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Nanopatterning of a Covalent Organic Framework Host-Guest System

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1. Synthesis of COF-1 on HOPG

Commercially available benzene-1,4-diboronic acid (≥ 95%) and 1-octanoic acid (≥ 99%) were used as received (Sigma-Aldrich). HOPG samples were purchased from Momentive performance materials Inc., quality grade ZYB.

A 1.5 mg/ml suspension of benzene-1,4-diboronic acid in octanoic acid was sonicated for 15 minutes before dropcasting on a freshly cleaved HOPG sample. The sample was placed inside a glass reactor together with a glass vial containing 1 ml of water. On the vial is a cap with a small hole, which allows slow evaporation of water from the vial inside the reactor.

The reactor is built up out of two petri dishes, 1 large and 1 small, glued together with PDMS. Small grains of sand fill the empty space between the two petri dishes. Before starting the reaction, a third petri dish with a diameter that fits between the other two is placed up-side down and pressed into the sand. In this way, the water vapor that is released in the reactor does not evaporate immediately but slowly diffuses through the sand. This ensures equilibrium conditions and reversibility of the reaction.

The reaction is done at ≈ 100°C for approximately 1 hour. When all visual signs of water have disappeared from the reactor, the sample is removed and analyzed with STM.
2. Geometry optimization of COF-1 and calculation of adsorption energy of C\textsubscript{60} on COF-1/graphite using DFT

Method
All the calculations were done with CP2K software\textsuperscript{1} using density functional theory (PBE functional\textsuperscript{2}) with empirical dispersion correction (D2) proposed by Grimme.\textsuperscript{3} Pseudopotentials by Goedecker, Teter and Hutter\textsuperscript{4} from the CP2K database were used. Double-\(\zeta\) valence polarized basis sets (DZVP) were used for all atoms. Additionally, five of the lowest-energy structures of the di-boronic acid network were tested using triple-\(\zeta\) valence polarized basis sets (TZVP); their relative energies differed from the DZVP results by no more than 0.5 kJ mol\(^{-1}\), and binding energies with respect to diboronic acid and gaseous water differed from DZVP by no more than 4 kJ mol\(^{-1}\); this confirms that the DZVP basis set is sufficient. The cutoff for electron density in the auxiliary basis set was 600 Ry.

The structure of the diboronic acid derived covalent organic framework (COF) was modelled using periodic boundary conditions by maintaining the hexagonal symmetry of the unit cell (the angle \(\alpha\) between the lattice vectors A and B was kept equal to 60°). Adsorption of COF and C\textsubscript{60} on graphite was modelled using a single layer of graphite, within periodic boundary conditions. The lattice of the COF was adjusted to be commensurate with graphite; two commensurate cells were constructed in this way: one unit cell of the COF on a 6x6 graphite lattice (114 atoms, or 174 atoms when C\textsubscript{60} was added) and a 2x2 replicated unit cell of the COF on a 13x13 graphite lattice (506 atoms, or 746 atoms when four C\textsubscript{60} molecules were added) – see the following Section for the details of the structures. The \(z\) parameter of the cell was set to 12 Å (20 Å if the C\textsubscript{60} was present), to ensure a large enough vacuum layer above the adsorbate. The graphite layer was fixed at the ideal atom positions, while the COF and the C\textsubscript{60} were allowed to relax. Adsorption energies were calculated and corrected for the basis set superposition error (BSSE) using the counterpoise method.\textsuperscript{5}

Geometry optimization

Table S1: Energies of the benzene-1,4-diboronic acid-based COF. (The data for the most stable structure are in bold).

<table>
<thead>
<tr>
<th>Lattice parameter, Å</th>
<th>Energy relative to the minimum-energy structure, kJ mol(^{-1})</th>
<th>Enthalpy of formation relative to benzene-1,4-diboronic acid and gas-phase water, kJ mol(^{-1})</th>
</tr>
</thead>
<tbody>
<tr>
<td>14.6</td>
<td>86.4</td>
<td>127.2</td>
</tr>
<tr>
<td>14.7</td>
<td>56.4</td>
<td>97.1</td>
</tr>
<tr>
<td>14.76</td>
<td>41.5</td>
<td>82.2</td>
</tr>
<tr>
<td>14.8</td>
<td>32.9</td>
<td>73.6</td>
</tr>
</tbody>
</table>
The optimum lattice parameter of COF-1 was found by varying the value of the parameter $A$ (=B) between 14.6 and 16.1 Å in steps of 0.1 Å, while maintaining the hexagonal symmetry of the cell (with the angle between A and B equal to $60^\circ$, figure S2). The lowest-energy value of the lattice parameter was found to be 15.1 Å.

