Ferromagnetism in Cr-doped topological insulator TISbTe$_2$

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We have synthesized a new ferromagnetic topological insulator by doping Cr to the ternary topological-insulator material TISbTe$_2$. Single crystals of Tl$_{1-x}$Cr$_x$SbTe$_2$ were grown by a melting method and it was found that Cr can be incorporated into the TISbTe$_2$ matrix only within the solubility limit of about 1%. The Curie temperature $T_C$ was found to increase with the Cr content but remained relatively low, with the maximum value of about 4 K. The easy axis was identified to be the $c$-axis and the saturation moment was 2.8 $\mu_B$ (Bohr magneton) at 1.8 K. The in-plane resistivity of all the samples studied showed metallic behavior with $p$-type carriers. Shubnikov-de Hass oscillations were observed in samples with the Cr-doping level of up to 0.76%. We also tried to induce ferromagnetism in TIBiTe$_2$ by doping Cr, but no ferromagnetism was observed in Cr-doped TIBiTe$_2$ crystals within the solubility limit of Cr which turned out to be also about 1%. © 2015 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution 3.0 Unported License.

Three-dimensional (3D) topological insulators (TIs) are a class of materials characterized by a nontrivial $Z_2$ topology of the bulk wave function, where an insulating bulk hosts a linearly dispersing surface state protected by the time-reversal symmetry. Theoretical studies showed that the thallium-based ternary chalcogenides TlSbTe$_3$, TlBiSe$_2$, and TlBiTe$_2$ are 3D TIs with a single-Dirac-cone surface state at the $\Gamma$ point. Experimentally, TlBiSe$_2$ and TlBiTe$_2$ have been confirmed to be topological insulators; in particular, it was found that the surface-state structure of TlBiTe$_2$ is similar to that in Bi$_2$Se$_3$, making it suitable for studying the Dirac-cone physics in a simple setting with a large bulk band gap or $\sim$0.35 eV. Interestingly, a topological phase transition was found in the TlBi(Se$_{1-x}$S$_x$)$_2$ solid solution, which provides a platform for realizing the 3D Dirac semimetal. Furthermore, if one could induce ferromagnetism at the topological phase transition point of this solid-solution system by doping a magnetic element, the broken time-reversal symmetry would lead the emergence of a Weyl semimetal. Theoretical studies have also shown that quantum anomalous Hall effect could occur in TI-based TIs when doped with transition metals (TMs). Therefore, TM doping to TI-based TIs would be important for the pursuit of novel quantum states of matter. In the past, TM doping to various TI materials has been tested: Successful observations of ferromagnetism were reported for Mn-doped Bi$_2$Te$_3$, V- or Cr-doped Sb$_2$Te$_3$, Fe-doped Bi$_2$Te$_3$, and V- or Cr-doped (Bi,Sb)$_2$Te$_3$. So far, such ferromagnetic TIs have been found only in tetradymite TI materials, and ferromagnetism has never been observed in a TI-based ternary TI. Here, we report our explorations of ferromagnetism in TISbTe$_2$ and TIBiTe$_2$ by TM doping. We found that Cr doping can induce ferromagnetism in TISbTe$_2$, but not in TIBiTe$_2$.

The single crystals of Cr-doped TISbTe$_2$ were grown by a melting method using elemental shots of Tl (99.99%), Sb (99.9999%), and Te (99.9999%) as powders of Cr (99.9%) as
starting materials. Mixtures of those materials with the nominal composition of Tl$_{1-x}$Cr$_x$SbTe$_2$ ($x = 0.00, 0.007, 0.01, 0.02, 0.03$) were prepared with the total weight of 4.0 g and were sealed in evacuated quartz tubes; we also prepared a batch of Te-rich composition, Tl$_{0.98}$Cr$_{0.02}$SbTe$_2$, for comparison. The quartz tubes were heated and kept at 700 °C for 48 h with intermittent shaking to ensure homogeneity of the melt, followed by cooling slowly to 450 °C. Single crystals with the lateral dimension of up to a few centimeters can be obtained by cleaving along the (00l) plane. We also synthesized Tl$_{1-x}$Cr$_x$BiTe$_2$ ($x = 0.01, 0.02, 0.03$) crystals with the same method (Bi purity was 99.9999%). Note that before the synthesis of our samples, we performed surface cleaning procedures to remove the oxide layers formed in air on the raw shots of Tl, Sb, and Bi: Tl shots are annealed in hydrogen atmosphere at 230 °C for 2 h; Sb and Bi shots are washed with diluted HNO$_3$.

