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1 **Leaching behaviour of co-disposed steel making wastes: Effects of aeration**
2 **on leachate chemistry and vanadium mobilisation**

3

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18 **Keywords: Steel slag; vanadium; leaching behaviour; alkaline wastes**

19

20 **Highlights**

- 21 • Steel making by-products unsuitable for recycling stored in landfill
- 22 • Leaching mechanism is primarily controlled by air availability
- 23 • Air-exclusion led to Ca-rich, high pH leachates, with low V concentrations
- 24 • Carbonate formation during aeration led to lower pH but higher V concentrations

25

26 **Abstract**

27 Steelmaking wastes stored in landfill, such as slag and spent refractory liners, are often
28 enriched in toxic trace metals (including V). These may become mobile in highly alkaline
29 leachate generated during weathering. Fresh steelmaking waste was characterised using XRD,
30 XRF, and SEM-EDX. Batch leaching tests were performed under aerated, air-excluded and
31 acidified conditions to determine the impact of atmospheric CO₂ and acid addition on leachate
32 chemistry. Phases commonly associated with slag including dicalcium silicate, dicalcium
33 aluminoferrite, a wüstite-like solid solution and free lime were identified, as well as a second
34 group of phases including periclase, corundum and graphite which are representative of
35 refractory liners. During air-excluded leaching, dissolution of free lime and dicalcium silicate
36 results in a high pH, high Ca leachate in which the V concentration is low due to the constraint
37 imposed by Ca₃(VO₄)₂ solubility limits. Under aerated conditions, carbonation lowers the
38 leachate pH and provides a sink for aqueous Ca, allowing higher concentrations of V to
39 accumulate. Below pH 10, leachate is dominated by periclase dissolution and secondary phases
40 including monohydrocalcite and dolomite are precipitated. Storage of waste under saturated
41 conditions that exclude atmospheric CO₂ would therefore provide the optimal environment to
42 minimise V leaching during weathering.

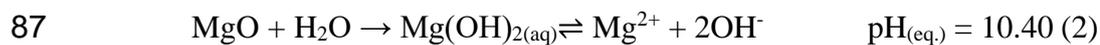
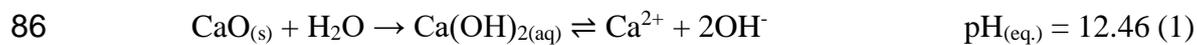
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44 1 Introduction

45 Steel slag is a ubiquitous byproduct of the steelmaking industry that is produced in large
46 quantities worldwide. It is estimated that 160-240 million tonnes of steel slag were produced
47 in 2016 (Ober, 2017) which corresponds to approximately 10-15 % of crude steel output
48 (Piatak et al., 2014). Primary steelmaking (i.e. conversion of iron to steel) produces steel slag
49 via two principal production methods; 1) basic oxygen furnace (BOF) steelmaking in which
50 molten iron from a blast furnace is used and 2) electric arc furnace (EAF) steelmaking which
51 uses a combination of scrap steel, directly reduced iron and pig iron. In both processes lime (or
52 limestone) is added to the furnace as a fluxing agent to remove impurities from the molten
53 metal (Eloneva et al., 2010, Piatak et al., 2014). The composition of BOF and EAF slag is
54 broadly similar and consistent across location and process (Tossavainen et al., 2007, Yildirim
55 and Prezzi, 2011, Proctor et al., 2000). These predominantly consist of Ca, Mg, Fe and Al
56 oxides and silicates (Proctor et al., 2000); the relative proportions of which will vary according
57 to the raw materials used during manufacture. Secondary steelmaking slags are formed during
58 secondary steelmaking where both BOF and EAF derived steels are further processed in ladle
59 furnaces, producing BOF(L) or EAF(L) slags respectively. These are much more variable in
60 composition and are also relatively enriched in Mg and Al due to additives used in the process
61 (Shi, 2002). In addition to slag production, a wide variety of refractories (MgO-C; Al-silicate;
62 MgO-Al₂O₃-C) are used as furnace liners during steelmaking to protect the furnace (Quaranta
63 et al., 2014). Refractories that are in contact with molten slag wear over time and the build-up
64 of solidified slag above the melt (due to sputtering) and in conduits also require regular
65 removal. Therefore, periodic renewal of the entire liner is required. The result is a mixed waste
66 containing both slag and refractories which is difficult to separate and are often co-disposed
67 (Hanagiri et al., 2008).

68 Primary steelmaking slags are recycled where possible, usually as aggregate in civil
69 engineering applications, such as road construction and as a general fill material due to its
70 stability (Geiseler, 1996, Yi et al., 2012). However, in some cases elevated concentrations of
71 free lime (CaO) and periclase (MgO), which expand on hydration, preclude reuse in
72 engineering applications. Ladle slags and refractories can also be recycled during primary
73 steelmaking (as an alternative source of CaO or MgO flux) but virgin materials are often
74 preferred due to their more uniform composition and the increased effort that would be required
75 in slag sorting and processing (Kwong and Bennett, 2002, de Sa et al., 2007). For this reason,
76 and also because supply frequently exceeds the demand for secondary aggregates, steelmaking
77 byproducts are often stored either in landfill or in open ‘heaps’. However, as recycling rates
78 increase, materials with problematic properties (e.g. high metal content, high % of CaO or
79 MgO, or simply uncertain or variable composition) will make up an ever greater proportion of
80 materials stored in landfill. It is therefore increasingly import to understand the leaching
81 behaviour of such non-standard by-products as they become a significant part of the disposed
82 inventory.

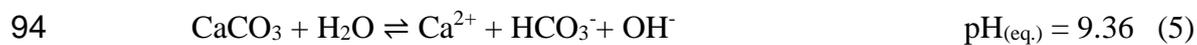
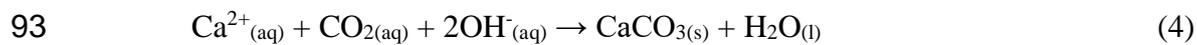
83 Steel slags contain free lime (CaO) and periclase (MgO; also in present in refractories)
84 that hydrate to form portlandite (Ca(OH)₂) or brucite (Mg(OH)₂). These phases dissolve in
85 water to generate high pH (10.5-12.5) leachate (Mayes et al., 2008):



88 Alkalinity may also be produced by the dissolution of Ca-silicates, (e.g. β-dicalcium silicate,
89 larnite; Roadcap et al. (2005)):



91 In addition, under aerated conditions, reactions involving the in-gassing of CO₂ results in
92 conversion of Ca/Mg hydroxide based alkalinity to carbonate alkalinity.



95 Whilst steel slag has historically been considered to be a non-hazardous waste, concerns
96 have been raised in recent years regarding the high concentrations of potential toxic trace
97 metal(loid)s (including Al, Cr, Pb, Mo and V) present in the slag (Tossavainen et al., 2007,
98 Proctor et al., 2000, Matern et al., 2013) which may be mobilised in the alkaline leachate
99 (Cornelis et al., 2008). V leaching in particular has received a lot of recent attention (Chaurand
100 et al., 2006, De Windt et al., 2011, Huijgen and Comans, 2006, Navarro et al., 2010) due to its
101 relative enrichment in steel slags and regulatory concern over high V concentrations in
102 leachates (Environment Agency, 2014). Steelmaking wastes at disposal sites are often
103 deposited in both saturated and unsaturated settings, however, few data currently exist
104 concerning slag leachate generation and chemistry under aerated conditions (Bayless and
105 Schulz, 2003, Roadcap et al., 2005, Mayes and Younger, 2006). Under aerated conditions,
106 CaCO₃ precipitation results in a drop in solution pH as OH⁻ ions are consumed (equation 4).
107 Changes in both pH and redox have significant implications for the mobility of many
108 potentially toxic metals; for example Al, Cr, and V mobility and toxicity are all highly
109 dependent on their speciation (with higher oxidation states generally regarded as more toxic
110 and mobile) (Pourbaix, 1966).

