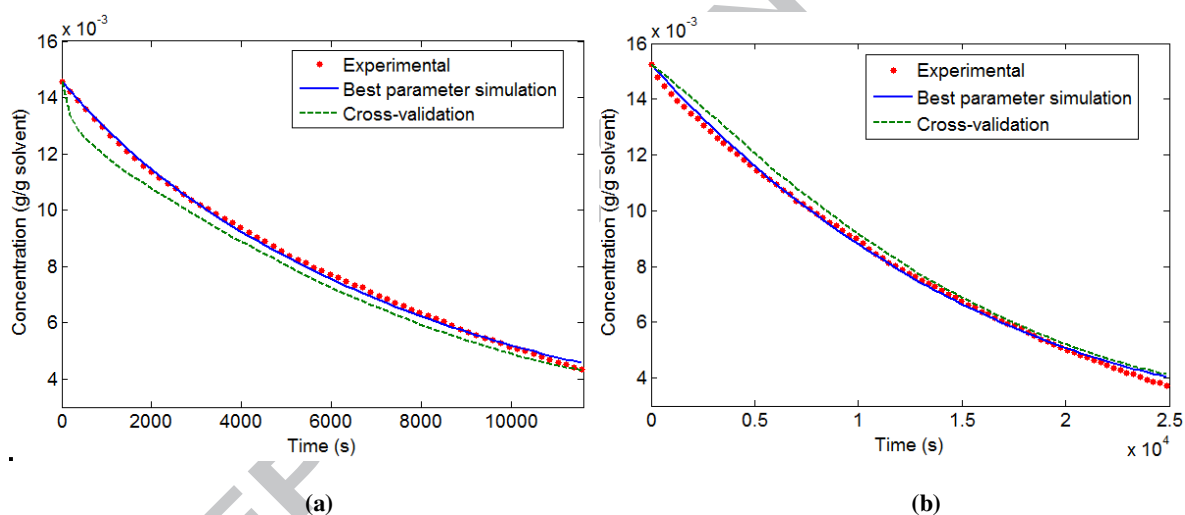
Figure 4 shows, for each dissolution experiment performed with form I, the experimental data for concentration together with the best fit (obtained using all the available experiments) and the calculated concentration for the cross validation (calculated using the parameters obtained by leaving that experiment out of the estimation).



**Figure 5: Dissolution experiments for form II: (a) Experimental, fitted and cross-validation concentration profile for the dissolution of form II at 10 °C (b) Experimental, fitted and cross-validation concentration profile for the dissolution of form II at 15 °C (c) Experimental, fitted and cross-validation concentration profile for the dissolution of**

form II at 25 °C (d) Experimental, fitted and cross-validation concentration profile for the dissolution of form II at 35 °C.

