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Surface Complexation of Cu on Birnessite $(\delta$ -MnO₂): Controls on Cu in the Deep Ocean

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1	Surface Complexation of Cu on Birnessite (δ -MnO ₂):
2	Controls on Cu in the Deep Ocean
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27 ABSTRACT

28

29 Hexagonal birnessite (-MnO₂) is a close analogue to the dominant phase in hydrogenetic 30 marine ferromanganese crusts and nodules. These deposits contain ~0.25 wt. % Cu which is 31 believed to be scavenged from the overlying water column where Cu concentrations are near 0.1 μ g/L. Here, we measured the sorption of Cu on -MnO₂ as a function of pH and surface 32 33 loading. We characterized the nature of the Cu sorption complex at pH 4 and 8 using EXAFS 34 spectroscopy and find that, at pH 4, Cu sorbs to birnessite by inner-sphere complexation over the vacancy site to give a 3- to 4-fold coordinated complex with 6 Mn neighbors at ~3.4 Å. 35 At pH 8, however, we find that some Cu has become structurally incorporated into the MnO₂ 36 layer by occupying the vacancy sites to give 6 Mn neighbors at ~2.91 Å. Density functional 37 calculations on $CuMn_{18}O_{24}(OH)_{30}(H_2O)_3^{-4}$ and $CuMn_{18}O_{21}(OH)_{33}(H_2O)_3^{-1}$ clusters 38 39 predict a 3-fold coordinated surface complex and show that the change from surface 40 complexation to structural incorporation is a response to protonation of oxygens surrounding 41 the vacancy site. Consequently, we propose that the transformation between sorption via 42 surface complex and vacancy site occupancy should be reversible. By fitting the Cu sorption 43 as a function of surface loading and pH to the formation of the observed and predicted surface 44 complex, we developed a surface complexation model (in the basic Stern approximation) for the sorption of Cu onto birnessite. Using this model, we demonstrate that the concentration of 45 46 inorganic Cu in the deep ocean should be several orders of magnitude lower than the observed 47 total dissolved Cu. We propose that the observed total dissolved Cu concentration in the 48 oceans reflects solubilization of Cu by microbially generated ligands.

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CC

50 1. INTRODUCTION

51 The depth-concentration profile of Cu in the oceans is intermediate between that 52 characteristic of scavenged and nutrient elements (Bruland & Lohan, 2004). Dissolved 53 copper is strongly scavenged by sorption to colloidal iron and manganese oxides; these 54 accumulate to form ferromanganese crusts and nodules with Cu concentrations enriched by 10⁶ over average seawater concentration (Arrhenius, 1963). In addition to being scavenged, 55 dissolved Cu⁺² is complexed by microbially generated ligands which may play a role in 56 57 detoxifying Cu (Coale and Bruland, 1988; Bruland and Lohan, 2004). There is recent 58 evidence that such ligands may be significant for Cu even in the deep ocean (Moffat and 59 Dupont, 2007). 60 Within ferromanganese crusts, Cu tends to be more strongly associated with the 61 manganese oxide phase (Cronan, 1969; Kumar et al., 1994). The nature of the manganese 62 oxide phases and the mechanism by which such minerals become highly enriched in trace 63 metals is still unclear. The dominant manganese oxide phase in marine ferromanganese 64 precipitates is birnessite (Burns and Burns, 1979). Birnessite is a phyllomanganate consisting of sheets of edge-sharing MnO₆ octahedra; the sheets yield X-ray diffraction lines 65 at 2.4 Å and 1.4 Å. In well-crystalline birnessites, the sheets of edge-sharing MnO₆ 66 octahedra are stacked in the c-direction to give additional diffraction lines at ~7 Å (001) and 67 ~3.6 Å (002). However, marine birnessites usually show only diffraction lines at 2.4 and 1.4 68 Å and are termed "vernadite" or δ -MnO₂. The absence of the ~7 Å (001) and ~3.6 Å (002) 69 70 lines is due to a lack of coherent stacking order of the MnO₆ sheets in the c-direction. 71 In addition to the degree of stacking order, natural and synthetic birnessites show 72 variations in the oxidation state and fraction of vacancies within the MnO₆ sheet. Natural 73 marine birnessites have all Mn as Mn(IV) (Murray et al., 1984) and hexagonal symmetry 74 (Jones and Milne, 1956; McKenzie, 1989; Post, 1992). Drits et al. (1997) showed that this 75 symmetry results from ordered vacancies in the octahedral sites. Sorption of Ni on synthetic 76 hexagonal-birnessites occurs via complexation over vacancy sites (Peacock and Sherman, 77 2007a; Manceau et al., 2007), as does sorption of Cu (Manceau et al., 2002). With increasing 78 pH, however, surface-complexed Ni cations move into the vacancy site and become 79 structurally incorporated (Peacock and Sherman, 2007a; Peacock and Sherman, 2007b). 80 Significantly, this transition between surface complexation and structural incorporation has

81 been demonstrated to be reversible (Peacock, 2009). For -MnO₂, Villalobos et al. (2006)

report 0.06 vacancies per Mn octahedron although a more recent study indicates the number
of vacancies per octahedron may be higher (Grangeon et al., 2008). This suggests that
hexagonal birnessites may sorb transition metals in excess of 6 mole % via complexation over
vacancy sites.