111 This study investigates leaching behaviour of co-disposed steelmaking waste under
112 aerated and air-excluded conditions. These represent contrasting conditions present either near
113 the surface (good contact with atmosphere) and below the water table deeper within waste
114 heaps. The waste has been characterised using x-ray diffraction (XRD) and fluorescence (XRF)

115 techniques, as well as scanning electron microscopy (SEM) with energy-dispersive x-ray
116 spectroscopy (EDX) to determine the mineral phases present. Fully aerated and air-excluded
117 water leaching tests have been performed to determine how phase dissolution behaviour,
118 secondary mineral formation and trace metal release are affected by changes in pH and
119 availability of air. Results will assist prediction of metal release from waste when stored in
120 landfill above and below the water table, enabling effective environmental risk assessment and
121 cost-effective long-term management of the waste.

122 2 Methods and Materials

123 2.1 Sample collection and characterisation

124 Samples were collected within one week of deposition from the Yarborough Landfill
125 (British Steel, Scunthorpe, UK) in May 2013 (LAT 53°35'22.24" LONG 0°35'41.52"). The
126 sample consisted of 50-500 g blocks (~100 kg total). A sub-sample of the collected material
127 (approx. 500 g consisting of 50 g pieces) were brushed to remove any fines and crushed to
128 provide a homogenised powder consisting of 20-100 µm particles. The crushed waste was
129 stored in a polythene bag within an airtight glass jar containing soda lime to prevent weathering
130 due to contact with atmospheric CO₂ and moisture.

131 Elemental analysis of the powdered waste was undertaken using a PANalytical Axios
132 Advanced X-ray Fluorescence (XRF) spectrometer (data corrected for loss on ignition at 1050
133 °C). Samples were prepared for major element analysis as fused beads with lithium
134 metaborate/tetraborate flux (Johnson Matthey Spectroflux JM100B) (0.6 g sample; 3 g flux).
135 For minor/trace element analysis pressed pellets were prepared containing ~10 g of dried waste
136 using ~10–20 drops of 6.6 % w/v polyvinyl alcohol in a 1:6 mix of methanol and distilled
137 deionized water as a binder (Moviol 88 solution). The elemental limit of detection (LoD) was
138 generally < 0.02 wt% and the analytical uncertainty (versus certified reference standards) was
139 < ±10% of the data value. Mineralogical analysis (LoD was approximately 3 wt% for
140 crystalline phases) was undertaken by powder X-ray diffraction (XRD) using a Bruker D8
141 diffractometer, where powder samples were mounted on silicon slides and scanned between 2°
142 and 70° 2θ using Cu K_α radiation. Diffraction peaks from unknown samples were then matched
143 to known standard reference patterns using Diffrac.Suite Eva v3.0 software using the
144 International Centre for Diffraction Data (ICDD) PDF2 database (powder diffraction file (PDF)
145 reference numbers are reported for identified phases).

146 Two polished blocks were prepared by first cutting waste pieces to size under water
147 and setting the resultant $\sim 2 \text{ cm}^3$ blocks into epoxy resin with the cut surface exposed. This
148 surface was then polished using a water-free diamond paste to remove the top 1-2 mm of
149 material potentially exposed to water during cutting. Electron micrographs were subsequently
150 collected on a FEI QUANTA 650 FEG ESEM, which was equipped for Oxford Instruments
151 INCA 350 energy-dispersive X-ray spectroscopy (EDS) system/80 mm X-Max silicon drift
152 detector. EDS spectra and elemental maps were collected and analysed using Oxford
153 Instruments AZtec software. The element and sample specific LoD for EDS analysis was
154 between 0.1 and 0.5 wt%.

155 2.2 Acid Neutralisation Capacity (ANC)

156 Homogenised powdered waste (0.4 g) was mixed with 40 mL HCl with concentrations
157 ranging from 1 M to 0.001 M in 50 mL polypropylene Oak Ridge tubes (Nalgene, USA).
158 Experiments were performed in triplicate. Headspaces of the 50 ml tubes were flushed with N_2
159 gas prior to sealing. All tubes were subsequently stored in airtight 2 L glass jars (Le Parfait,
160 France) filled with N_2 gas and also containing $\sim 100 \text{ g}$ soda lime (as a CO_2 absorbent) to prevent
161 any CO_2 infiltration from atmosphere to the experimental tubes. After 1 and 50 days
162 equilibration the 50 ml tubes were centrifuged in a Sigma 2-16 centrifuge using a 12151 rotor
163 at 8000 rpm (6000 g) for 5 minutes to separate aqueous and solid phases. 1 mL of the
164 supernatant was removed and immediately added to 9 mL 0.1 M HNO_3 prior to ICP analysis,
165 and the pH of the remaining solution was measured (pH measurement is described below).

166 2.3 Leaching tests

167 Triplicate aerated experiments containing 1 g homogenised powdered waste and 100
168 mL deionised water were established in open 500 mL PETG Erlenmeyer flasks. These were
169 gently agitated on a Stuart Scientific SSL1 orbital shaker at 175 rpm to allow for equilibrium

170 of the reaction solution with air (i.e. dissolution of atmospheric CO₂ and O₂). At regular
171 intervals over a 50 day period 3 mL aliquots of slurry were removed from the flasks and
172 centrifuged in 2 x 1.5 ml Eppendorf tubes in a Spectrafuge 16M microcentrifuge at 14,000 rpm
173 (16,000 g) for 5 minutes to separate aqueous and solid phases. 1 mL of supernatant was
174 removed and acidified in 0.1 M HNO₃ and the pH of the remaining supernatant was determined.
175 The moist solid samples were stored at -20 °C prior to drying at 40 °C overnight for further
176 analysis.

177 Replicate air-excluded experiments were established containing 0.4 g homogenised
178 powdered waste and 40 mL deionised water in 50 mL Oak ridge tubes. All tubes were
179 anaerobically handled as described above for the ANC tests. Periodically over a 50 day period
180 3 tubes were sacrificially sampled. Solid and solution samples were taken and stored following
181 the procedures above.

182 2.4 Aqueous Analysis

183 Solution pH (± 0.2 pH units) was measured using an Orion DualStar pH/ISE benchtop
184 meter (Thermo Scientific, USA) with an electrode that was calibrated daily at pH 4, 7 and 10.
185 Nitrogen gas was bubbled through the sample tube during pH measurements made in solutions
186 from the air-excluded experiments to prevent contact with atmosphere. Metal concentrations
187 in acidified aqueous samples from the air-excluded leaching experiments were determined
188 (with an analytical uncertainty of $< \pm 3\%$) using a Thermo iCAP 7400 ICP-OES ion-coupled
189 plasma, optical emission spectrometer (ICP-OES; for Ca, Si, Mg; LoDs were $< 10 \mu\text{g L}^{-1}$) and
190 Thermo iCAP Qc ion-coupled plasma, mass spectrometer (ICP-MS; for V, Mn, and Cr with
191 LoDs $< 0.1 \mu\text{g L}^{-1}$; for Li and Fe with LoD $< 2 \mu\text{g L}^{-1}$). Calibration was against certified multi-
192 element standards and check samples were analysed every 10 samples to check for calibration
193 drift. Aqueous elemental concentrations from the aerated leaching experiments were

194 determined using a Perkin Elmer Optima 5300DV ICP-OES for all elements (LoDs were < 100
195 $\mu\text{g L}^{-1}$ for all elements). The 5300DV was calibrated with 3 standards (0.1, 1, 10 mg L^{-1}) and a
196 blank which were run every 15 samples and the calibration cross-checked with a trace metal
197 Certified Reference Material (NIST1646A) with all values within $\pm 5\%$.
198

199 **3 Results**

200 3.1 Material Characterisation

201 The elemental composition (Table 1) of the powder sample was dominated by Ca, Fe,
202 Mg, Si and Al with Cr and Mn as minor constituents. Trace elements included P, V, K, Ti
203 and S.