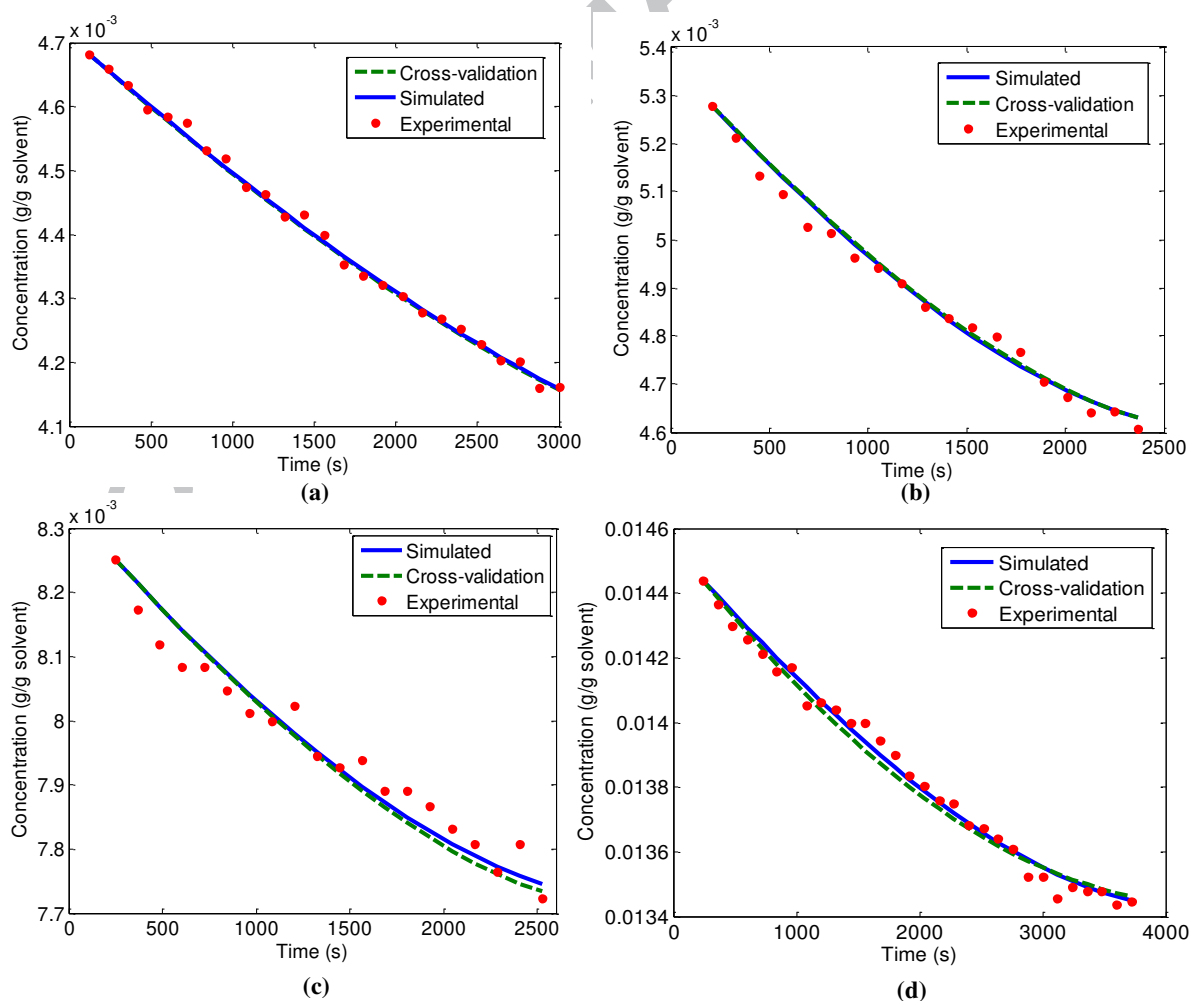
The same experimental, fitted and cross-validation concentrations are shown in Figure 5 for the dissolution of form II, in Figure 6 for the growth of form I and, finally, in Figure 7 for the growth of form II. The simulated concentrations for the dissolution of both forms seem to follow well the experimental values and the 95% confidence interval for all the estimated values are narrow (as shown in the third and fourth column of Table 3).



**Figure 6: Growth experiments for form I: (a) Experimental, fitted and cross-validation concentration profile for the growth of form I in a desupersaturation experiment performed with  $-0.1$  °C/min cooling rate (b) Experimental, fitted and cross-validation concentration profile for the growth of form I in a desupersaturation experiment performed with  $-0.05$  °C/min cooling rate.**