The enthalpy change for the formation of COF-1 and gaseous water from benzene-1,4-diboronic acid is positive. However, the entropy term is likely to be large and negative, since water vapour is produced in the reaction. Therefore, we expect that the free energy for the formation of COF-1 will be negative, despite the positive enthalpy of formation.

<table>
<thead>
<tr>
<th>$A$ (Å)</th>
<th>$B$ (Å)</th>
<th>$A+B$ (Å)</th>
</tr>
</thead>
<tbody>
<tr>
<td>14.9</td>
<td>15.7</td>
<td>56.5</td>
</tr>
<tr>
<td>15.0</td>
<td>4.8</td>
<td>45.6</td>
</tr>
<tr>
<td><strong>15.1</strong></td>
<td><strong>0.0</strong></td>
<td><strong>40.8</strong></td>
</tr>
<tr>
<td>15.2</td>
<td>1.2</td>
<td>42.0</td>
</tr>
<tr>
<td>15.3</td>
<td>7.9</td>
<td>48.6</td>
</tr>
<tr>
<td>15.4</td>
<td>19.8</td>
<td>60.6</td>
</tr>
<tr>
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<td>36.8</td>
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<td>15.6</td>
<td>58.8</td>
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<td>85.5</td>
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<td>15.8</td>
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<td>157.5</td>
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<tr>
<td>15.9</td>
<td>152.3</td>
<td>193.1</td>
</tr>
<tr>
<td>16.0</td>
<td>191.8</td>
<td>232.6</td>
</tr>
<tr>
<td>16.1</td>
<td>235.5</td>
<td>276.3</td>
</tr>
</tbody>
</table>
Figure S2: 3×3 extended unit cell of the diboronic acid-based covalent network (a unit cell is highlighted).

C₆₀ adsorption on COF-1/graphite

To model periodic COF-1 network on periodic graphite, the network must be commensurate with graphite. The smallest commensurate system corresponds to one COF-1 unit cell on 6×6 graphite unit cells (figure S3). In this case, the network lattice parameter should be compressed by 2%, from 15.1 Å to 14.76 Å (= 6 × 2.46 Å). This compression costs 41.5 kJ mol⁻¹ per unit cell (Table S1).

- 1x1 COF and 1 C₆₀ on 6×6 graphite:
  - -145.2 kJ mol⁻¹ – adsorption energy of one C₆₀ onto COF+graphite.

Alternatively, if we instead expand the lattice parameter of the COF-1 network, the best commensurate structure will be a 2×2 replicated network unit cell on a 13×13 graphite unit cell (figure S4). In this case, the network lattice parameter should be increased by 6%, from 15.1 Å to 16.0 Å (13 x 2.46 Å = 31.98 Å). This expansion costs 191.8 kJ mol⁻¹ per unit cell (Table S1).
Because of this large energy cost, this commensurability seems less likely than the compressed structure (but in practice, the two networks do not have to be commensurate; commensurability is necessary to model a periodic structure).

- 2x2 COF and 4 $C_{60}$ on 13x13 graphite system:
  - -123.2 kJ mol$^{-1}$ – adsorption energy of one $C_{60}$ onto COF+graphite.

![Image](image.png)

Figure S4: 2x2 COF and 4 $C_{60}$ on 13x13 graphite.

These calculations clearly show that the adsorption energy of $C_{60}$ onto the COF-1/graphite network depends on the periodicity of the network.