204 [TABLE 1]

205 Mineralogical analysis of the crushed waste using XRD (Fig. 1a) identified the presence
206 of periclase (MgO; PDF #44-946), corundum (Al₂O₃; PDF #10-173), larnite (dicalcium
207 silicate, β-Ca₂SiO₄; PDF #33-302), brownmillerite (dicalcium aluminoferrite, Ca₂(Al,Fe)₂O₅;
208 PDF #30-226), wüstite (FeO; PDF #46-1312), brucite (Mg(OH)₂; PDF #44-1482) and
209 elemental carbon (graphite-C; PDF #23-64). The principal XRD peak of free lime (CaO; PDF
210 #37-1497) overlaps with a secondary peak of corundum at ~37.5° 2θ, however, the secondary
211 peaks at ~ 32 and 54 ° 2θ were present, suggesting its presence. There was no calcite peaks in
212 the XRD pattern for the unreacted waste (n.b. the principle calcite peak is at ~29 ° 2θ).

213 [FIGURE 1]

214 SEM analysis of a polished block (Fig. 2) showed a material that was composed of
215 intergrown 10-30 μm crystallites, EDS elemental mapping indicated that there were three
216 dominant compositions. The dominant composition by area (~ 65 % of the sample viewed) was
217 a Ca-Si-O rich phase containing trace amounts of Al, P, Ti, V and Fe by EDS analysis,
218 consistent with the larnite phase identified by XRD. The second most abundant composition
219 by area (~25 %) was a Mg-Fe-O rich phase containing trace amounts of Na, Ca, Cr and Mn,
220 corresponding to the wüstite phase identified by XRD. The third most abundant composition
221 by area (~10 %) was a Ca-Fe-Al-O rich phase containing trace amounts of Ti, V, Cr and Mn,
222 corresponding to the brownmillerite phase identified by XRD. Regions of the dicalcium silicate

223 phase contain lamellae identified as CaO rich regions associated with dicalcium silicate. The
224 second polished block was composed of a single Al-Si-O rich composition containing no
225 significant trace elements (data not shown).

226 [FIGURE 2]

227 3.2 Acid Neutralisation Tests.

228 For crushed waste, the relationship between the final solution pH value and the amount
229 of acid added was similar after 1 day and 50 days of equilibration (Fig. 3a). The acid
230 neutralisation capacity (to pH 7) was very similar at the two time-points (11.0 and 9.5 M H⁺
231 kg⁻¹ waste, respectively). In tests where the final pH value was greater than 7, the pH value
232 increased with time. The solution Ca and Mg concentrations at a particular pH value were
233 broadly similar after 1 day and 50 days (Fig. 3c, d), however, Si (Fig. 3b) was present in
234 solution at all final pH values < 10 after 1 day, but was present at significantly reduced
235 concentrations at 50 days at all pH values. Calcium was released to solution at all pH values,
236 with generally higher concentrations at lower pH values. Magnesium was only present in
237 solution below ~pH 10, but also has generally higher concentrations at lower pH values.

238 [FIGURE 3]

239 3.3 Batch leaching tests

240 During air-excluded leaching (Fig. 4), the leachate pH rose rapidly within the first hour
241 to a value of 11.4, and then increased more slowly to a maximum value of 11.9 after 52 days.
242 The Ca concentration increased rapidly to ~2.5 mmol L⁻¹ over the first 24 hrs, and then more
243 steadily to ~3.0 mmol L⁻¹ after 5 days. Between 5 days and 10 days the Ca concentration
244 decreased to ~2.5 mmol L⁻¹, but thereafter increased steadily to ~3.4 mmol L⁻¹ after 50 days.
245 The Mg concentration showed no trend with time and was less than 0.4 mmol L⁻¹ throughout
246 the experiment. The Si concentration increased to ~0.15 mmol L⁻¹ over the first 24 hrs, and

247 then to $\sim 0.25 \text{ mmol L}^{-1}$ after 5 days. After about 10 days the Si concentration decreased to
248 $\sim 0.15 \text{ mmol L}^{-1}$ but subsequently varied between ~ 0.15 and $\sim 0.25 \text{ mmol L}^{-1}$ until the end of
249 the test. The V concentration gradually increased from $\sim 0.002 \text{ mmol L}^{-1}$ after 24hrs to ~ 0.007
250 mmol L^{-1} after 50 days. Aqueous Fe concentrations range from 0.002 to 0.07 mmol L^{-1} and
251 were just above the limit of detection by ICP-MS ($0.0018 \text{ mmol L}^{-1}$), which increases scatter
252 in the data. However it is still discernible that Fe accumulated in solution during the air-excluded
253 experiments and did not in parallel aerated experiments. There was very little change in XRD
254 patterns collected from the solid residue over time, except that portlandite (PDF #44-1481) was
255 detected in samples collected from the experiment end points (Fig 1b).

256 [FIGURE 4]

257 During the aerated experiments (Fig. 4), pH increased rapidly within the first hour to a
258 maximum value of 11.6. This was followed by a decrease to pH 8.9 after 1 day and then by a
259 second rise to pH 10.0 after 6 days. Subsequently pH steadily declined to 9.3 after 50 days of
260 leaching. The Ca concentration increased rapidly to $\sim 2 \text{ mmol L}^{-1}$ within the first two hours, but
261 then decreased to $\sim 0.25 \text{ mmol L}^{-1}$ after 6-7 days. Initially the Mg concentration was low but it
262 increased rapidly after day 2 to reach a concentration above 12.5 mmol L^{-1} by day 20, which
263 persisted until the end of the test. The Si concentration was $\sim 0.15 \text{ mmol L}^{-1}$ after 1 hour,
264 decreased to $\sim 0.03 \text{ mmol L}^{-1}$ after 2 days but then rose to $\sim 0.25 \text{ mmol L}^{-1}$ at 6 days.
265 Subsequently Si concentrations remained above 0.2 mmol L^{-1} for the remainder of the test. The
266 V concentration increased steadily over the duration of the test to reach a maximum of ~ 0.065
267 mmol L^{-1} after 48 days. The aqueous Fe concentrations were initially around 0.05 mmol L^{-1}
268 and decreased over 5 days to $< 0.002 \text{ mmol L}^{-1}$ for the remainder of the tests. XRD patterns
269 collected from solid residue after 1 day show that calcite (PDF #24-27) was present (Fig. 1c),
270 after 6 days the periclase and larnite peaks had reduced in relative intensity and
271 monohydrocalcite (PDF #29-306) was detected (Fig 1d). After 50 days monohydrocalcite was

272 the dominant secondary phase detected (although some dolomite, PDF #36-426, was also
273 present) and periclase and larnite peaks were absent (Fig. 1e).

274 Concentrations of other trace elements considered to be potential environmental risk
275 drivers for steel slags (e.g. Mn, Cr, Li) were low in both the aerated and the air-excluded tests
276 (and were present at concentrations close to or below to the limit of detection).

277

278 4 Discussion

279 4.1 Waste characterisation

280 The bulk chemical and mineralogical composition of the waste was distinct from BOF
281 slag samples collected at the Yarborough site (Table 1). Compared to BOF slag the material
282 was relatively depleted in elements such as Ca, Fe, Mn and V; but enriched in Mg, Al, Ti and
283 Cr. The waste mineralogy could be split into two distinct groups. The first group included a
284 range of phases commonly occurring in BOF slag (i.e. larnite, brownmillerite, lime, and
285 wüstite; Yildirim and Prezzi (2011), Proctor et al. (2000), Geiseler (1996)), which were
286 observed in SEM as discrete particle assemblages within an intergrown matrix of 20-50 µm
287 crystallites. The second group of minerals identified by XRD included corundum, periclase
288 and graphite that are rare in BOF slag and are more commonly associated with refractory liner
289 materials (e.g. in MgO-C and Al_2O_3 -MgO-C refractories; Rovnushkin et al. (2005)).
290 Aluminosilicate was also observed in SEM analysis. This was most likely the high temperature
291 phase mullite ($Al_6Si_2O_{13}$; Schneider et al. (1994), Chesters (1973), Mazdiyasi and Brown
292 (1972), Tamari et al. (1993)) which is also used as a refractory material or can form in slags
293 from reaction of corundum and Si (Dana and Das, 2004, Zhao et al., 2014). Therefore, the
294 waste can be characterised as a mixed steelmaking waste containing both BOF slag and
295 refractory materials. The potential for re-use of mixed wastes is low due to uncertainties about
296 their chemical and physical behaviour, however, it is important to understand their potential
297 leaching behaviour during disposal.