86 The sorption of Cu to birnessite is rapid (compared to its oceanic residence time) and 87 we hypothesize that dissolved Cu in the oceans is in chemical equilibrium with birnessite. 88 To develop this hypothesis further, we need a thermodynamic model for the uptake (scavenging) of Cu by phases such as -MnO₂. This model would enable us to predict how 89 90 dissolved Cu concentrations might respond to anthropogenic perturbations and may help us 91 decipher how paleochemical records indicate past chemical perturbations (e.g., changes in pH 92 and inputs due to hydrothermal fluxes). In the work presented here, we report results of 93 sorption experiments of copper to synthetic birnessite as a function of pH and concentration. 94 We characterize the nature of the sorption complexes using EXAFS spectroscopy and *ab* 95 initio (density functional) predictions of the structures and energetics of model clusters that 96 correspond to possible sorption complexes. We then fit the sorption edges to a surface 97 complexation model based on the sorption complexes determined from EXAFS and ab initio 98 simulations. From this model, we attempt to predict the equilibrium distribution of Cu 99 between seawater and ferromanganese crusts.

100

101 2. EXPERIMENTAL AND COMPUTATIONAL METHODS

- 102 2.1. Mineral preparation and characterisation
- 103 As shown by Murray et al. (1984), the Mn oxides which occur in marine

104 ferromanganese nodules and crusts have Mn entirely as Mn(IV). Synthetic birnessites can

105 have variable oxidation states of Mn with charge compensation provided by H⁺ or Na, K

106 cations. The $^{\text{TM}}$ -MnO₂ form of birnessite was prepared by following the synthesis procedure 107 outlined in Villalobos et al. (2003) based on the reaction:

108

$$109 \quad 2KMnO_4 + 3MnCl_2 + 4NaOH \rightarrow 5MnO_2 + 2KCl + 4NaCl + 6H_2O \tag{1}$$

110

111 to yield a product which Villalobos et al. (2003) demonstrated had all of the Mn as Mn(IV).

112 The Mn K-edge XANES of the $^{\text{TM}}$ -MnO₂ used here is identical to that of the -MnO₂ phase in

113 Villalobos et al. (2003). Iodometric titration of our product was unable to detect Mn⁺³, this is

consistent with the synthesis product of Villalobos et al. (2003). However, Grangeon et al.

115 (2008) have reanalyzed the Villalobos et al. (2003) birnessite using a Mohrs salt titration 116 method and found an average oxidation state of 3.9. The significance of any trace Mn^{+3} to 117 the sorption of metals to birnessite is unclear. X-ray diffraction patterns of the dried product 118 show asymmetric peaks at ~7 Å, 2.4 Å and 1.4 Å in agreement with the product described in Villalobos et al. (2003). The surface area (measured using the BET method), is $107 \text{ m}^2/\text{g}$. 119 120 As will be shown below, however, this surface area greatly underestimates the sorption 121 capacity of the birnessite. We measured the experimental $pH_{_{PZC}}$ of our $-MnO_2$ by potentiometric titration at two 122 salt concentrations (0.01 and 0.1 M NaNO₃). Freshly prepared, air-dried -MnO₂ (0.83 g/L) 123 124 was suspended in pre-boiled, nitrogen purged (<1 ppm $CO_2(g)$) 18.2 m Ω MilliQ water, and nitrogen purged (<1 ppm CO₂ (g)) for several hours. Titrations were performed at 25 °C in an 125 126 air-tight vessel with constant stirring to avoid settling. Base (NaOH, free from carbonate), 127 acid (HNO₃) and salt solutions (NaNO₃) were added via an automated titrator. A nitrogen 128 atmosphere (<1 ppm CO_2 (g)) was maintained throughout the experiment. Electrolyte was 129 added to bring the ionic strength to 0.01 M and then acid added gradually to lower the pH to 130 ~1.5. Incremental addition of base then produced a titration from acid pH to pH ~9. After 131 each incremental addition of base, up to 5 mins were allowed for pH equilibration. The 132 suspension was returned to acid pH by reverse acid titration, electrolyte added to adjust the 133 ionic strength to the next level and the titration repeated following the same method. No 134 significant hysteresis was observed between the base and acid legs. We find $pH_{PZC} = 2.3 \pm$ 0.2 and this is in good agreement with previous acid-base titrations of -MnO₂ (e.g., Catts and 135 Langmuir, 1986). 136

