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1 **Characterization of room temperature AlGaAs soft X-ray mesa photodiodes**

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9 10 **Abstract**

11
12 Results characterizing a set of nine prototype Al_{0.8}Ga_{0.2}As p⁺-i-n⁺ mesa photodiodes (400 μm
13 diameter, 1.7 μm i layer) are presented. The results show the performance of the devices as room
14 temperature spectroscopic photon counting soft X-ray detectors. The responses of the photodiodes to
15 illumination with an ⁵⁵Fe radioisotope X-ray source were measured using a low noise charge sensitive
16 preamplifier; the energy resolutions measured with the devices were consistent with each other and
17 had a mean FWHM at 5.9 keV of 1.27 keV. The devices are the thickest (highest quantum efficiency)
18 AlGaAs X-ray spectroscopic mesa photodiodes reported in the literature to date. They also have
19 better energy resolution than all previously reported non-avalanche AlGaAs X-ray detectors of the
20 same area.

21
22 **Keywords:** AlGaAs; detector; X-ray; spectroscopy; photodiode

23 24 25 **1. Introduction**

26
27 Narrower bandgap materials such as silicon (the material most commonly used for semiconductor X-
28 ray detectors) offer excellent energy resolutions when cooled to ≤ 20 °C [1], but this performance
29 degrades at high temperatures due to increased thermal charge carrier generation [2]. In some
30 extreme cases, the optimum operating temperatures for silicon X-ray detectors can be as low as -130
31 °C [3], although cooling to more modest temperatures is more common. Under certain circumstances,
32 such as spaceflight, cooling detectors to low temperatures can be impractical or undesirable since it
33 increases the mass, volume and power requirements of the instrument.

34
35 Spectroscopic photon counting X-ray photodiodes made from materials such as SiC [4,5], GaAs
36 [6,7,8] and AlGaAs [9,10] offer the ability to operate uncooled in high temperature environments
37 (>>20 °C) due to the low thermally induced leakage currents present in wide band gap materials
38 giving rise to correspondingly small parallel white noises in the detector system [2]. Potential uses for
39 high temperature spectroscopic photon counting detectors include X-ray fluorescence spectroscopy
40 for geological applications on hot planetary surfaces (e.g. Mercury, Venus, parts of Earth) as well as
41 for various terrestrial applications in industrial instrumentation and process control.

42
43 The first reported use of AlGaAs for X-ray detection was by Lauter et al. in 1995 [11] as part of an
44 AlGaAs/GaAs X-ray APD. However, it is only in more recent years that AlGaAs has started to
45 receive significant attention for X-ray applications with results reported showing the response of
46 single pixel detectors to X-rays [9,10-13], beta particles and electrons [9,14] and alpha particles [12],
47 with the subsequent measurement of key parameters relevant to high temperature AlGaAs detector
48 physics such as the electron-hole pair creation energy [15,16] and the temperature dependence of the
49 impact ionization coefficients [17]. Another challenge in the development of AlGaAs for X-ray
50 detector applications has been the historically low good device yield (low leakage, acceptable and
51 consistent spectral performance) from growth and fabrication runs. The effect of this has been that
52 until now researchers have presented results from only single or very few AlGaAs X-ray devices
53 rather than reporting characterisation of multiple devices on the same die as would be required if
54 monolithic AlGaAs pixel arrays are to become a reality.

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In this paper, results are presented from a new nine diode set of $\text{Al}_{0.8}\text{Ga}_{0.2}\text{As}$ X-ray mesa photodiodes (400 μm diameters, 1.7 μm i layers) operated at room temperature and coupled to a charge sensitive preamplifier of custom design. The devices were randomly selected from a ~ 5 mm by ~ 5 mm die from a wafer grown by Molecular Beam Epitaxy. Unlike previous work, every diode tested showed a low leakage current and a spectroscopic photon counting response with consistent energy resolution. This is a significant step towards the realisation of AlGaAs mesa X-ray photodiode pixel arrays since the device yield is now demonstrably high enough for the production of small arrays (e.g. 3 by 3 pixels) to be a reality. The devices also have the highest quantum efficiency for spectroscopic photon counting AlGaAs X-ray mesa photodiodes reported in the literature to date, the best spectral resolution for non-avalanche AlGaAs devices of this size reported in the literature to date, and for the first time, a quantitative estimate of charge trapping noise in AlGaAs X-ray detectors is calculated.