298

299 4.2 Acid neutralisation behaviour

300 The acid neutralisation capacity experiments were conducted under air-excluded
301 conditions and provide information on phase dissolution in the waste as a function of the acid

302 addition and final solution pH. At low acid additions ($< 1 \text{ mol H}^+ \text{ kg}^{-1}$) the waste buffered pH
303 to values > 11.5 due to the dissolution of free lime and dicalcium silicate (Eqns. 1 and 3), with
304 dissolution of brucite from periclase hydration also consuming acid when the pH < 10.4 (Eqn.
305 2). Ca was released to solution at all pH values with the amount of release increasing with
306 decreasing pH (Fig. 3c), and Mg was released to solution when pH values < 10 with the amount
307 of release also increasing with decreasing pH (Fig. 3d), which is consistent with the dissolution
308 behaviour of these phases (Eqns 1 and 2). The alkalinity producing phases were progressively
309 exhausted by larger acid additions and the final pH decreased steadily until neutral values were
310 reached with an acid addition of $\sim 8 \text{ mols H}^+ \text{ kg}^{-1}$. CaO hydration is a very fast reaction (Shi et
311 al., 2002), whereas dicalcium silicate hydration and dissolution is generally considered to be
312 slower (Taylor, 1986). However, here we find that both are involved in buffering the pH in the
313 first 24 hrs as Si is released to solution. Nevertheless, the 10:1 ratio in the aqueous Ca and Si
314 concentrations indicated an excess of Ca release relative to pure dicalcium silicate dissolution
315 (i.e. Ca : Si $\approx 2:1$) suggesting that free lime was the dominant Ca bearing phase that was
316 dissolving over 24 hrs.

317 The difference in the final pH of the 1 and 50 day tests with the same acid addition
318 indicates that part of the alkalinity generation occurs over longer time periods. At high pH this
319 was most likely associated with continued dissolution of dicalcium silicate, although the Si
320 concentrations decreased between day 1 and 50. This Si decrease was most likely associated
321 with the formation of secondary Si-containing phases. These phases were most probably
322 calcium-silicate-hydrate (Ca-Si-H) phases at pH values > 9 (Walker et al., 2016, Costa et al.,
323 2016) and amorphous silica ($\text{SiO}_2_{(\text{am})}$) at lower pH values (Langmuir, 1997). The increase in
324 Mg concentration between day 1 and 50 at pH values < 9 suggests periclase hydration also
325 continued beyond 24 hours.

326

327 4.3 Leaching behaviour under aerated conditions

328 At the initial sampling point (1 hr) during the aerated leaching tests, the pH value was
329 11.5 ± 0.2 and both Ca and Si were released to solution. This rapid initial release of alkalinity,
330 Ca and Si was due to the hydration and dissolution of both free lime and dicalcium silicate
331 phases present in the waste. Previous slag leaching tests using granulated BOF slag pieces
332 found dicalcium silicate dissolution to be significantly slower than free lime dissolution
333 (Stewart et al., 2018). The use of crushed powder samples in these tests which will have
334 contained high surface area fines must have promoted rapid initial dissolution of dicalcium
335 silicate. However, as dicalcium silicate dissolution releases Ca and Si to solution in an
336 approximately 2:1 ratio (Hobson et al., 2017), the much higher initial Ca/Si ratios in solution
337 (between 13 and 20) indicates that free lime dissolution was most likely to be the predominant
338 source of rapidly leached alkalinity in these tests.

339 In the 2 days after the initial release of alkalinity Ca and Si to solution, the pH reduced
340 to 8.9 contemporaneously with rapid Ca and Si removal. The decrease in pH and Ca
341 concentrations coincides with the appearance of calcite peaks in the XRD plots after 1 hour
342 (Fig. 1) indicating that the initial spike in pH was buffered down to 8.9 due to in-gassing of
343 atmospheric CO_2 and the subsequent precipitation of CaCO_3 , which consumes both OH^- and
344 Ca^{2+} ions (Eqn. 4). The contemporaneous decrease in Si concentrations may be evidence of the
345 formation of a calcium silicate hydrate phase (Ca-Si-H). Indeed, low Ca/Si ratio (< 1) Ca-Si-H
346 phases are predicted to form under the observed initial pH and Ca concentrations on a timescale
347 of 24-48 hours (Walker et al., 2016). The peaks associated with dicalcium silicate (larnite) in
348 XRD patterns became less prominent over time and were ultimately absent by the end of the
349 experiments, indicating continued dicalcium silicate dissolution beyond day 6. Despite this no
350 further increase in Ca concentration was observed after 2 days, probably because most of the
351 Ca released by dicalcium silicate dissolution was precipitated as CaCO_3 under aerated

352 conditions (monohydrocalcite peaks became increasingly prominent in the XRD pattern over
353 time). Si concentration increased to a maximum value between 2-6 days contemporaneously
354 with a second observed peak in pH. There are only modest changes in Si concentrations
355 observed after 6 days, suggesting that equilibrium with secondary Si-containing phases such
356 as Ca-Si-H or amorphous SiO₂ (Langmuir, 1997, Costa et al., 2016, De Windt et al., 2011) was
357 limiting Si concentrations in these experiments.

358 XRD analysis indicates that there was progressive loss of periclase from the solids over
359 time, such that it was absent from the final XRD pattern. There were also small brucite peaks
360 in the XRD patterns from all time points. Periclase hydrates in water to form brucite
361 (Mg(OH)₂), which readily dissolves at pH values < 10.4 (Eqn. 2). No significant Mg release to
362 solution was observed during the first 24 hours of leaching because brucite is relatively
363 insoluble at the then prevailing high pH value that was imposed by CaO and calcium silicate
364 weathering. After the initial spike in the pH value on the first day, the solution pH was lower
365 than the brucite equilibrium value, and the aqueous Mg concentration increased as brucite
366 dissolved. As a result, periclase hydration and brucite dissolution provided an additional source
367 of alkalinity to the system. Dissolution of brucite leads to a switch in solution chemistry from
368 a Ca to an Mg dominated system. Monohydrocalcite was observed in XRD patterns after day
369 6, after which it becomes the dominant carbonate phase, demonstrating a switch from calcite
370 formation in the Mg-free early part of the experiments to predominately monohydrocalcite
371 formation in the Mg-dominated system present after day 6. Recent studies have shown that an
372 Mg-rich environment, such as that observed after 5 days, may support the precipitation of
373 monohydrocalcite (CaCO₃·H₂O) into which Mg may be incorporated (Rodriguez-Blanco et al.,
374 2014). At later time points dolomite (CaMg(CO₃)₂) was detected in XRD analysis, which may
375 also form due to the high Mg concentrations suppressing calcite or aragonite formation during
376 recrystallisation of monohydrocalcite (Rodriguez-Blanco et al., 2014).

377

378 4.4 Leaching behaviour under air-excluded conditions

379 At the first sampling point at 1 hr the pH value and the Ca and Si concentrations were
380 very similar to those observed in the aerated experiments (pH 11.5 ± 0.2 , [Ca] 1.5 ± 0.5 mmol
381 L⁻¹ and [Si] 0.15 ± 0.025 mmol L⁻¹) indicating similar processes were occurring over the first
382 hour of leaching in both experiments (i.e. rapid dissolution of free lime and fine grained
383 dicalcium silicate particles). As a saturated Ca(OH)₂ solution will reach a pH of ~12.5 (Eqn. 1)
384 and experimental pH values were between 11 and 12 at all-time points, it suggests that the
385 magnitude of the initial rise in pH was mass limited (i.e. controlled by the amounts of CaO and
386 reactive dicalcium silicate that are readily available for dissolution) rather than Ca(OH)₂
387 solubility limited. Between the first sampling point and day 5 there was a slow increase in pH,
388 Ca and Si which was probably the result of continued dicalcium silicate dissolution.