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138

2.2. Cu vs. pH adsorption edge experiments

139 Batch sorption edge experiments were prepared with Cu(II) aqueous solution using 140 AR grade reagents and 18.2 m Milli-Q water. All solutions and resulting suspensions were 141 purged with N_2 (g) (<1 ppm CO₂ (g)) and all adsorption experiments were conducted at 25 °C. 142 pH measurements were calibrated to ±0.05 pH units using Whatman NBS grade buffers. 143 Copper(II) stock solution was prepared at 1000 ppm from Cu(NO₃), 3H₂O in 0.1 M 144 HNO₃. NaNO₃ stock solution was prepared at 0.1 M for use as background electrolyte. 145 Adsorption experiments at 143, 77 and 32 ppm [Cu]_{total} were prepared by adding 5, 2.5, and 1

146 ml of 1000 ppm Cu stock solution to 0.100 g solid δ -MnO₂ to 30 mL of 0.1 M NaNO₃,

147 respectively. Adsorbent concentration was therefore 3.3 g/L, with the adsorbent containing ~

148 5, 2.5 and 1 wt.% Cu at 100% adsorption, respectively. Stock solution was added

149 incrementally after ~pH 5.5 to avoid the possible precipitation of solid Cu (hydr)oxide phases

150 (Cu speciation in our experimental solutions was calculated using PHREEQC (Parkhurst and

151 Appelo, 1999) and the MINTEQ.V4 database (Charlton and Parkhurst, 2002)). The resulting

152 suspensions were immediately shaken. Suspension pH was varied between pH 1-7 by the

153 dropwise addition (<1 mL) of either HNO₃ or NaOH and recorded after stabilization to two

154 decimal places. Adsorption edge experiments were then shaken continuously for 1 week

during which pH values were adjusted using NaOH or HNO₃. Final pH measurements were

156 obtained for the experimental suspensions before centrifugation or filtering.

157 All adsorption suspensions were separated by centrifugation (~3000 g for 10 min) into 158 an adsorption sample (thick paste) and a clear supernate for determination of total Cu 159 concentration. Thick paste adsorption samples at pH 4 and 8 and with 5 wt.% Cu were 160 subject to EXAFS after storing at 1 - 4 °C for a maximum of 48 hours before scanning. 161 Supernates were filtered using $0.2 \,\mu m$ cellulose nitrate membrane filters, acidified with 1% 162 HNO₃ and analyzed for Cu by inductively-coupled plasma atomic emission spectroscopy 163 (ICP-AES). Supernates from the experiments used for EXAFS (at pH 4 and 8 and with 5 164 wt.% Cu) were analysed for Cu and Mn by ICP-MS. Cu and Mn analytical detection limits 165 were better than 10 ppb. Aqueous Mn was <100 ppb for both EXAFS sample supernates.

166

167 2.3. EXAFS spectra of Cu sorption complexes

168 EXAFS data were collected at the CCLRC Synchrotron Radiation Source at 169 Daresbury Laboratory, U.K. Spectra of the Cu K-edge (8979 eV) were collected on station 170 16.5, which was designed for measurements on ultra-dilute systems. During data collection, 171 storage ring energy was 2.0 GeV and the beam current varied between 130 and 240 mA. 172 Adsorption samples were presented to the X-ray beam as a wet paste held in a 2 mm-thick 173 Teflon slide with a 4 x 10 mm sample slot. Small sheets of 250 µm-thick Mylar were placed 174 on either side of the Teflon slide and sealed with a small amount of vacuum grease to hold the 175 wet pastes in place and prevent drying of the samples. EXAFS data were collected from up to 176 20 fluorescence mode scans using an Ortec 30-element solid-state detector. 177 EXAFS data reduction was performed using Daresbury Laboratory software

178 (EXCALIB and EXBACK, Dent and Mosselmans, 1992). EXCALIB was used to calibrate