2. The photodiode design

$\text{Al}_{0.8}\text{Ga}_{0.2}\text{As}$ $\text{p}^+\text{-i-n}^+$ epilayers (**Table 1**) were grown by Molecular Beam Epitaxy on GaAs n^+ substrates at the EPSRC National Centre for III-V Technologies, Sheffield, UK. Mesa diodes with diameters of 400 μm were photolithographically etched from the wafer by researchers at University of Sheffield, UK. The p^+ -side Ohmic contact covered 45 % of each diode's face and was formed from Au, Zn, Au layers with thicknesses of 5 nm, 10 nm and 200 nm, respectively. The capacitance of the devices was measured at University of Sheffield using an HP 4275 LCR meter (AC test voltage signal magnitude of 50 mV rms, frequency of 1 MHz). At reverse biases ≥ 5 V, the capacitance of each diode was 7 pF and there was no reduction in capacitance at higher applied fields. This indicated that the i layer was fully depleted at biases of 5 V and higher. This assumption was subsequently confirmed by X-ray measurements. Nine randomly selected diodes on a single die were subsequently gold ball bonded to a standard TO package for I-V and X-ray characterisation.

Figure 1 shows the calculated quantum efficiency of the devices assuming that the i layer is the only active region of the detector. For comparison, the quantum efficiencies of two other previously reported AlGaAs devices [9,17] are also plotted in **Figure 1** with the same assumption. It should be noted that whilst the detectors reported here are the thickest spectroscopic photon counting AlGaAs X-ray mesa photodiodes reported to date, the detectors are still thin and intended for device research, rather than as 'user-ready' detectors which would normally require thicker active layers and thinner dead layers to give higher quantum efficiencies.

[TABLE 1]

[FIGURE 1]

3. Experimental method and results

3.1 Leakage current

After packaging, each diode's leakage current was measured as a function of applied reverse bias up to 10 V using a Keithley picoammeter. The measurements were carried out in a dry N_2 environment at a temperature of 20 $^\circ\text{C}$. Plots of the measured leakage currents as a function of applied reverse bias are presented in **Figure 2**. At 5 V, the normal operating reverse bias of the detectors, the mean leakage current of the nine detectors was $5.4 \text{ pA} \pm 2.1 \text{ pA}$ (rms deviance) corresponding to a leakage current density of $4.72 \text{ nA cm}^{-2} \pm 1.67 \text{ nA cm}^{-2}$ (rms deviance). These leakage current densities are smaller than some that have been reported for AlGaAs X-ray devices at full depletion (e.g. 17.5 nA cm^{-2} and 13.7 nA cm^{-2} [10]), but larger than the best reported in the literature (2.2 nA cm^{-2} [9]). Ensuring low leakage currents is important in order to minimise the contribution from white parallel noise (§3.3.2) to the achievable energy resolution; we highlight that the leakage current densities so far reported for AlGaAs mesa devices are much greater than those reported for high quality 4H-SiC Schottky devices (e.g. 1 pA cm^{-2} [18]).

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[FIGURE 2]

3.2 X-ray measurements

3.2.1 Measurements at 5 V reverse bias

To obtain X-ray spectra, the diodes were each reverse biased at 5 V and connected to a single channel low noise charge sensitive preamplifier in turn. The preamplifier used a silicon JFET (NJ26, capacitance ~ 2 pF) as the input transistor. The preamplifier was of a feedback resistorless design similar to ref. [19]. The preamplifier was connected to an Ortec 571 shaping amplifier (shaping time constant = 3 μ s) and multi-channel analyser (MCA). The MCA lower input discriminator was set at 1.8 keV to limit counts from the zero energy noise peak of the preamplifier. An ^{55}Fe radioisotope X-ray source, giving characteristic Mn K α (5.9 keV) and Mn K β (6.49 keV) lines, was positioned above the diodes. As was the case for the leakage current measurements, the diodes and preamplifier were operated in a dry N₂ environment at a temperature of 20 °C. Spectra with live time limits of 300 seconds were accumulated for the devices.