$C_{60}$ intermolecular interaction energy

The energy of $C_{60}$-$C_{60}$ interaction at distances relevant to this host-guest network can be approximately evaluated by placing (i) 4 $C_{60}$ molecules on the 13x13 graphite surface, in the positions corresponding to $C_{60}$@COF-1 (this structure is similar to Figure S4 but without the COF), and (ii) one $C_{60}$ molecule on the same 13x13 graphite surface. These systems with different $C_{60}$ coverage have different intermolecular distances and therefore different strengths of intermolecular interaction. In system (i), the shortest distance between $C_{60}$ molecules is 8.9 Å and there may be some intermolecular interaction. In system (ii), the shortest distance between $C_{60}$ molecules is 24.9 Å, and the molecules can be considered non-interacting. The binding energy, with respect to an isolated $C_{60}$ and graphite, is -73.1 kJ mol$^{-1}$ per one $C_{60}$ for system (i) (high coverage) and -63.1 kJ mol$^{-1}$ for system (ii) (low coverage). Therefore the interaction between $C_{60}$ molecules in the $C_{60}$@COF-1 system causes additional stabilization of ~10.0 kJ mol$^{-1}$.

3. Details of STM measurements

Scanning Tunneling Microscopy (STM, PicoLE, Agilent) measurements were performed in constant current mode at the liquid-solid interface at room temperature (20-25°C). Mechanically cut Pt/Ir wire (80/20, 0.25mm diameter) were used as STM tips. All measurements (including measurements of COF-1 samples) were done in 1-phenyloctane (98%, Sigma Aldrich). To determine unit cell parameters, the images were drift-corrected using the underlying graphite lattice as a reference. All STM images were processed using SPIP software (Image metrology).
4. Simultaneous visualization of COF-1 and C$_{60}$

Figure S5: STM image of C$_{60}$ self-assembled on top of COF-1 (c = 1.39 mmol/l in phenyloctane). $V_{\text{bias}} = -0.800$ V, $I_{\text{set}} = 0.018$ nA.

In figure S5a, the red arrow points along the direction of C$_{60}$ molecules and we can see that this direction indeed corresponds to the position of the pores in the network. The different colors in figure S5b give a better contrast between C$_{60}$ and COF, which makes it easier to visualize their position. The overlaid white lattice is centered at the C$_{60}$ positions. When we follow the lattice points, again in the direction of the red arrow, we can see that the C$_{60}$ positions correspond with the blue areas. These are the lowest points in the apparent height image and correspond to the pores of the network. Representing the data in this way makes it easier to see the identity between the C$_{60}$ and the COF-1 lattices.
5. Visualizing defects in the COF-1 monolayer

Figure S6: a) STM image (120 × 120 nm) of C$_{60}$ on top of COF-1 self-assembled from a saturated solution in phenyloctane. $V_{\text{bias}} = -0.900$ V, $I_{\text{set}} = 0.02$ nA. The four rectangular digital zoom-ins correspond to the insets, each having its own specific colored edge.

The orientation of the different C$_{60}$ domains was determined by overlaying the corresponding lattice in the analysis software. The red dot in the inset is the reference domain that was used as the starting point. Comparison of the position of each lattice point relative to the position of the C$_{60}$ molecules in the image shows that all domains have the C$_{60}$ molecules at a different position compared to the reference domain. This means that the C$_{60}$ domains are a consequence of the domain structure of the COF-1 monolayer, otherwise they would have the same orientation. This makes it possible to judge the quality of the COF-1 layer using the C$_{60}$ molecules for contrast enhancement.
6. Tentative Models of a) multilayers of C$_{60}$ on top of COF-1 and b) COF-C$_{60}$-COF sandwich structure

Figure S7: Tentative Models of a) multilayers of C$_{60}$ on top of COF-1 and b) COF-C$_{60}$-COF sandwich structure.
7. Clustering of $C_{60}$ molecules at low concentration

Figure S8: STM image of $C_{60}$ self-assembled on COF-1 from a solution in phenyloctane at $4.47 \times 10^{-5}$ mol/l. $V_{\text{bias}} = -0.900$ V, $I_{\text{set}} = 0.02$ nA.

Looking at the STM image, the area marked with white arrow 1 is bare graphite surface without COF-1. The areas marked with white arrow 2 can be assigned to COF-1 without $C_{60}$ adsorption. Arrow 3 points at a cluster of $C_{60}$ molecules.

We can clearly see that the $C_{60}$ molecules have a tendency to cluster together on top of the COF-1 network. All molecules that can be visualized are located in small clusters, indicating some sort of interaction between the molecules.
8. Invasive STM on multilayers of C\textsubscript{60} on COF-1/graphite

Figure S9: STM images of C\textsubscript{60} self-assembly on top of COF-1 network from a saturated solution in phenyloctane. \( V_{\text{bias}} = -0.900 \) V, \( I_{\text{set}} = 0.02 \) nA.