179 from monochromator position (millidegrees) to energy (eV) and to average multiple spectra 180 from individual samples. EXBACK was used to define the start of the EXAFS oscillations 181 (determined from the inflection point on the K-edge) and perform background subtraction. 182 The pre-edge was fit to a linear function and the post-edge background to two 2nd-order 183 polynomial segments. The phase-shifts and potentials were calculated in the small atom 184 (plane-wave) approximation and we allowed for multiple scattering as coded in EXCURV98 185 (Binsted, 1998). The phase-shift functions used in the curve fitting were derived by *ab initio* 186 methods in EXCURV98 using Hedin-Lundqvist potentials (Hedin and Lundqvist, 1969) and 187 von Barth ground states. No Fourier filtering was performed during the data analysis. 188 Typical errors associated with EXAFS modelling over the k-range used here are 15 % and 25 189 % for first and second shell coordination numbers, respectively, ± 0.02 and 0.05 Å for first and second shell distances, respectively, and 15 % and 25 % for first and second shell Debye-190 191 Waller factors, respectively (Binsted, 1998). 192 193 2.4. Density functional theory calculations 194 Quantum mechanical calculations on CuMn₁₈O₂₄(OH)₃₀(H₂O)⁻⁴ and CuMn₁₈O₂₁(OH)₃₃(H₂O)⁻¹ clusters were used to predict bond lengths and energies of 195 196 proposed Cu sorption complexes. Calculations were done using the ADF code of te Velde et 197 al. (2001). ADF implements density functional theory for finite clusters and molecules using 198 the linear combination of atomic orbital formalism. For all atoms except hydrogen, we used 199 frozen core orbitals (i.e., 1s, 2s, 2p, 3s and 3p for Mn and Cu; 1s for O). Molecular orbitals in 200 the ADF code are constructed from Slater type atomic orbitals which consist of a cartesian part $r^{kr}x^{kx}y^{ky}z^{kz}$ with $k_x + k_y + k_z = l$ (l = angular momentum quantum number) and an 201 202 exponential part e^{-r} . For all atoms, we used an uncontracted, triple-zeta basis set with 203 polarization functions (Van Lenthe and Baerends, 2003). No optimization of the basis set 204 was done. With this basis set, counterpoise corrections for basis set superposition energy, evaluated by replacing a Cu with a ghost atom in a simple $Cu(H_2O)_6^{+2}$ cluster, are found to 205 206 be negligible (< 1 kJ/mole). 207 The calculations were done using the PBE version (Perdew et al., 1996) of the

208 generalized gradient approximation (GGA) of the exchange-correlation functional (Perdew et

al., 1992). All calculations were done using the spin-unrestricted formalism to account for the

210 three unpaired 3d-electrons of Mn and the 1 unpaired 3d-electron of Cu; in each cluster, the

211 Mn and Cu atoms were set up in a ferromagnetic configuration. The geometries of the

212 clusters were optimized using a Newton-Raphson method and Broydon-Fletcher update of the

213 Hessian matrix as coded in ADF. The total energies of each complex during geometry

- 214 optimizations were converged to \pm 0.3 kJ/mole.
- 215

216 2.5. Surface complexation modelling

217 Surface complexation modelling was done using a FORTRAN90 program EQLFOR 218 (Sherman et al., 2008; Sherman, 2009) based on the original "tableau" speciation algorithm 219 described by Morel and Morgan (1972). Modifications to the mass balance, Jacobian and 220 convergence routines were made to include the surface and diffuse layer charges of the basic 221 (2 layer) Stern model (Sherman, 2009). Equilibrium constants for surface complexes were 222 derived by fitting all sorption edges simultaneously. The speciation routines are called by a 223 gradient search fitting algorithm (Bevington and Robinson, 2002) to optimize the equilibrium constants. The fitting was obtained by minimizing χ^2 , which is calculated assuming a relative 224 225 error of 2 % in concentrations/pH.

226 In the Morel and Morgan (1972) method, the speciation is calculated using mass 227 balance constraints to give the molal concentrations of surface species. The correct 228 thermodynamic formulation (i.e., one that gives the correct ideal configurational entropy of a 229 species) of the activity of a surface species is in terms of the mole fraction of surface sites 230 occupied by the species. The resulting stability constants are obtained from the apparent 231 (fitting to mass balance) stability constants by converting the concentration of each surface 232 species to its mole fraction. Activity corrections to the stability constants for aqueous species 233 were made using the Davies equation.

234

235 3. RESULTS AND DISCUSSION

236 3.1. Sorption of Cu on birnessite

The sorption of Cu to birnessite at several different surface loadings is shown in Figure 1. The results obtained here are very different from those obtained by Catts and Langmuir (1986) and Fu et al. (1991). In those experiments, the Cu sorption edge occurs at ~2 pH units higher for similar degrees of loading (Figure 1). Both of those studies used a

241 birnessite or δ -MnO₂ phase synthesized by reacting solutions of Mn⁺² and KMnO₄. X-ray

242 diffraction of the phases used by these authors are reported to have the same diffraction lines 243 as those obtained here; however, the intensities of the (001) and (002) reflections are 244 unknown. A highly crystalline birnessite in which the phyllomanganate sheets are stacked in 245 the c-direction will have a smaller fraction of vacancy sites relative to edge-sites available for 246 surface complexation. The discrepancies between the results obtained here and those reported 247 by previous authors may reflect differences in the crystallinity of the birnessite. Marine 248 birnessites typically show no (001) reflection and, consequently, surface complexation should 249 be dominated by vacancy rather than edge-site sorption.