Figure 3 shows the ^{55}Fe spectrum accumulated at 5 V reverse bias with one of the devices, Diode 3. Double Gaussians (dashed lines in **Figure 3**) have been fitted to the observed peak representing the Mn K α (5.9 keV) and K β (6.5 keV) peaks from the radioisotope source in their accepted ratios [20]. The Mn K α and K β lines were too close together to be resolved given the energy resolution of the detector. The MCA low energy cut off (1.8 keV) was set as a compromise between minimising the number of noise counts from the tail of zero energy noise peak and maintaining the low energy response of the detector system. The upwards curve on the low energy tail at low energies is from the right hand side of the zero energy noise peak (which has not been completely eliminated by the 1.8 keV low energy cut off) extending under the low energy tail of the combined Mn K α , Mn K β peak. The spectra were calibrated in energy terms by using the position of the zero energy noise peak and the position of the fitted Mn K α peak at 5.9 keV for each spectrum as points of known energy on the MCA's charge scale and assuming a linear variation of detected charge with energy.

[Figure 3]

The energy resolution of the system (as measured by the FWHM at 5.9 keV) for each diode when reverse biased at 5 V is shown in **Figure 4**. Also shown in the figure are FWHM for a subset of the devices operated at 10 V reverse bias (see §3.2.2). Every diode functioned as a spectroscopic photon counting detector and all diodes had remarkably similar FWHM suggesting uniform quality of the wafer material across the die. At 5 V, the mean energy resolution was $1.27 \text{ keV} \pm 0.04 \text{ keV}$ (rms deviance) at 5.9 keV. The majority of the variability of the FWHM comes from the different leakage currents of the devices giving rise to differently sized parallel white noises (see §3.3.1). The FWHM are significantly better than that previously reported with a 400 μ m diameter AlGaAs X-ray photodiode (FWHM at 5.9 keV of 1.95 keV [9]), but the energy resolutions are not as good as the best currently reported for 200 μ m AlGaAs devices (1.07 keV at 5.9 keV) [10]. The 200 μ m devices in ref. [10] had an area four times smaller than the presently reported devices and consequently benefitted from a reduced capacitance and correspondingly lower series white noise giving improved energy resolution.

[Figure 4]

Low energy tailing on the combined Mn K α , Mn K β peak due to partial charge collection can also be seen in the spectrum presented in **Figure 3**. This is hypothesized to be from charge created by X-ray photon interactions in the non-active layers of the device diffusing towards the active region of the device and consequently having some limited collection. The valley-to-peak (V/P) ratio is one method of quantifying of the amount of low energy tailing, and the V/P ratio for the detectors we report (mean = 0.08) is better than that of previous AlGaAs devices even of smaller area (0.13 [10]),

165 but is not as good as has been reported for thicker room temperature and cooled GaAs detectors (V/P
 166 = 0.04, [8,21]) or cooled silicon DEPFETs (V/P ~ 0.0001) [22]. If the low energy tailing in the
 167 AlGaAs detectors reported here is primarily a consequence of partial charge collection from photons
 168 absorbed in the non-active regions of the devices, then the V/P ratio is likely to improve when thicker
 169 (higher quantum efficiency) AlGaAs detectors are produced.

171 The number of counts per second in the combined Mn K α Mn K β fitted peaks for each of the nine
 172 diodes when operated at 5 V was $139 \text{ s}^{-1} \pm 12 \text{ s}^{-1}$ (rms deviance); this will be commented on further in
 173 §3.2.2.