The STM images in figure a), b) and c) were recorded at the same location. a) is the first scan, b) the seventh and c) is the eight scan in a row of consecutive images. In figure a), there is a lot of material/C\textsubscript{60} present on the surface what makes it difficult to clearly identify the structures. Scanning at the same location (8 times in a row) seems to remove molecules from the scan area. In figure b) the imaging conditions are already more stable and we can clearly see the size and shape of different domains. The sudden tip-change in figure c) is caused by applying a voltage pulse. Only then it is possible to identify single C\textsubscript{60} molecules. It appears that the STM tip is responsible for the removal of the excess of material. A possible explanation for this observation is the presence of multilayers of C\textsubscript{60}. STM is not effective in resolving these multilayers, but after removing the excess of material (with the STM tip) we can see the molecules in the first (and second) layer.
9. Multilayer formation of COF-1

Figure S10: a) STM image of C\textsubscript{60} self-assembled on top of COF-1 from a solution in phenyloctane containing \( c = 1.39 \text{ mmol/l} \) C\textsubscript{60}. b) Line profile of the white line in figure a).

We can clearly observe contrast variations when we compare different domains of C\textsubscript{60} molecules. Domain 1 in figure S10a for instance, appears to be lower in height compared to domain 2 next to it. The apparent height difference between these two domains is approximately 0.35 ± 0.10 nm, which can be expected as a height difference between a mono- and bilayer of COF-1. (The experimental value reported in literature is 0.33 nm.)\textsuperscript{6} The contrast variation in combination with the apparent height difference is a good indication for the formation of bilayers of COF-1 underneath the C\textsubscript{60} layer.
10. Increased local coverage of $C_{60}$ by tip-scanning

Figure S11: STM images of $C_{60}$ on top of COF-1 self-assembled from a $4.47 \times 10^{-5}$ mol/l solution in phenyloctane. $V_{\text{bias}} = -0.900$ V, $I_{\text{set}} = 0.02$ nA. Figure b) was taken 9 minutes after figure a) and the area was scanned continuously during this time (8 consecutive images). Figure c) is a zoomed-out image of the same area taken after the image in figure b) was recorded.

Due to the scanning procedure, the coverage of $C_{60}$ increased from 14% in figure S6a to 32% in figure S11b. In figure S11c we can see that this is a local effect, caused by the scanning procedure. The coverage in the area surrounding the scanned area is clearly not as high (figure S11c).

11. Ideal shape of scratched area following the symmetry of the COF-1 network

Figure S12: STM image of $C_{60}$ on top of COF-1 self-assembled from a saturated solution in phenyloctane. $V_{\text{bias}} = -0.900$ V, $I_{\text{set}} = 0.02$ nA. The black hexagon indicates an area where no COF is present.

The spontaneously created empty area in figure S12 is more or less hexagonal. This means that preferentially, the symmetry of the COF-1 network is followed.
12. Uncontrolled scratching due to defects in the COF-1 network

Figure S13: STM images of self-assembly of C\textsubscript{60} from a saturated solution in phenyloctane on top of COF-1 network. $V_{\text{bias}} = -0.900$ V, $I_{\text{set}} = 0.02$ nA. A scratching procedure at $V_{\text{bias}} = -0.100$ V, $I_{\text{set}} = 0.5$ nA was applied to the area in the black square.

The red arrows in figure S13a are pointing at defects in the COF structure. We found that scratching in the vicinity of these defects, and close to the domain border in this case, has a dramatic effect on the quality of the scratched shape.

The blue curved arrow in figure S13a points out a piece of the COF that has been moved going from figure a) to b) as a consequence of scratching.