250

251 3.2 EXAFS of Cu sorbed to birnessite

252 EXAFS spectra of Cu sorbed to birnessite (5.0 wt. %) at pH 4 and 8 are shown in 253 Figure 2. The Fourier transform of the EXAFS of Cu-birnessite at pH 4 indicates a first 254 coordination shell containing ~4 oxygen atoms near 1.95 Å and a further coordination shell 255 corresponding to ~6 Mn near 3.4 Å. We interpret this as indicating a Cu surface complex 256 formed over a vacancy site; consequently, we forward-modelled the EXAFS spectrum using 257 the cluster shown in Figure 3a. During the fit, we refined a total of 9 parameters (the 258 correction to the Fermi energy (EF), 2 Cu-O distances, 2 Cu-Mn distances and 4 Debye-259 Waller factors (DWF's). These are given in Table 1. Stern's rule (Stern, 1993) limits the 260 number of independent data parameters N_{ind} as $2\Delta k\Delta R/\pi + 2$ (Booth and Hu, 2009) where Δk 261 and ΔR are the range in k- and R-space actually fitted. For our experiment, N_{ind} = 15; hence 262 our model should not be over-fitting the data. Note that the apparent resolution of the two Mn 263 shells at 3.39 and 3.43 Å is not statistically significant since such small differences in Cu-Mn 264 distances cannot be resolved with the k-range used. The two distances simply reflect the 265 symmetry of the cluster used.

Our model for Cu-birnessite at pH 4 differs somewhat from the interpretation of Manceau et al. (2002); they fit their EXAFS of Cu-birnessite at pH 4 (described in Lanson et al., 2002) to 4 O at 1.96 Å and two O at 2.23 Å, with a much larger DWF, as expected for Cu in a Jahn-Teller distorted environment. However, we cannot resolve the second oxygen shell in our fit. Moreover, we do not expect Cu to be in 6-fold coordination when complexed to a surface vacancy. First, our predicted geometries from density functional calculations on a Cu surface complex cluster (discussed below) do not predict 6-fold coordination of Cu.

273 Secondly, it is now realized that the coordination number of Cu^{+2} in aqueous solution is 274 actually 5 (Pasquarello et al., 2001; Amira et al., 2005). On the other hand, we are in good

agreement with the interpretation of Manceau et al. (2002) regarding the outer shell of 6 Mn
atoms near 3.43 Å and we also find no evidence for Cu sorption at edge sites or for Cu
surface precipitation.

278 The EXAFS of Cu sorbed at pH 8 is similar to that at pH 4 except that there is an additional scattering due to Mn atoms near 2.9 Å. This indicates a second coordination 279 280 environment of Cu; we interpret this as Cu occupying a vacancy site. To forward-model the 281 EXAFS of Cu sorbed at pH 8, we used two clusters simultaneously: a surface complex 282 cluster (Figure 3a) and a cluster corresponding to Cu occupying the vacancy site (Figure 3b). 283 From the fit, we estimate that ~20 % of the Cu has occupied the vacancy site. The calculated 284 fit parameters are given in Table 1. The Cu EXAFS at pH 8 was refined with total of 12 285 parameters (EF, 2 Cu-O distances, 3 Cu-Mn distances, 5 DWF's and the fraction of Cu occupying a vacancy (see Table 1)). Since this is less than N_{ind} for the k- and r-ranges used, 286 we have not over-fit the data. However, as with the surface complex cluster, the apparent 287 resolution between the two Mn shells at 2.81-2.91 Å in cluster 2 is not statistically significant 288 289 since such small differences in Cu-Mn distances cannot be resolved with the k-range used.

By analogy with Ni (Peacock and Sherman, 2007a), a surface complex formed on the edge-sites of birnessite (as opposed to a complex formed over a vacancy), would have a Cu-Mn distance intermediate between that found for the vacancy complex and solid solution. Including such an edge complex is not necessary to fit the EXAFS; consequently, although edge-complexes may be present they cannot be resolved.

295

296 3.3 DFT Optimized geometries and energetics of surface complexes

297 The optimized structures and energies of clusters corresponding to Cu sorbed as a 298 surface complex and Cu occupying a vacancy sites are shown in Figures 4 and 5. The 299 geometry of the surface complex predicts the Cu will be in 3-fold coordination in an 300 approximately trigonal arrangement with an average Cu-O bond length of 1.93 Å and an average Cu-Mn distance of 3.29 Å. Note that the water molecules in Figure 5 are > 3Å away 301 302 from the Cu atom and not part of its coordination shell. The predicted average Cu-O distance 303 is in good agreement with that observed (1.95 Å) using EXAFS, although the EXAFS were fit 304 to four nearest neighbor oxygens. However, in the EXAFS fit, the DWF of the fourth oxygen 305 is quite large (0.02). The difference in coordination number between that predicted and that 306 derived from the EXAFS may simply reflect the uncertainty in the DWF's and coordination 307 numbers derived from EXAFS. The predicted average Cu-Mn distance (3.29 Å) is 4 %