175 3.2.2 Measurements at 10 V reverse bias

177 To investigate the performance of the detectors when operated at increased reverse bias, X-ray
 178 measurements were repeated at 10 V for a random selection of two-thirds (six) of the diodes, for
 179 convenience these diodes were numbered 1–6 for this manuscript. At this elevated reverse bias, the
 180 mean of the leakage currents for the subset of detectors had increased to $14.1 \text{ pA} \pm 5.3 \text{ pA}$ (rms
 181 deviance) from $5.1 \text{ pA} \pm 1.3 \text{ pA}$ (rms deviance) at 5 V. The FWHM measured with the devices when
 182 they were reversed biased at 10 V are shown in **Figure 4**; the FWHM were increased compared with
 183 those measured at 5 V, indicating that the larger parallel white noise at 10 V outweighs any positive
 184 aspects which may be brought from operation higher reverse bias, such as reduced charge trapping
 185 (see Section 3.3.1). The V/P ratios for spectra accumulated with the detectors reverse biased at 10 V
 186 were the same as when the detectors were biased at 5 V.

188 To assess whether there was any difference in the quantum efficiency of the devices when operated at
 189 10 V compared to 5 V, the count rate in the combined Mn K α , Mn K β fitted peaks for each detector
 190 measured at 10 V was calculated. Had the quantum efficiency or width of the depletion region been
 191 significantly increased at 10 V compared with at 5 V, a higher count rate would have been
 192 correspondingly expected. No such increase was observed. For the subset of diodes characterised at
 193 10 V the mean count rate was $137 \text{ s}^{-1} \pm 12 \text{ s}^{-1}$ (rms deviance); at 5 V the mean count rate for the same
 194 subset of devices was $136 \text{ s}^{-1} \pm 14 \text{ s}^{-1}$ (rms deviance). Consequently it can be concluded there was no
 195 additional extension of the depletion region or increase in quantum efficiency at 10 V. This
 196 corroborates the assumption made from the C-V measurements that the devices were fully depleted at
 197 5 V and that there was no extension of the depletion region into the p⁺ or n⁺ layers at biases up to 10
 198 V.

200 3.3 Noise analysis

202 The fundamental ‘Fano-limited’ energy resolution (FWHM) of a non-avalanche photodiode is given
 203 by

$$205 \quad \Delta E = 2.355\omega \sqrt{\frac{FE}{\omega}} \quad (\text{Eq. 1})$$

206 where ω is the average energy consumed in the generation of an electron-hole pair (recently measured
 207 to be 5.1 eV for X-rays in Al_{0.8}Ga_{0.2}As at room temperature [15][16]), F is the Fano factor (still to be
 208 experimentally measured for AlGaAs), and E is the energy of the photon. In practice, this resolution
 209 is degraded by electronics noise from the connection of the preamplifier to the detector and the
 210 preamplifier itself, and noise arising from imperfect charge transport processes in the semiconductor
 211 detector, such that

$$214 \quad \Delta E = 2.355\omega \sqrt{\frac{FE}{\omega} + a^2 + r^2} \quad (\text{Eq. 2})$$

215

216 where a is the equivalent noise charge contributed by electronics noise (comprised of many
 217 constituent noise sources) and r is the equivalent noise charge contributed by charge trapping and
 218 collection inefficiencies (both in units of e^- rms). Even though the energy resolutions of the detectors
 219 reported here are better than comparable AlGaAs detectors previously reported, the detectors have
 220 energy resolutions far poorer than if they were Fano limited (FWHM = 142 eV at 5.9 keV, assuming
 221 $F = 0.12$). As such, it is informative to consider the relative contributions of the noise components
 222 that are broadening the energy resolution beyond the Fano limit.

223

224 3.3.1 Charge trapping noise

225

226 Very large amounts of charge trapping can manifest itself in X-ray spectra obtained with
 227 semiconductor devices as significant deviations in peak shape from the expected Gaussian forms [23].
 228 The peaks in the spectra accumulated with the detectors are Gaussian and do not show morphology
 229 suggestive of trapping being the dominant noise source. Another common ‘quick test’ for the
 230 presence of changes in the charge collection efficiency as a function of reverse bias was also
 231 performed: the positions on the MCA’s charge scale (in units of MCA channel number) of the 5.9
 232 keV peaks in the 10 V spectra were compared to the positions of the 5.9 keV peaks in the 5 V spectra.
 233 A significant change in the amount of charge collected from the absorption of a 5.9 keV photon in the
 234 active region would have revealed itself as a change in peak position. No such shift was present.