The crystal structure of each sample was checked by powder X-ray diffraction (XRD) using Rigaku Ultima-IV diffractometer with Cu $K\alpha$ emission, which was performed on powders obtained by crushing the crystals. The actual Cr content in the samples was analyzed by inductively coupled plasma atomic-emission spectroscopy (ICP-AES). Magnetization measurements were carried out using a SQUID magnetometer (Quantum Design Magnetic Property Measurement System) and a vibrating sample magnetometer (Quantum Design Physical Property Measurement System). The in-plane transport properties were measured in magnetic fields up to 14 T with a standard six-probe method to record the longitudinal resistivity $\rho_{xx}$ and the Hall resistivity $\rho_{yx}$ simultaneously. The single crystal samples for transport measurements were cut into a rectangular shape with a typical size of $2 \times 0.5 \times 0.2$ mm$^3$ and electrical contacts were made by using room-temperature-cured silver paste.

Motivated by a theoretical proposal that ferromagnetism should be induced by TM doping in TlBiX$_2$ ($X = \text{Te and Se}$) and that Cr would be the most promising element to induce ferromagnetic order, we started our explorations by growing Tl$_{1-x}$Cr$_x$BiTe$_2$. Figure 1(a) shows the powder XRD patterns of the grown Tl$_{1-x}$Cr$_x$BiTe$_2$ samples with nominal $x$ values of 0.01 and 0.02. One can see that the $x = 0.01$ sample is single phase and all the diffraction peaks can be well indexed to the rhombohedral structure of TlBiTe$_2$ with space group R-3m (we use the hexagonal notation). However, peaks from an impurity phase, Cr$_2$Te$_3$, show up in the data for $x = 0.02$ as indicated by asterisks in Fig. 1(a). This phase is known to be ferromagnetic with the Curie temperature of 165 K. In the magnetization data shown in Fig. 1(b), one can see that the $x = 0.02$ sample indeed presents ferromagnetism below 165 K with a clear magnetic hysteresis [inset of Fig. 1(b)]. On the other hand, no clear ferromagnetism was observed down to 1.8 K in the $x = 0.01$ sample which is free from the Cr$_2$Te$_3$ impurity phase. Therefore, one may conclude that TlBiTe$_2$ has a relatively low solubility limit of about 1% for Cr and that ferromagnetic order is not established above 1.8 K.
within this solubility limit. For TlBiTe$_2$, we also tried doping of other TM elements, Mn, Fe, and Ni, but none of them were found to induce ferromagnetism.

After obtaining these negative results on TlBiTe$_2$, we switched to work on TlSbTe$_2$. Although the surface state observation has not been successful for TlSbTe$_2$ by angle-resolved photoemission spectroscopy because of its $p$-type nature, there is no reason to doubt its topological nature. Figure 2 shows the powder XRD patterns of Tl$_{1-x}$Cr$_x$SbTe$_2$, which were obtained on crushed crystals. For this Tl$_{1-x}$Cr$_x$SbTe$_2$ system, we show the actual $x$ values determined by the ICP-AES analysis (see Table I) except for the nominal $x = 0.03$ sample which was found to contain the Cr$_2$Te$_3$ impurity phase; all other samples with $x \leq 0.0092$ are single phase with the expected rhombohedral structure of TlSbTe$_2$ (space group R-3m). The samples with the highest actual composition of Cr in the present series, $x = 0.0092$, were obtained from the batches with the nominal $x$ value of 0.02. This suggests that the solubility limit of Cr in TlSbTe$_2$ is about 1%, which is similar to the case of TlBiTe$_2$. We note that the sample indicated as “$x = 0.0092$ (TR)” was grown from the Te-rich nominal composition of Tl$_{0.99}$Cr$_{0.02}$SbTe$_2$ and is expected to contain more Te antisite defects compared to other samples. Indeed, the ICP-AES analysis (Table I) suggests that some of the Tl sites are occupied by Te in this sample; also, as we show later, its hole density was found to be the highest among the present series. The purpose of growing the $x = 0.0092$ (TR) sample was to see the effect of hole density on the Curie temperature in ferromagnetic samples.