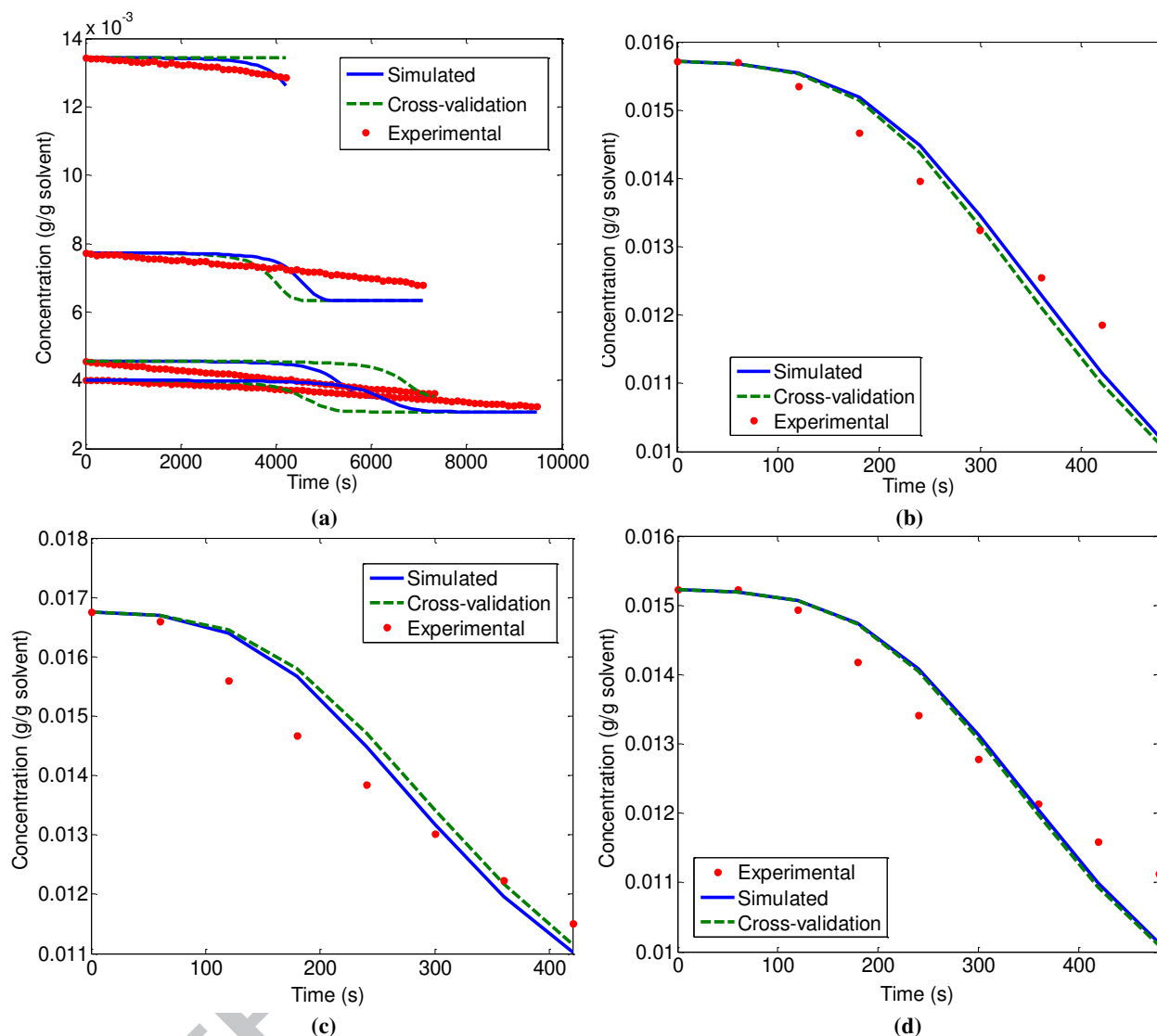
The cross validation still follows reasonably well the experimental data with the exception of the dissolution of form I at 10 and 35 °C (see Figure 4a and d). In these two cases, the trends of the cross-validation concentration slightly differ from the experimental values. There are two possible reasons for this behaviour; the first is simply the approximation of the crystal

size distribution to a Gaussian function that might generate an error in the evaluation of the initial crystal size distribution for these two specific experiments. The second reason might be a reduced sensitivity to the temperature change during the estimation performed leaving out the highest and lowest temperatures, due to the smaller temperature range in which the parameters are calculated (20 and 15 °C instead of 25 °C for the cross-validation of the experiments at 15 and 25 °C). Figure 6a also shows a deviation of the calculated cross-validation concentration from the experimental values, especially at the beginning of the profile. In this case, the difference is most certainly due to an experimental error on the determination of the initial crystal size distribution or to the approximation of the distribution itself with a Gaussian function, as the deviation is located close to the initial period.