389 Between day 5 and 10 in the air-excluded experiments there was a decrease in Ca
390 concentrations that coincided with a decrease in Si concentrations, which suggests that a Ca-
391 Si-H phase formed. The pH, Ca and Si concentrations present at the time of formation in the
392 leaching experiments were consistent with precipitation of a Ca-Si-H phase with Ca/Si ratio of
393 close to 1 (Walker et al., 2016). The continuing slow rise in pH, Ca and Si concentrations
394 during the remainder of the experiment indicate that under air-excluded conditions, the solution
395 composition slowly evolves towards dicalcium silicate solubility limits over time (also
396 observed by De Windt et al. (2011)). However, Ca-Si-H gel formation can cover particle
397 surfaces, making alkalinity generation a diffusion-limited process (Hobson et al., 2017, Costa
398 et al., 2016, Nikolić et al., 2016), slowing the dissolution of the remaining reactive solid phases,
399 and leading to the incomplete dissolution of larnite (dicalcium silicate) observed at the end of
400 these experiments (Fig. 1b). No Mg was released to solution under air-excluded conditions

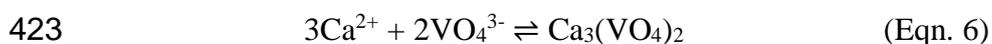
401 because the pH remained above 10.4 throughout the experiments, and therefore, dissolution of
402 the Mg containing phases was inhibited.

403

404 4.5 Control of trace metal release

405 Fe can be released to solution during dissolution of both free lime and dicalcium silicate
406 (in which it can be a minor constituent; Hobson et al. (2017)), and therefore Fe is present in the
407 leachate throughout the air-excluded experiments (The slightly lower Fe concentrations after 5
408 days under air-excluded conditions may be due in part to incorporation of some Fe into Ca-Si-
409 H phases). Conversely, under aerated conditions Fe is only present in significant concentrations
410 during the first 3 days during which time the leachate pH falls from 11.5 to 9. Under aerated
411 conditions, any Fe(II) released is likely to be readily oxidised to Fe(III) and precipitated either
412 as an insoluble hydroxide or incorporated into spinel-like secondary phases (e.g. magnetite,
413 Fe₃O₄; De Windt et al. (2011)) at pH 9, limiting Fe accumulation in solution.

414 Vanadium release in the air-excluded experiments is low (maximum of ~0.007 mmol
415 L⁻¹) compared to that observed in the aerated experiments (maximum of ~0.065 mmol L⁻¹).
416 Previous work on BOF slag weathering suggests that it is V(V) associated with dicalcium
417 silicate which is most readily leached to solution (Hobson et al., 2017), which is present in high
418 pH solution as the vanadate oxyanion (VO₄³⁻; Wehrli and Stumm (1989)). Thus, V
419 concentrations observed during BOF slag leaching are likely to be controlled by Ca₃(VO₄)₂
420 solubility limits (K_{sp} = 10^{-17.97}; Huijgen and Comans (2006), Cornelis et al. (2008), Allison et
421 al. (1991)) which impose an inverse relationship between Ca and V concentrations in the
422 leachate (Fig. 5).



424 Under air-excluded conditions Ca released from CaO and dicalcium silicate weathering
425 accumulated in solution, producing a leachate with high Ca concentrations. The leachate
426 solution, therefore, quickly reached $\text{Ca}_3(\text{VO}_4)_2$ solubility limits, preventing further release of
427 V to solution and limiting V concentrations in the leachate.

428 Aerated conditions, however, allow in-gassing of atmospheric CO_2 and associated
429 formation of secondary carbonate minerals. This process provides a sink for Ca, lowering
430 aqueous Ca concentrations. The V that is released therefore persists in solution (due to the
431 lower $[\text{Ca}^{2+}][\text{VO}_4^{3-}]$ ion activity product), leading to much higher concentrations than those
432 seen under air-excluded conditions. Monohydrocalcite is relatively soluble ($K_{\text{sp}} = 1 \times 10^{-7.1}$;
433 Kralj and Brečević (1995)), therefore, the equilibrium phase controlling Ca concentrations in
434 these experiments is probably dolomite ($\text{CaMg}(\text{CO}_3)_2$; $K_{\text{sp}} = 1 \times 10^{-17.2}$; Sherman and Barak
435 (2000)). Indeed, the predicted Ca concentration in solutions equilibrated with dolomite and
436 atmospheric pCO_2 are similar to those found at the end of these experiments (Fig 5; (Langmuir,
437 1997)). In the aerated experiments, the overall

438 [FIGURE 5]

439 leachate chemistry remains undersaturated with respect to $\text{Ca}_3(\text{VO}_4)_2$, which is consistent with
440 the observed depletion of the V-hosting dicalcium silicate phase by the end of the aerated
441 experiments.

442

443 4.6 Implications for waste management

444 Co-disposed wastes such as those used in this study are unattractive for reuse in
445 furnaces due to their highly variable composition making it difficult to control the steelmaking
446 processes. Furthermore, the presence of MgO makes their reuse in civil engineering
447 applications difficult, since periclase hydration to form brucite leads to significant volumetric

448 expansion and cracking. Given these problematic characteristics, combined steelmaking wastes
449 are generally stored in landfill. Over time, rainwater inevitably infiltrates and reacts with
450 landfilled waste to generate leachate. Therefore, the storage environment plays a key role in
451 establishing leachate composition. There is a key interaction between Ca concentrations and
452 the solution pH, which controlled by the equilibrium with the dominant Ca-phase present (i.e.
453 portlandite, Ca-Si-H or CaCO₃). The presence of CO₂ is a key factor in determining which
454 secondary Ca phases are present during leaching and also the degree to which the primary
455 dicalcium silicate was leached (therefore, indirectly affecting the release of trace metals such
456 as V).

457 Under air-excluded conditions, the leaching profile of the combined waste is similar to
458 that of BOF slag (Hobson et al., 2017, De Windt et al., 2011, Huijgen and Comans, 2006, Costa
459 et al., 2016). Rapid dissolution of Ca-bearing phases without in-gassing of CO₂ leads to a high
460 pH, Ca(OH)₂ dominated leachate. High Ca concentrations limit V release due to Ca₃(VO₄)₂
461 solubility limits. Consequently, water saturated environments with limited opportunity for CO₂
462 ingress will provide the safest environment for waste storage. Leachate could either be
463 recirculated (and reach equilibrium with Ca(OH)₂) or be removed and treated off site to lower
464 its pH and allow carbonation without triggering V release (Gomes et al., 2017). Under aerated
465 conditions, lower alkalinity and Ca concentrations allow much higher V concentrations to
466 accumulate in leachate, as well as allowing dissolution of MgO to form a Mg(OH)₂ dominated
467 leachate which favours the formation of monohydrocalcite and dolomite as secondary
468 carbonate phases. Leaching under aerated conditions rapidly produces leachate with V
469 concentrations in excess of acute toxicity thresholds (280 µg L⁻¹; Fig 5) and all experiments
470 display V concentrations higher than chronic toxicity thresholds (19 µg L⁻¹, Buchman, 2008).
471 New proposals for V exposure based on recent reviews of ecotoxicology data (Smit, 2012)
472 have suggested new water quality standards for V of 1.2 and 3.0 µg L⁻¹ for chronic and acute

473 exposure respectively. Almost all slag leachates reported (Fig 5) would exceed these more
474 stringent limits, therefore, leachate generated from co-disposed steel making wastes would
475 require careful management (and treatment prior to discharge) to avoid the potential for
476 environmental harm.

477 Extrapolating from experimental results to field scale predictions must always be done
478 with caution. For example, these experiments were performed on crushed powder samples at
479 much lower solid to liquid ratios than would typically be found in a heap leaching scenario. It
480 is much more likely that during heap leaching (due to high solid : solution ratios present) free
481 lime and dicalcium silicate dissolution will control leachate quality resulting in higher Ca
482 concentrations (i.e. leachate saturated with respect to $\text{Ca}(\text{OH})_2$) and therefore lower V
483 concentrations compared to those seen in experimental systems (indeed most real site leachates
484 commonly contain V at order of magnitude lower concentrations than found in laboratory
485 experiments; Fig 5). However, the experimental results indicate that although carbonation
486 reactions are helpful in reducing pH and alkalinity over time, the Ca concentrations produced
487 may allow higher V concentrations to accumulate in leachates over time. Therefore leachates
488 produced from co-disposed steelmaking wastes will likely require monitoring and secondary
489 treatments (e.g. by employing wetlands or cascade systems; Gomes et al., 2017) for decades
490 after their initial disposal.