The temperature dependences of the magnetization $M$ measured in 0.1 T are shown in Fig. 3(a) for the single-phase samples of Tl$_{1-x}$Cr$_x$SbTe$_2$. The rapid increase of the magnetization at low temperature points to a ferromagnetic ordering. The Curie temperature $\theta_C$ can be determined from

![FIG. 2. Powder XRD patterns of the series of Tl$_{1-x}$Cr$_x$SbTe$_2$ samples grown in this work; asterisks mark the peaks from the Cr$_2$Te$_3$ impurity phase, which were observed only in samples with nominal $x$ values larger than 0.02. The vertical axis in panel (a) is linear, while that in panel (b) is logarithmic.](image-url)

**TABLE I.** Actual compositions of the Cr-doped TlSbTe$_2$ crystals determined from ICP-AES analysis. Since ICP-AES analysis only gives compositional ratios of the constituent elements, the composition values within each sample are determined by setting their sum to be 4.

<table>
<thead>
<tr>
<th>Nominal composition</th>
<th>Tl</th>
<th>Cr</th>
<th>Sb</th>
<th>Te</th>
</tr>
</thead>
<tbody>
<tr>
<td>Tl$<em>{0.99}$Cr$</em>{0.007}$SbTe$_2$</td>
<td>0.9788</td>
<td>0.0049</td>
<td>1.0453</td>
<td>1.9710</td>
</tr>
<tr>
<td>Tl$<em>{0.99}$Cr$</em>{0.01}$SbTe$_2$</td>
<td>0.9865</td>
<td>0.0076</td>
<td>1.0219</td>
<td>1.9840</td>
</tr>
<tr>
<td>Tl$<em>{0.98}$Cr$</em>{0.02}$SbTe$_2$</td>
<td>0.9816</td>
<td>0.0092</td>
<td>1.0220</td>
<td>1.9873</td>
</tr>
<tr>
<td>Tl$<em>{0.98}$Cr$</em>{0.02}$SbTe$_2$.2</td>
<td>0.9025</td>
<td>0.0092</td>
<td>1.0778</td>
<td>2.0106</td>
</tr>
</tbody>
</table>
the Curie-Weiss law,

\[ M = \frac{C}{T - \theta_C} + M_0, \]

by plotting \(1/(M - M_0)\) vs \(T\) (\(C\) is a constant and \(M_0\) is the background determined from the high temperature data). For example, the data for the \(x = 0.0049\) sample plotted in this way [Fig. 3(b)] can be well fitted by a straight line, whose intercept on the \(T\) axis gives \(\theta_C\) of 0.8 K; this \(\theta_C\) increases to 2.5 and 3.1 K for \(x = 0.0076\) and 0.0092, respectively. The relationship between \(\theta_C\) and \(x\) for those three samples is shown in the inset of Fig. 3(b), which shows a nearly linear trend. Importantly, the \(x = 0.0092\) sample which is expected to have a higher hole density presented the highest \(\theta_C\) of 4.1 K, suggesting that \(\theta_C\) is determined not only by the density of local moments but also by the density of mobile carriers which would mediate the coupling between local moments. Similar results were reported for Cr- or V-doped \((\text{Bi}_{1-x}\text{Sb})_2\text{Te}_3\).

To corroborate the establishment of ferromagnetism in \(\text{Tl}_{1-x}\text{Cr}_x\text{SbTe}_2\), we measured \(M(B)\) curves; Fig. 4 shows the data for all the single-phase samples at 1.8 K. Clear magnetic hysteresis was observed in all samples except for \(x = 0.0049\); note that \(\theta_C\) obtained for \(x = 0.0049\) was less than 1 K, and hence a hysteresis is not expected for this sample at 1.8 K. The \(x = 0.0092\) sample
sample having the highest $\theta_C$ of 4.1 K presents the largest remnant magnetization of $\sim 0.3$ $\mu_\text{B}/\text{Cr}$ and the coercive field $B_C$ of 23 mT. This $B_C$ is comparable to that in Mn-doped Bi$_2$Te$_3$ (35 mT)\textsuperscript{19} and in Cr-doped Sb$_2$Te$_3$ (10 mT)\textsuperscript{20} but is much smaller than that in V-doped Sb$_2$Te$_3$ (1.2 T)\textsuperscript{21}. The inset of Fig. 4 shows the $M(B)$ curve at 1.8 K for $x = 0.0092$ measured up to 9 T applied parallel to the $c$-axis; the saturated magnetic moment is 2.8 $\mu_\text{B}/\text{Cr}$, which is a bit smaller than the expected magnetic moment of Cr$^{3+}$ (3.9 $\mu_\text{B}$). Note that Cr is antiferromagnetic\textsuperscript{28} and its possible clustering cannot explain the observed ferromagnetism.