**Figure 7: Growth experiments for form II: (a) Experimental, fitted and cross-validation concentration profile for the growth of form II at 10 °C (b) Experimental, fitted and cross-validation concentration profile for the growth of form II at 15 °C (c) Experimental, fitted and cross-validation concentration profile for the growth of form II at 25 °C (d) Experimental, fitted and cross-validation concentration profile for the growth of form II at 35 °C.**

However, the presence of an estimation error due to an imprecise initial crystal size distribution is not surprising considering how difficult is to obtain good and reliable measurement of the crystal size distribution with standard techniques such as the Malvern Mastersizer or 2D image analyses (Su et al. 2017; Ma et al. 2016). Figure 8 shows the simulated primary and secondary nucleation for form I.



**Figure 8: (a) Secondary nucleation of form I during transformation (four isothermal experiments): continuous line is simulated and dots are experimental data (b,c,d). Primary nucleation of form I after seeding (three experiments, similar conditions): continuous line is simulated and dots are experimental data.**

The difference between simulated and experimental concentration is higher compared to the experiments with growth and dissolution of both forms and the 95% confidence intervals are also broader (as shown in Table 3). This is due to the difficulty in estimating the kinetics parameters for a stochastic process such as nucleation and also because of the limited number of experimental data available.

One of the APFC experiments performed (Simone et al. 2014) was used to validate the set of parameters estimated. Seeding and dissolution cycle were simulated using the initial conditions shown in Table 4.

**Table 4: Initial conditions for the model validation shown in Figure 9 and 10.**

Validation initial conditions parameters	Value
Seeding temperature [ $^{\circ}C$ ]	37.26
Solute concentration [ $\frac{g}{g \text{ solvent}}$ ]	0.0151
Mass of seed crystals [ $g$ ]	0.61
Form II in the seed crystals [% w/w]	60
Mass of solvent [ $g$ ]	400
Form I sigma, $\sigma_I$ [ $\mu m$ ]	6
Form I mean, $\mu_I$ [ $\mu m$ ]	69
Form II sigma, $\sigma_{II}$ [ $\mu m$ ]	225
Form II mean, $\mu_{II}$ [ $\mu m$ ]	75

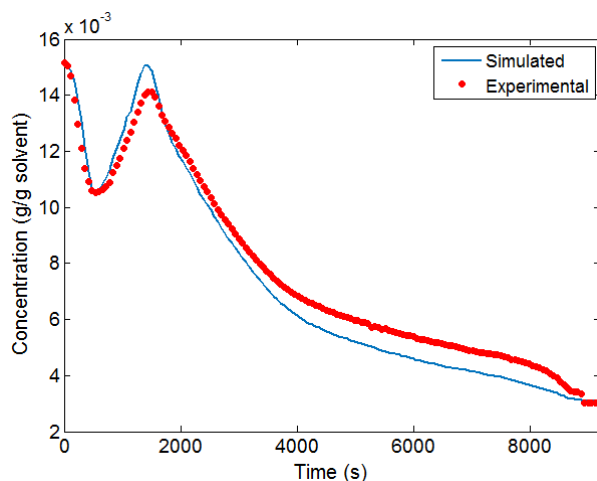
The mean  $\mu$  and the sigma  $\sigma$  of the crystal size distribution for the validation experiment were estimated as follows:

$$\mu = \frac{L_{max} + L_{min}}{2} \quad (19)$$

$$\sigma = \frac{L_{max} - L_{min}}{2} \quad (20)$$

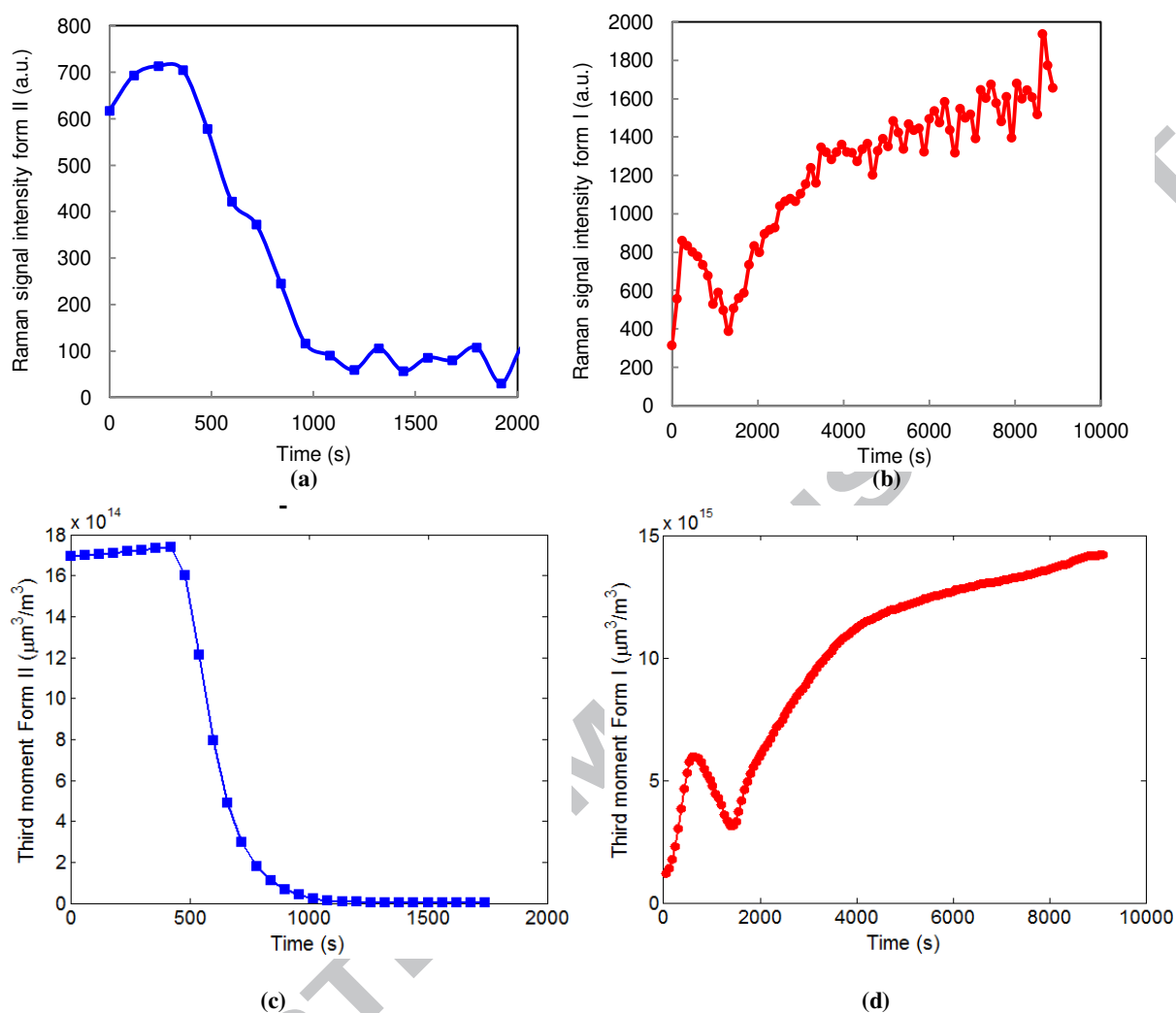
where  $L_{max}$  and  $L_{min}$  are the maximum and minimum sizes of the sieves used to separate the seeds (the Malvern Mastersizer was not used for this sample).

The results of the validation experiment are shown in Figure 9 and 10.