235

236 However, a more quantitative assessment of trapping noise was also made, which showed that
 237 significant levels of trapping were present. Other than charge trapping noise, the only noise source
 238 which varies with detector reverse bias is the parallel white noise (Section 3.3.2). By subtracting the
 239 parallel white noise from the equivalent noise charge of the measured FWHM at each respective
 240 reverse bias, in quadrature, and then subtracting the remaining value for 10 V reverse bias from that
 241 for 5 V, in quadrature, the additional charge trapping noise that is present in the 5 V spectra compared
 242 with the 10 V spectra can be estimated.

243

$$244 \quad \Delta r^2 = \left(\frac{\Delta E_{5V}}{2.355\omega} \right)^2 - N_{5Vpw}^2 - \left(\left(\frac{\Delta E_{10V}}{2.355\omega} \right)^2 - N_{10Vpw}^2 \right) \quad (\text{Eq. 3})$$

245

246 Computing Eq. 3 suggests a mean additional charge trapping noise of 26 e^- rms at 5.9 keV when
 247 operating at 5 V compared to 10 V. Consequently it can be said that there is at least 26 e^- rms
 248 equivalent noise charge attributable to charge trapping at 5 V.

249

250 The measured amount of trapping noise is significant, but small compared to the noise from the
 251 electronics currently used. If the electronics noise remained the same but the trapping noise was
 252 eliminated, the FWHM at 5.9 keV would improve from 1.27 keV to 1.23 keV. However, the
 253 significance of the trapping would be more readily apparent if the electronics noise was negligible and
 254 the detector system was Fano limited apart from trapping noise; from Eq. 1, a Fano limited resolution
 255 (FWHM) of 142 eV at 5.9 keV was predicted, however the energy resolution achievable in the
 256 presence of 26 e^- rms of trapping noise would be 343 eV.

257

258 The presence of significant levels of charge trapping in compound semiconductors is well known,
 259 with much having been published on trapping and polarisation in many compound semiconductor
 260 materials, including GaAs [24] and CdZnTe [25]. That an emerging and relatively young material (for
 261 X-ray spectroscopy use) such as AlGaAs suffers trapping is not a surprise, however, to the authors’
 262 knowledge this is the first reported measurement and quantification of trapping in AlGaAs.

263

264 3.3.2 Electronics noise

265

266 The electronics noise was the most significant contributor to the measured energy resolution in the
 267 system we report. It is comprised of parallel white noise, series white noise (including induced gate

current noise), $1/f$ series noise, and dielectric noises. An introduction to these noise components in X-ray photodiodes coupled to charge sensitive preamplifiers can be found in Ref. [26]. The calculated contributions of these noise sources for each diode when reverse biased at 5 V are shown in **Figure 5**.

With the exception of the parallel white noise, the electronics noise components are the same for each diode characterized. The values in **Figure 5** assume that any remaining charge trapping noise beyond the $26e^-$ rms ENC calculated in Section 3.3.1 was small compared to the other noise sources contributing to the total energy resolution (Eq. 1). Only a minimum bound for the value of the series white noise could be readily calculated since it depends on the total capacitance load on the input transistor of the preamplifier. Due to the prototype nature of the preamplifier used, there were significant stray capacitances with unknown values in addition to the capacitances that were known or readily estimable. The same is true for the dielectric noises; dielectric noise contributions from the detector, JFET and feedback capacitor were readily estimable, but other lossy dielectrics in proximity to the preamplifier may have also added to the noise.

[Figure 5]

By subtracting the predicted Fano noise contribution, the calculated charge trapping noise and electronics noise contributions from the measured energy resolution in quadrature, the remainder can be attributed to the total noise from the stray capacitances and dielectrics that were not included previously; however, their separate contributions cannot easily be detangled. The results shown in **Figure 5** are consistent with previous findings for comparable detectors using similar preamplifiers [27] in that they suggest that the noises from these sources (likely to be dominated by contributions from the packages of the input JFET and detector) are the most significant source of noise in systems of such design. Using an unpackaged (die form) input JFET is known to significantly reduce the noise [19]. If the noises from the stray capacitances and the additional dielectrics could be eliminated, it is predicted that a mean energy resolution of 1.0 keV FWHM at 5.9 keV would be achieved with detectors and system.