308 smaller than that observed using EXAFS. The origin of this discrepancy is unclear but might,

- 309 for example, reflect longer range structural relaxation associated with next nearest-neighbor
- 310 vacancies. The discrepancy might also reflect a dependence of the Cu-Mn distance on the
- 311 protonation state of the vacancy oxygen as discussed below.
- 312 If Cu occupies the vacancy site (solid solution) instead of forming a surface complex, 313 the expected Jahn-Teller distortion gives Cu-O distances of 2.0 (x2), 2.1 (x 2) and 2.2 (x2) to 314 give an average distance of 2.1 Å which is about 8 % higher than those observed with 315 EXAFS; however, the experimental Cu-O distance is poorly constrained due to overlap with 316 the Cu-O scattering in the surface complex. The predicted average Cu-Mn distance of 2.92 Å 317 is in good agreement with the EXAFS result of ~2.9 Å where this peak is well resolved.
- The cluster calculations, moreover, provide insight on sorption via surface

319 complexation vs. sorption via solid solution. When the oxygen atoms surrounding the 320 vacancy site are not protonated, the surface complex cluster is calculated to be 40 kJ/mole 321 less stable than the cluster with Cu occupying the vacancy site. However, if the opposite 322 three oxygens surrounding the vacancy site are protonated as in Figure 4, the surface complex 323 is 40 kJ/mole more stable than the complex having Cu occupy the vacancy site. This could 324 explain why Cu forms only a surface complex at low pH but occupies a vacancy site at high 325 pH. A more quantitative understanding of the relative proton affinities of the surface oxygen 326 sites on birnessite is needed before we can predict the protonation of the surface oxygens as a 327 function of pH. Unfortunately, we still have an incomplete understanding of the surface 328 protonation of goethite ((-FeOOH) even though the goethite surface has received far more 329 attention than that of birnessite (e.g., Sherman, 2009 and references therein).

330

331 *3.4. Surface complexation modelling*

332 The batch sorption experiments show that the Cu sorption edge occurs below pH 4. 333 **EXAFS** results indicate that sorbed Cu forms a surface complex at this pH but starts to 334 occupy vacancy sites at high pH. The optimized structures and energies of clusters 335 corresponding to Cu sorption complexes suggest that the transformation between surface 336 complex and incorporation into the vacancy sites is a response to changes in the protonation 337 state of the vacancy site oxygens. Given this molecular understanding of Cu sorption, we can 338 now fit the sorption edges to a surface-complexation model. To this end, we need to know 339 the surface site densities and the protonation of the birnessite surface oxygens.

340

341 3.4.1 Surface site densities

342 As discussed above, at the loading used here, Cu sorbs to birnessite by forming an 343 inner-sphere complex above (and in) the vacancy sites. An idealized birnessite consisting of 344 randomly stacked H₄Mn₅O₂₄ layers with a vacancy structure, as described in the model for 345 the 1H phase of hexagonal birnessite (Lanson et al., 2000), would have 2.22 vacancy sites per nm^2 . However, the structural model of Villalobos et al. (2006) for δ -MnO₂ has only 0.8 346 vacancies per nm². Assuming a maximum reactive surface area of $611 \text{ m}^2/\text{g}$ (i.e., only half 347 348 of the surface area of a single phyllomanganate sheet is counted assuming only 1 Cu atom can 349 complex on each vacancy), we find the maximum sorption capacity for Cu would be 5.15 wt. %. However, given the measured BET surface area of $107 \text{ m}^2/\text{g}$ for the birnessite used here, 350 351 the sorption capacity should be only 0.9 wt% Cu. This is well below that observed in the 352 batch absorption experiments; the EXAFS at 5 wt. % loading do not show evidence for any Cu(OH)₂ phase or resolvable evidence for complexes on particle edges. Consequently, we 353 conclude that the BET surface greatly underestimates the reactive surface area of birnessite. 354 355 This has been shown to be the case for ferrihydrite (Dzombak and Morel, 1990). Because of 356 our uncertainty in the reactive surface area and in the number of available vacancy sites on the 357 {001} surfaces, the surface complexation modelling discussed below was done by using the 358 number of reactive sites as an adjustable parameter. Since sorption edges at three different 359 loadings were measured, it was possible to optimize the fit to the number of reactive sites. 360