491 5 Conclusions

492 The co-disposed waste investigated in this study was composed of BOF steelmaking
493 slag containing dicalcium silicate, wüstite, dicalcium aluminoferrite and free lime; and
494 refractory oxides comprising corundum, periclase, graphite and high temperature
495 aluminosilicate. V was predominately associated with the dicalcium silicate and dicalcium
496 aluminoferrite phases. During leaching, alkalinity was produced by dissolution of free lime and
497 dicalcium silicate. Under air-excluded conditions high Ca concentrations and the inverse
498 relationship between Ca and V concentrations imposed by $\text{Ca}_3(\text{VO}_4)_2$ solubility limits
499 restricted V release to the leachate. Under aerated conditions in-gassing of CO_2 promoted
500 carbonation reactions and secondary carbonate formation. Leachate pH and Ca concentrations
501 were reduced and MgO hydration and dissolution was promoted leading to a switch from a Ca
502 to an Mg dominated leachate and precipitation of monohydrocalcite and dolomite. V
503 concentrations in leachate were higher under aerated conditions where formation of carbonate
504 minerals provides a sink for aqueous Ca following in-gassing of CO_2 . Under these conditions,
505 the inverse relationship imposed by $\text{Ca}_3(\text{VO}_4)_2$ solubility limits allows higher concentrations
506 of V to accumulate in leachate than those seen under air-excluded conditions. Therefore, when
507 considering long-term leaching behaviour, it is important that risk assessments consider the
508 expected in situ environmental chemistry of specific waste storage environments.

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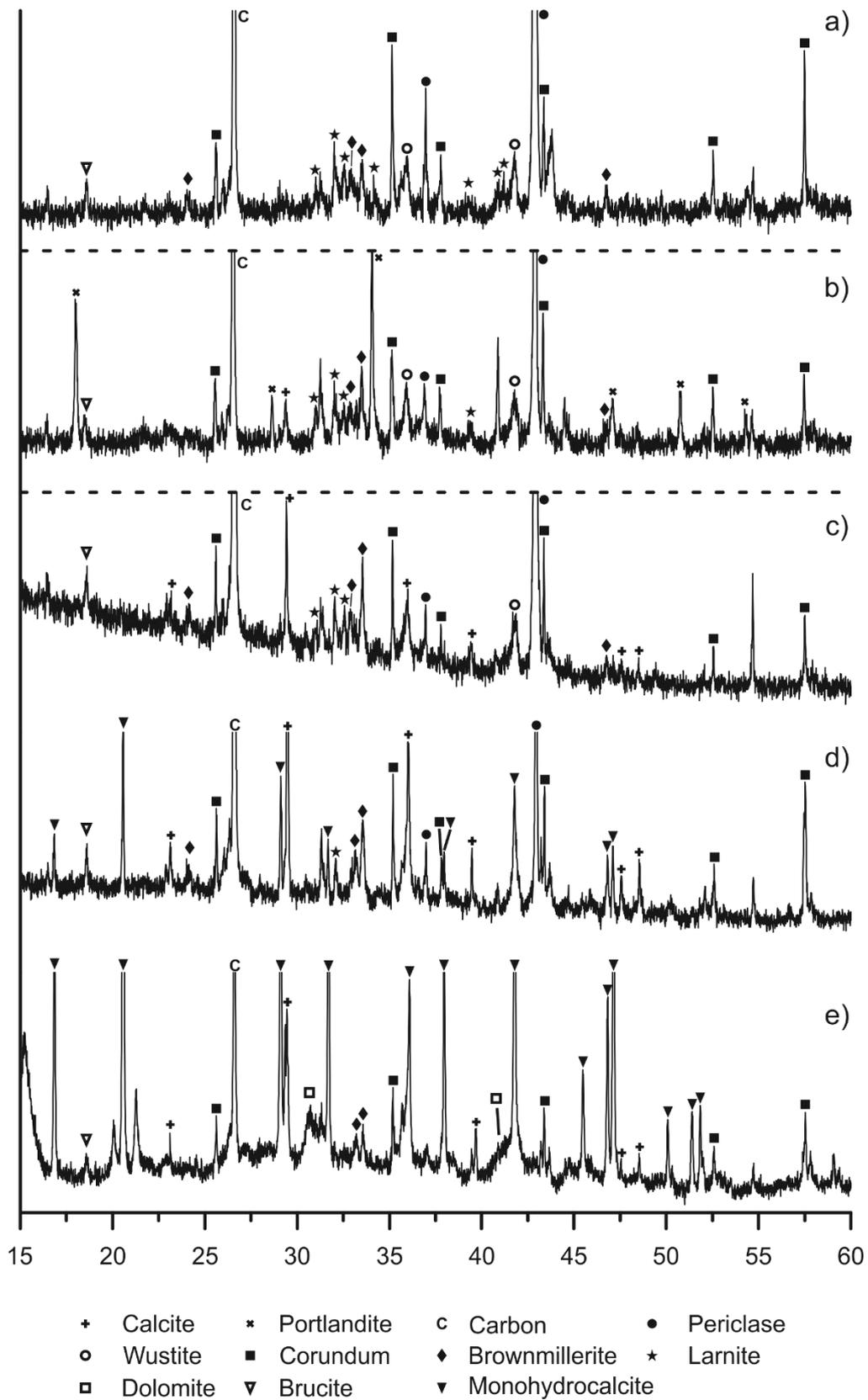
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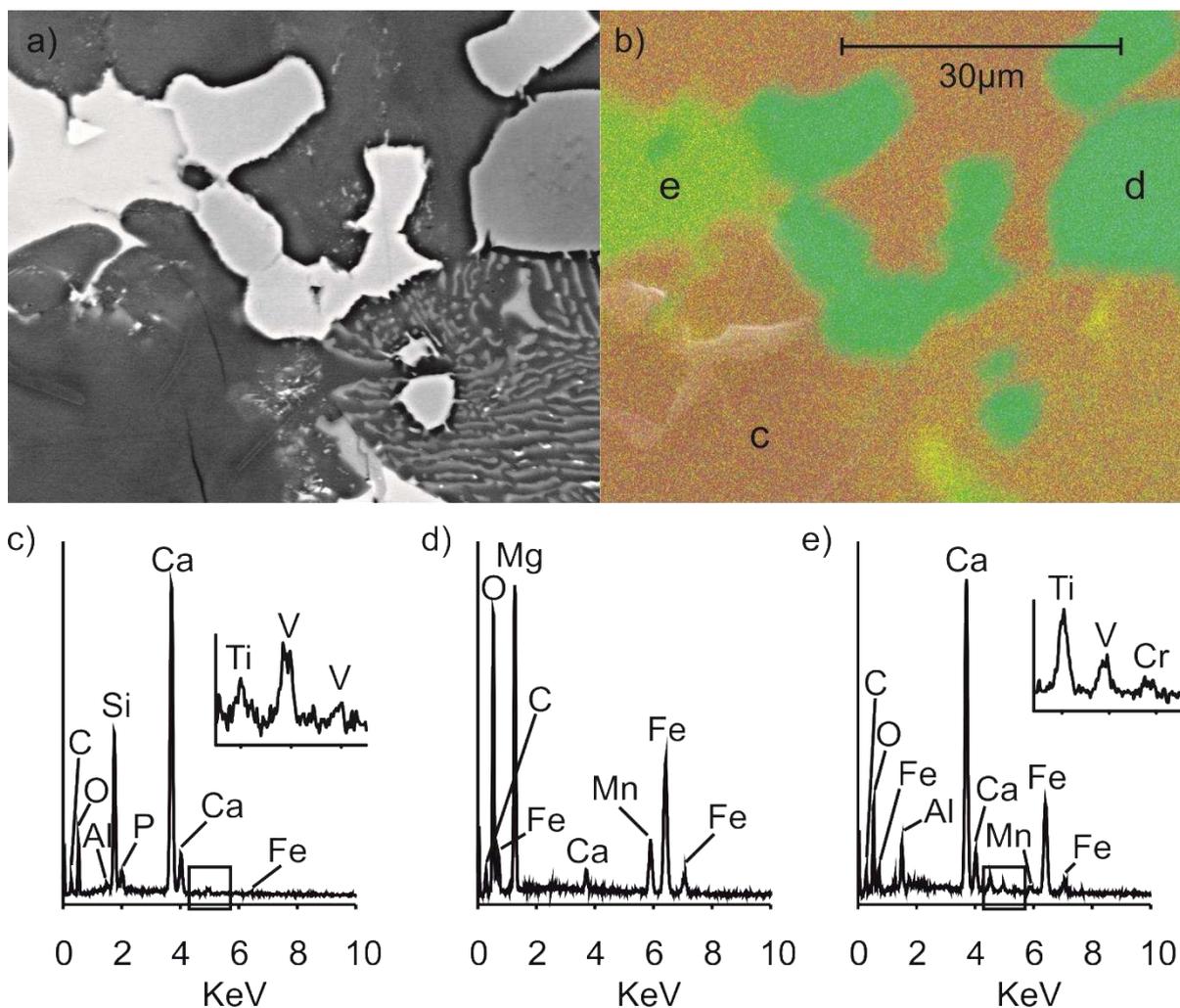
653 **Table 1.** Chemical composition of BOF steel slag samples from the Yarbrough Repository,
 654 Scunthorpe UK, and the powder sample used in leaching tests.