**Figure 9: Model validation: experimental and simulated data for an APFC experiment. Temperature plotted with experimental and simulated concentrations.**

Despite the difficulty in obtaining a good estimation for the nucleation kinetics the simulated concentration for the validation experiments correctly follows the experimental data and the trend of the first moment of form II is similar to the corresponding Raman signal. A discrepancy is present in the cooling section and it is probably due to the uncertainty in the estimation of primary nucleation (the simulated first moment of form I is higher than the actual one, and therefore, the growth is overestimated). However, the maximum percentage error on the concentration measurement is around 16% and it is localized in the cooling section, and the time of complete dissolution of the undesired form is calculated correctly.



**Figure 10: Model validation: experimental and simulated data for an APFC experiment. Raman signal intensities for form I and II during the experiment (a and b) compared to the simulated third moment of both polymorphs.**

Figure 10 shows the third moments of form I and II compared to the Raman signals of the specific peaks of those forms. The Raman intensity is proportional with the amount of solid in suspension and, therefore, can be directly compared with the third moments of the polymorphs. In fact, the third moment of the distribution ( $\mu_3$ ) is proportional to the specific volume of crystal population ( $V_c$ ) and is one of the infinite moments of CSD, generally defined as:

$$\mu_k = \int_0^{\infty} L^k f(L, t) dL \rightarrow V_c = k_v \mu_3 \quad (21)$$

This means that the model is able to simulate and predict well the APFC dissolution cycle so it is suitable for optimization. Form II slightly grows during the first seconds after seeding and then is dissolved by the heating cycle. The amount of form I increases because of nucleation, then decreases during the dissolution cycle because of partial dissolution of form I with form II and then increases again due to growth.

### 3.2 Process optimization for polymorphic crystallization

Optimization was performed using the kinetic parameters to find the optimal temperature profile that eliminates form II and maximizes the size of the crystals of form I at the end of the batch. The batch time was discretized in fifty time intervals of equal duration and the temperature profile optimization was performed by applying the *ES-CMA* global optimization algorithm. The results were further refined by performing a second optimization using the global optimizer's crude optimum as starting point, applying the Matlab *fmincon* function (SQP algorithm). The initial temperature of seeding was fixed at around 37 °C. The problem is formulated as follows:

$$\min_{T(k)} (-\bar{L}_{I,end}) \quad (22)$$

Subject to:

$$-0.5 \leq \frac{dT}{dt} \leq 0.5 \quad (23)$$

$$11 \leq T \leq 45 \quad (24)$$

$$C_{end} \leq C_{max,end} = 0.005 \text{ g/g solvent} \quad (25)$$

$$\mu_{II1,end} = 0 \quad (26)$$

where  $T$  is the temperature defined in  $^{\circ}\text{C}$ ,  $\frac{dT}{dt}$  the heating/cooling rates in  $^{\circ}\text{C}/\text{min}$ ,  $C_{end}$  the solute concentration at the end of the batch (g/g solvent) and  $\mu_{II1,end}$  the second moment of form II at the end of the batch.

A 20 minutes stabilization time was applied: the final temperature was kept constant to allow the consumption of the remained supersaturation. The initial conditions used for the optimization are shown in Table 5.

**Table 5: Initial conditions for the optimization.**

Validation initial conditions parameters	Value
Seeding temperature [ $^{\circ}\text{C}$ ]	37.26
Solute concentration [ $\frac{\text{g}}{\text{g solvent}}$ ]	0.015
Mass of solid [ $\text{g}$ ]	0.6
Form II in the seed crystals [% w/w]	40
Mass of solvent [ $\text{g}$ ]	400
Form I sigma, $\sigma_I$ [ $\mu\text{m}$ ]	10
Form I mean, $\mu_I$ [ $\mu\text{m}$ ]	50
Form II sigma, $\sigma_{II}$ [ $\mu\text{m}$ ]	10
Form II mean, $\mu_{II}$ [ $\mu\text{m}$ ]	50

The results of the optimization (shown in Figure 11a to c) demonstrate that a heating step is not only required to eliminate form II but also allows larger crystal size of form I at the end of

the batch: imposing only cooling in the optimization code ( $-0.5^{\circ}\frac{C}{min} \leq \frac{dT}{dt} \leq -0.001^{\circ}C/min$ ) resulted in lower crystal size, although all the metastable form naturally converted to the stable one by the end of the batch.