13. Self-assembly of ISA-OC14 after scratching of C\textsubscript{60}/COF-1

Synthesis of ISA-OC14

**Synthesis of Dimethyl 5-(tetradecyloxy) isophthalate**

To a solution of dimethyl 5-hydroxyisophthalate (2.0 g, 9.52 mmol) in DMF (50 mL), K\textsubscript{2}CO\textsubscript{3} (6.5 g, 47.6 mmol) was added and the reaction mixture was stirred at room temperature for 30 min, then 1-bromotetradecane (7.0 mL, 23.8 mmol) was added and the reaction mixture was heated at 110ºC for 12 hours. After being cooled to room temperature the solid was filtered and solvent was concentrated under reduced pressure. The residue was dissolved in CH\textsubscript{2}Cl\textsubscript{2} and the organic layer was washed with water and brine, dried over anhydrous MgSO\textsubscript{4} and concentrated under vacuum to obtain dimethyl 5-(tetradecyloxy) isophthalate 1 as white solid (2.9 g, 75%). MS (ESI) $m/z = 407$ [MH]\textsuperscript{+}; \textsuperscript{1}H NMR (300 MHz, CDCl\textsubscript{3}): $\delta$ 8.25 (s, 1H, ArH), 7.73 (s, 2H, ArH), 4.03 (t, $J = 6.6$ Hz, 2H, OCH\textsubscript{2}), 3.93 (s, 6H, COOCH\textsubscript{3}), 1.82-1.77 (m, 2H, CH\textsubscript{2}), 1.48-1.44 (m, 2H, CH\textsubscript{2}), 1.39-1.26 (m, 20H, CH\textsubscript{2}), 0.87 (t, $J = 6.4$ Hz, 3H, CH\textsubscript{3}).
Synthesis of 5-(tetradecyloxy) isophthalic acid

To a suspension of dimethyl 5-(tetradecyloxy) isophthalate 1 (2.9 g, 7.14 mmol) in methanol (30 mL) was added a solution of NaOH (1.42 g, 35.7 mmol) in 15 mL water. The reaction mixture was heated to reflux for 5 hours. After completion of the reaction, the organic phase was evaporated under reduced pressure. The aqueous phase was acidified with concentrated HCl. A white solid precipitated from the solution and the solid was collected by filtration. After recrystallization from hot methanol 5-(tetradecyloxy) isophthalic acid 2 (2.2 g, 81%) was obtained as a white solid. MS (ESI-) m/z = 376 [M-H]; ^1^H NMR (300 MHz, DMSO-d_6): δ 8.06 (s, 1H, ArH), 7.62 (s, 2H, ArH), 4.06 (t, J = 6.2 Hz, 2H, OCH_2), 1.74-1.70 (m, 2H, CH_2), 1.41-1.36 (m, 2H, CH_2), 1.35-1.23 (m, 20H, CH_2), 0.89 (t, J = 6.4 Hz, 3H, CH_3).

The details of ISA-OC14 self-assembly are described elsewhere.\(^7\)

Self-assembly of ISA-OC14 in scratched areas

Figure S14: STM images of C_60 self-assembly on COF-1 network from a saturated solution in phenyloctane. V\(_{\text{bias}}\) = -0.900 V, I\(_{\text{set}}\) = 0.02 nA. Figure a) is the network before scratching, b) after applying a scratching procedure at V\(_{\text{bias}}\) = -0.001 V, I\(_{\text{set}}\) = 1.100 nA in the white square in a) and c) is a zoomed-in current image of the scratched area in b).

Figure S14b is again a good example of uncontrolled scratching when there are defects in the COF network. The defects marked by red arrows in a) cause the formation of a trench towards the edge of the COF domain. The current image of the scratched area in figure c) gives less contrast between domains of C_60 and ISA-OC14 which makes it easier to visualize the self-assembly in the scratched area. The scratched area contains the linear phase of ISA-OC14, while the area above the COF network also shows formation of the porous phase.
14. Selective removal of COF-1 sheets when multilayers are formed

Figure S15: STM images of COF-1 double layer. The white numbers depict the number of COF sheets. $V_{\text{bias}} = -0.600 \text{ V}$, $I_{\text{set}} = 0.06 \text{ nA}$. The tunneling conditions for the scratching from a) $\rightarrow$ b) are $V_{\text{bias}} = -0.100 \text{ V}$, $I_{\text{set}} = 0.700 \text{ nA}$ and from b) $\rightarrow$ c) are $V_{\text{bias}} = -0.001 \text{ V}$, $I_{\text{set}} = 2.000 \text{ nA}$.

Figure S15a is the COF double layer before scratching. In figure S15b a small area of the double layer has been removed and the COF monolayer is exposed. In figure S15c a small patch of the exposed monolayer in S15b has been completely removed and the pristine graphite surface has been exposed. Due to the invasive setting that are necessary for the scratching of the monolayer COF, an additional part of the double layer has been removed, but part of it is still present.

References