361 3.4.2 Surface protonation

362

We modeled the protonation of $\{001\}$ surface oxygens on δ -MnO₂ using the

363 equilibrium

$$>Mn_2O^{-2/3} + H^+ = >Mn_2OH^{+1/3}$$
 (2a)

$$K_a = \frac{\{> Mn_2OH^{+1/3}\}}{\{> Mn_2O^{-2/3}\}[H^+]} \exp\left(-\frac{\Psi_0F}{RT}\right) \frac{366}{367} \qquad (2b)$$
where Ψ_0 is the electrostatic potential at the 0-

plane, *R* is the gas constant, *F* is the Faraday constant and *T* is temperature. We use the log K_a = 2.3 for reaction (2a) from the potentiometric titration done described above. Here, and in what follows, we define the (ideal) activity of a surface complex as the mole fraction of the surface sites that it occupies (this is relative to a standard state of complete coverage). This

372	definition gives the correct treatment of the configurational entropy and for multidentate
373	complexes this approach is essential (e.g., Tadanier and Eick, 2002). The surface
374	electrostatic potential \neg_0 is defined from the surface charge distribution using the basic Stern
375	model (Westall and Hohl, 1980). In the basic Stern model, the Stern layer (i.e., that between
376	the 0-plane and 1-plane) has a capacitance of $C_{stern} = \varepsilon_0 \varepsilon_s / d$ where ε_0 is the permittivity of a
377	vacuum (8.854 x 10^{-12}), ε_s is the Stern layer dielectric constant and <i>d</i> is the distance of charge
378	separation in meters (assumed to be the average distance of approach of hydrated counter
379	ions). Assuming d is ~ 4 Å and that the dielectric constant is intermediate between that of
380	pure water (78) and water at dielectric saturation (6), we estimate a Stern layer capacitance of
381	1 F/m^2 . This is comparable to that used by Boily et al. (2001) for goethite. However,
382	changing the capacitance by ± 0.5 F/m ² had a negligible effect on the goodness of fit or the
383	equilibrium constants for the sorption edges measured here.
384	For the surface protonation reactions, the change in charge is assigned to the 0-plane.
385	Following the approach used by previous workers (e.g., Hiemstra and Van Riemsdijk, 1996)
386	for iron oxides, we assign a formal reference charge $-2/3$ to oxygens in the Mn ₂ O surface
387	groups based on Pauling's second rule. (Note that it is only the change in surface charge that
388	contributes to the free energy of sorption; the assignment of a formal charge to surface sites is
389	simply to enable charge conservation in the mass-balance equations used when fitting the

- 390 sorption data to a surface complexation model).
- 391

392 *3.4.3 Cu surface complexation model*

Based on the results from EXAFS spectroscopy, we will neglect any complexes forming at the particle edges and assume that Cu^{+2} only forms complexes with the three > $Mn_2O^{-2/3}$ oxygens associated with vacancy sites on the {001} surface. We assume that the > Mn_3O^0 surface sites on {001} do not protonate or form complexes with Cu^{+2} . The formation of the surface complex is then:

398

399
$$3(>Mn_2O^{-2/3}) + Cu^{+2} = (>Mn_2O)_3Cu^0$$

400

401 with conditional (I = 0.1 M) equilibrium constant:

402

(3a)

403
$$K_s = \frac{\{(>Mn_2O)_3Cu^0\}}{\{>Mn_2O^{-2/3}\}^3[Cu^{+2}]} \exp\left(-\frac{2\Psi_0F}{RT}\right)$$
 (3b)

404

where $\{(>Mn_2O)_3Cu^0\}$ is the mole fraction of $\{001\}$ surface $>Mn_2O$ sites occupied by the 405 $(>Mn_2O)_3Cu^0$ complex. We assign all of the change in surface charge due Cu+2 adsorption 406 (+2) to the change in charge of the 0-plane. The optimized value of log $K_8 = 5.63$. The one-407 408 site surface complexation model fits to the Cu-birnessite adsorption data are shown on Figure 1. The sorption edges (pH < 5) are fit to only the $(>Mn_2O)_3Cu^0$ surface complex. The 409 410 transformation of the surface complex to the occupied vacancy site at high pH is not included 411 in the sorption equilibria. This means that we are not completely describing the surface 412 electrostatic contribution to the surface sorption at high pH. If Cu atoms occupying the 413 vacancy sites can disorder to form a complete solid solution with Mn, then we are not 414 correctly including the entropy of mixing. Neglect of these effects suggests that our model 415 will underestimate the sorption of Cu to birnessite at high pH (e.g., as in seawater).