Major Elements Nominal Oxide wt%	BOF slag (Hobson et al., 2017)	Powder sample (this study)
CaO	40 ±5	19 ±2
FeO	32 ±9	16 ±2
SiO ₂	14 ±3	11 ±1
MgO	5.2 ±1	28 ±3
MnO	4.5 ±1	2.1 ±0.2
Al ₂ O ₃	1.2 ±0.4	18 ±2
P ₂ O ₅	1.3 ±0.4	0.87 ±0.09
V ₂ O ₅	0.81 ±0.24	0.33 ±0.03
TiO ₂	0.30 ±0.13	0.86 ±0.09
Cr ₂ O ₃	0.24 ±0.13	3.3 ±0.3
SO ₃	0.23 ±0.09	0.33 ±0.03
K ₂ O	n.d.	0.12 ±0.01
Na ₂ O	n.d.	0.18 ±0.02
SrO	n.d.	0.14 ±0.01
ZrO ₂	0.02 ±0.01	0.14 ±0.01
BaO	n.d.	0.01 ±0.01
NiO	0.02 ±0.01	n.d.
CuO	0.01±0.01	n.d.
ZnO	n.d.	0.02 ±0.01
PbO	n.d.	0.03 ±0.01
LOI	n.d.	0.96
TOTAL	98.9	100.9

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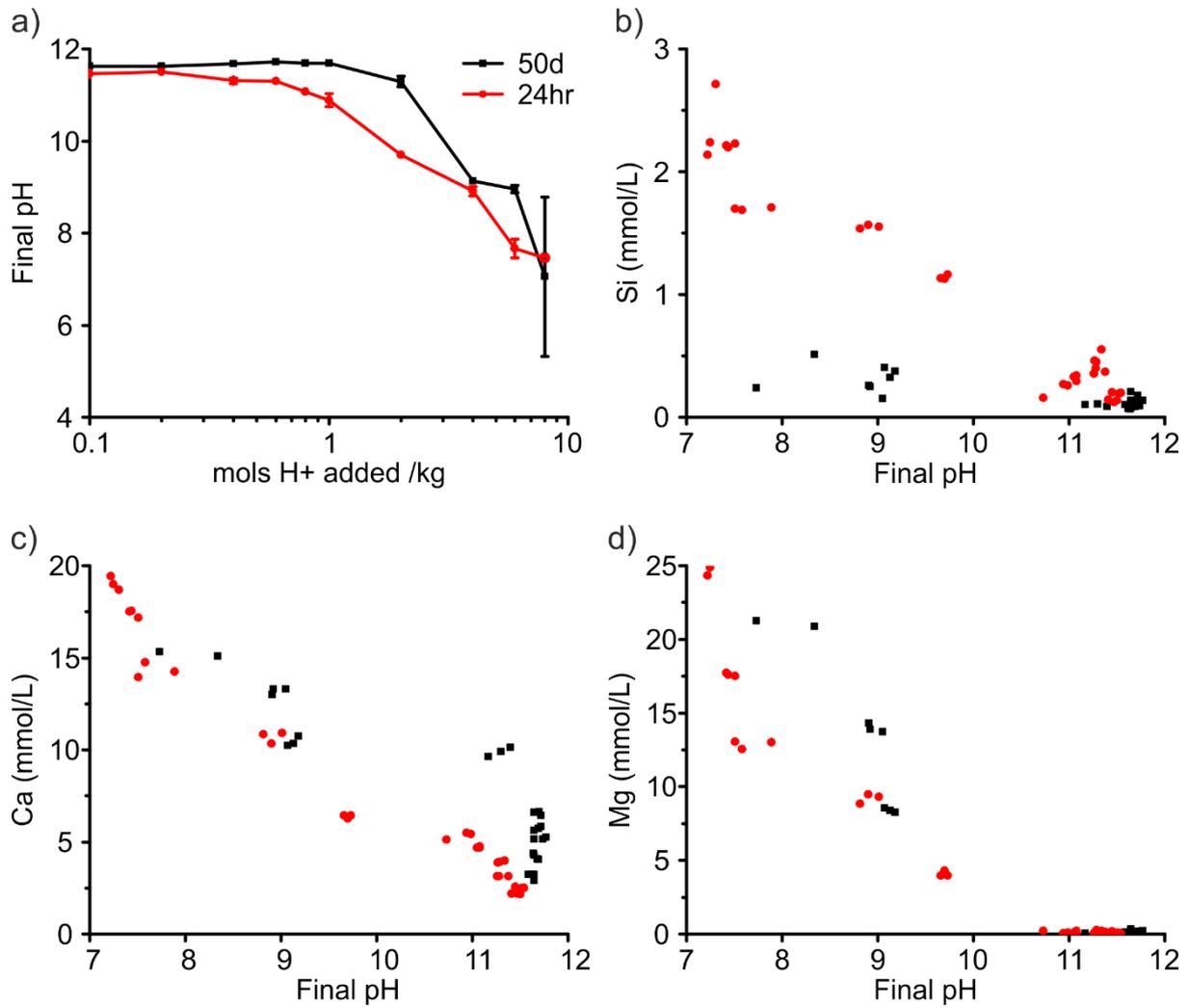


657 **Figure 1.** XRD patterns from (a) unweathered steelmaking waste, (b) steelmaking waste
658 leached for 50 days under air-excluded conditions, and steelmaking waste leached under
659 aerated conditions for (c) 1 hour, (d) 6 days and (e) 50 days