Figure 5(a) shows the $M(B)$ curves for $x = 0.0092$ (TR) measured at 1.8 K with the magnetic field directions of $B//ab$ and $B//c$ from which one can easily see that the magnetic easy axis is the $c$-axis. This easy axis direction is the same as that reported for Mn-doped Bi$_2$Te$_3$ (Ref. 19) and for V- or Cr-doped Sb$_2$Te$_3$\textsuperscript{20,21}. We also measured the $M(B)$ curves for $x = 0.0092$ (TR) at various temperatures as shown in Fig. 5(b); in these measurements, the sample was first heated to 20 K and then cooled to the target temperature in 0 T to guarantee perfect demagnetization. The hysteresis disappears between 4.5 and 6 K, which is consistent with $\theta_C = 4.5$ K determined from the $M(T)$ data.

Now, we briefly discuss the transport data of Tl$_{1-x}$Cr$_x$SbTe$_2$. The temperature dependences of $\rho_{xx}$ in 0 T and the magnetic-field dependences of $\rho_{y,x}$ at 1.8 K are shown in Fig. 6. The absence of a clear anomalous Hall signal in our $\rho_{y,x}(B)$ data is probably due to the very small magnetization associated with ferromagnetism (up to $\sim 0.003$ $\mu_\text{B}$/f.u. at 1.8 K). It is worth noting that the $x = 0.0092$ (TR) sample shows the smallest positive slope of $\rho_{y,x}(B)$, which means that the hole density is the largest among all the samples. To be concrete, the hole density $p$ estimated

**FIG. 5.** (a) $M(B)$ curves observed in the $x = 0.0092$ (TR) sample at 1.8 K for $B//ab$ and $B//c$. (b) $M(B)$ curves for $B//c$ in the same sample at various temperatures.

**FIG. 6.** (a) Temperature dependences of $\rho_{xx}$ in the Tl$_{1-x}$Cr$_x$SbTe$_2$ crystals. (b) Magnetic-field dependences of $\rho_{y,x}$ at 1.8 K.
from $1/eR_H$ ($R_H$ is the low-field Hall coefficient and $e$ is the elementary charge) is $3.2 \times 10^{19}$, $1.8 \times 10^{19}$, $0.79 \times 10^{19}$, and $3.9 \times 10^{19}$ cm$^{-3}$ for $x = 0.0000$, 0.0049, 0.0076, and 0.0092(TR), respectively. Correspondently, the mobilities for these samples are calculated to be 1502, 1509, 2038, 2232, and 1282 cm$^2$/Vs. The decreasing trend in $\rho$ with increasing $x$ is reasonable, because Cr$^{3+}$ substitution for Ti$^4+$ leads to electron doping. The increase in mobility in samples with higher $x$ [except for $x = 0.0092$ (TR)] suggests that the electron-electron scattering is dominant over the impurity scattering on Cr$^{3+}$ ions.

We observed clear Shubnikov-de Hass (SdH) oscillations in samples with $x \leq 0.0076$. Figure 7 shows the oscillations in $d\rho_y/dB$ in which the main oscillation frequency $F$ is 128–137 T and does not change much with $x$; the corresponding hole density (assuming a spherical Fermi surface) is $0.8–0.9 \times 10^{19}$ cm$^{-3}$. As is most obvious in the data for $x = 0.0049$, the SdH oscillations present beating, suggesting the existence of more than one Fermi surfaces with similar sizes.

In conclusion, ferromagnetism was observed in Tl$_{1-x}$Cr$_x$SbTe$_2$ but not in Tl$_{1-x}$Cr$_x$BiTe$_2$ above 1.8 K. The solubility limit of Cr in both TlSbTe$_2$ and TiBiTe$_2$ is found to be about 1% and the Cr$_2$Te$_3$ impurity phase appears when the Cr content exceeds this solubility limit. The Curie temperature $\theta_C$ of Tl$_{1-x}$Cr$_x$SbTe$_2$ increases with both $x$ and the hole density. The highest $\theta_C$ of about 4 K was observed in $x = 0.0092$ (TR) sample with the hole concentration of $3.9 \times 10^{19}$ cm$^{-3}$.

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