416

417 *3.5 Application to the Deep Ocean*

Using our surface complexation model, we can estimate an upper limit for the 418 dissolved concentration of inorganic Cu^{+2} (e.g., Cu^{+2} , $CuOH^+$ and $CuCO_3$) that would be in 419 420 chemical equilibrium with a ferromanganese crust with 0.25 wt. % Cu. For a marine birnessite with 1.1×10^{-3} moles of vacancy sites/g, as estimated for the birnessite used in this 421 422 study (Table 2), this corresponds to a mole fraction of ~0.035 surface sites complexed to Cu. 423 To estimate the concentration of dissolved Cu in seawater, we implemented our surface 424 complexation model in PHREEQC (Parkhurst and Appello, 1999) together with the 425 MINTEQ.V4 database (Charlton and Parkhurst, 2002) and a solution with the major ion composition of seawater. We find that, at pH 7-8, a birnessite with 0.25 wt. % Cu would be 426 in equilibrium with a dissolved inorganic Cu⁺² concentration $< 10^{-6}$ nmolkg⁻¹. This is 427 many orders of magnitude below that observed ($\sim 4 \text{ nmolkg}^{-1}$); we conclude that essentially 428 429 all dissolved Cu in deep ocean seawater must be present in chelated form. This is consistent 430 with the observed deep-water complexation of Cu observed by Moffet and Dupont (2007).

431

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560 Figure Captions

561

501	
562	FIGURE 1: Sorption of Cu to birnessite $(\delta$ -MnO ₂) as a function of surface loading (1 wt. %,
563	2.5 wt. % and 5 wt. % Cu maximum adsorbed) and pH. Solid lines are fits from the surface
564	complexation model obtained in this work. Also shown (hollow symbols) are results from
565	previous experiments (Catts and Langmuir, 1986; Fu et al., 1992).
566	
567	FIGURE 2: EXAFS spectra (a) and their Fourier transforms (b) of Cu on birnessite at pH 4
568	and pH 8. Spectra were modeled using the clusters shown in Figure 3 with the parameters
569	given in Table 1. Note that the Fourier transforms calculated using EXCURV98 (Binsted,
570	1998) are corrected by Cu-O scattering phase shifts to give approximate actual distances.
571	
572	FIGURE 3: Clusters used to model the EXAFS of Cu on birnessite a) cluster with pseudo- C_{3v}
573	symmetry to model Cu as a surface complex over the vacancy b) cluster with pseudo- C_{2v}
574	symmetry to model Cu occupying a vacancy site. The distances were obtained by fitting to
575	the EXAFS spectra.
576	
577	FIGURE 4: Optimized geometry of Cu surface complex over a vacancy site on birnessite
578	calculated using density functional theory a) view down z-axis b) view down x-axis c) close-
579	up viewed down the, z-axis showing Cu-O and Cu-Mn distances. Note that oxygens
580	surrounding the vacancy site are protonated unless bonded to Cu.
581	
582	FIGURE 5: Optimized geometry of cluster corresponding to Cu occupying a vacancy site.
583	This is the minimum energy structure if all of the the oxygens surrounding the vacancy are
584	deprotonated .
585	

Table 1: Cu coordination environment from fits to Cu K-edge EXAFS spectra.

	10	30	3 Mn**	3 Mn**
R	1.95	1.95	3.39	3.43
φ	0	120	141	123
θ	0	0, ±120	±90	±30
2σ ²	0.024	0.011	0.02	0.02

Cluster 1: Surface Complex at pH 4 and 8

Cluster 2: Occupied vacancy complex at pH 8

PCC

	30	30	2 Mn**	2 Mn**	2 Mn**
R	1.87	1.96	2.81	2.94	2.91
φ	120	60	90	90	90
θ	0, ±120	180, ±60	±90	±30	±150
2σ ²	0.019	0.019	0.012	0.012	0.012

*Values in italics are constrained. φ and θ are the spherical coordinates defined as the angles from the z and x axes in Figure 3. **Note that, with the k-range used, distance differences < 0.15 Å between shells are not statistically significant.

Surface area $(m^2/g)^a$	107	
C _{stern} (F/m ²) ^b	1.0	
$[Mn_2O^{-2/3}] (mol/g)^c$	3.30×10^{-3}	
		2
Mineral surface reaction	Equilibrium constant log K	
$>Mn_2O^{-2/3} + H^+ = >Mn_2OH^{+1/3}$	2.3 ^d	
$3 > Mn_2O^{-2/3} + Cu^{+2} = (>Mn_2O)_3Cu^0$	5.6 ^c	
Solution Speciation		
$Cu^{+2} + H_2O = CuOH^+ + H^+$	-7.497°	
$Cu^{+2} + 2H_2O = Cu(OH)_2 + 2H^+$	-16.196 ^e	
$2Cu^{+2} + 2H_2O = Cu_2(OH)_2^{+2} + 2H^+$	-10.594 °	

Table 2: Surface complexation modelling parameters

^aDetermined from BET analysis (this study). ^bFixed (this study). ^cDetermined from fitting Cu sorption data (this study) ^dDetermined from potentiometric titration (this study). ^e From MINTEQ.V4 database (Charlton and Parkhurst, 2002).

Sherman and Peacock. Figure 1

(one column wide)







Sherman and Peacock Figure 3 (one column wide)



Sherman Figure 4 (one column wide)





Sherman and Peacock. Figure 5 (one column wide)