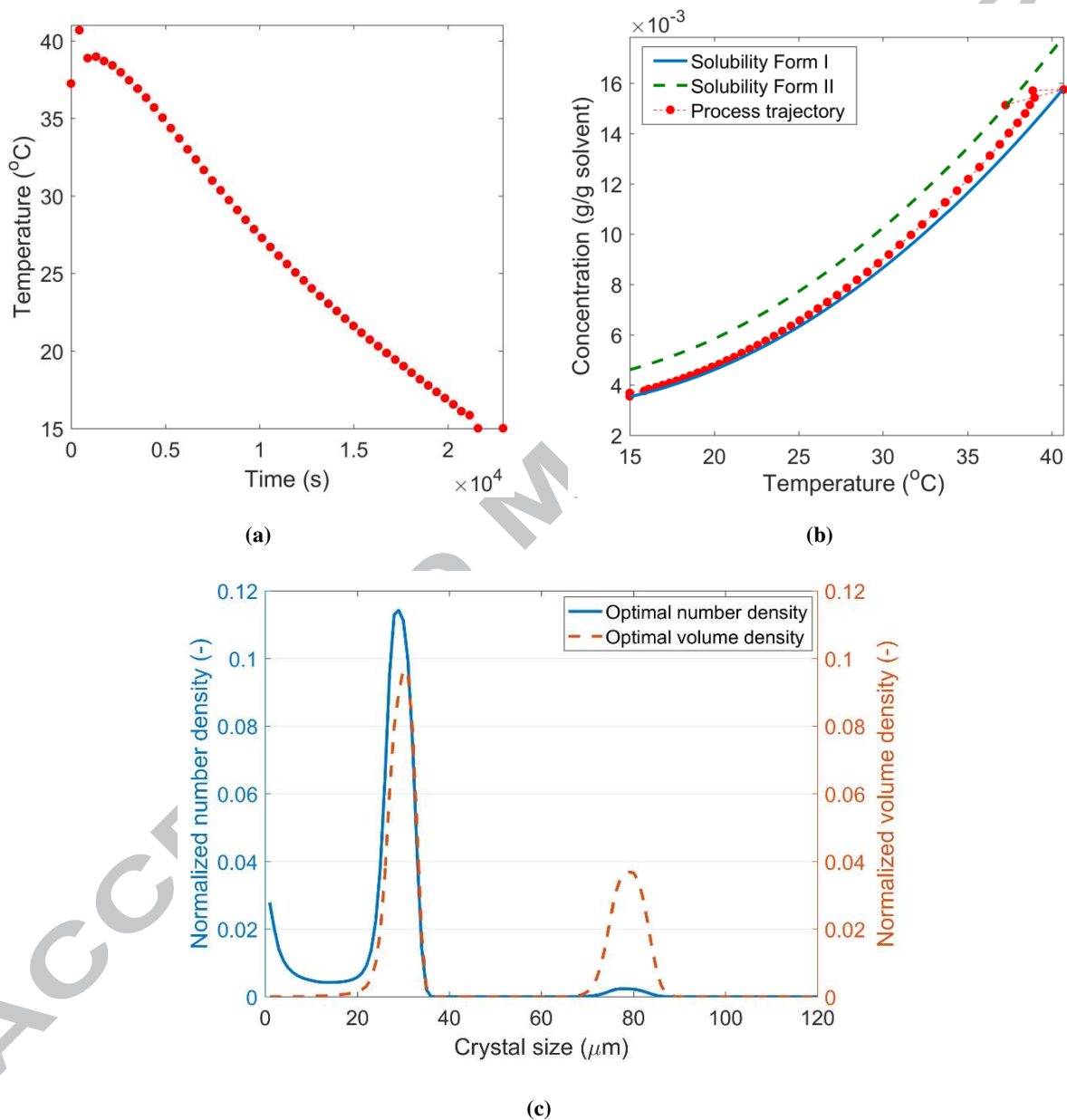


Figure 11: (a) Optimal temperature profile plotted against time; (b) Optimal operating trajectory plotted in the phase diagram; (c) Optimal crystal size distribution (CSD) obtained at the end of the batch.