660

661 **Figure 2.** Backscattered electron micrograph (top left) with corresponding false colour element
 662 map (top right) and EDS spectra from c) dicalcium silicate, d) wüsite and e) dicalcium
 663 aluminoferrite phases. The laminae containing phase (bottom right) also contains discrete CaO
 664 laminae.
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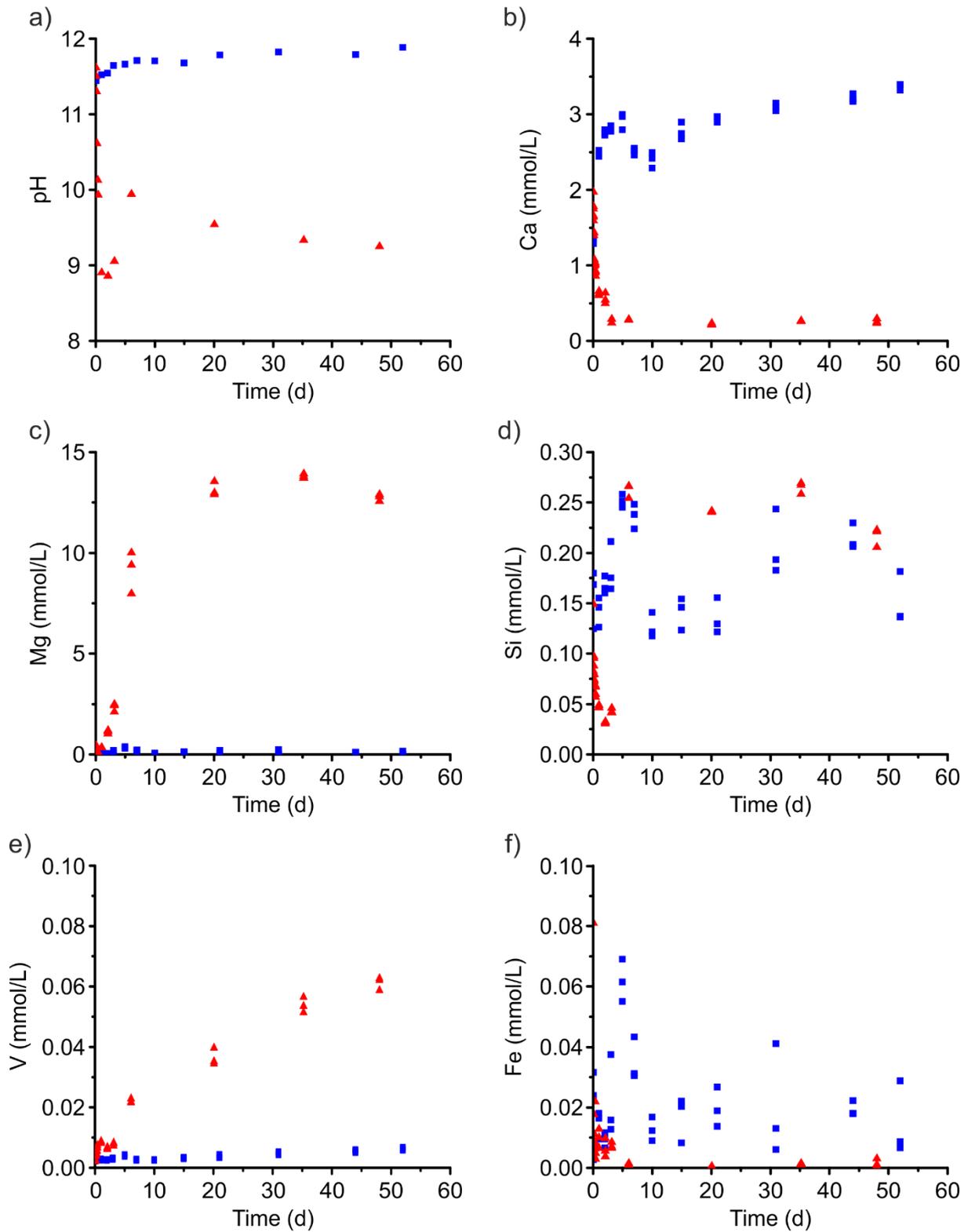
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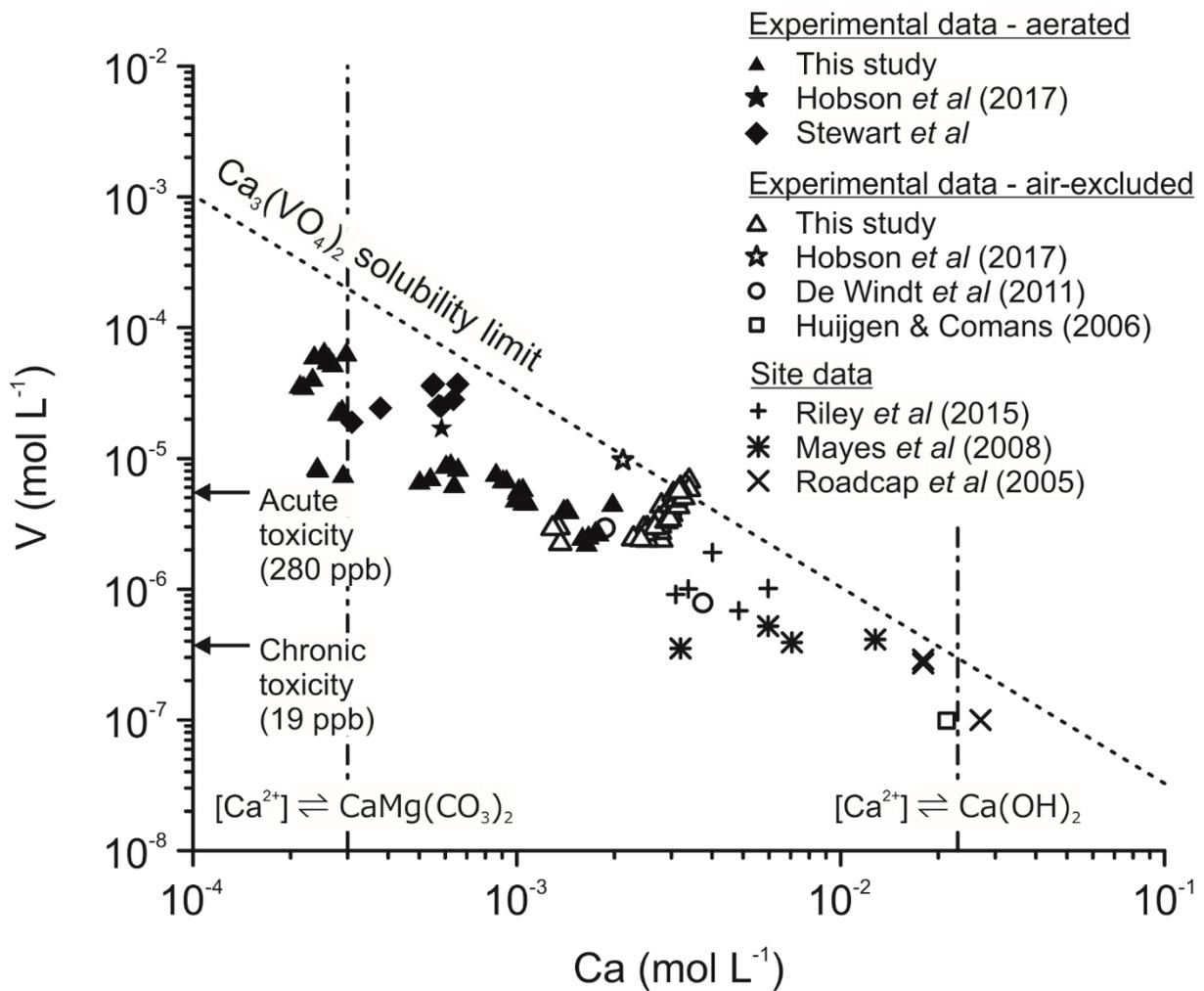
Fig. 3. Acid neutralisation results showing (a) final pH values after 24 hours (red circles) and 50 days (black squares) as a function of initial acid concentration (each point represents the mean value and error bars $\pm 1\sigma$ from triplicate samples) and (b)-(d) metal concentrations leached from waste over 24 hours and 50 days as a function of final pH. Initial test conditions in all cases were 10g waste L^{-1} .



673

674 **Figure 4.** Leachate composition during the aerated (red triangle) and air-excluded (blue square)
 675 batch leaching test on steelmaking waste (initial test conditions: 10g solid/L deionised water).
 676 Data from triplicate experiments shown separately.

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679 **Figure 5.** Plot of [V] versus [Ca] for selected experimental and site data (site data from Riley
 680 and Mayes (2015), Roadcap et al. (2005), Mayes et al. (2008)). The dashed line marks the
 681 solubility limits for Ca₃(VO₄)₂ at 20 °C (Allison et al., 1991). Data plotting below the solubility
 682 limit is undersaturated with respect to that phase. Vertical dashed lines indicate [Ca] in
 683 solutions in equilibrium with dolomite in contact with atmospheric CO₂, or with Ca(OH)₂,
 684 respectively (both at 20 °C). Horizontal arrows indicate acute and chronic freshwater toxicity
 685 guideline limits (Buchman, 2008) (Figure redrawn after Hobson et al. (2017)).