4. Discussions, conclusions and further work

A set of nine 400 μm diameter $\text{Al}_{0.8}\text{Ga}_{0.2}\text{As}$ mesa p^+i-n^+ photodiodes with 1.7 μm i layers have been characterised as room temperature soft X-ray detectors. The devices were found to be fully depleted at low reverse bias (5 V) and they showed low leakage currents (mean = 5.4 pA at 5 V). The X-ray performance of the detectors was investigated by coupling them to a low noise charge sensitive preamplifier; the devices functioned as photon counting spectroscopic X-ray detectors with moderate energy resolution (FWHM = 1.27 keV at 5.9 keV). Whilst the energy resolutions are modest compared to those achievable at colder temperatures and with more mature technologies such as silicon DEPFETs (e.g. 135 eV at 5.9 keV at -10°C [22]) and the best GaAs devices (e.g. 266 eV at 5.9 keV at -31°C [28]), the resolutions now reported are the best so far recorded for AlGaAs devices of this size at room temperature, surpassing the previous best of 1.95 keV [9]; in both cases the AlGaAs devices were investigated using preamplifiers of similar design. The improved performance reported here is likely attributable to lower white series noise in the present detectors due to their lower capacitance and improvements in device fabrication and wafer quality.

Operation at increased reverse bias was investigated for a randomly selected sub-set of two-thirds of the devices. The count rates of the devices were consistent and the same at 10 V as they were at 5 V, indicating that there was no increase in quantum efficiency at the higher reverse bias. This supported the assumption from the C-V measurements that the devices were fully depleted at 5 V. Interestingly, the FWHM of the devices at 10 V had broadened less than would have been expected from the measured increase in leakage current at 10 V. Since the electric fields developed in the devices at reverse biases of 5 V and 10 V are too small to have produced avalanche multiplication, and since the electronic noise sources other than the leakage current driven parallel white noise are unchanging with increased reverse bias, the reduction in noise at higher reverse bias may be attributed to reductions in charge trapping noise. The measurements suggested there was $26e^-$ rms equivalent noise charge more

323 charge trapping noise at 5 V than 10 V. However, the FWHM were still larger at 10 V than at 5 V
324 because the increase in parallel white noise from the leakage current was greater than the reduction in
325 suspected charge trapping noise. This is the first time there has been sufficient data to be able to
326 quantitatively estimate the trapping noise in AlGaAs X-ray detectors.

327

328 If the preamplifier had had a pulse generator test signal input (as per ref. [27]), it would have been
329 possible to measure the total electronics noise directly since it would have been given by the width of
330 the peak from the test signal. This could then have been subtracted in quadrature, along with the
331 expected Fano noise, from the experimentally measured FWHM of the 5.9 keV peak from the ⁵⁵Fe
332 radioisotope X-ray source to produce an absolute measurement of the trapping noise in the detectors
333 rather than a lower bound. Unfortunately the preamplifier did not have a test signal input so this was
334 not possible. We plan to build a preamplifier with a test input to enable such a measurement in future.

335

336 The results reported here are the first investigation that characterises multiple randomly selected
337 AlGaAs X-ray diodes from the same semiconductor die, rather than simply reporting characterisation
338 of single ‘known good’ devices selected by pre-screening. By characterising nine randomly selected
339 devices across a die size of ~5 mm by ~5 mm, and finding that every diode is functional, with
340 consistent C-V, I-V and X-ray performances, it has been shown that AlGaAs X-ray detector
341 technology is now at a stage suitable for research to move to the production of small pixel arrays (e.g.
342 3 by 3 pixels). This is an important advance for AlGaAs X-ray detectors and we anticipate reporting
343 on the wafer growth, device fabrication and characterisation of such arrays in the near future as part of
344 the ongoing semiconductor research at University of Sussex. Further investigations of charge
345 trapping in AlGaAs as well as the growth of wafers using metal organic chemical vapor deposition
346 (rather than MBE) in order to achieve thicker i layers are also anticipated.

347

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349

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