The value of the objective function calculated was quite low compare to experimental results

( $\frac{\mu_{1,opt}}{\mu_{0,opt}} = 26.5 \mu m$ , corresponding to a  $\frac{\mu_{4,opt}}{\mu_{3,opt}} = 44.9 \mu m$  versus around 100-150  $\mu m$  obtained

experimentally at the end of the batch) but this most probably is due to the uncertainty of the parameters estimated for nucleation of form I. The heating step at the very beginning of the optimal temperature profile is not only has the effect of dissolving form II, but it is also beneficial to improve the crystal size distribution of form I.

The presence of heating steps in optimized batch crystallization processes was already found by other authors (Majumder, Nagy 2013; Qamar et al. 2010; Yeom et al. 2013; Nagy et al. 2011) as a result of the inclusion of the dissolution kinetics in the PBEs. In those cases heating can correct a non-optimal seeding and allows a better final CSD. After the heating step in the optimal profile calculated in this work the temperature is kept high in order to allow growth of the form I crystal and then drops in the end to reach the desired yield. Figure 11b shows the optimized temperature profile in the phase diagram: the solute concentration is kept below the solubility of the metastable form to avoid its further nucleation and above the solubility of the stable form to allow its growth during the cooling phase.

#### 4. Conclusions

The active polymorphic feedback control (APFC) is a strategy that detects and eliminates the metastable polymorph after nucleation of a mixture or contaminated seeding (Simone et al. 2014). The approach uses a combination of Raman spectroscopy to detect the metastable polymorph and trigger a dissolution cycle to eliminate it, and then applies ATR-UV/Vis spectroscopy to grow the remaining crystals of the stable form through supersaturation control. Despite being very efficient in obtaining the pure stable polymorph, this model-free control does not lead to optimal crystal size distribution of the product of stable polymorph at

the end of the batch. In fact, the size distribution of the crystals of the stable form after the dissolution cycle is not controlled and it might not be the optimal to allow a good quality CSD at the end of the supersaturation control. For this reason a model-based active polymorphic control (mbAPC), that allows both the elimination of the metastable form and larger crystals of the stable form at the end of the batch, was developed. The kinetic parameters that are needed to describe the mbAPC for ortho-aminobenzoic acid (dissolution and growth of form I and II, secondary nucleation of form I) were estimated and validated using the data from seeded experiments. A specific design of experiments was performed to estimate each parameter separately and therefore, to avoid correlations between them, as well as to simplify the parameter estimation. All the parameters estimated presented a narrow 95 % confidence interval, apart from the nucleation of the stable form, probably because of the stochastic nature of this phenomenon.

After the parameter estimation, optimization was performed. It was found that the dissolution cycle, normally induced by the APFC, not only allows the elimination of the metastable form II, but it is also beneficial to obtain larger crystals of form I at the end of the batch. This is in accordance with experiments as well as with the results of other optimization studies where dissolution was included in the model.

In conclusion, the proposed mbAPC can be useful for the design of batch crystallization processes of polymorphic systems as it allows obtaining large crystals of the stable form, even in case of in situ nucleation of a mixture of the stable and metastable polymorphs or erroneous seeding.

## Acknowledgements

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**Highlights:**

A model-based active polymorphic control for batch crystallization is proposed;

A high resolution finite volume method is used to solve population balance equations;

Optimization was performed to estimate the best temperature profile;

The dissolution cycle is beneficial to obtain larger crystals of the stable polymorph.

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